



9:00 - 9:20 a.m. 7/22/2012

# Direct Observation of Deformation and Fracture of Si Electrodes in the Nano-sized Lithium Ion Batteries

Xiaohua Liu

Center for Integrated Nanotechnologies (CINT)  
Sandia National Laboratories\*, Albuquerque, New Mexico

# ACKNOWLEDGMENT

## Financial Support

This work was supported by a Laboratory Directed Research and Development (LDRD) project at Sandia and by the Dept. of Energy Office of Basic Energy Science as part of an Energy Frontier Research Center (The NEES Center). This work was performed, in part, at the Sandia-Los Alamos Center for Integrated Nanotechnologies (CINT).



## Collaborators

CINT, Sandia N.L.

Jian Yu Huang, Yang Liu

John P. Sullivan, Nicholas S. Hudak, Arunkumar Subramanian

Hongyou Fan

Tom S. Picraux, Shadi A. Deyah, Jeong-Hyun Cho, Jinkyung Yoo

Ting Zhu, Shan Huang, Feifei Fan

Ju Li, Akihiro Kushima, Liang Qi

Scott X. Mao, He Zheng, Li Qiang Zhang, Jiang Wei Wang, Li Zhong

Sulin Zhang, Wen Tao Liang, Xu Huang

Chongmin Wang, Wu Xu

CINT, Los Alamos N.L.

Georgia Inst. Tech.

M.I.T.

Univ. Pittsburg

Penn State Univ.

Pacific Northwest N.L.

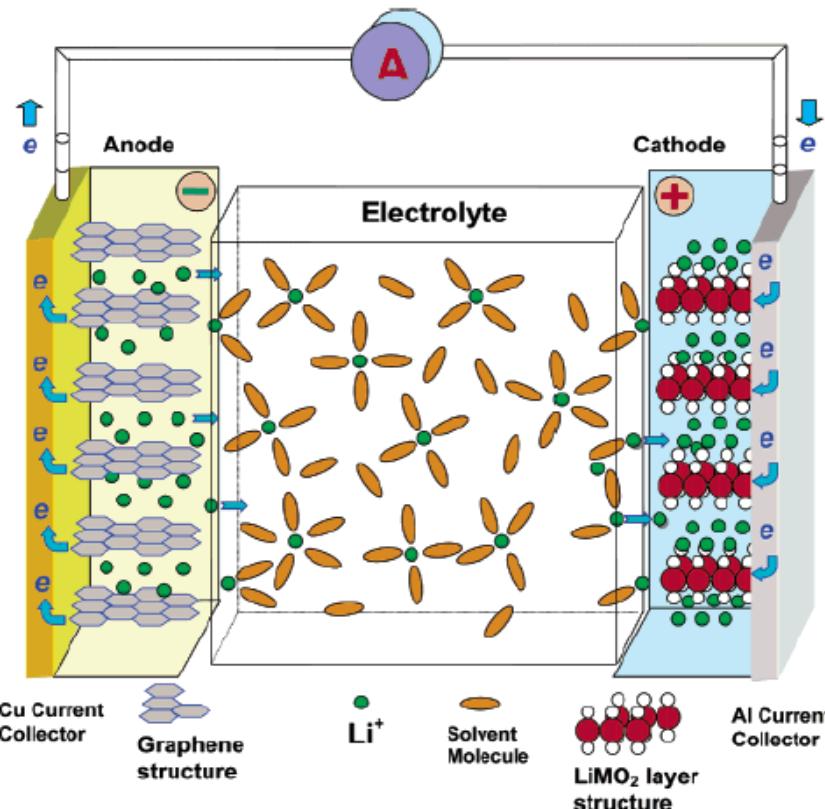
# OUTLINE

1. Motivation: Why Si? Why TEM?
2. Approach: TEM + nano-LIB
3. Results
  - 3.1 Anisotropic Swelling of Crystalline Si
  - 3.2 Size-Dependent Fracture of Si Nanoparticles
4. Conclusions

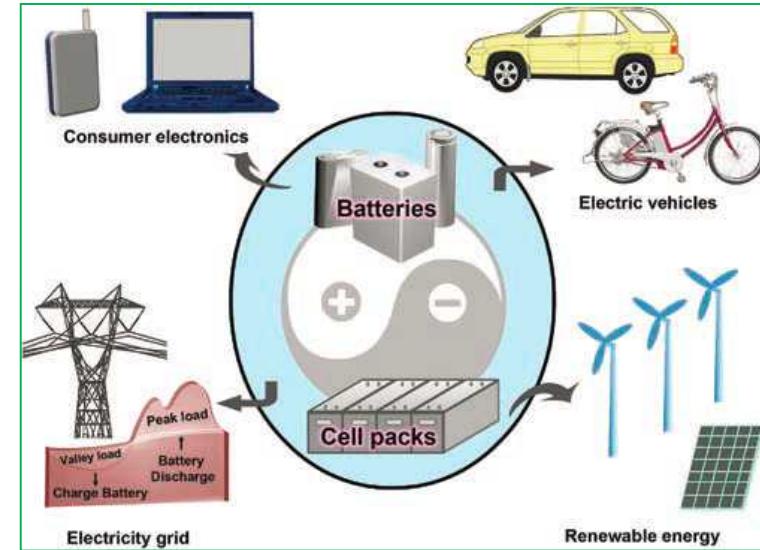
# 1. Motivation

## Lithium ion battery (LIB):

A complex system involves multiple phases, interfaces, charge and mass transport processes. Commercialized for > 30 years, having many applications, but not well understood.



A LIB during discharging



Prevailing and potential applications

## 1. Motivation

Global efforts are ongoing to improve LIB performance, targeting higher energy/power density, higher safety, and longer durability.

Mechanical degradation is one of the major problems of those high-energy LIBs under development.



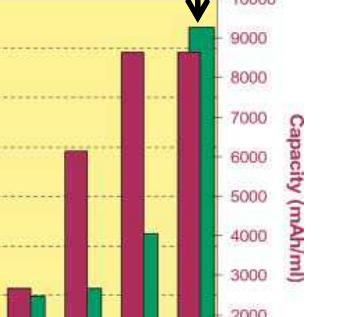
372 mAh/g



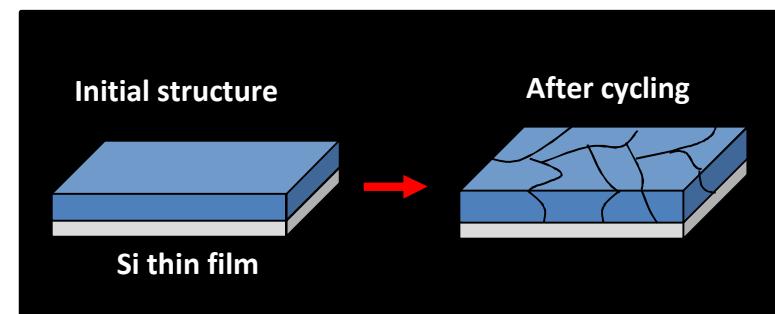
4200 mAh/g



Capacity of various anode materials for LIBs



Si electrodes crack and fracture after only a few cycles. There is an urgent need of understanding the mechanical responses upon electrochemical reactions.



# 1. Motivation

Large volumetric expansion is intrinsic to the high-energy anodes such as Si and Ge. But how does it actually occur accompanying the electrochemical reactions?

Nano-battery + TEM

Microscopically: ??

How is fracture initiated?  
Developed? Finalized?

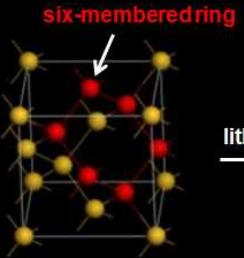
(length scale:  $\text{\AA} \sim \text{nm} \sim \mu\text{m}$ )

Initial structure

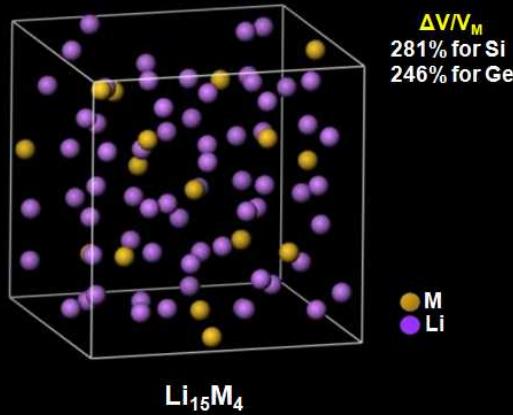
Si thin film

After cycling

Macroscopically: Crack and fracture  
(length scale: mm and up)



lithiation



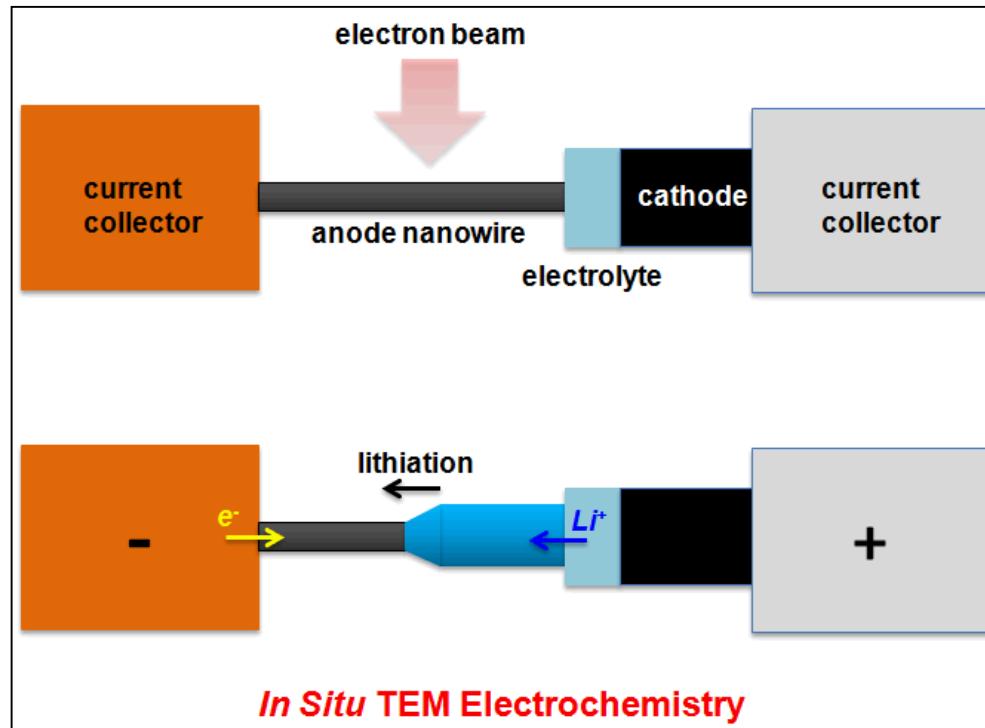
(M = Si, Ge)

Atomically: intrinsic volumetric changes and  
dilution of Si atoms by Li (length scale: sub  $\text{\AA} \sim 1 \text{ nm}$ )

## 2. Approach

Building a **nano-battery** compatible for TEM observation

→ Understand deformation at the atomic to micrometer length scale



Schematic illustration



**FEI Tecnai™ F30 TEM**  
Image of TEM of the same model obtained from <http://lims.msl.angstrom.uu.se/WebForms/Equipment/EquipmentView.aspx?toolId=135>

