



U.S. DEPARTMENT OF  
**ENERGY**

**Nuclear Energy**

# Structure-Property Relationship in Metal-Organic Frameworks for Enhanced Energy Related Applications

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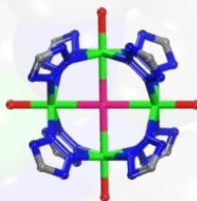
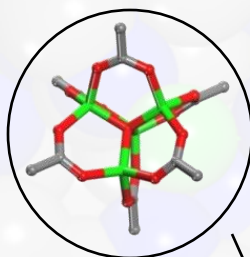
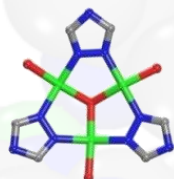
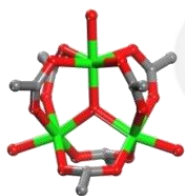
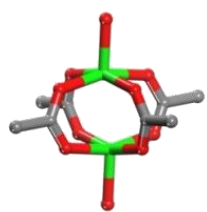
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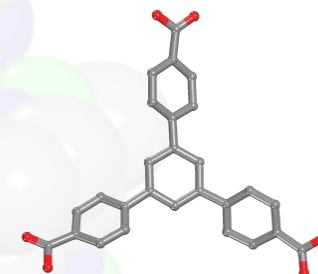
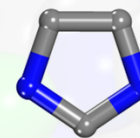
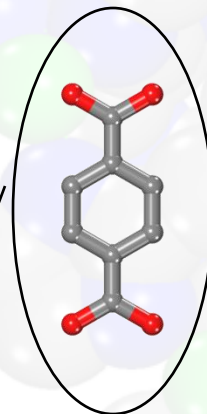




# MOFs are highly crystalline materials constructed from metal nodes and organic linkers



Common metal clusters  
(molecular building blocks)



Organic linking units

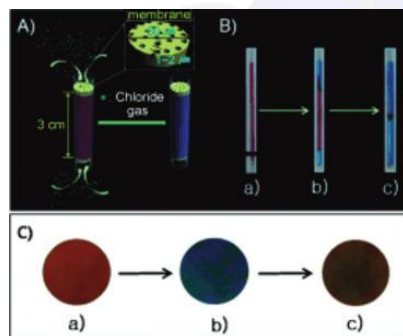




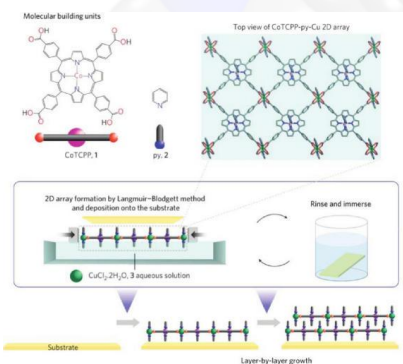
# MOFs are versatile materials that span diverse applications

*Chem. Rev.* **2012**, *112*, 673-1268.

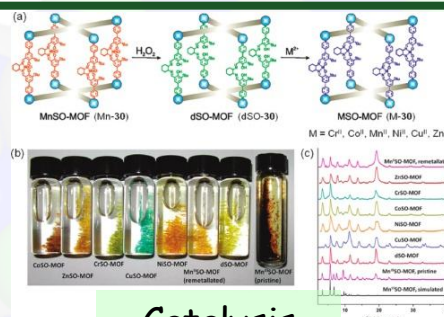
## Sensors



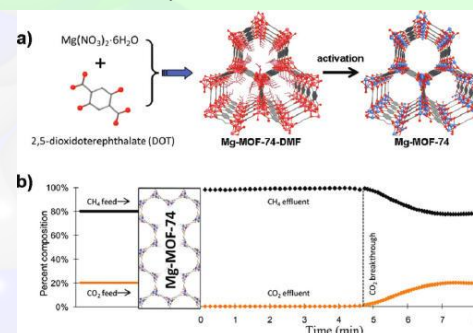
## Membranes



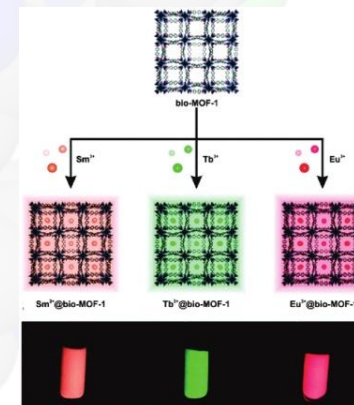
## Catalysis



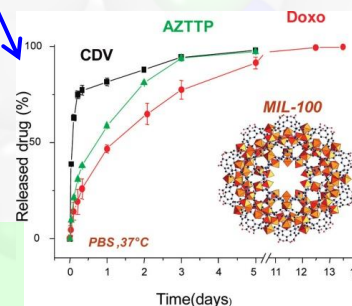
## Gas storage and separations



## Luminescence



## Biomedicine-drug delivery

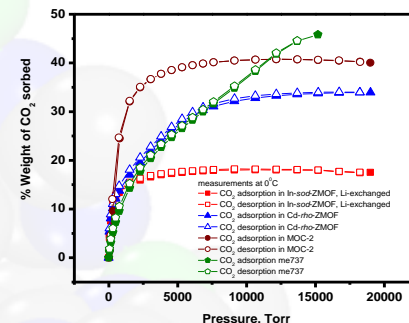
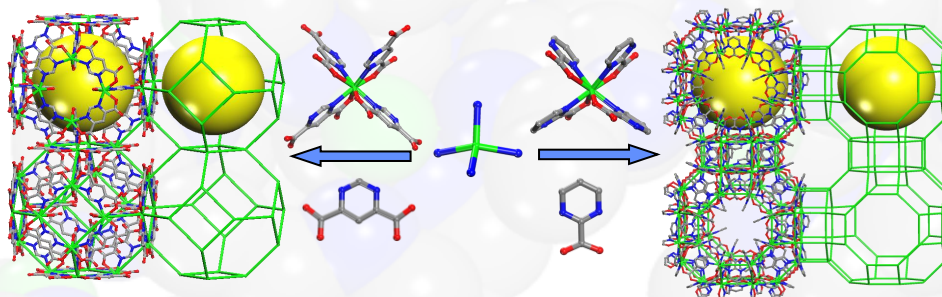






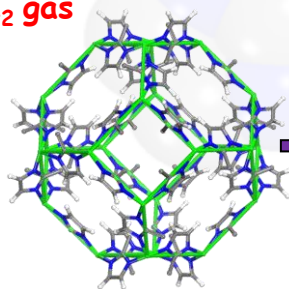
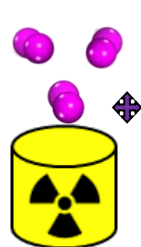
# Outline

