

# High Temperature Deformation Mechanisms in Ta<sub>2</sub>C

**Nicholas DeLeon<sup>1</sup>, Billie Wang<sup>1</sup>, Christopher R. Weinberger<sup>2</sup> and Gregory B. Thompson<sup>1</sup>**

<sup>1</sup> University of Alabama, Tuscaloosa

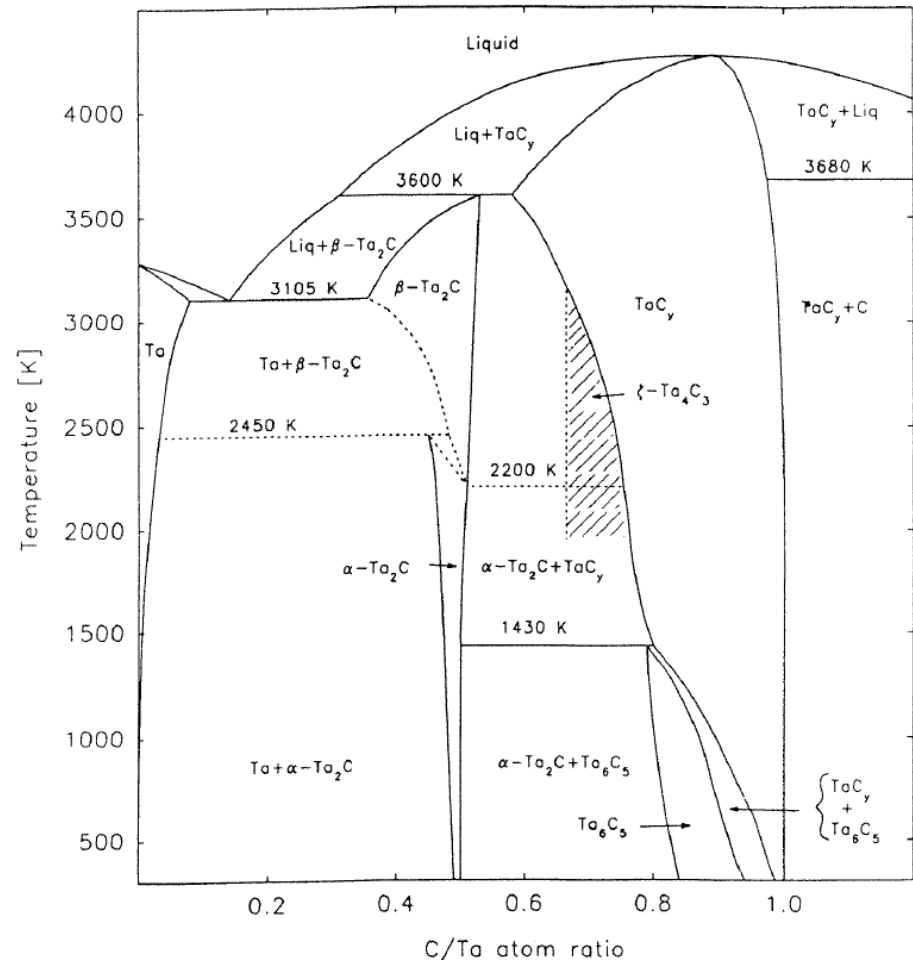
<sup>2</sup> Sandia National Laboratories



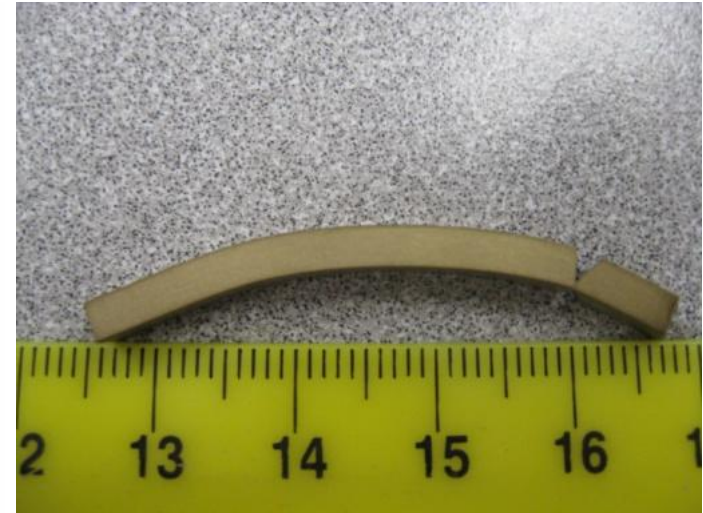
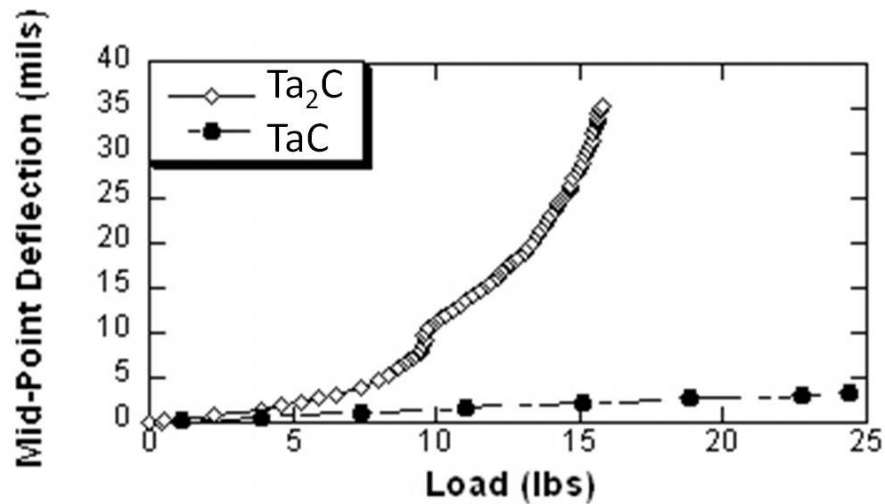
# Tantalum – Tantalum Carbon Phase Diagram