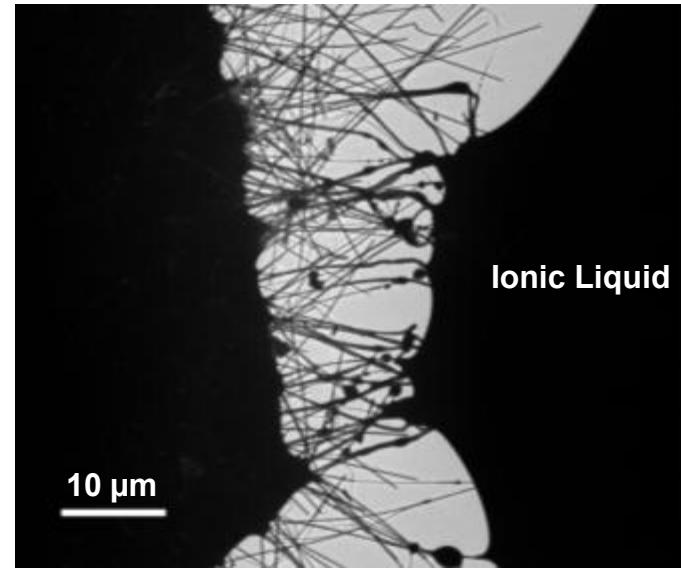
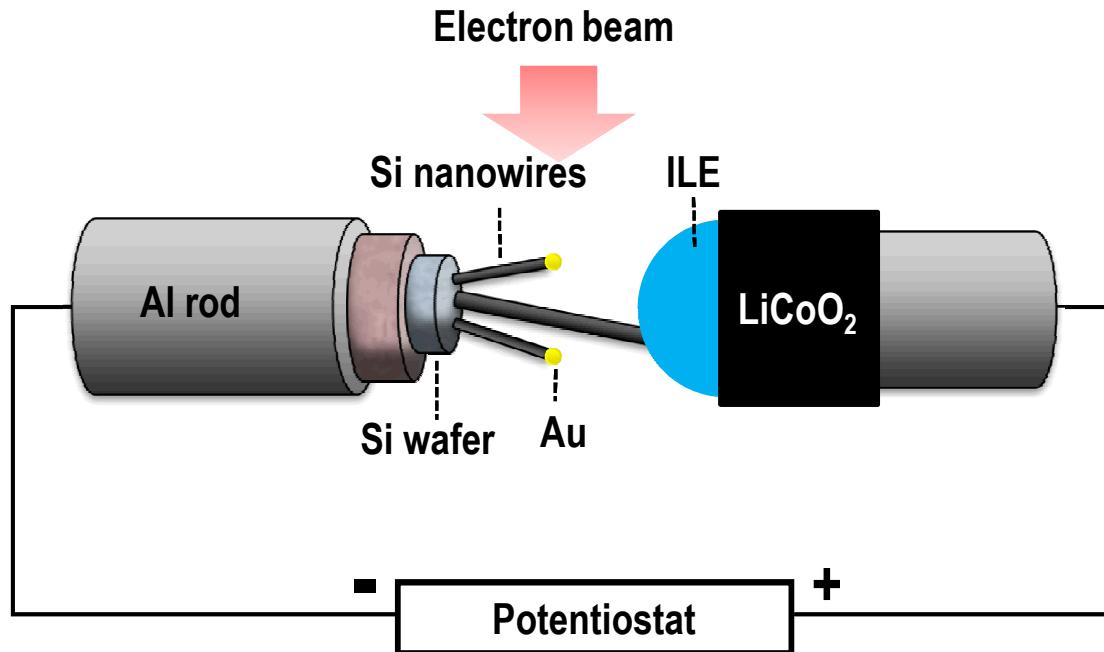


**Nanofactory™ STM-TEM holder**

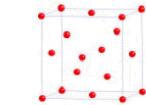
Instrumentation

# (1) Liquid Cell

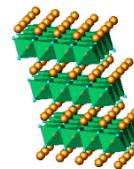
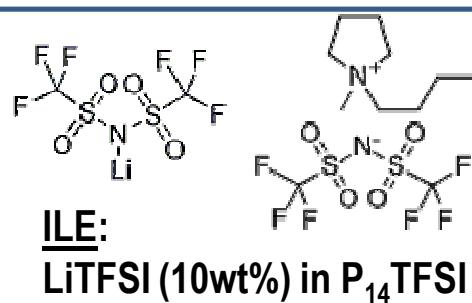
Using vacuum-compatible ionic liquid electrolyte (ILE)



A low-magnification view in TEM



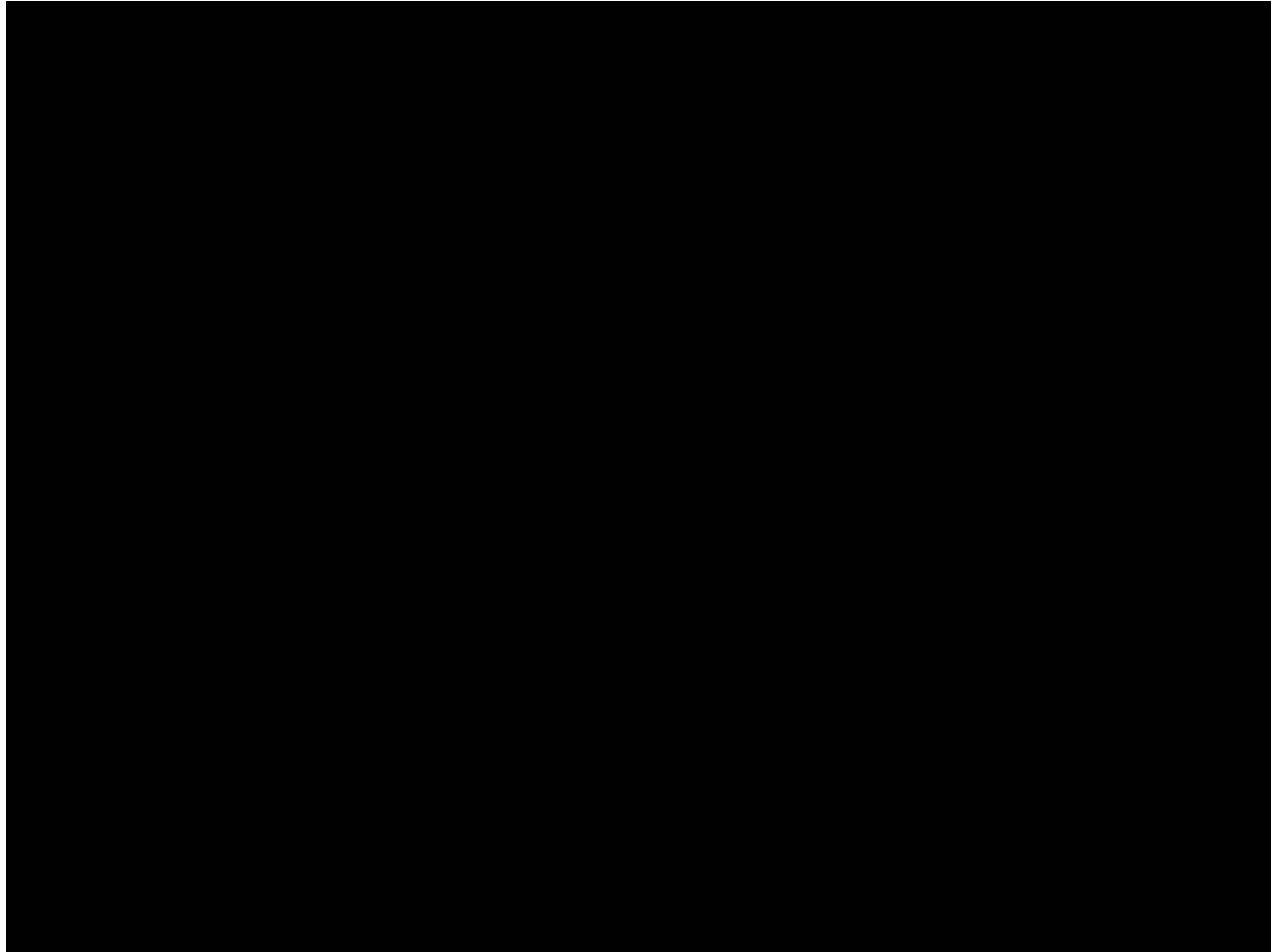
Anode: Si



Cathode: LiCoO<sub>2</sub>

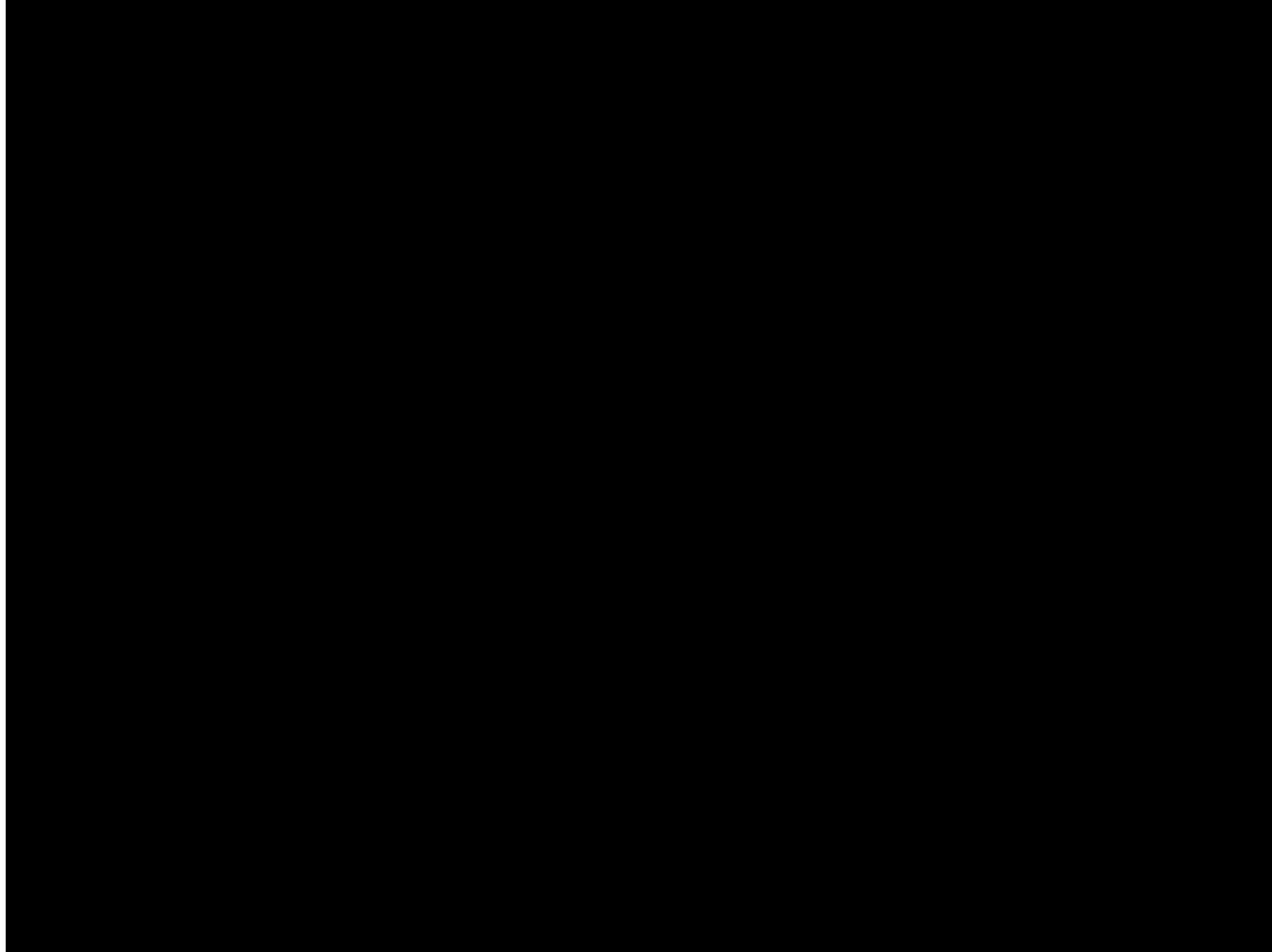
## (1) Liquid Cell

Lithiation of a  $\text{SnO}_2$  nanowire showing instant elongation and swelling