## I. Design strategies for the rational construction of MOFs with zeolitic topologies

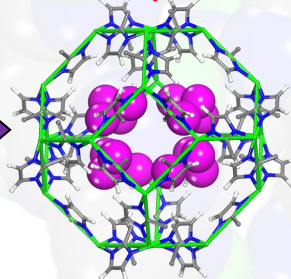


## II. Capture and Storage of Volatile Fission Gas Products from Reprocessing and/or Nuclear Accidents

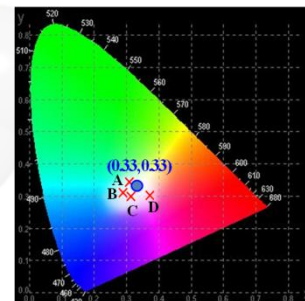
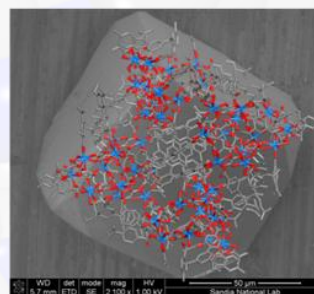
Volatile I<sub>2</sub> gas



I<sub>2</sub> capture



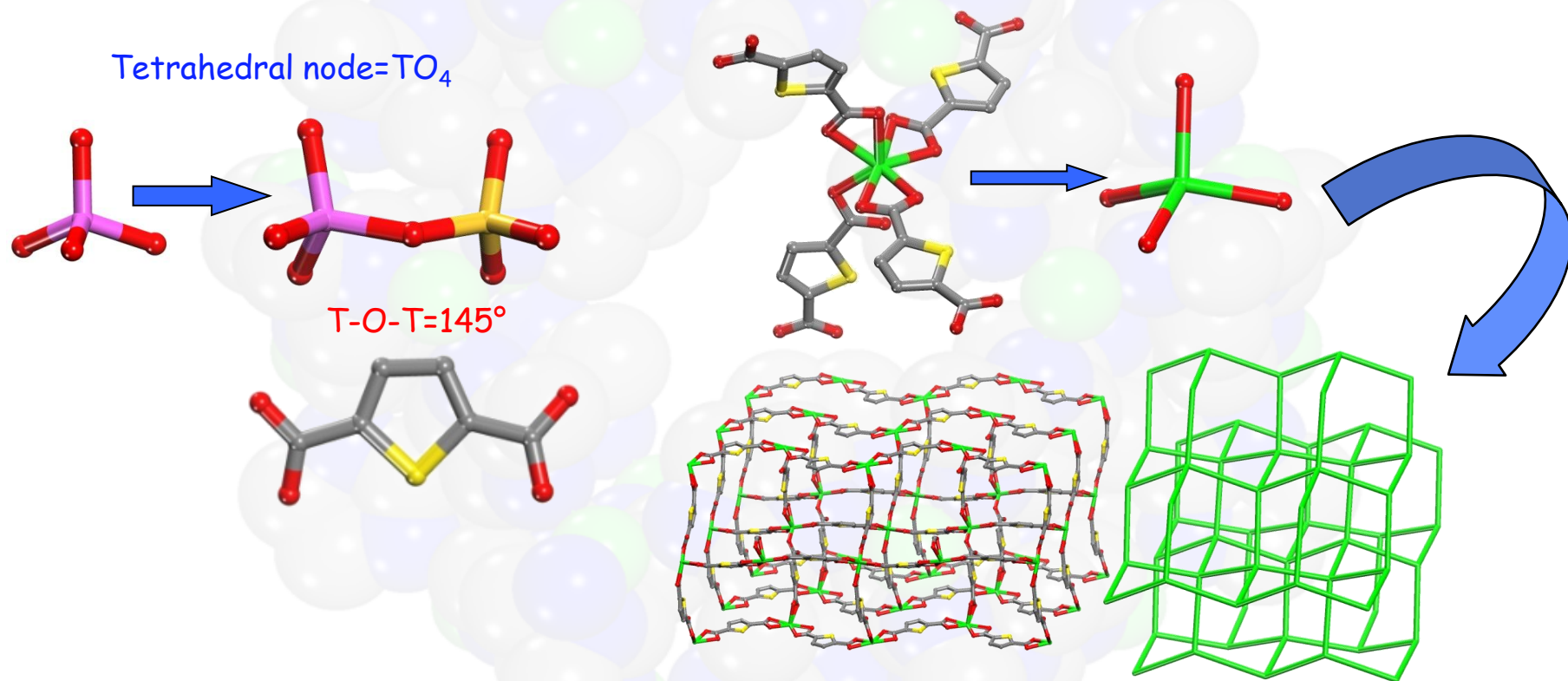
## III. Novel MOFs with Tunable Color Properties





# I. Design strategies for the construction of MOFs with zeolitic topologies

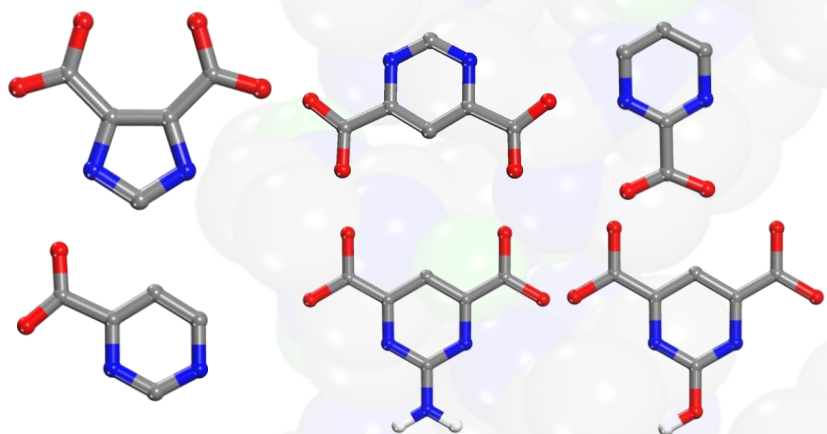
Top-down design and bottom-up approach for the “edge expansion” strategy for the construction of the metal-organic zeolitic analogues



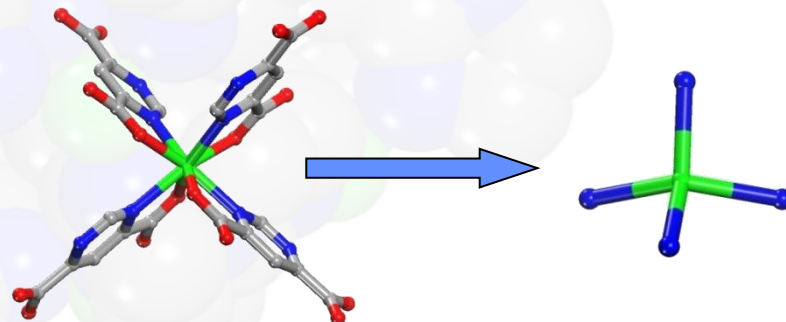
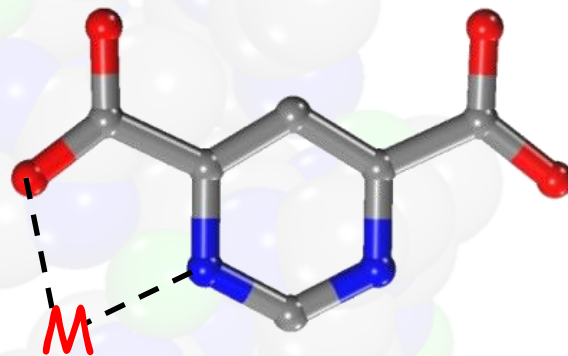