- Carbon – Diamond Sublimes at 3800C
- TaC melts at 3900 C
- Ta<sub>4</sub>HfC<sub>5</sub> melts at 4215 C

Tantalum Carbides are useful for high temperature structural applications.



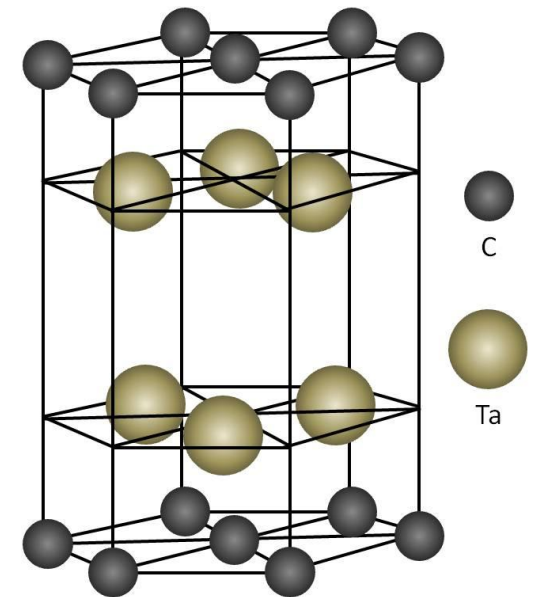
# Plasticity in Tantalum Carbides



Ta<sub>2</sub>C shows significantly more plastic deformation than TaC at ~1930K

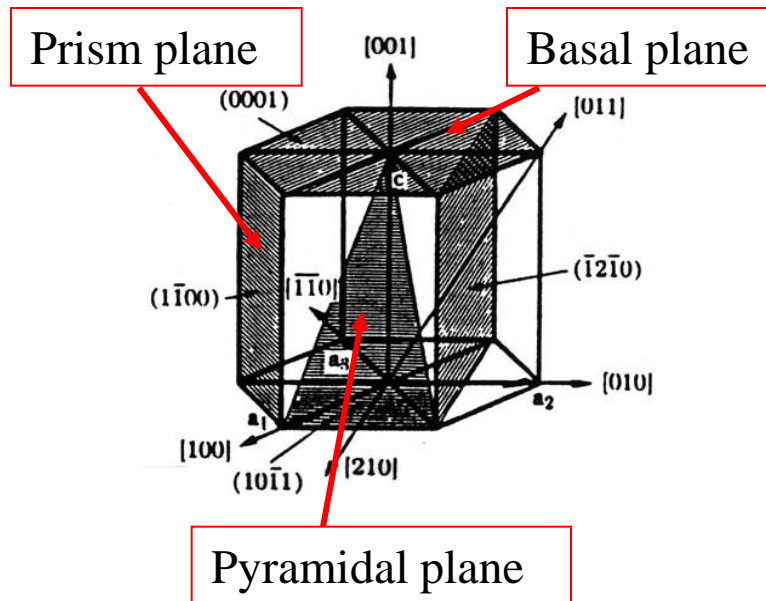
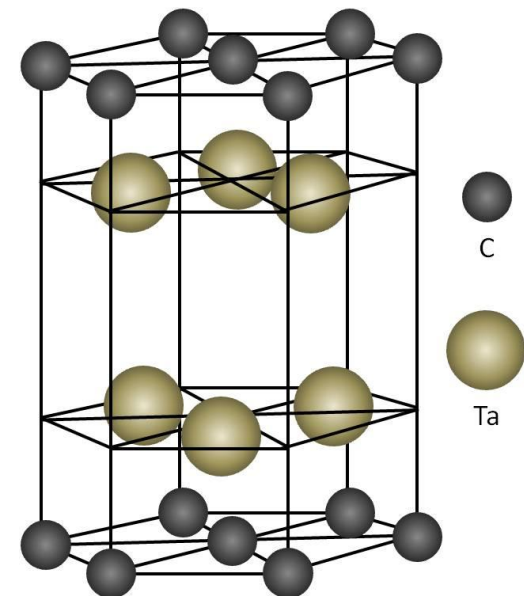
# Ta<sub>2</sub>C

- Structure: C6 Anti-CdI<sub>2</sub>
- Space Group: P-3m1
- Point Group is -3m : Trigonal
- Stacking on the basal plane:
  - AB $\gamma$ AB $\gamma$ AB $\gamma$
- Note that the stacking sequence involves the direct bonding of two tantalum layers.
- Exhibits mixed metallic-covalent bonding.