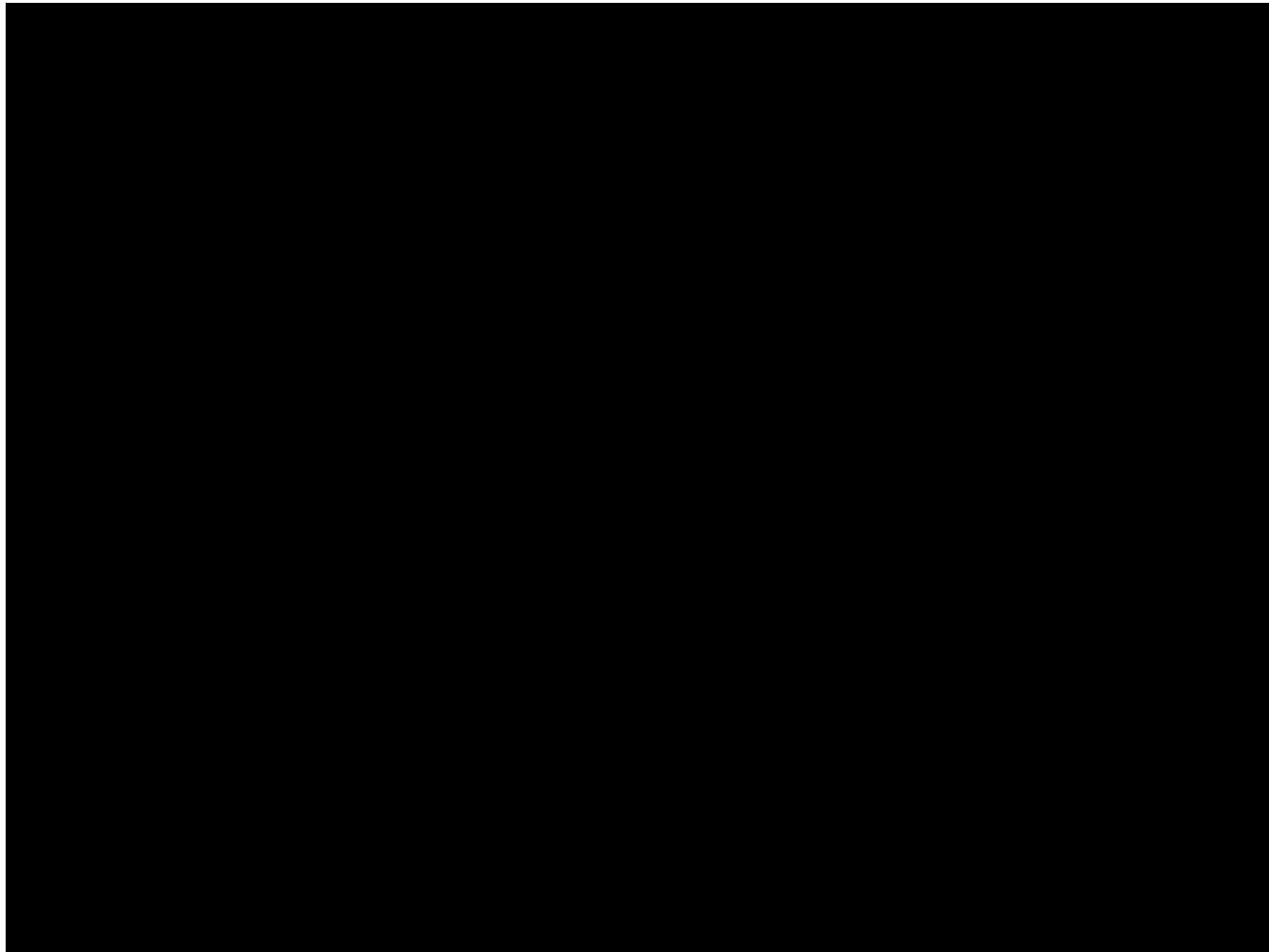
## (1) Liquid Cell

### Lithiation of $\text{SnO}_2$ - dislocation clouds and amorphization



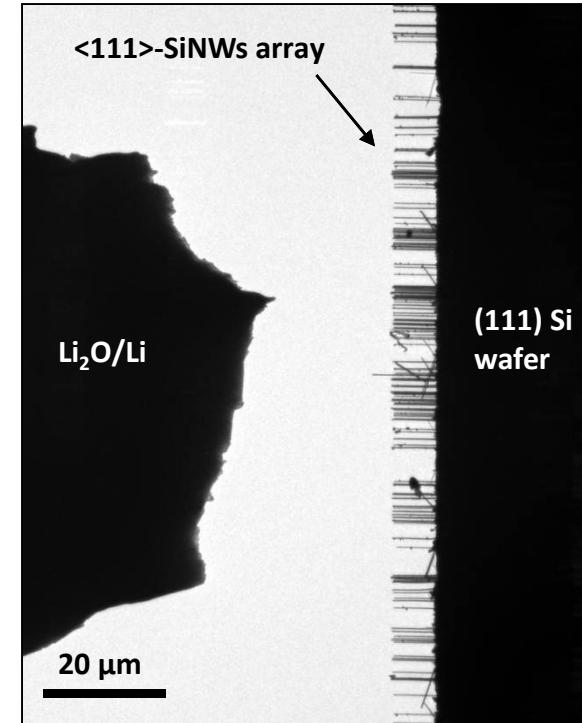
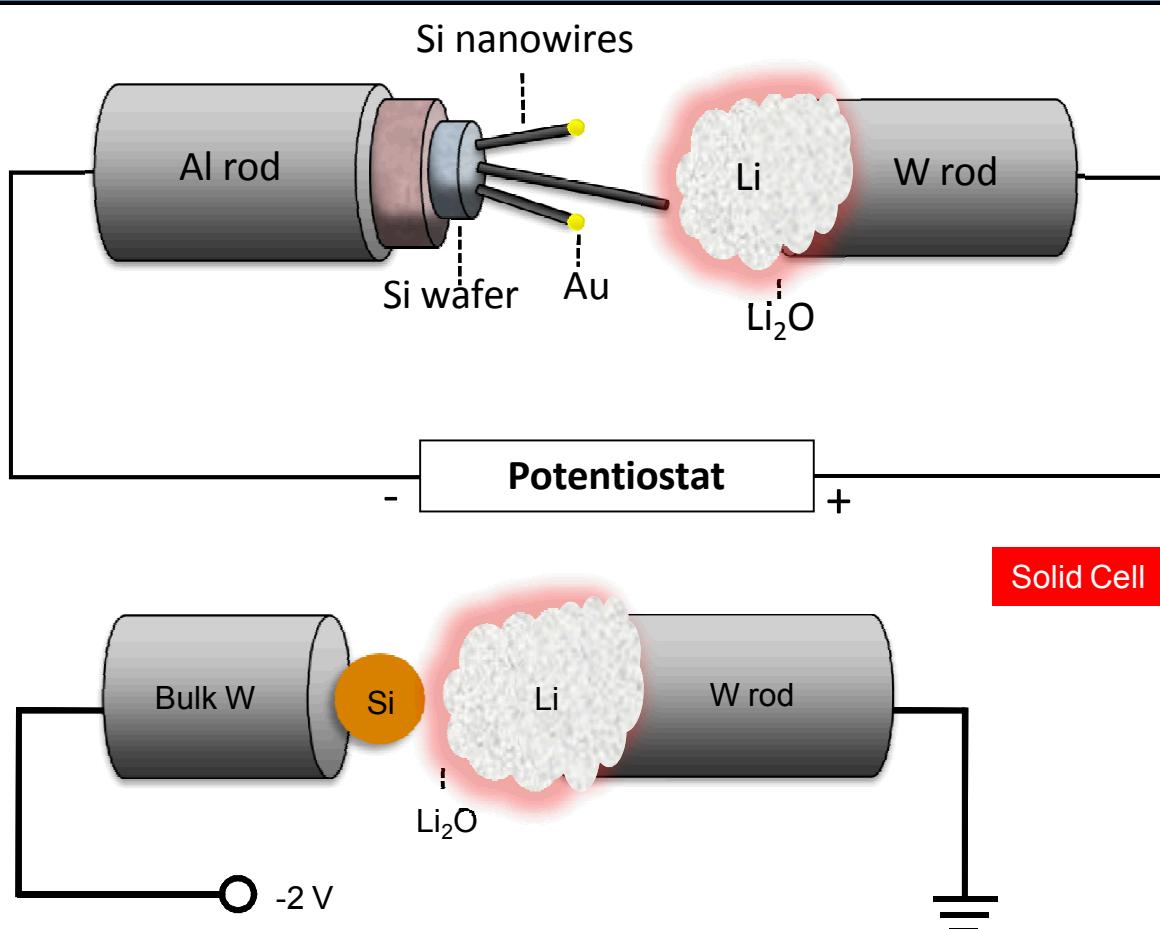
## (1) Liquid Cell

Ultrafast charging of a phosphorus-doped and carbon-coated Si nanowire



## (2) Solid cell based on $\text{Li}_2\text{O}/\text{Li}$

Half cell; Li counter-electrode;  $\text{Li}_2\text{O}$  solid-state electrolyte



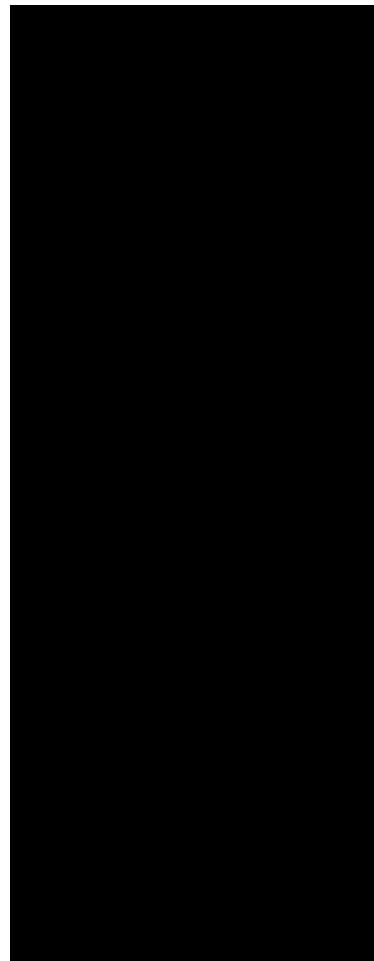
A low-magnification view in TEM

Advantages over the “liquid cell” configuration:

- (1) observation from the beginning;
- (2) sample size down to a few nanometers.

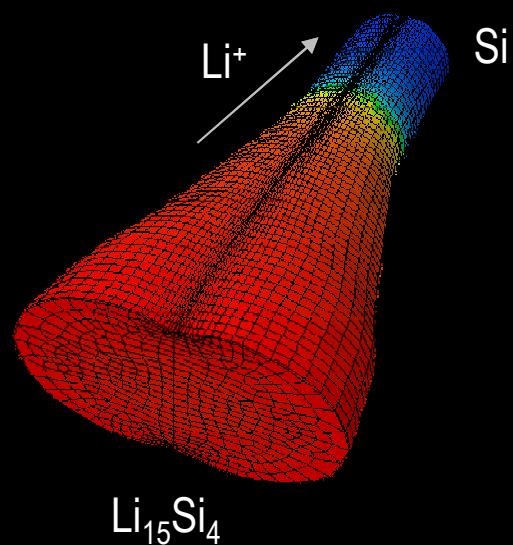
## (2) Solid Cell

### Lithiation of a crystalline Si nanowire



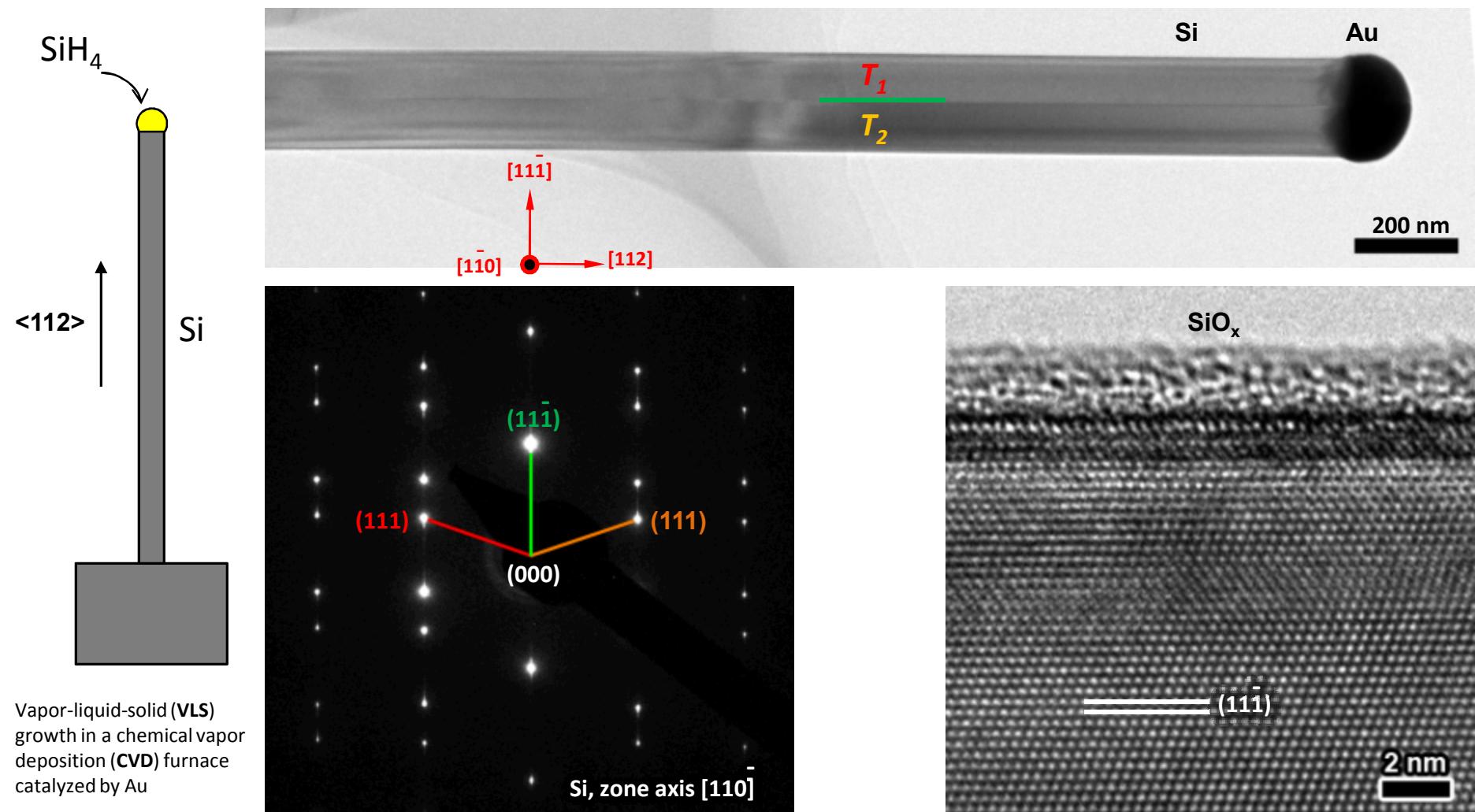
## Part 3.1

### Anisotropic Deformation of Crystalline Si during Lithiation



# Pristine <112>-Si nanowire

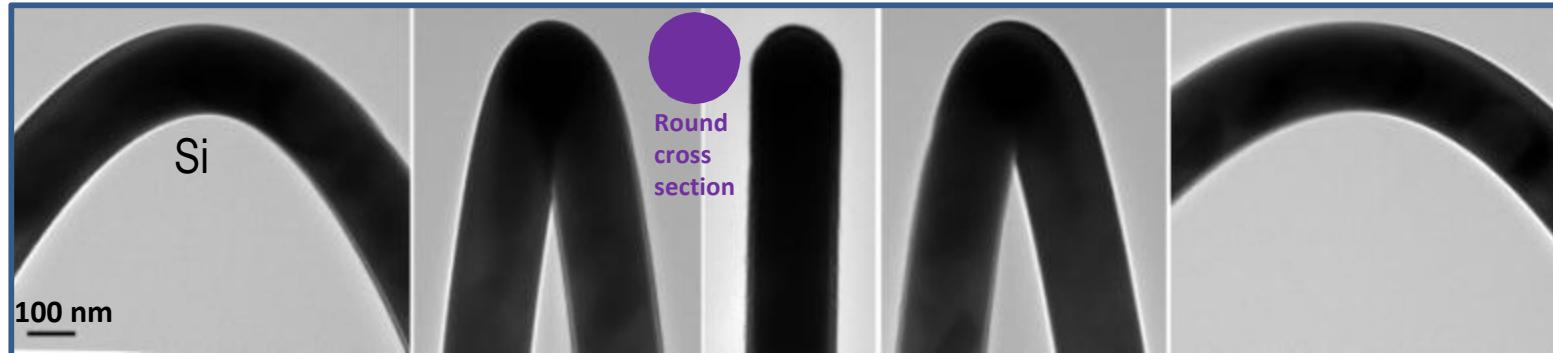
Perfect cylinder shape defined by the liquid droplet of Au catalyst  
Straight Si nanowires with a few twinning defects and a thin  $\text{SiO}_x$  layer