# Alternative approach via *rigid* and *directional* Tetrahedral Building Blocks

The deliberate construction of porous periodic solids, utilizing the  
single-metal-ion-based approach



Multifunctional heterochelating ligands  
render *rigid* and *directional*  
Molecular Building Blocks

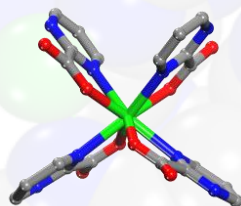
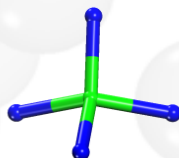
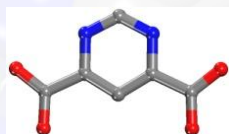
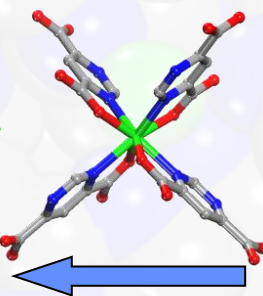
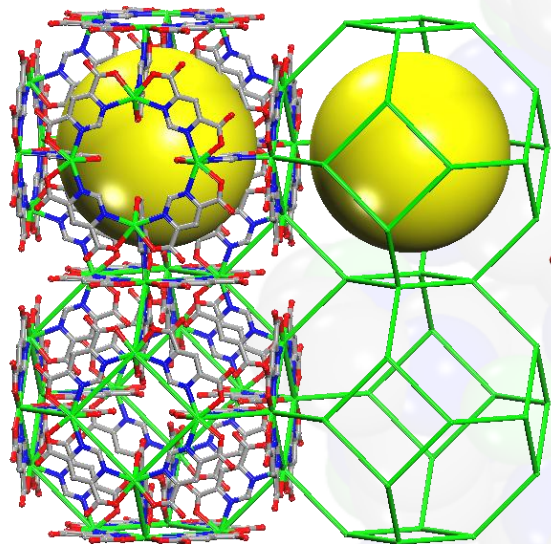




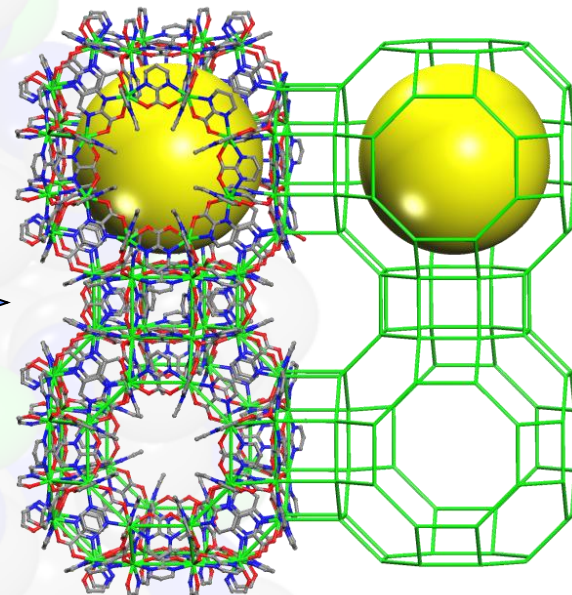


# Ia. Zeolite-like Metal–Organic Frameworks from pyrimidinecarboxylate bis-chelating bridging ligands

*sod*



*rho*



- (i) forbidden self interpenetration- design of readily accessible extra large cavities
- (ii) chemical stability in water (not common in MOFs) - applications for heterogeneous catalysis, separations, and sensors
- (iii) tune extra framework cations toward specific applications & removal/sequestration of toxic metal ions

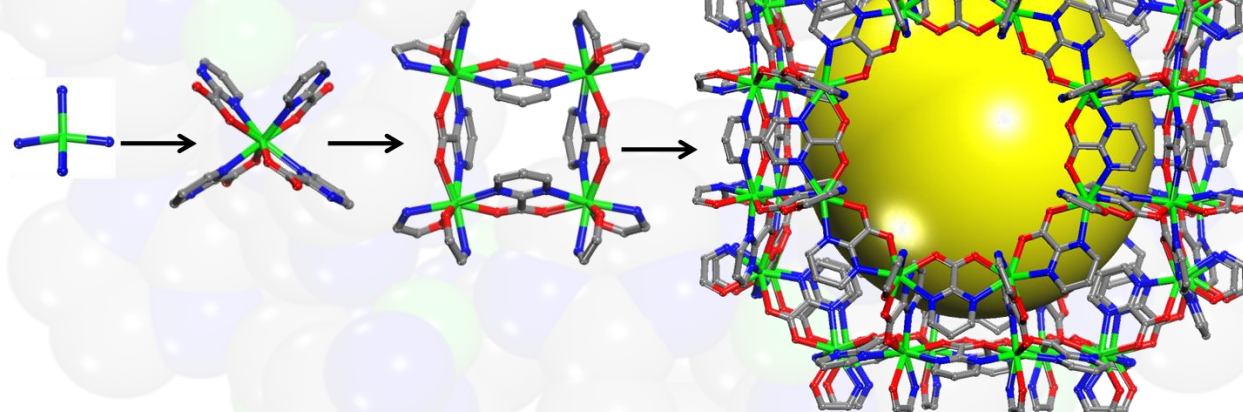
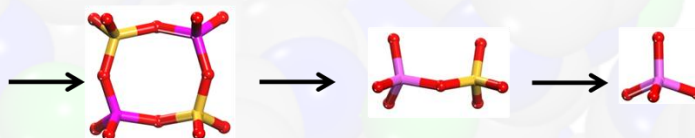
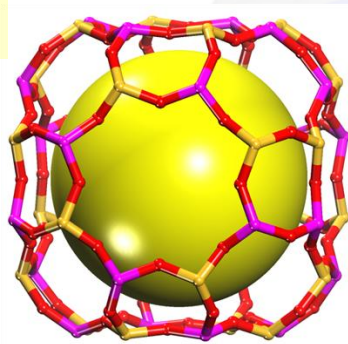