# Slip in Hexagonal & Trigonal Crystals

- Perfect slip via  $\frac{1}{3}\langle 11\bar{2}0 \rangle$  on  $\{0001\}$  (basal)
- Perfect slip via  $\frac{1}{3}\langle 11\bar{2}0 \rangle$  on  $\{10\bar{1}0\}$  (prism)
- Perfect slip via  $\frac{1}{3}\langle 11\bar{2}3 \rangle$  on  $\{11\bar{2}2\} \{10\bar{1}1\} \{hikl\}$  (pyramidal and banal)
- Partial slip via  $\frac{1}{3}\langle \bar{1}100 \rangle$  on  $\{0001\}$

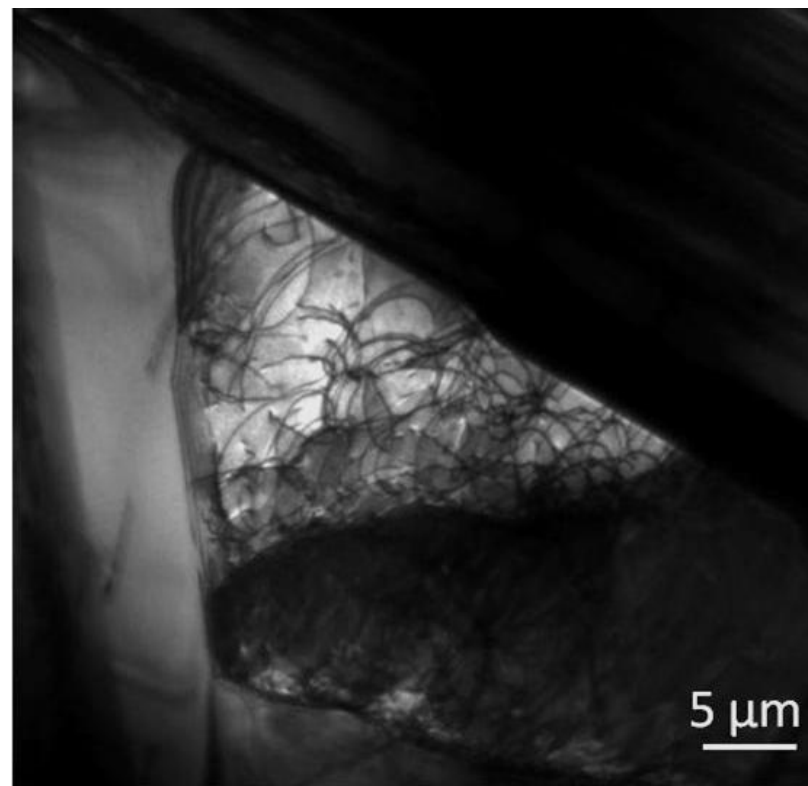
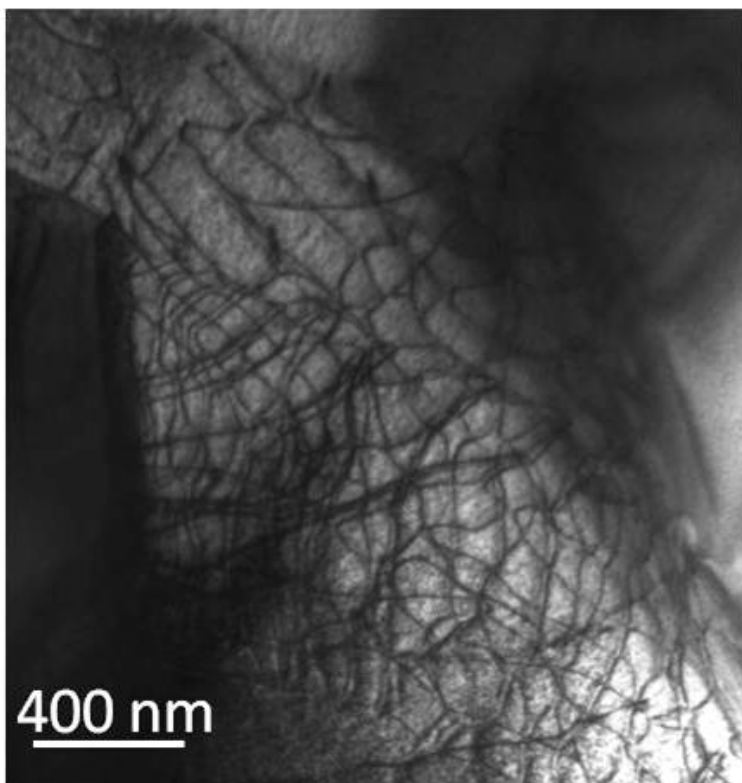