# Dramatic shape change after lithiation

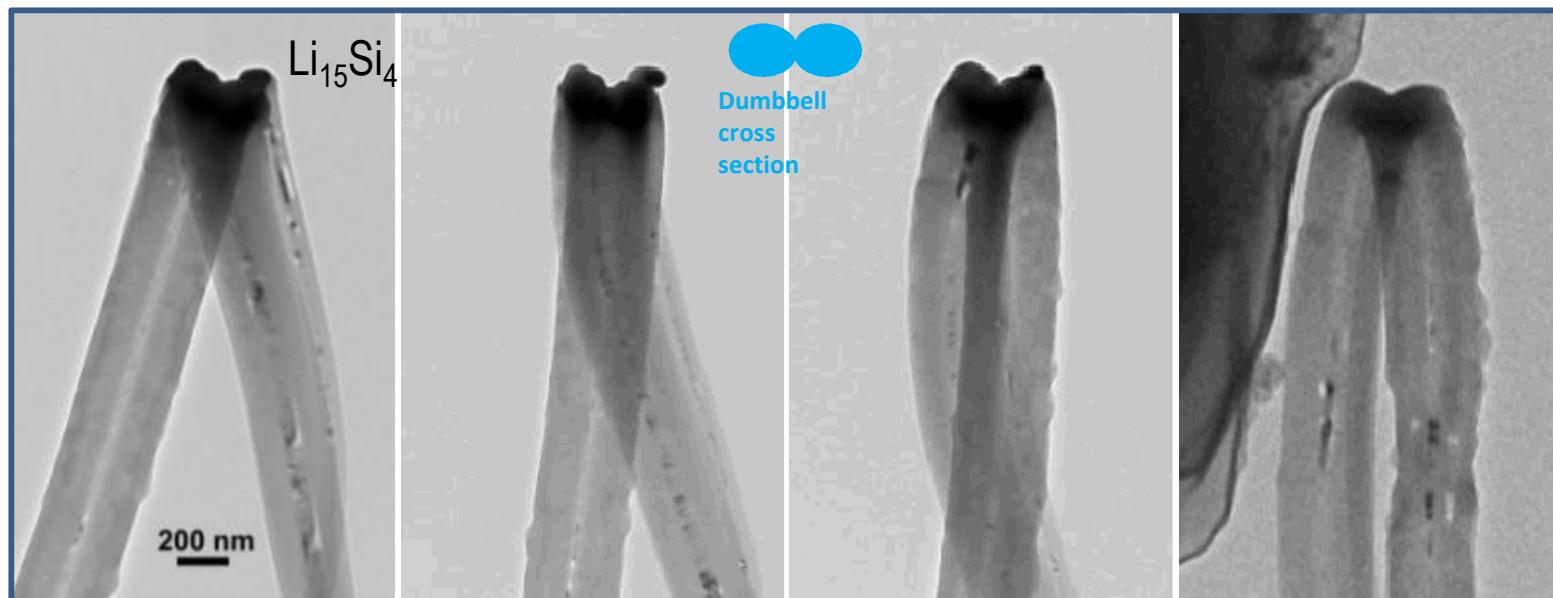
Perfect cylinder → dumbbell cross section

(a)



pristine

(b)

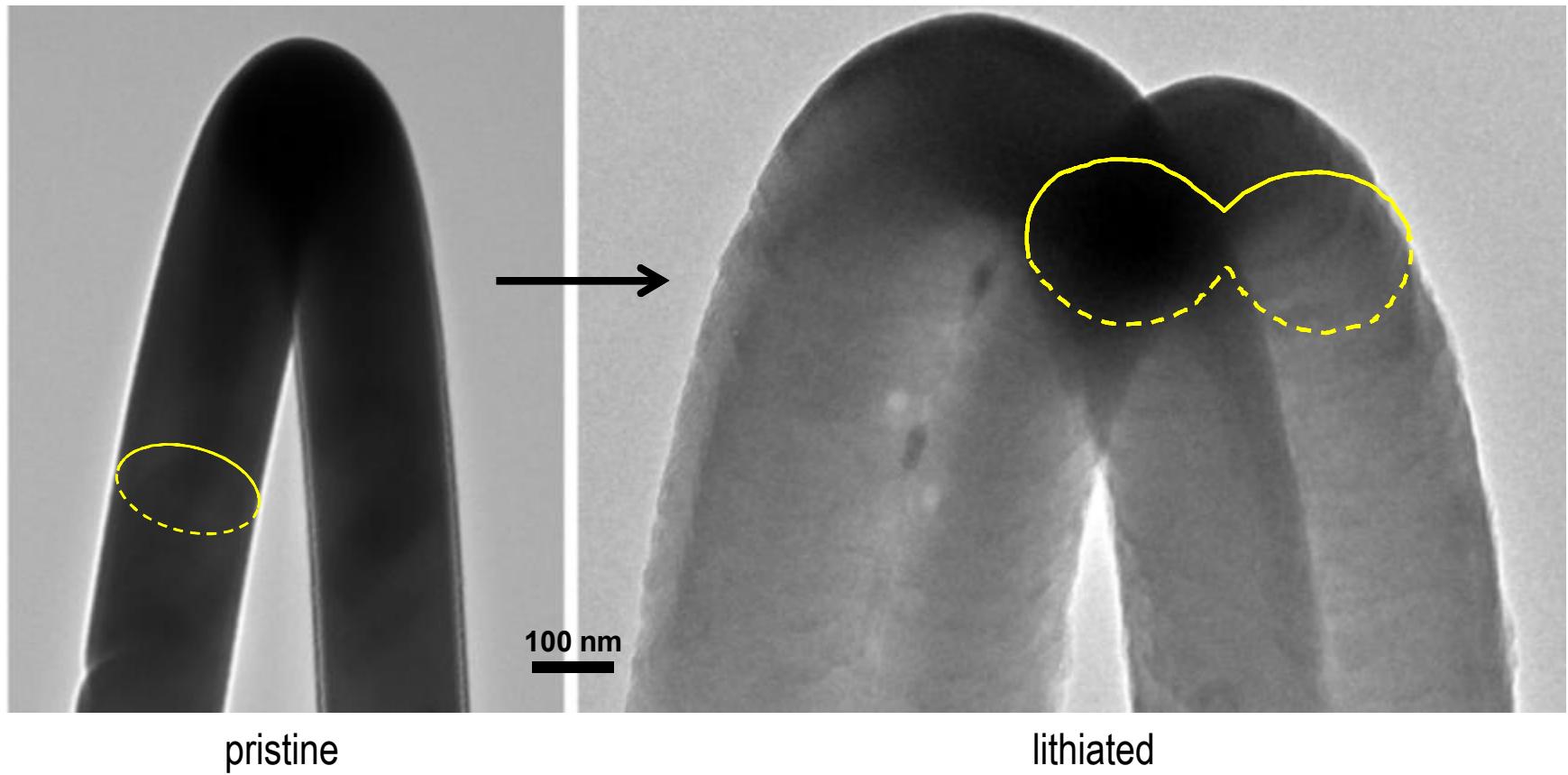


lithiated

Tilted series showing the shapes of the nanowires before and after lithiation.