# MOF zeolite *rho*: 8 times the volume and an order of magnitude higher surface area vs. inorganic RHO

Zeolite RHO



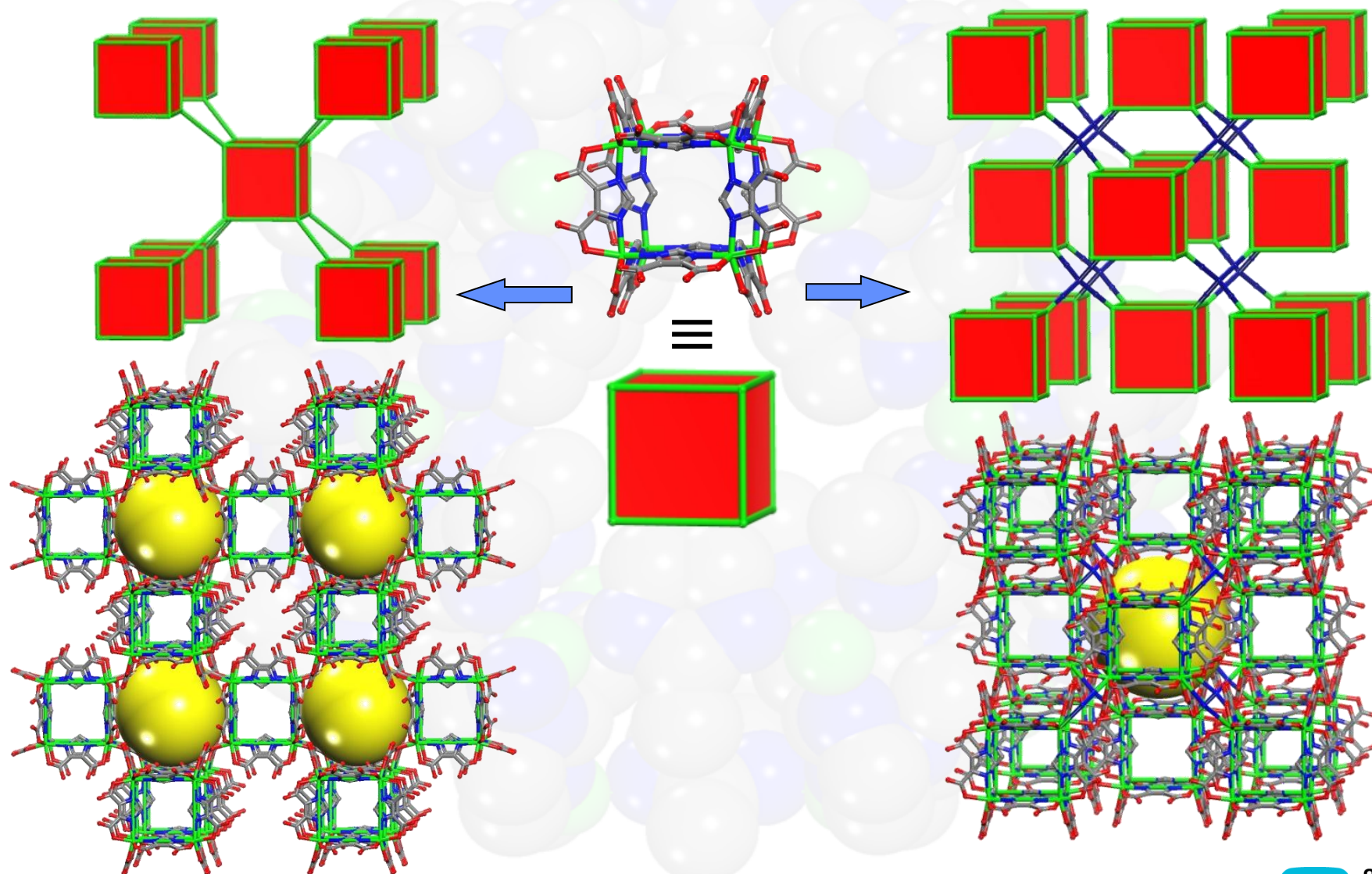
Metal-organic  
analogue *rho*

- The unit cell volume of the MOF *rho* framework is ~ **8 times larger** than the aluminosilicate analogue- unit cell edge is doubled (3 nm vs. 1.5 nm)

- Apparent surface area:  
inorganic RHO ~ 100-200 m<sup>2</sup>/g  
**MOF *rho* ~ 1100- 1800 m<sup>2</sup>/g**



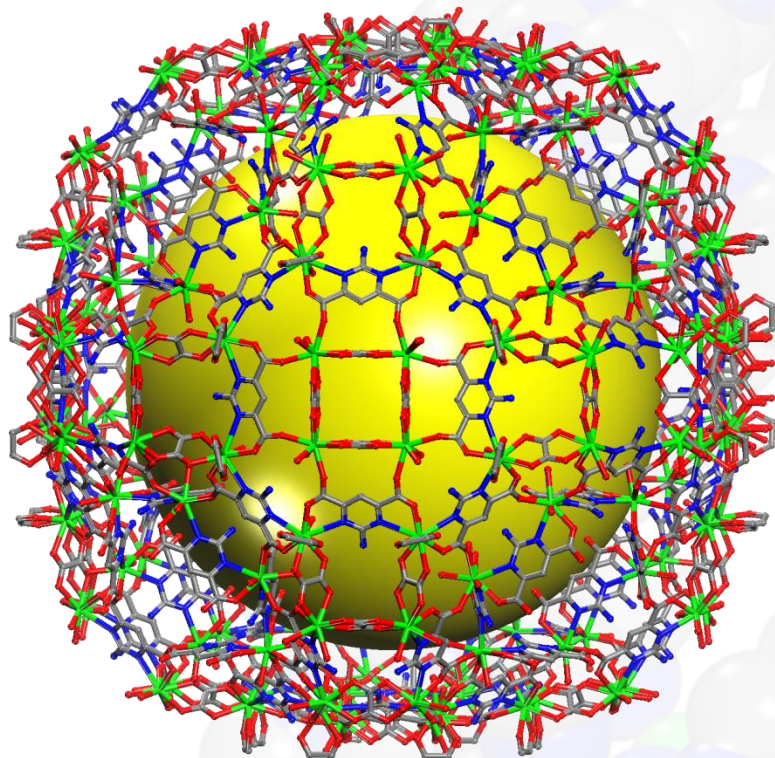
## Ib. MOFs with zeolitic topologies constructed from the assembly of Metal-Organic Cubes



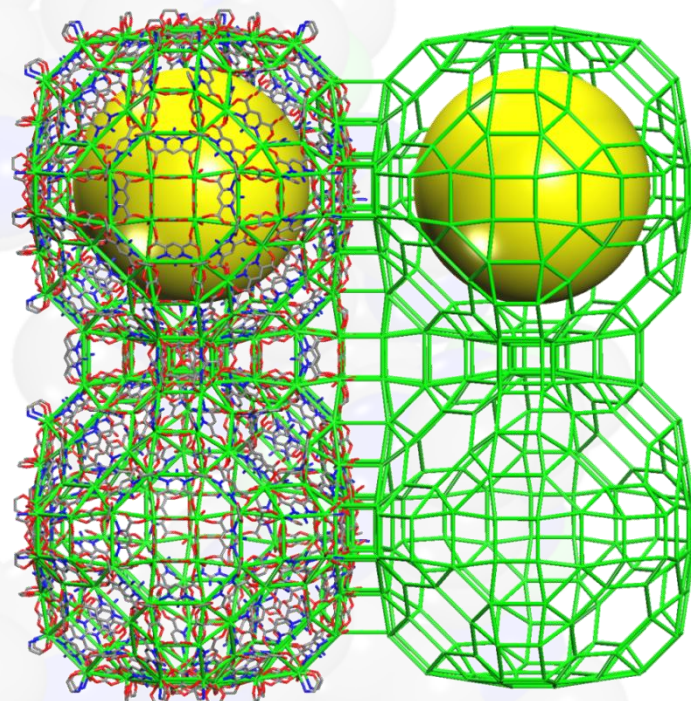




## Ic. Zeolite-like MOFs with extra-large cavities



diameter of  $\sim 38 \text{ \AA}$

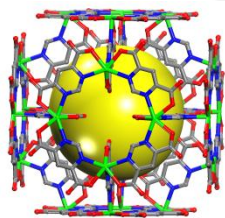


Material	Inner diameter	Outer diameter	Metal atoms per cage
This work	$38 \text{ \AA}$	$48 \text{ \AA} \times 53 \text{ \AA}$	168
Mes.MOF-1	$47.1 \text{ \AA}$	$59.5 \text{ \AA}$	84
MIL-101	$33.8 \text{ \AA}$	$46.7 \text{ \AA}$	126
ZIF-100	$35.6 \text{ \AA}$	$67.2 \text{ \AA}$	264