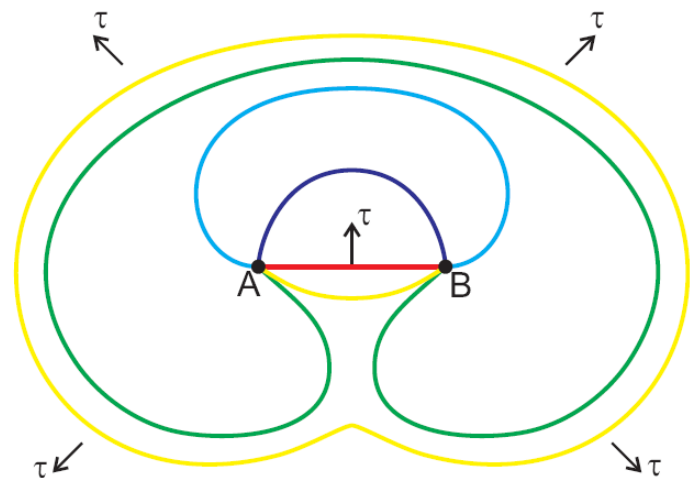
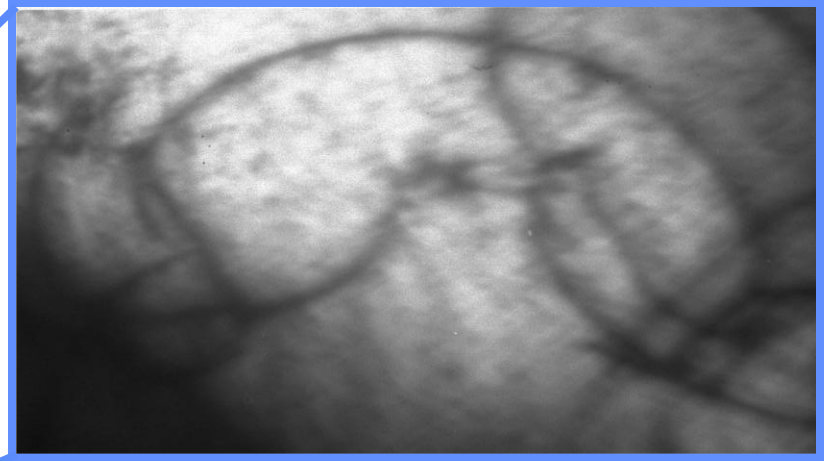
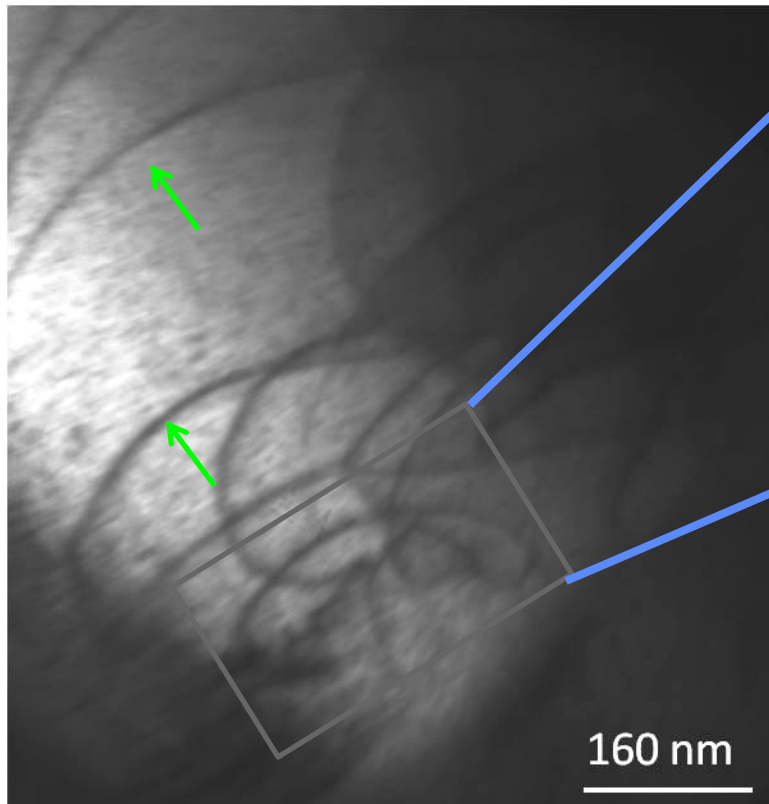


# Dislocations in Ta<sub>2</sub>C

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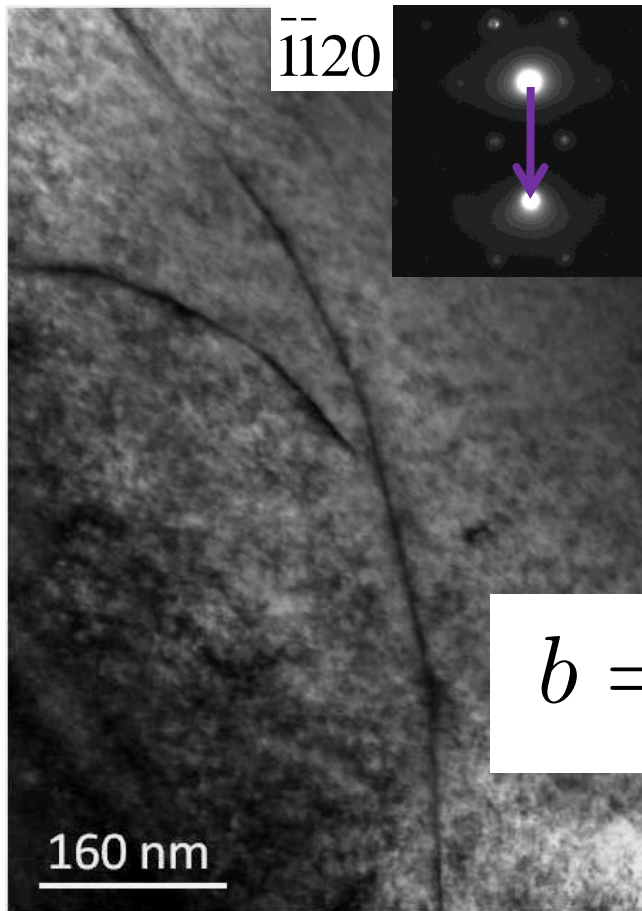