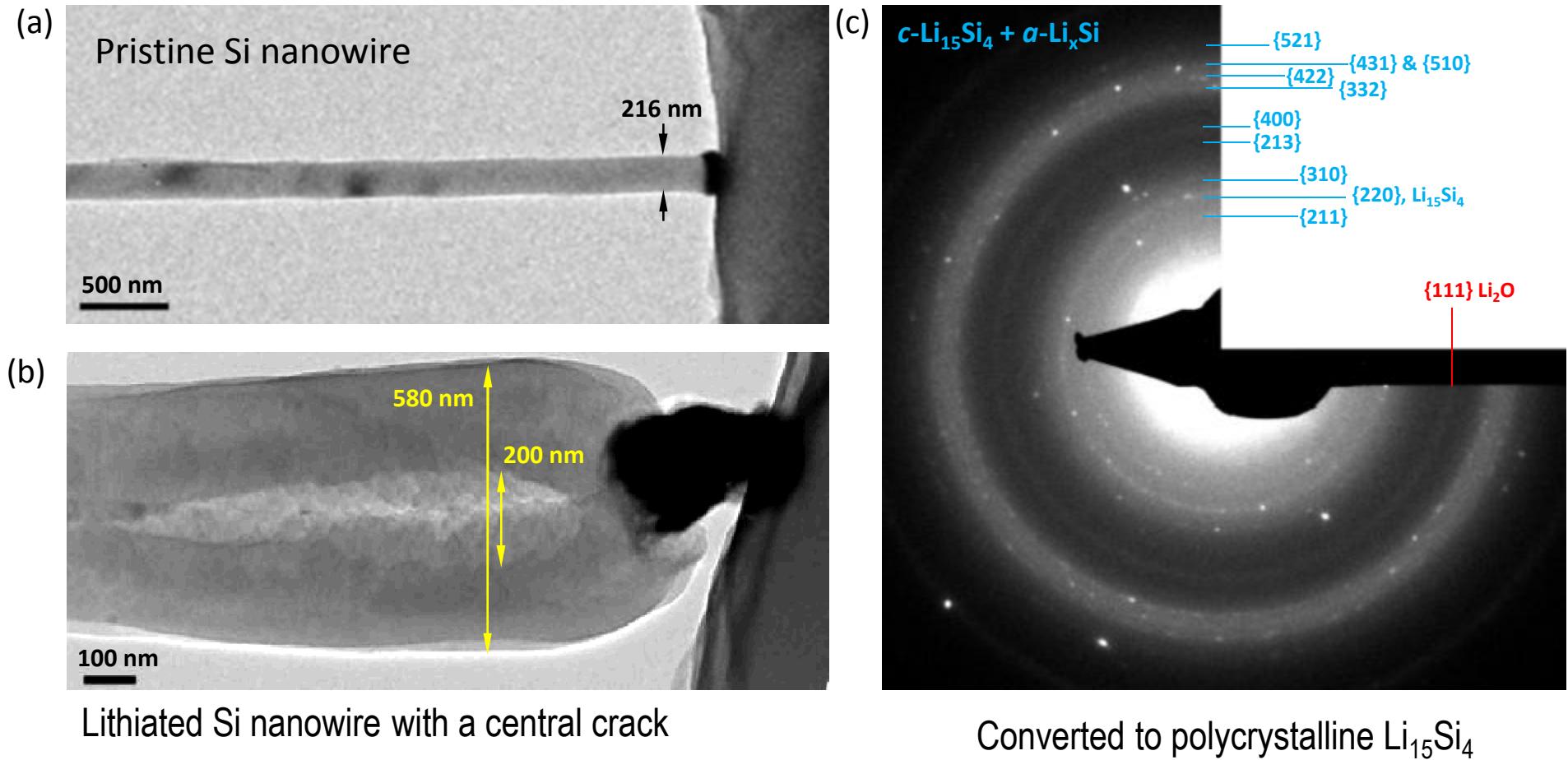
# Lithiated Si nanowire

Dumbbell shaped cross section



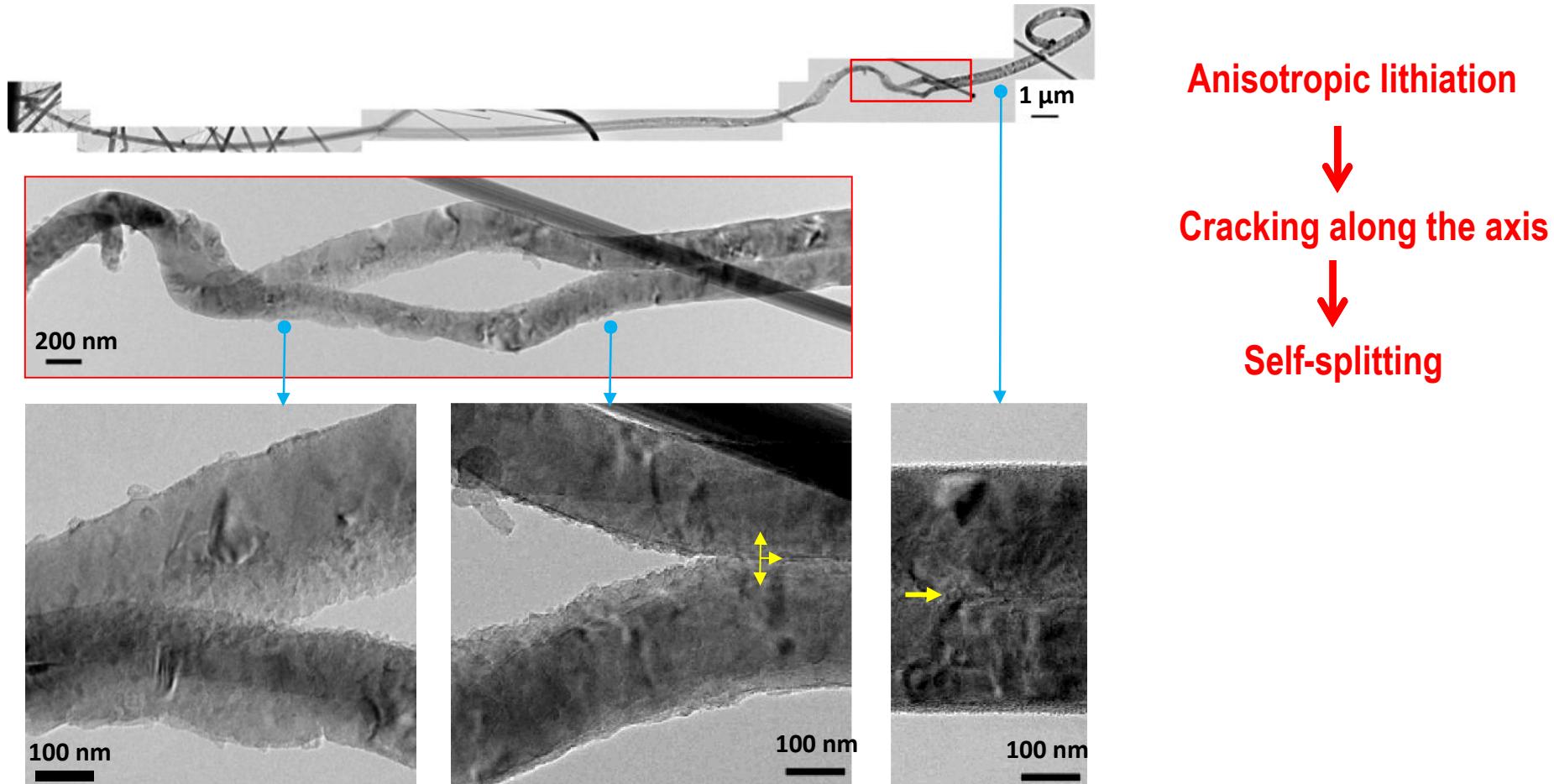
# Microstructure change

Center crack running along the axis of the nanowire due to uneven radial swelling