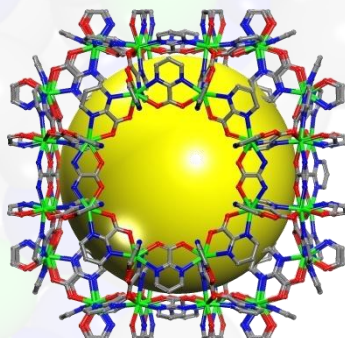




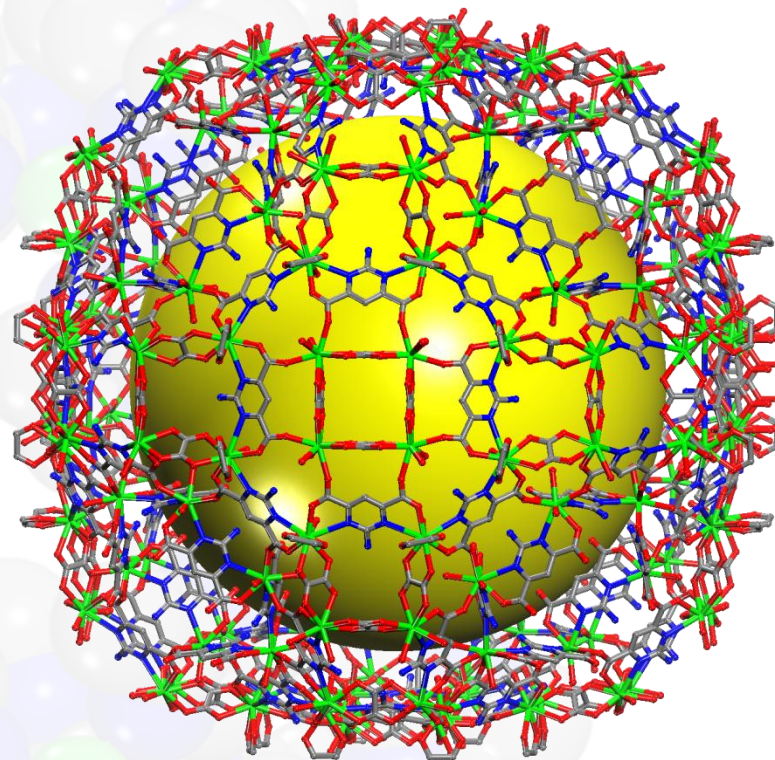
# Systematic variation of the organic components leads to increased complexity



*sod*- diameter  
of  $\sim 9.6 \text{ \AA}$



*rho* - diameter  
of  $\sim 16.4 \text{ \AA}$



diameter  
of  $\sim 38 \text{ \AA}$

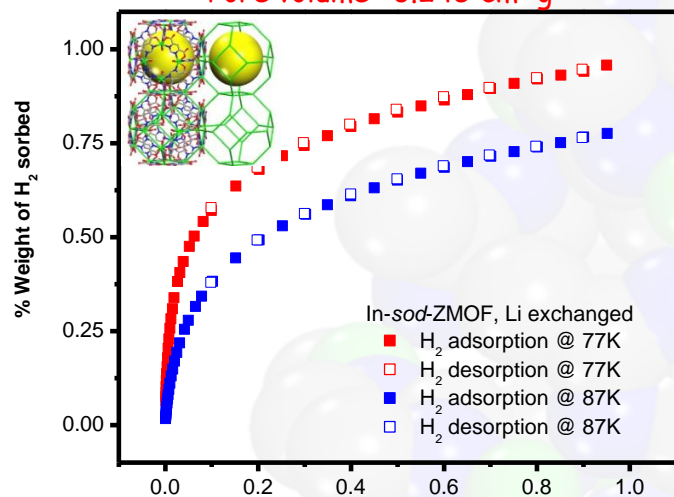




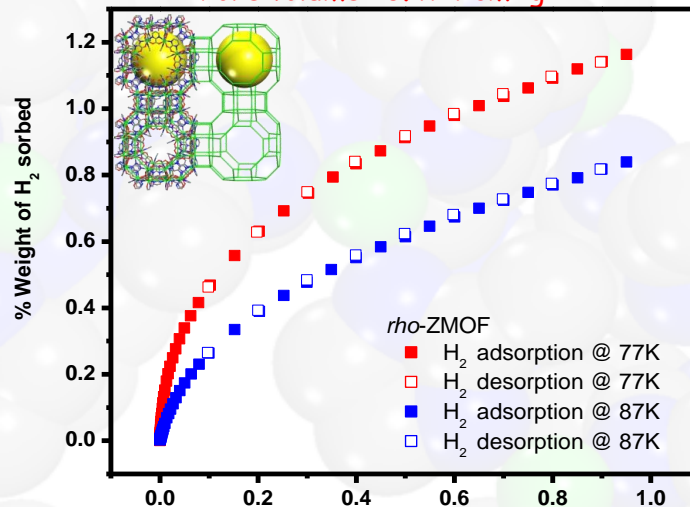


# H<sub>2</sub> sorption isotherms

Surface area= 616 m<sup>2</sup>·g<sup>-1</sup>  
Pore volume= 0.245 cm<sup>3</sup>·g<sup>-1</sup>

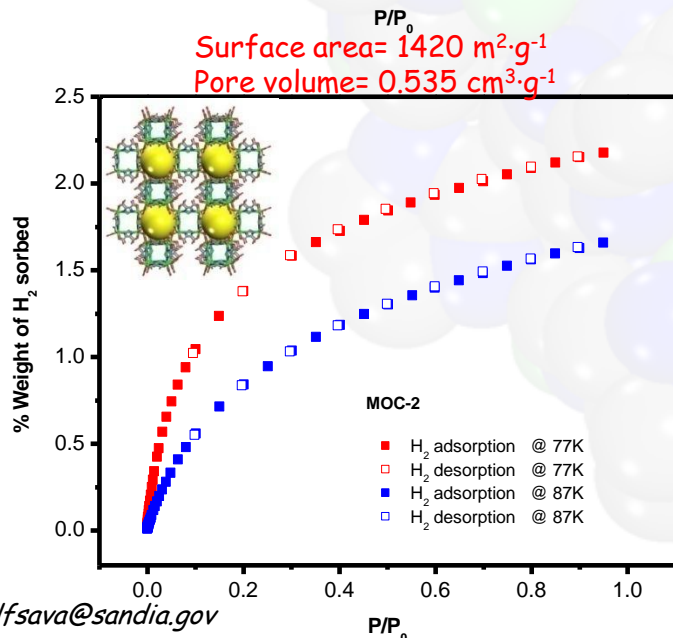


Surface area= 1168 m<sup>2</sup>·g<sup>-1</sup>  
Pore volume= 0.474 cm<sup>3</sup>·g<sup>-1</sup>