# More Dislocations

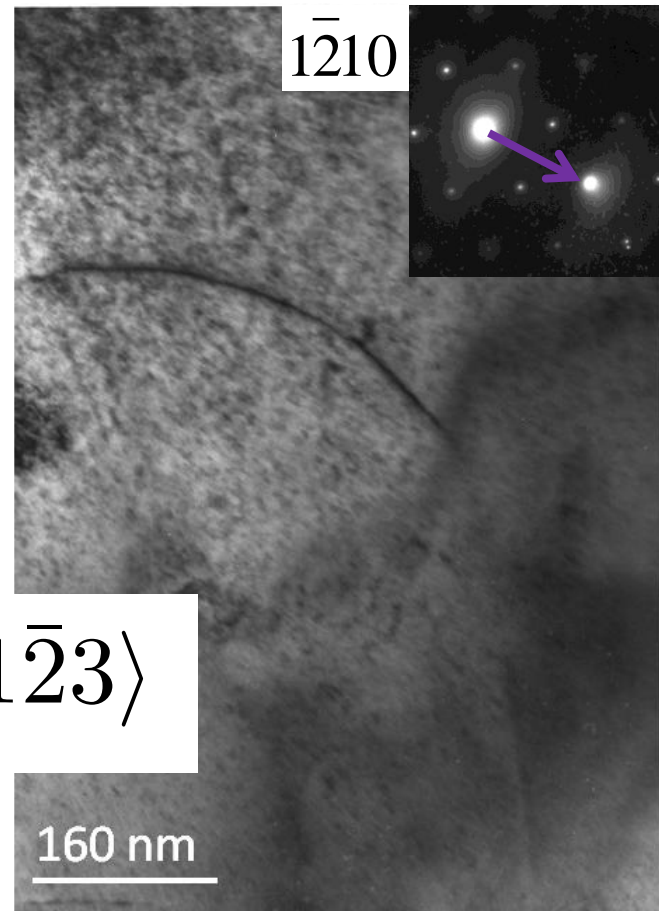


# Non Basal Slip

$$g \cdot b \neq 0$$



$$g \cdot b = 0$$

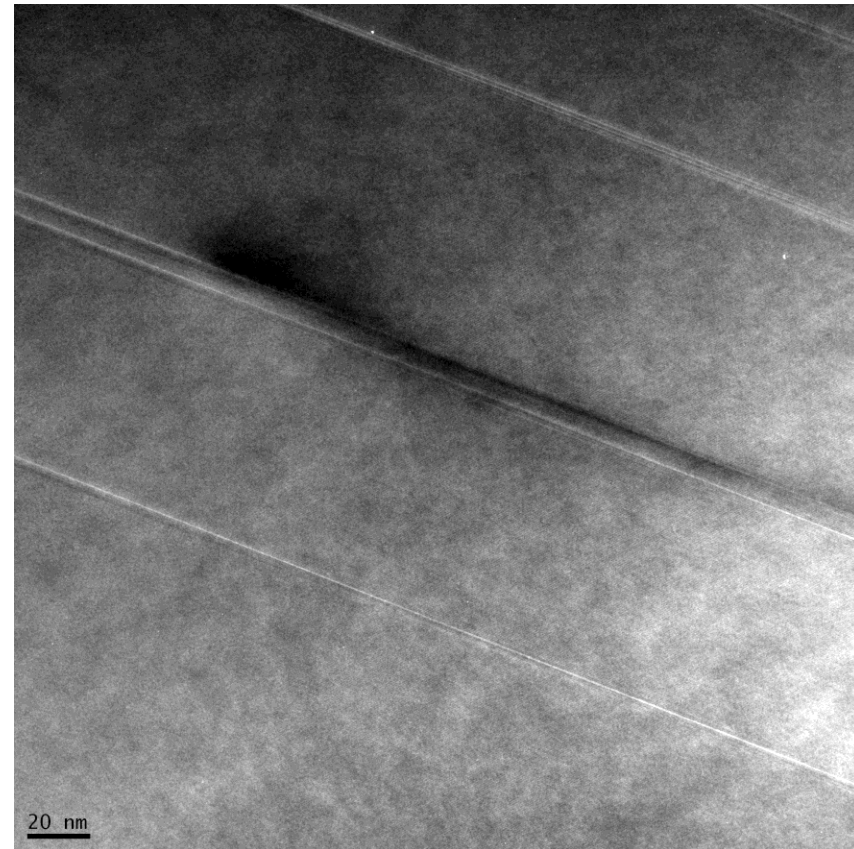
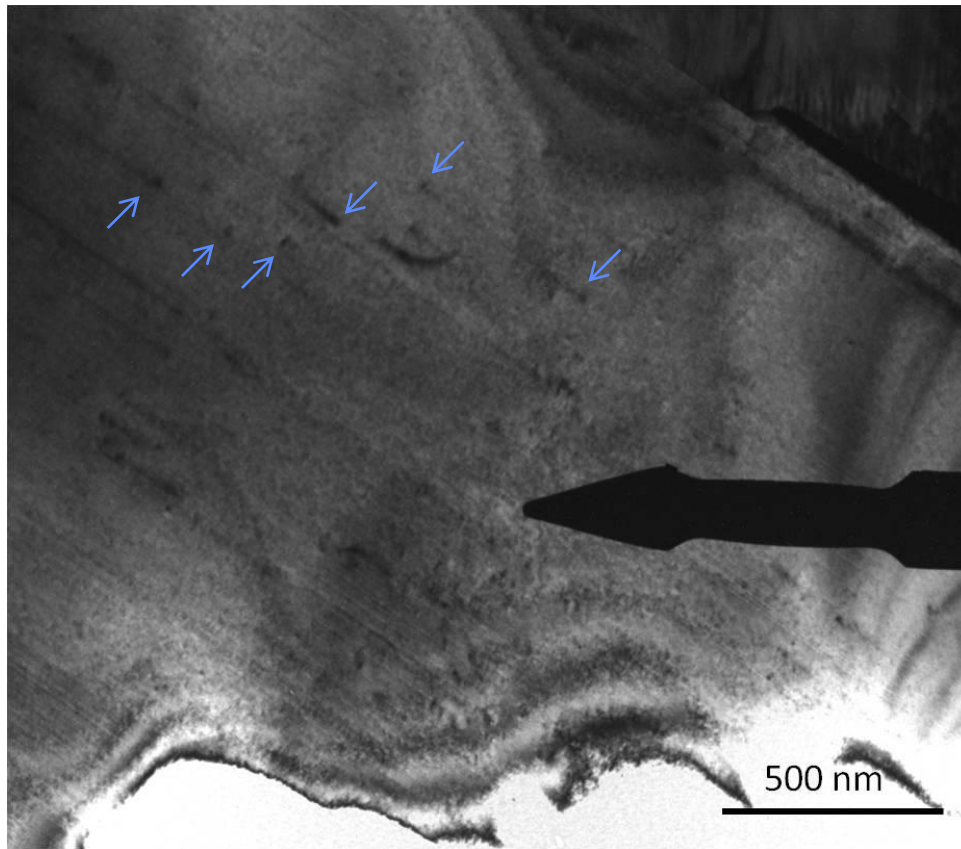