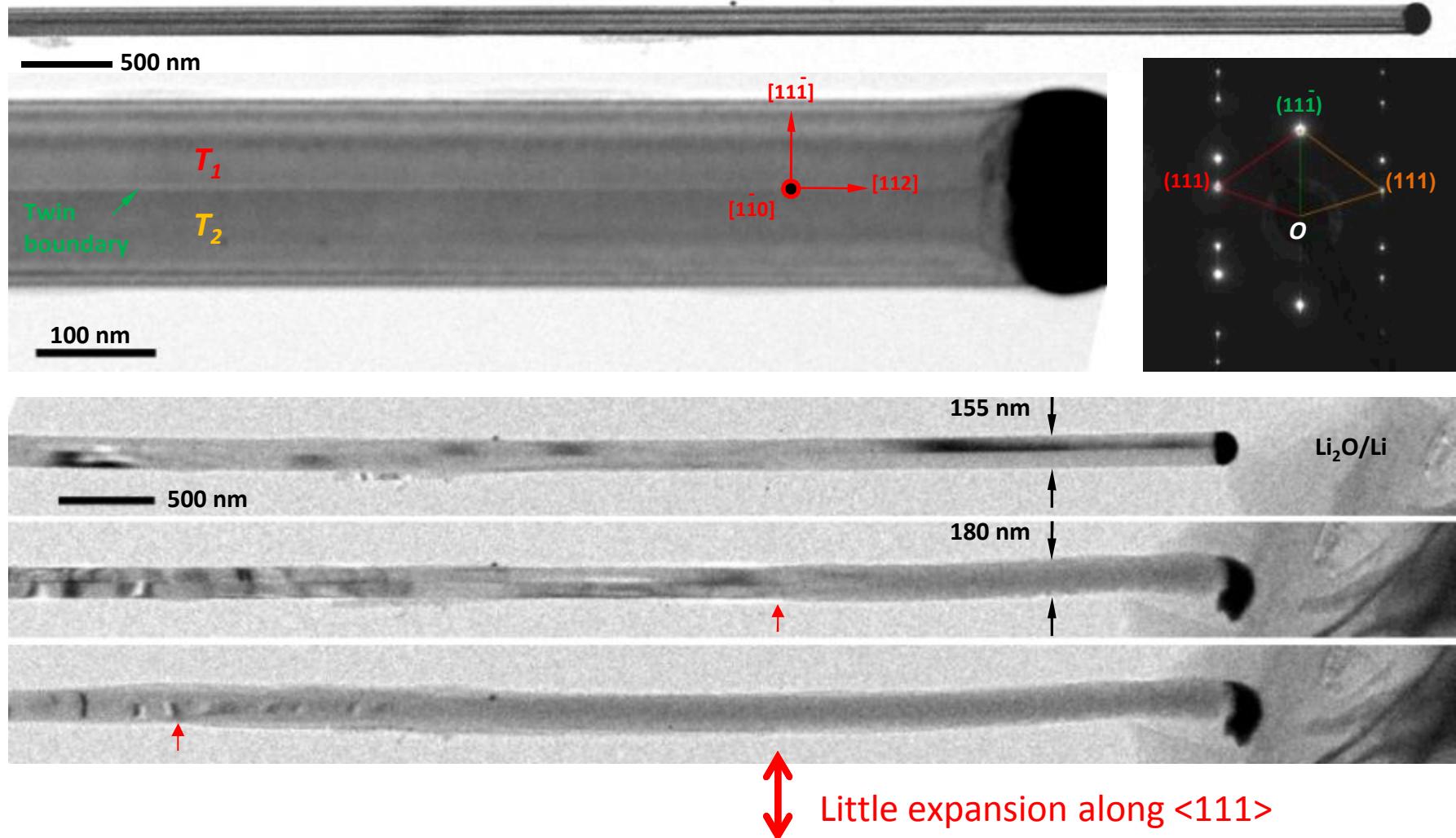
# Self-splitting

Anisotropic radial swelling sometimes split the nanowire into two sub-wires

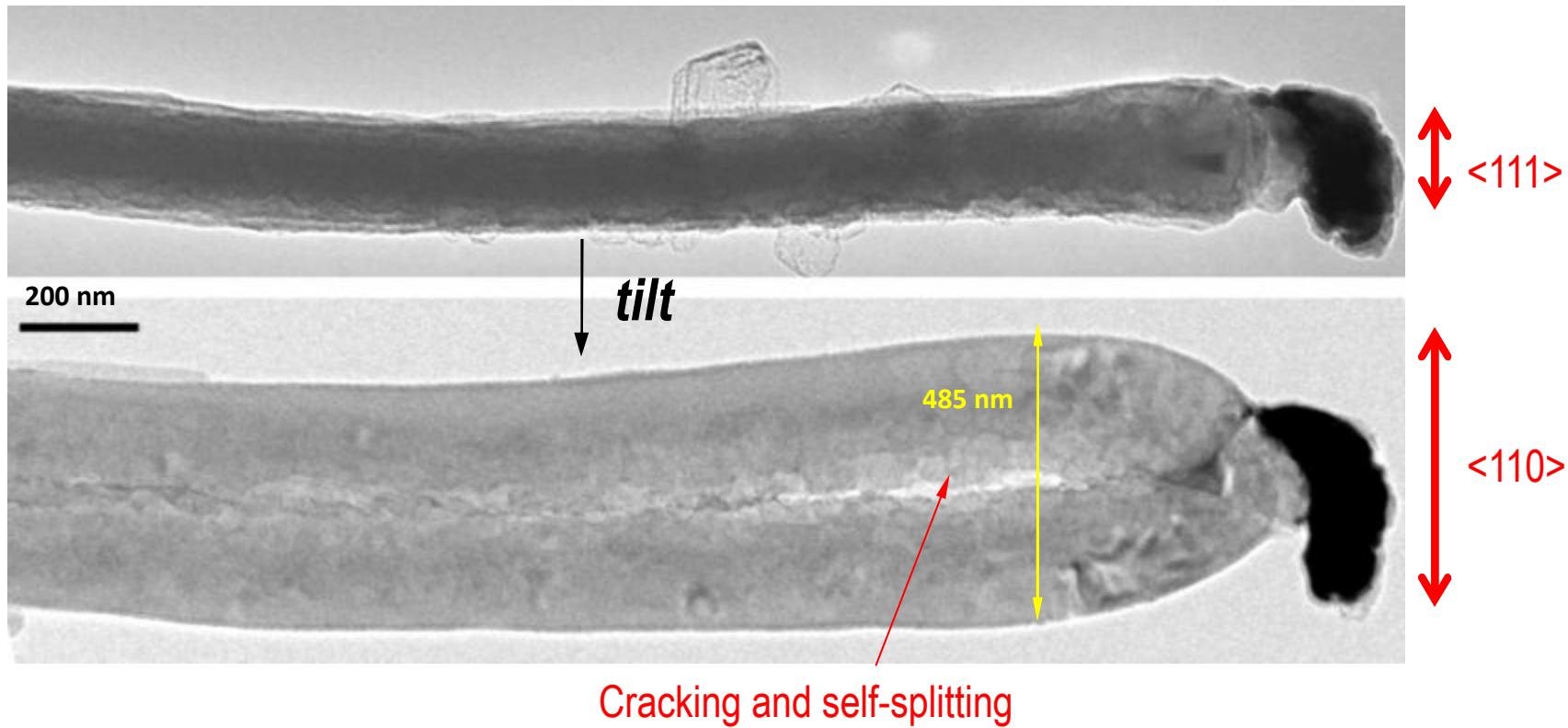


# Crystallography of the anisotropic lithiation of $<112>$ -Si nanowires

Little volume expansion was seen along  $<111>$

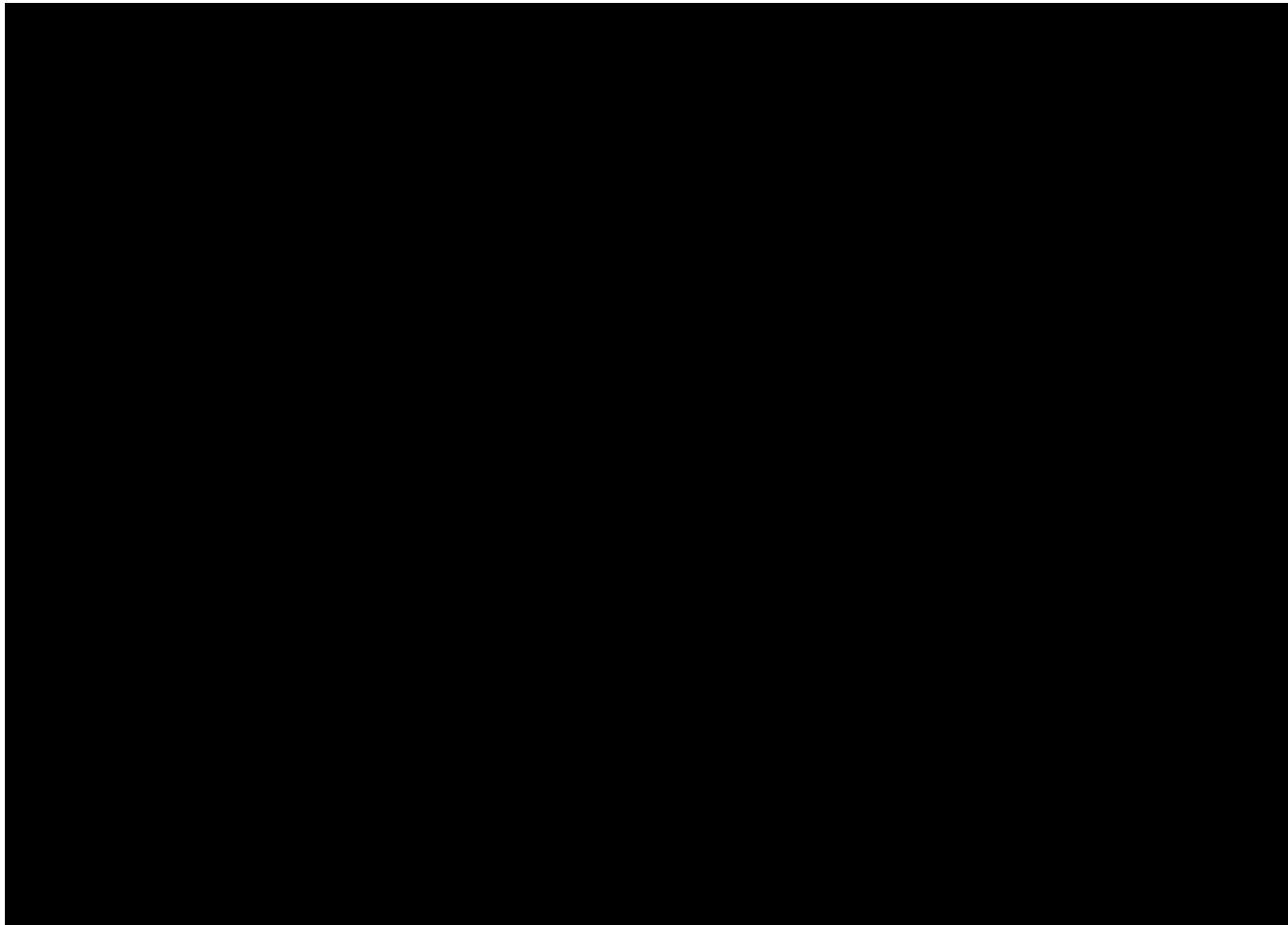


# Tilted images showing different views



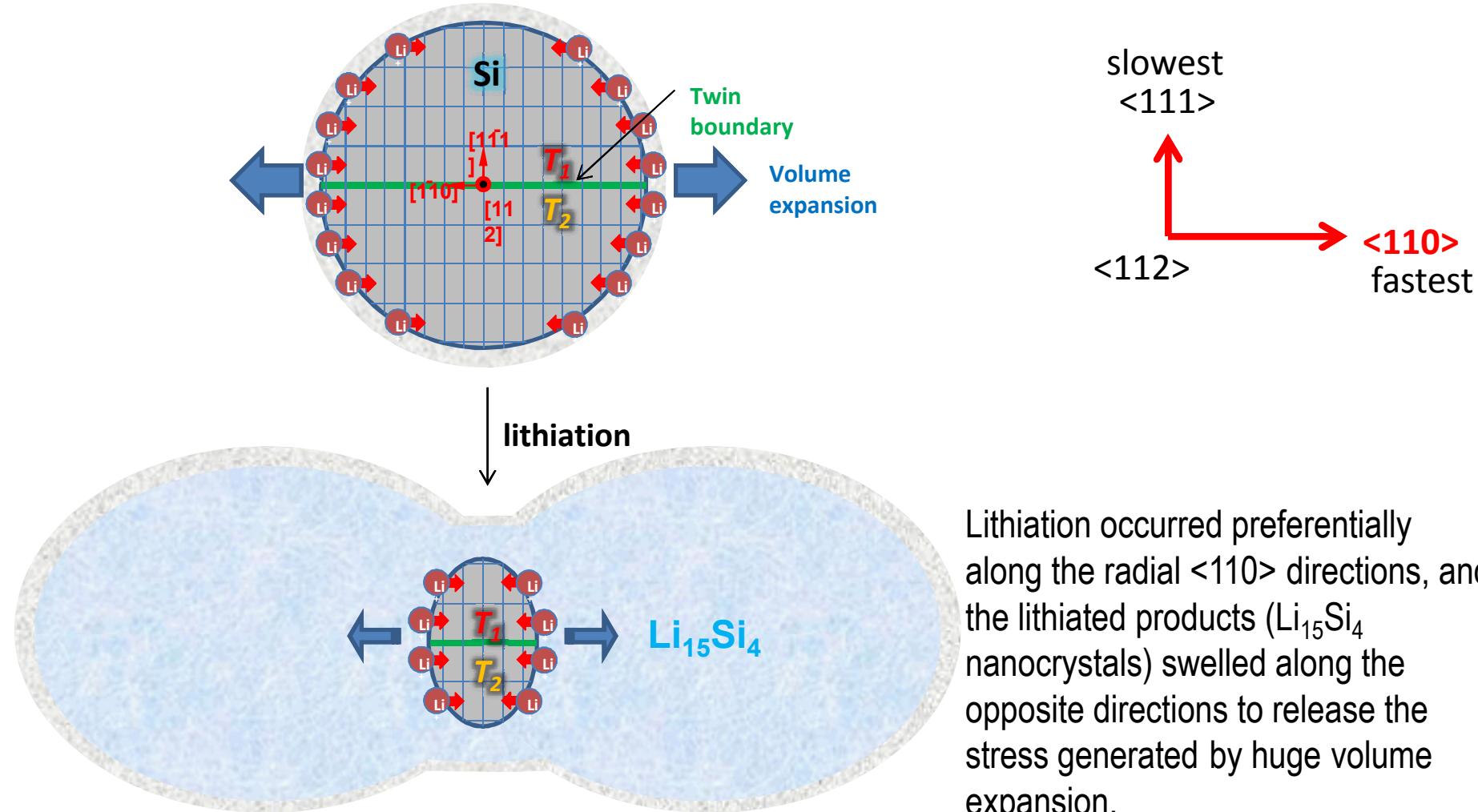
# Finite element simulation

Reproducing the anisotropic swelling of  $<112>$ -Si nanowires



# Crystallography of the anisotropic lithiation of $<112>$ -Si nanowires

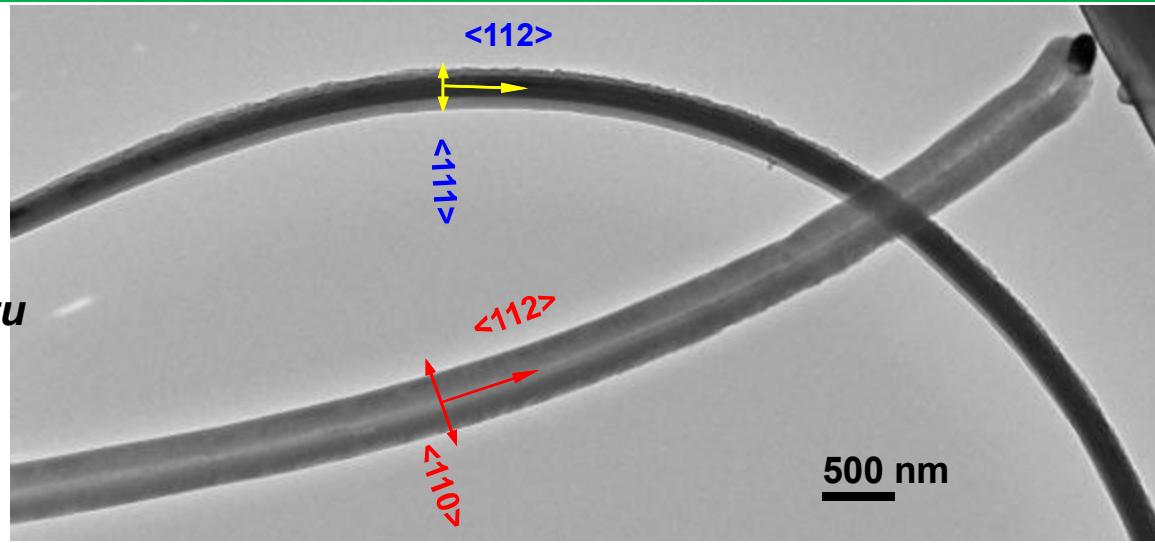
Much faster lithiation and swelling along the radial  $<110>$  directions than that along  $<111>$



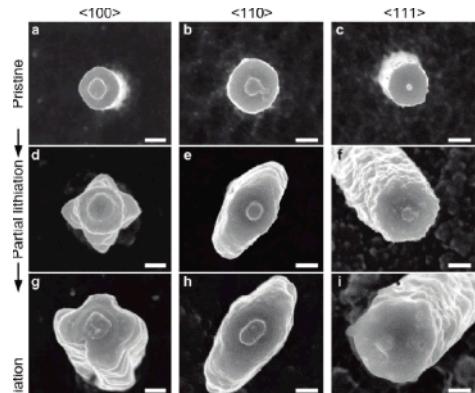
Lithiation occurred preferentially along the radial  $<110>$  directions, and the lithiated products ( $\text{Li}_{15}\text{Si}_4$  nanocrystals) swelled along the opposite directions to release the stress generated by huge volume expansion.

# Comparison of *in-situ* and *ex-situ* studies

Both showing fast lithiation along Si  $\langle 110 \rangle$  directions



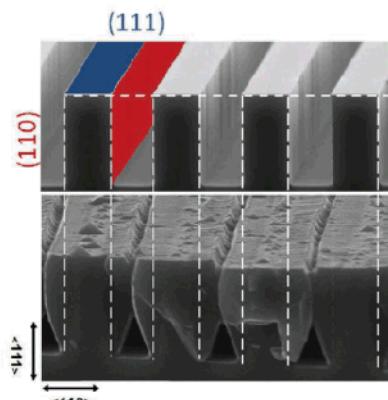
*in situ*



*ex situ*

Anisotropic shapes of Si pillars with different orientations.

Lee, et al. *Nano Lett.* **11**, 3034-3039 (2011).



Si microslabs swell on the  $\{110\}$  faces but not on  $\{111\}$ .

Goldman, et al. *Adv. Func. Mater.* **21**, 2412-2422 (2011).

The consistency between the *in-situ* and *ex-situ* studies also verifies the powerfulness of *in-situ* TEM and its relevance to real battery studies.

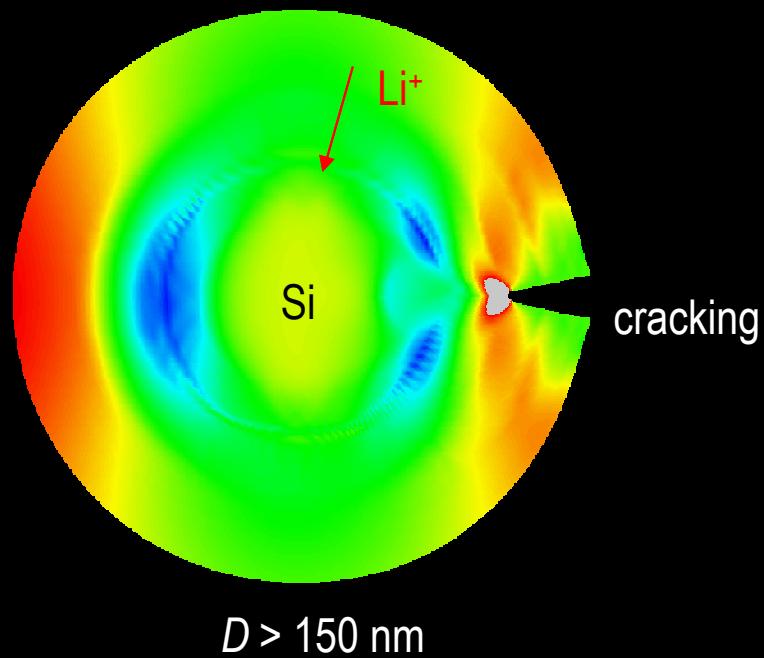
# A brief summary

on anisotropic lithiation of crystalline Si

1. For a  $<112>$ -oriented Si nanowires with a cylinder shape, lithiation and swelling occur predominately on the radial  $<110>$  directions;
2. Such anisotropic swelling leads to cracking and sometimes self-splitting;
3. Anisotropic lithiation should be considered in the design of Si electrodes.