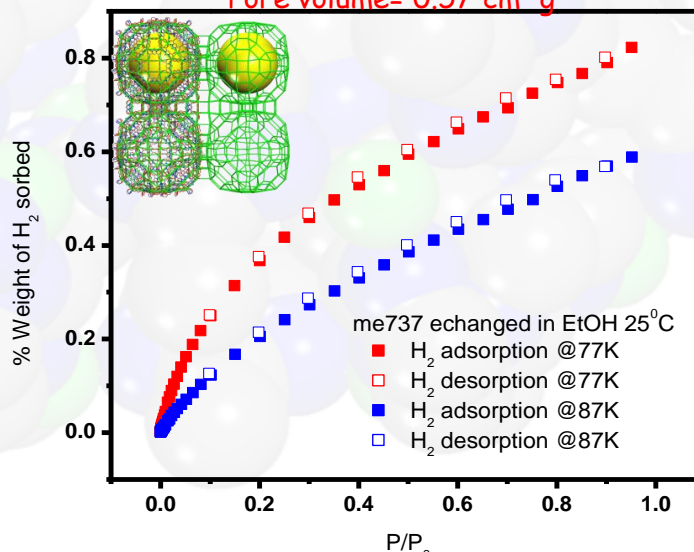


H<sub>2</sub> storage is dependant on surface area, pore size and pore volume, as well as on additional favorable *sites* for the gas adsorption to increase binding affinities

Surface area= 1420 m<sup>2</sup>·g<sup>-1</sup>  
Pore volume= 0.535 cm<sup>3</sup>·g<sup>-1</sup>



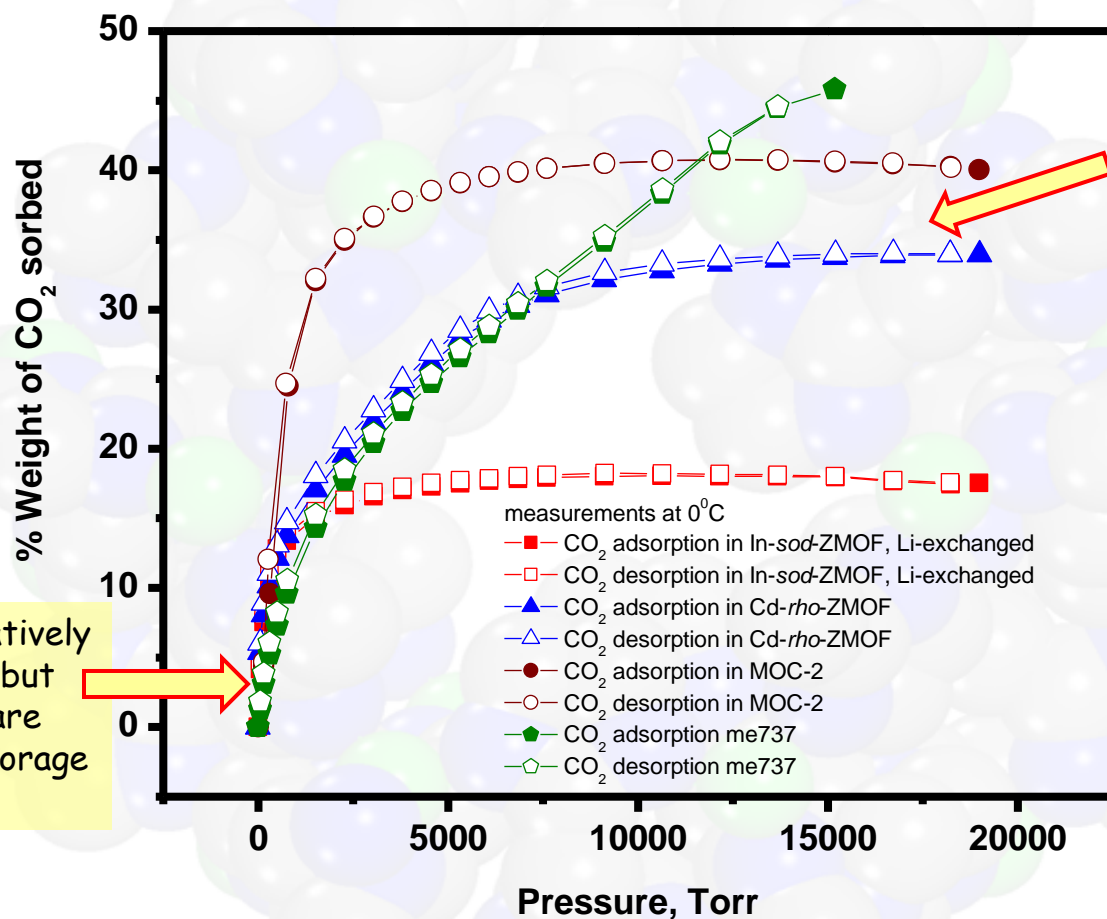
Surface area= 1476 m<sup>2</sup>·g<sup>-1</sup>  
Pore volume= 0.57 cm<sup>3</sup>·g<sup>-1</sup>



Materials with *high local charge densities* increase the binding affinities between H<sub>2</sub> and framework



# CO<sub>2</sub> sorption isotherms measured at 273K



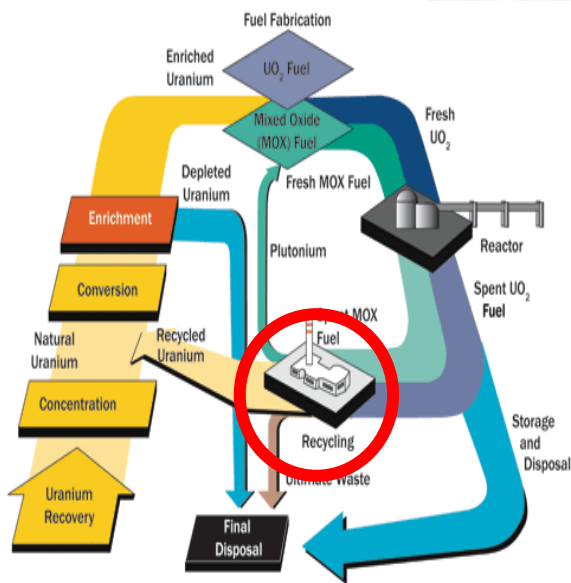
Materials with relatively narrow pore sizes, but large pore volume are suitable for CO<sub>2</sub> storage at low pressure

Compounds with both large and extra-large cavities and high pore volumes- important for applications conducted at high pressures