$$b = \frac{a}{3} \langle 11\bar{2}3 \rangle$$



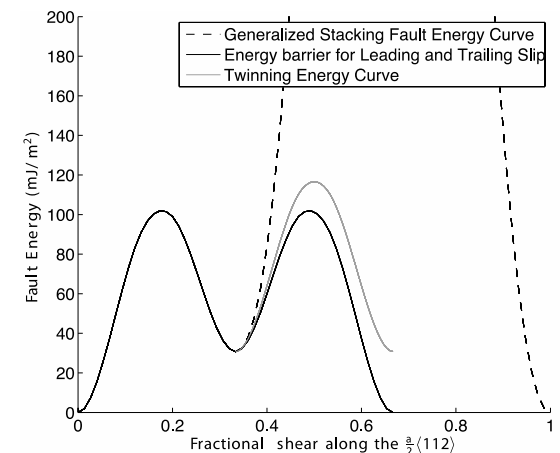
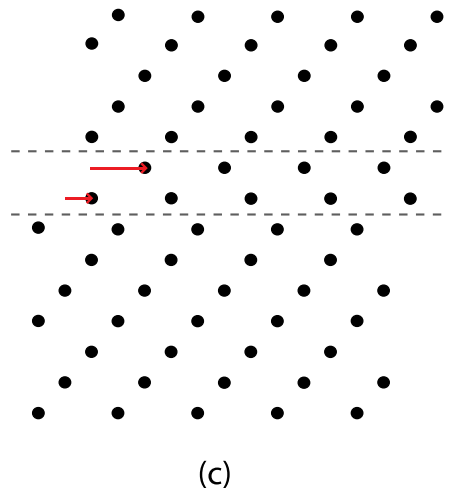
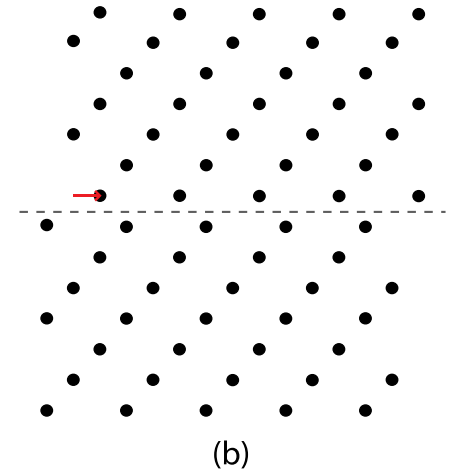
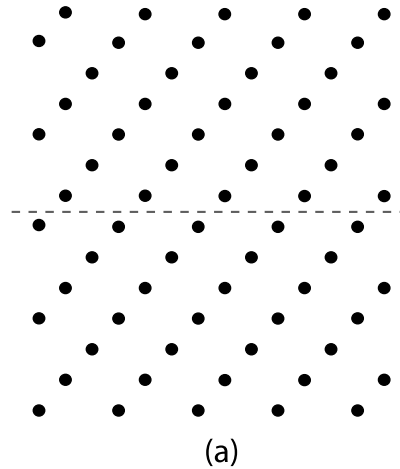