## Part 3.2

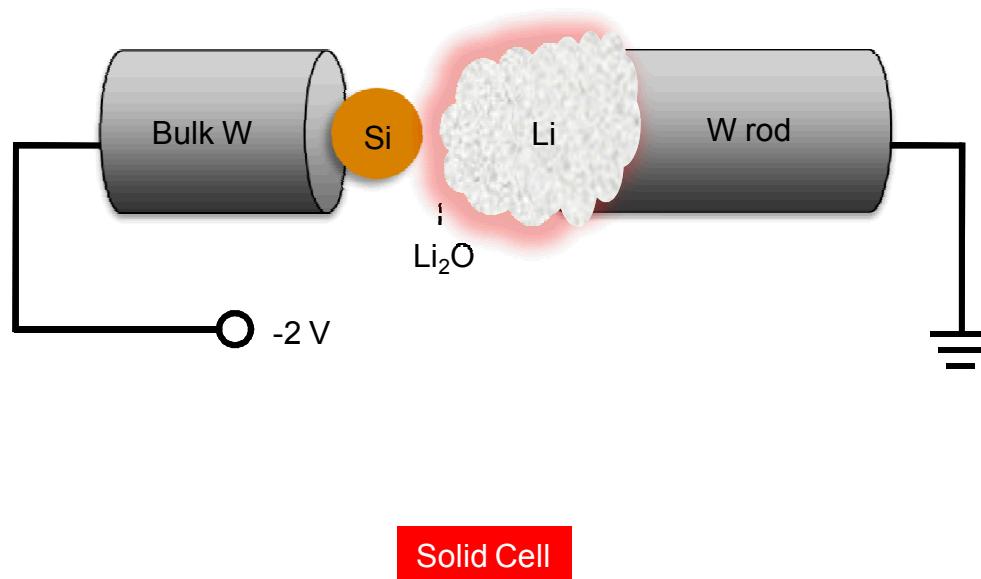
Size dependent fracture of Si nanoparticles during lithiation



# Experimental method

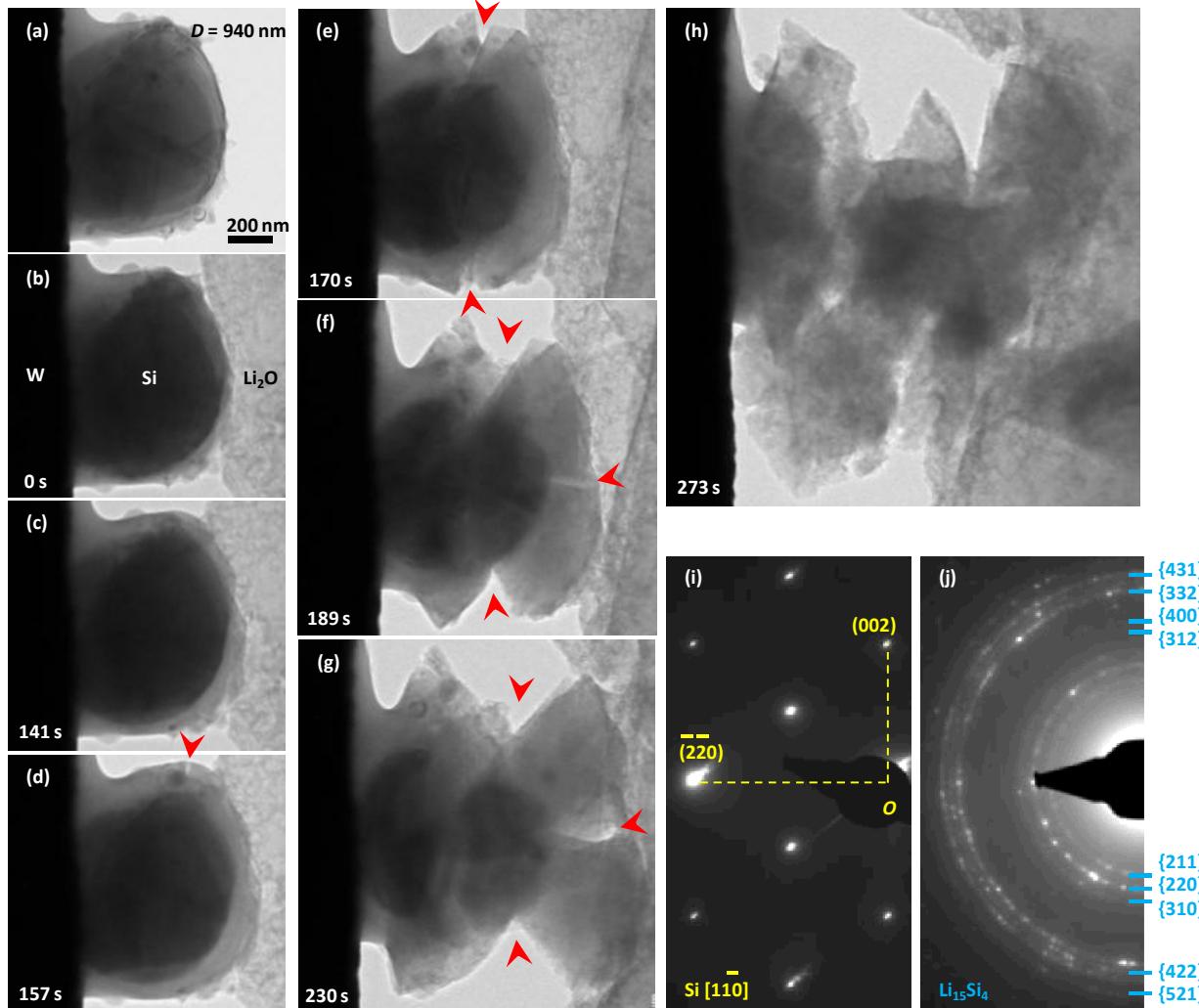
Si nanoparticles with various sizes were lithiated

1. Solid cell configuration using  $\text{Li}_2\text{O}/\text{Li}$  electrolyte/counter-electrode
2. Individual Si nanoparticles with  $\text{nm} \sim \mu\text{m}$  sizes



# Large Si nanoparticles

Always crack and fracture upon the first lithiation



Core-shell lithiation

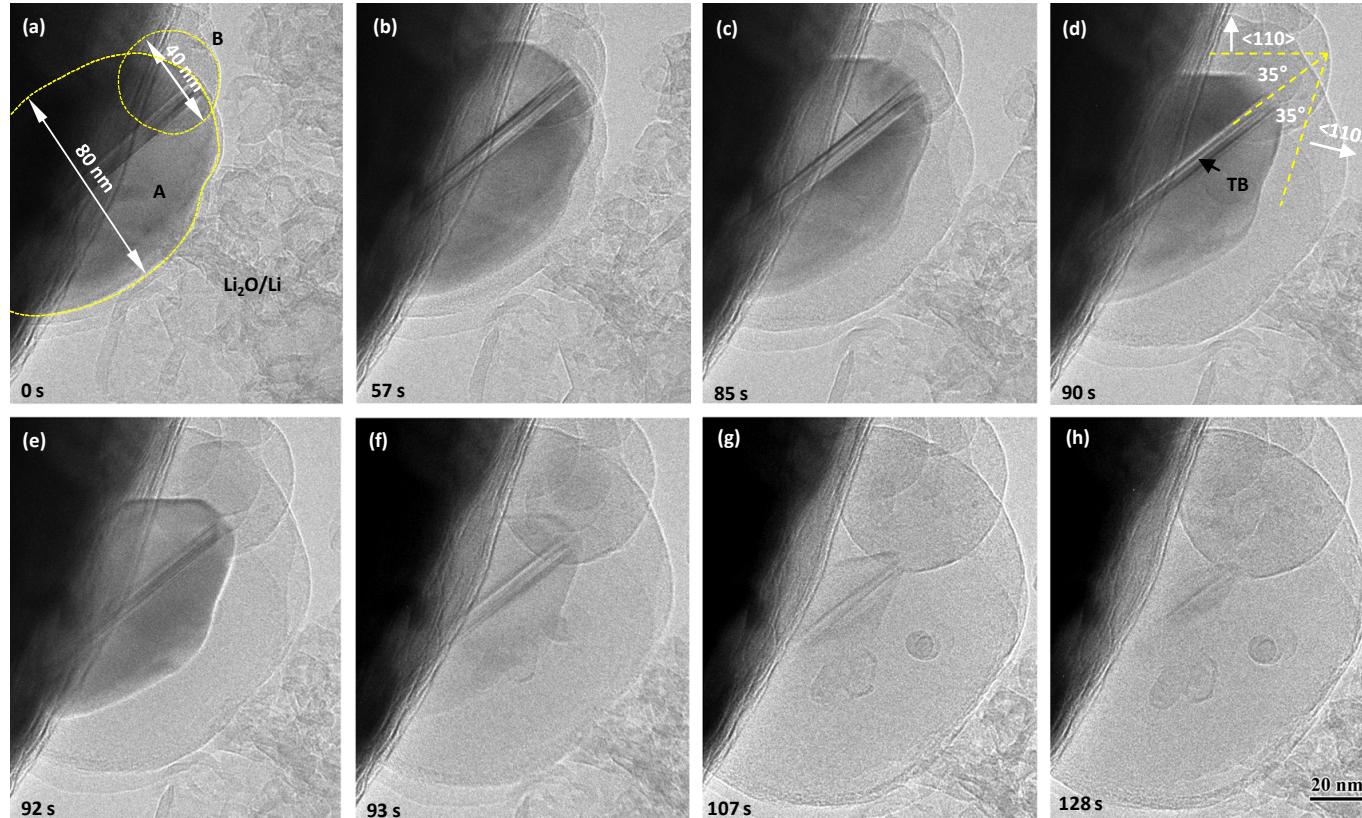
Two-phase ( $\text{Li}_x\text{Si}/\text{Si}$ ) mechanism

Broken into pieces after lithiation

Fully lithiated to  $\text{Li}_{15}\text{Si}_4$

# Small Si nanoparticles

Not cracking



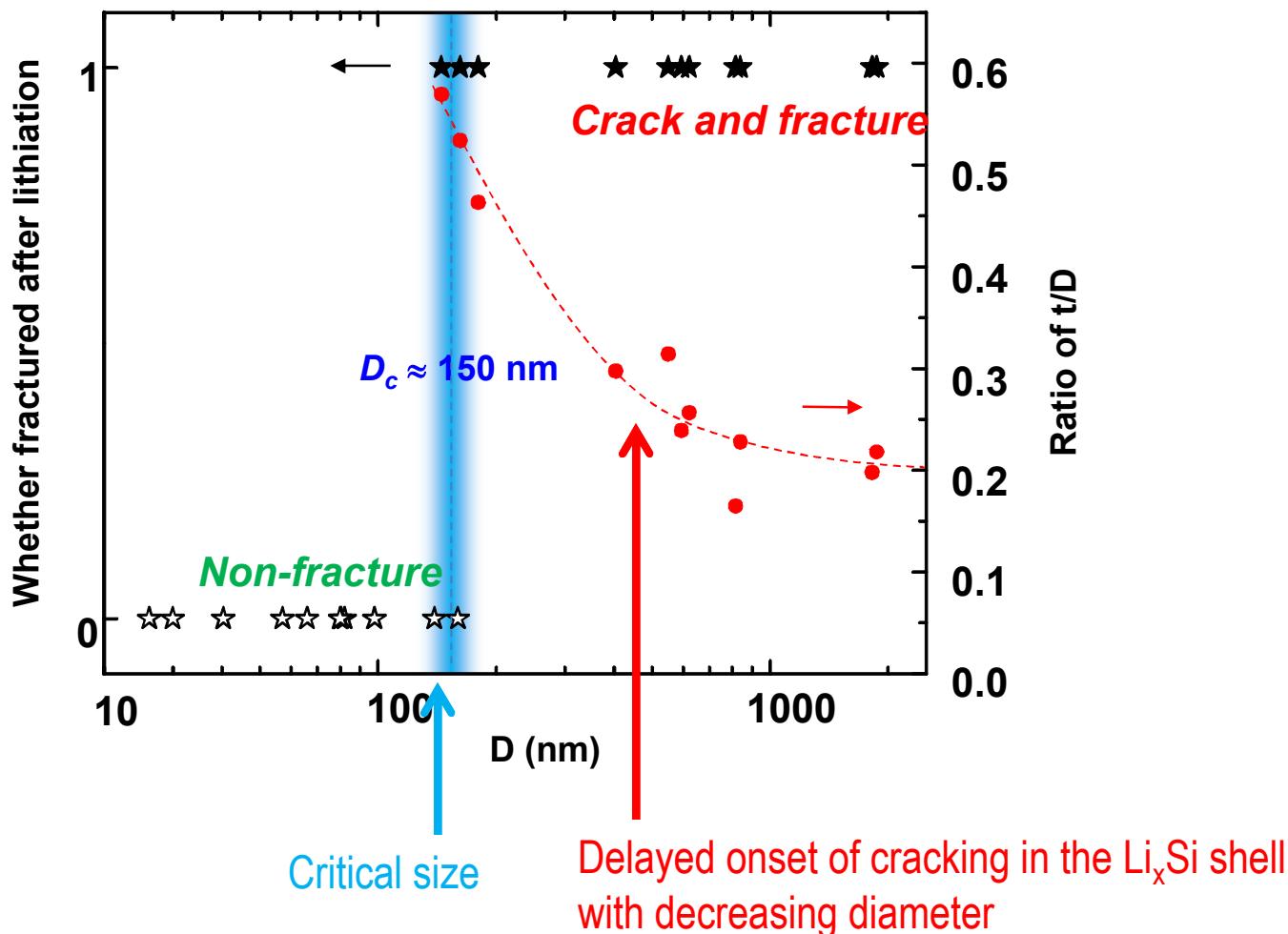
Also forming a core-shell structure

Like an inflating balloon

# Strong size dependence of fracture behavior

Critical size  $\sim 150$  nm

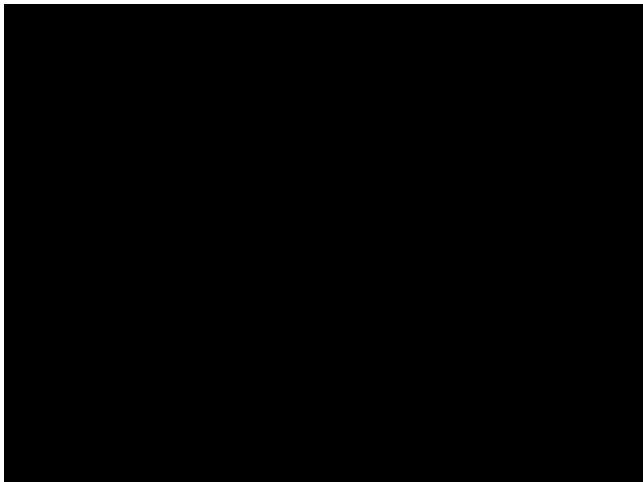
Statistics showing very reproducible size-dependent fracture behavior of Si nanoparticles upon first lithiation:



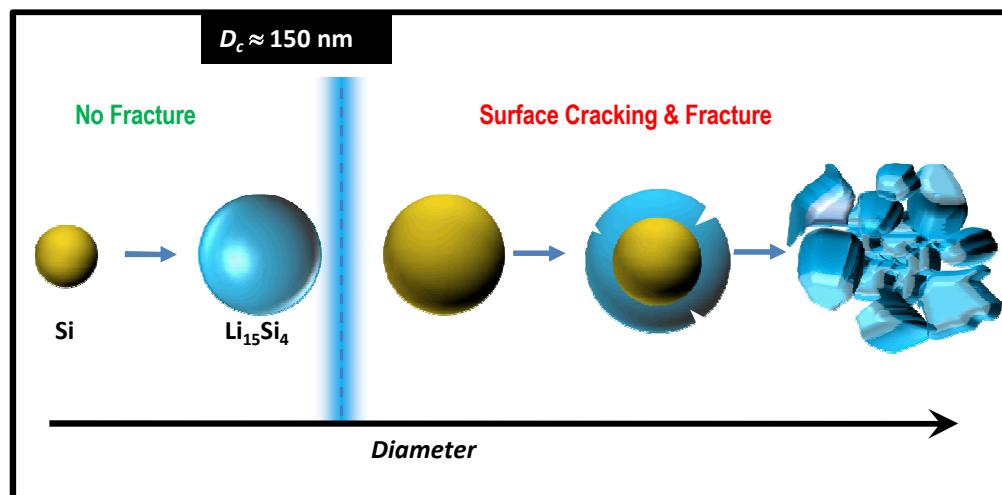
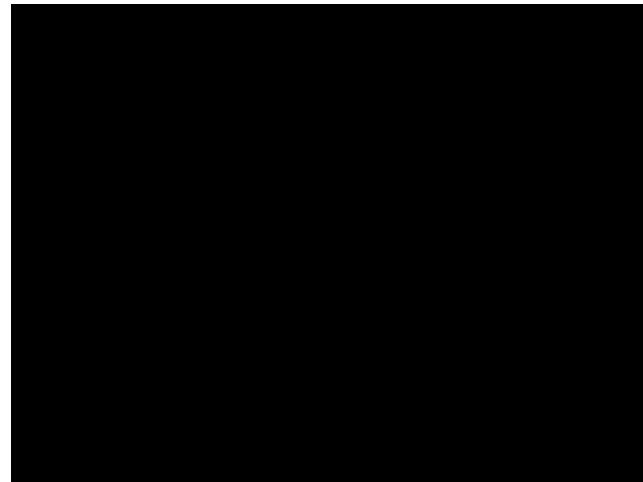
# Strong size dependence of fracture behavior

Critical size  $\sim 150$  nm

Fracture of large Si nanoparticles



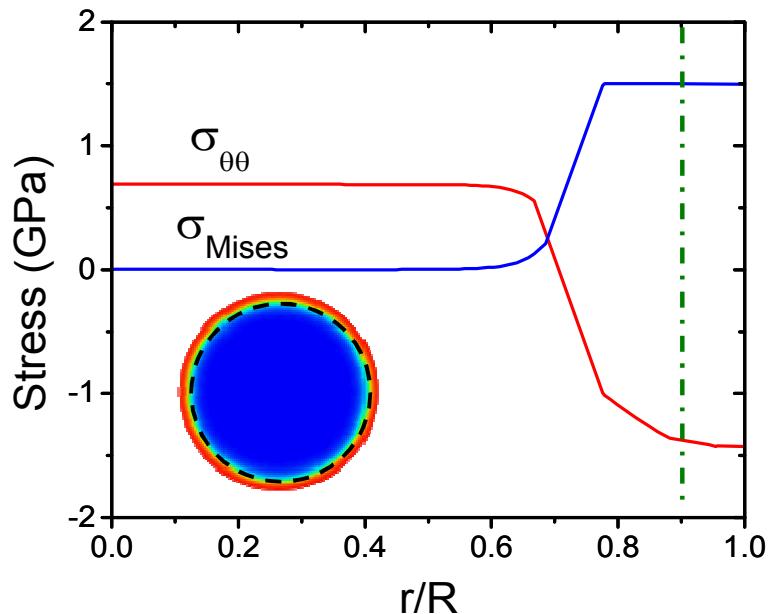
No fracture or cracking in small Si nanoparticles



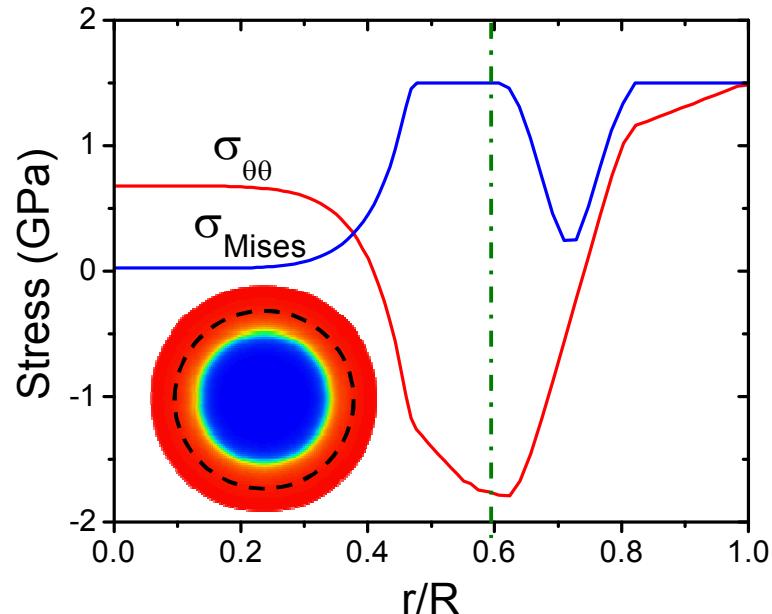
# Origin of the size effect

Hoop stress reversal in the  $\text{Li}_x\text{Si}$  shell as being pushed out: compression  $\rightarrow$  tension

Stress analysis of the Si nanoparticle in the different stages of lithiation:



Initial stage: compression in the lithiated shell



Later stage: hoop stress reverted to tension in the lithiated surface, because the inner and newly formed  $a\text{-Li}_x\text{Si}$  is pushing out

# Origin of the size effect

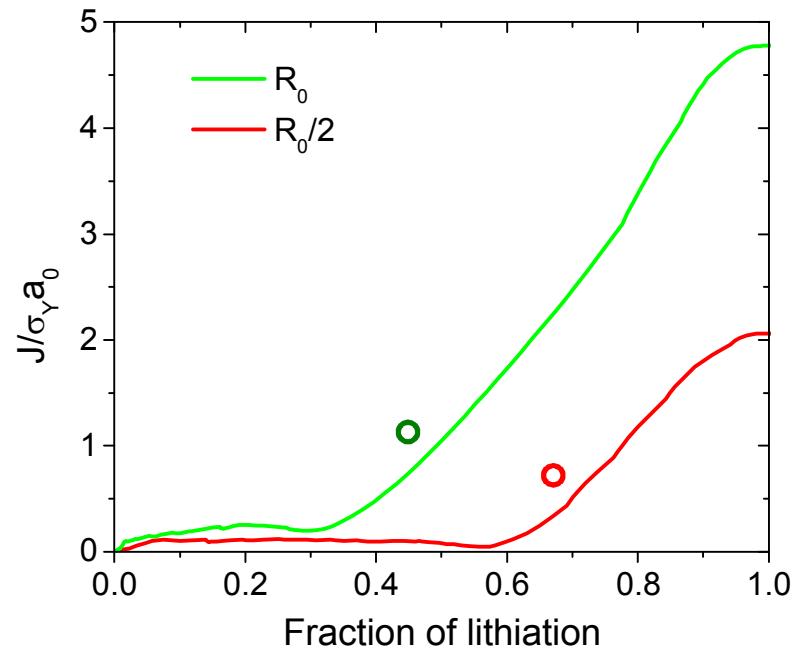
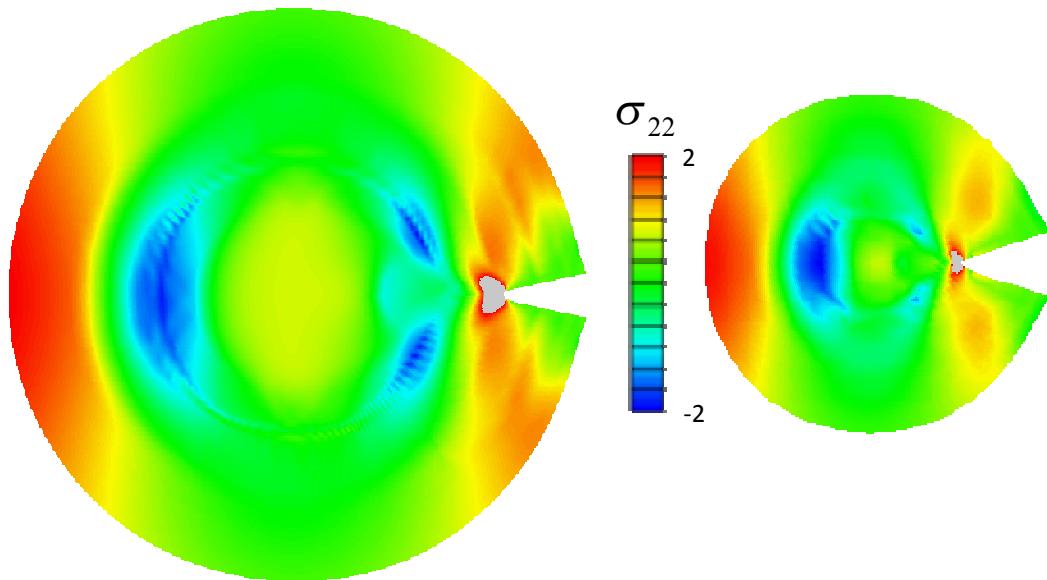
Hoop stress reversal in the  $\text{Li}_x\text{Si}$  shell as being pushed out: compression  $\rightarrow$  tension

Why cracking only in large particles?

Releasing mechanical energy by crack propagation

Versus

Increasing surface energy by generating new surface



In small particles, the driving force for crack propagation is insufficient.

# A brief summary

## Size-dependent fracture of Si nanoparticles

1. Larger Si nanoparticles ( $D > 150$  nm) always crack and fracture upon the first lithiation process;
2. Smaller Si nanoparticles avert fracture, suggesting that nano-Si may be good for fabricating durable Si electrodes (e.g., dispersing small Si particles in an elastomeric matrix to accommodate the volumetric changes);
3. Surface cracking was revealed, in contrast to the previously predicted center cracking models.

## 4. Conclusions on lithiation of crystalline Si

Based on the *in-situ* TEM experiments of various Si nanomaterials

1. Crystalline Si undergoes a two-step lithiation process, namely  $c\text{-Si} \rightarrow a\text{-Li}_x\text{Si} \rightarrow c\text{-Li}_{15}\text{Si}_4$ , accompanied by 281% volumetric change;
2. Si lithiation is the fastest along  $\langle 110 \rangle$  and slowest along  $\langle 111 \rangle$ ;
3. Si nanoparticles smaller than 150 nm do not crack or fracture upon lithiation;
4. *In situ* TEM is a powerful tool for the study of degradation mechanism of battery materials.

### Related publications:

- [1] Liu, X. H., *et al.* ***Nano Letters*** 11, (6), 2251-2258 (**2011**).
- [2] Liu, X. H., *et al.* ***Nano Letters*** 11, (8), 3312-3318 (**2011**).
- [3] Liu, X. H., *et al.* ***Nano Letters*** 11, (9), 3991-3997 (**2011**).
- [4] Liu, X. H., *et al.* ***Energy & Environmental Science*** 4, 3844-3860 (**2011**).
- [5] Liu, X. H., *et al.* ***Applied Physics Letters*** 98, (18), 183107 (**2011**).
- [6] Liu, X. H., *et al.* ***ACS Nano***, 6: 1522-1531 (**2012**).
- [7] Liu, X. H., *et al.* ***Adv. Energy Mater.***, DOI: 10.1002/aenm.201200024 (**2012**).

# **Thank you for your attention!**

---



Sandia National Laboratories



Center for Integrated Nanotechnologies (CINT)