## II. Capture and Storage of Volatile Fission Gas Products from Reprocessing and/or Nuclear Accidents

### Nuclear Fuel Reprocessing (NE)



Source: U.S. Nuclear Regulatory Commission

### Accidental Release



Fukushima Daiichi Nuclear Power Plant explosion; March 11, 2011

$I^{129}$ ,  $I^{131}$  volatile gas released  
(www.IAEA.org)

### The importance of capturing volatile radioactive iodine ( $I_2$ ) gas

- *Appropriate nuclear waste management*: a main concern for safety associated with nuclear energy
- Particularly challenging is the capture of volatile gaseous fission products from nuclear fuel reprocessing or inadvertent environmental release:  $^{129}I$  and  $^{131}I$ ,  $^3H$ ,  $^{14}CO_2$ , and  $^{85}Kr$
- Unique exposure problems for radio- $I_2$  isotopes:
  - $^{129}I$ : long-lived isotope (half-life of  $1.57 \times 10^7$  years), requiring *capture and reliable storage while it decays*
  - $^{131}I$ : short-lived (half-life of 8.02 days), but *requires immediate capture* as it directly affects human metabolic processes





# Judicious selection of an “ideal” candidate for I<sub>2</sub> sorption

## Pre-requisites

- Restrictive pore apertures to impart molecular selectivity for directional diffusion of iodine ( $\sim 3.35 \text{ \AA}$ )
- Large surface area and pore volume
- High *chemical, thermal, and moisture* stability







# Judicious selection of an “ideal” candidate for I<sub>2</sub> sorption

## Pre-requisites

- Restrictive pore apertures to impart molecular selectivity for a directional diffusion of iodine ( $\sim 3.35$  Å)
- Large surface area and pore volume
- High *chemical, thermal, and moisture* stability

## ZIF-8: Zn(2-methylimidazole)<sub>2</sub>

- ✓  $\beta$ -cages= 11.6 Å diameter,  
Pore apertures  $\sim 3.4$  Å
- ✓ Surface area ZIF-8 = 1,947 m<sup>2</sup> g<sup>-1</sup>  
Pore volume= 0.663 cc g<sup>-1</sup>
- ✓ Chemically stable in boiling solvents (including water), and thermally stable up to 550°C

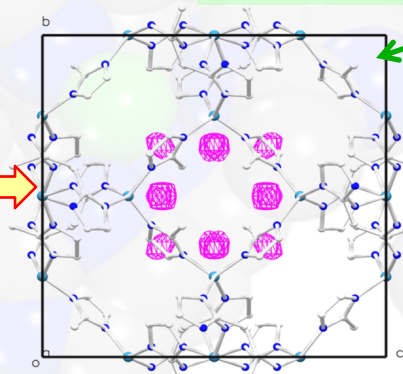
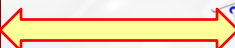
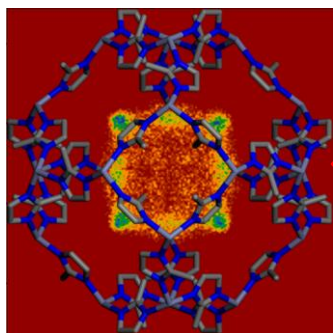




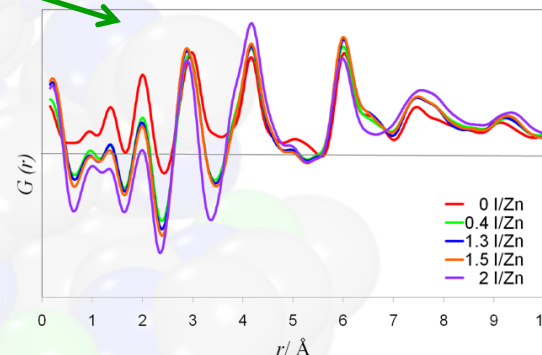
## IIa. Establishing structure-function relationship: integration of experiment and modeling

### Molecular modeling

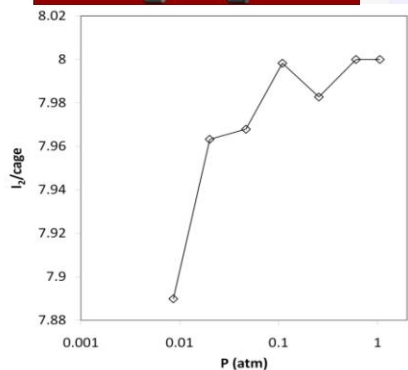
### Complementary local and long-range structural probes



High-resolution synchrotron data:  
Difference-Fourier analysis map  
confirms  $I_2$  location within cages



Pair Distribution Function  
analysis: histogram of all  
interatomic distances,  
independent of sample  
crystallinity