# Basal Slip



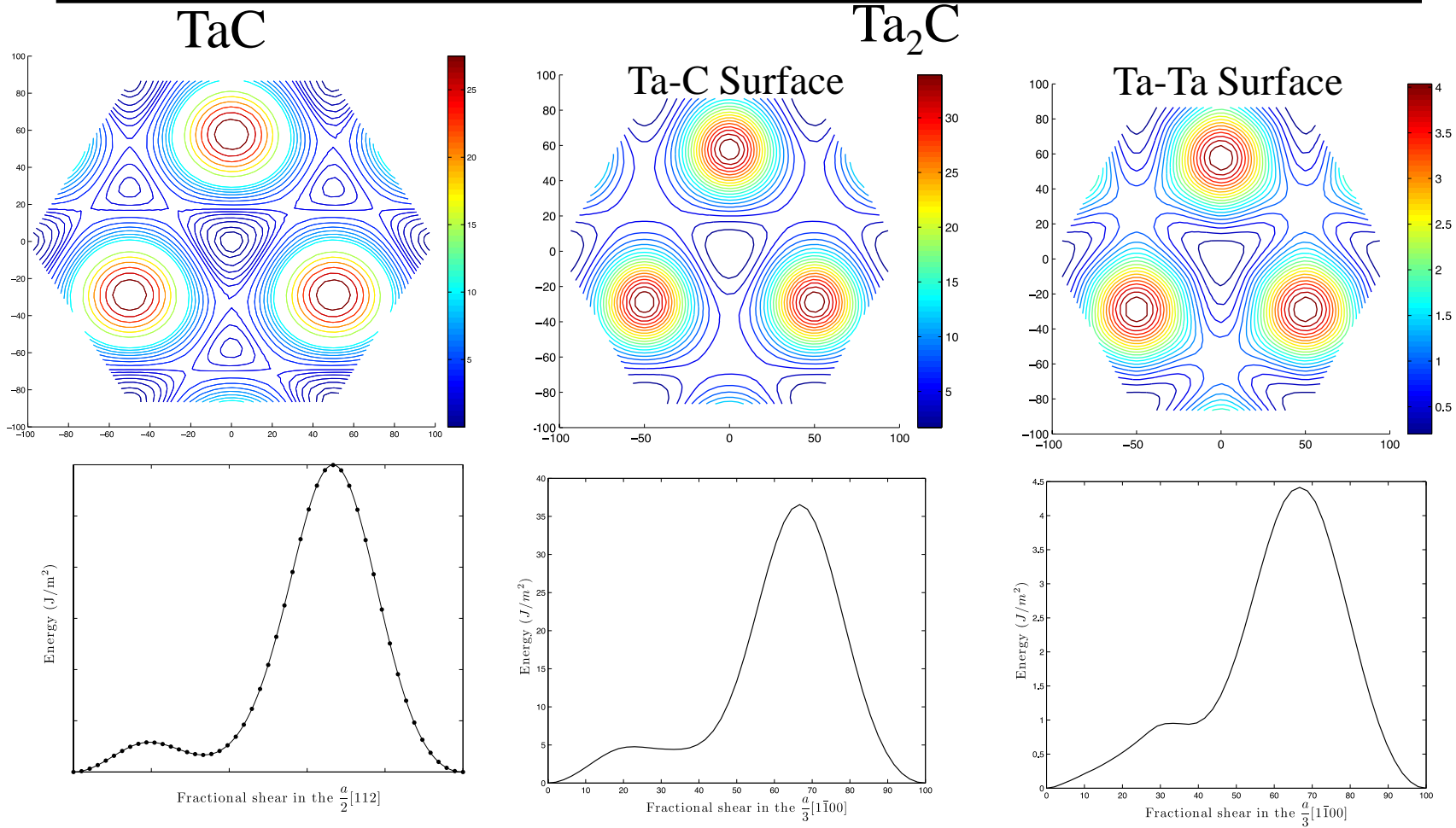
# Generalized Stacking Faults

- Start with the energy of a perfect crystal (a).
- Cut a plane and slide the crystal by an arbitrary amount  $x$  (b).
- The GSF curve represents the energy difference between the perfect registry and the faulted structure (d).



(d)

# Slip in TaC versus Ta<sub>2</sub>C



Slip between the closed-packed planes in Ta<sub>2</sub>C (Ta-Ta) is easier than that of closed packed planes in TaC.

# GSF Curves in Ta<sub>2</sub>C

Slip Plane	Slip Direction	Bonding	Maxima (J/m <sup>2</sup> )	Ideal Strength (GPa)
{0001}	<01-10>	Ta-Ta	0.95	6.7
{0001}	<01-10>	Ta-C	4.48	59
{0001}	<11-20>	Ta-Ta	1.89	63
{0001}	<11-20>	Ta-C	12.5	130
{10-10}	<-12-10>	Ta-Ta	4.5	40
{10-10}	<-12-10>	Ta-C	7.9	51
{10-10}	<0001>	Ta-Ta	16.8	160
{10-10}	<0001>	Ta-C	7.8	120
{10-11}	<-12-10>	Ta-Ta	3.2	63
{10-11}	<-12-10>	Ta-C	7.5	81
{10-1-1}	<-12-10>	Ta-Ta	11.9	120
{10-1-1}	<-12-10>	Ta-C	8.0	81
{10-11}	<-2113>	Ta-Ta	3.1	27
{10-11}	<-2113>	Ta-C	4.9	54
{10-1-1}	<-2113>	Ta-Ta	76	750
{10-1-1}	<-2113>	Ta-C	54	54

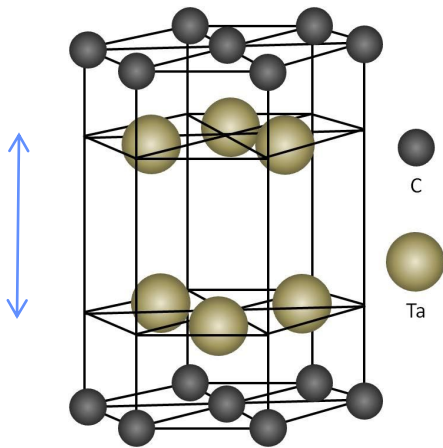
← Most favorable slip system.

GSF curves show that the most favorable slip systems agree with experimental observations  
<01-10>{0001} and  
<-2113>{10-11}

← Second most favorable slip system.

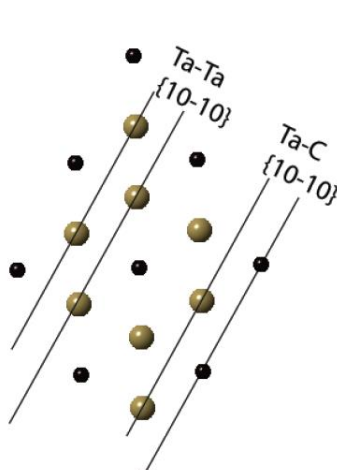
Slip is generally favored between Ta-Ta bonds

# Inter-planar Spacing

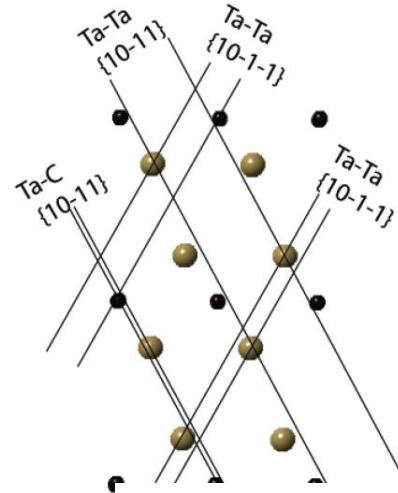


Basal Planes

The  $\{0001\}$  plane  
between Ta-Ta atoms  
has widest spacing



Prism Planes



Pyramidal Planes

The  $\{10-11\}$  plane  
between Ta-Ta atoms has  
second widest spacing

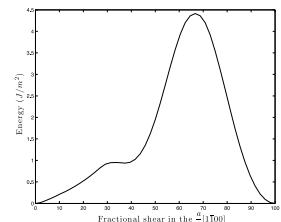
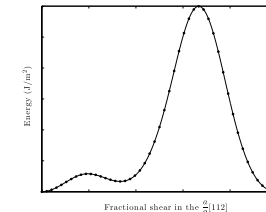
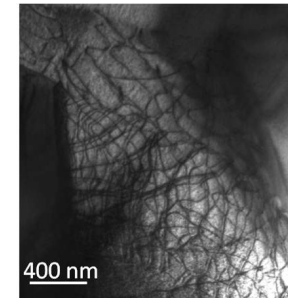
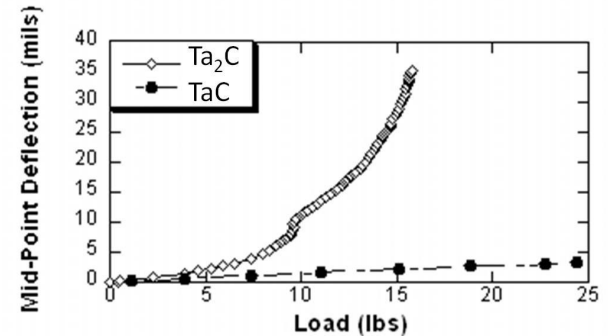
There is a strong positive correlation between GSF energies and inter-planar spacing.  
Inverse of the inter-planar spacing and max GSF energy are linearly related.

**Slip in these mixed-metallic covalent bonded materials occurs between the most widely spaced planes.**



# Conclusions

- Ta<sub>2</sub>C exhibits more plastic deformation than TaC at the same absolute temperature.
- Slip in Ta<sub>2</sub>C is dominated by basal slip, although pyramidal slip,  $\langle c+a \rangle$ , dislocations contribute.
- Slip on the basal plane occurs between the Ta-Ta bonds
- GSF curves show that slip is easier in Ta<sub>2</sub>C on the basal plane than  $\{111\}$  slip in TaC.
- Slip occurs between the most widely spaced planes.





# Acknowledgements

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