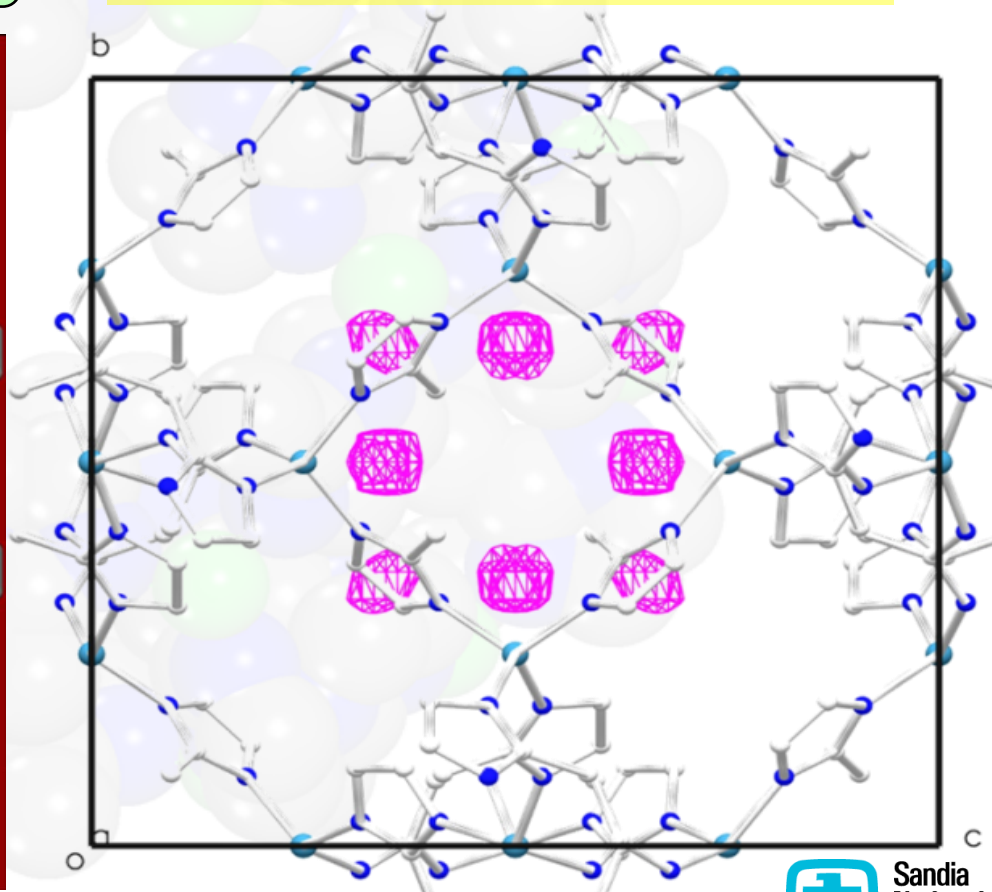
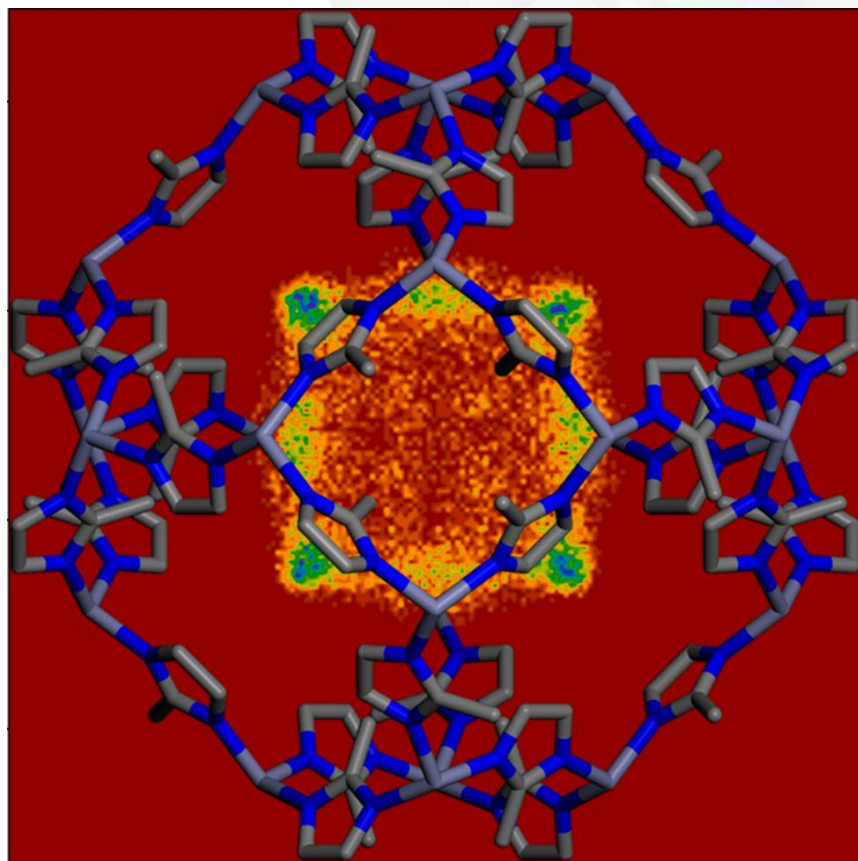
MD and GCMC simulations predict  $I_2$   
binding locations and maximum loading  
capacities



# MD simulations predict $I_2$ location and loading capacities within ZIF-8 cage

2 I/Zn (~6  $I_2$  per cage)

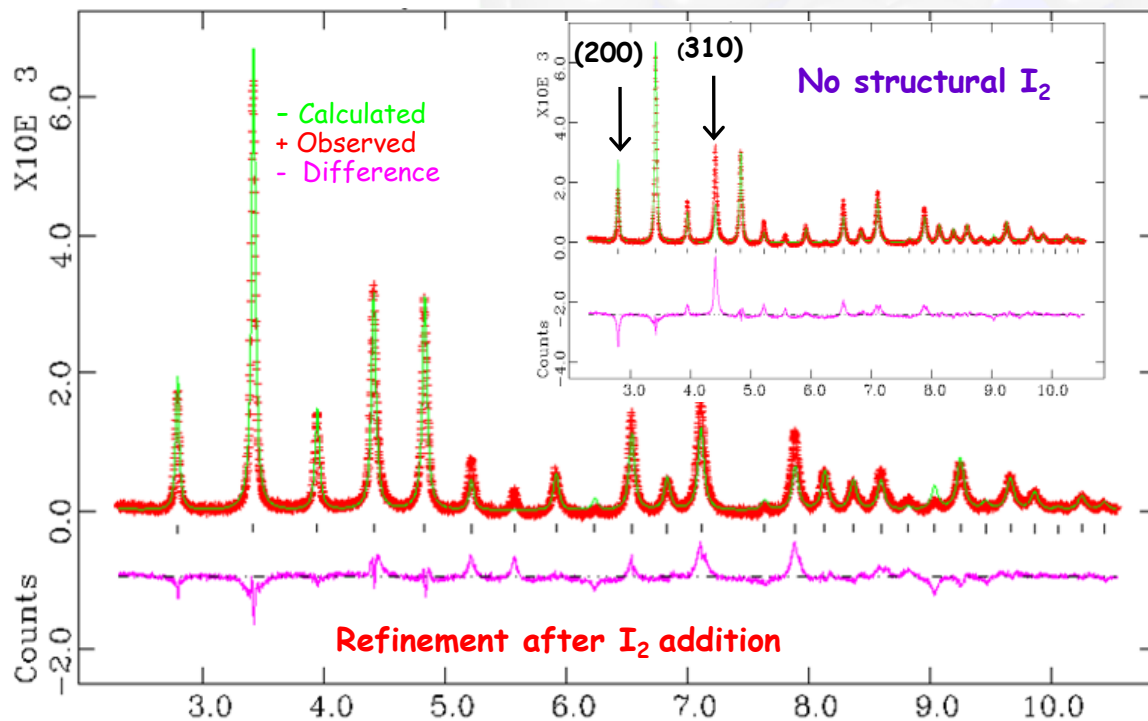
Time-averaged atomic density plot:  
increased density at positions closely  
related to those indicated in the  
Bragg analysis



Sava, D.F. et.al, *J. Am. Chem. Soc.* **2011**, 133, 12398.



# High-resolution synchrotron-based XRD



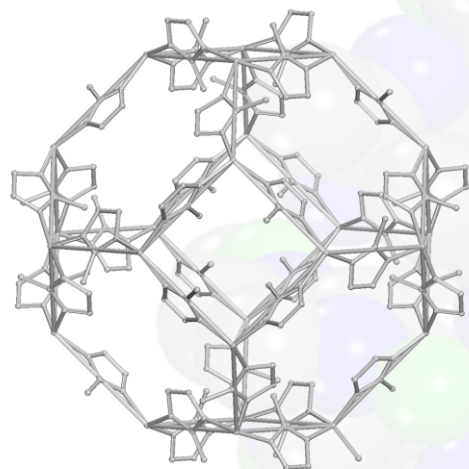
Calculated, observed and difference spectrum of **0.4 I/Zn (25 wt.% I<sub>2</sub>)** loading of I<sub>2</sub>@ZIF-8 after I<sub>2</sub> inclusion in structure refinement by Rietveld analysis (inset: before I<sub>2</sub> inclusion).

- Sample crystallinity is maintained up to ~1.3 I/Zn (70 wt.% I<sub>2</sub>) loadings
- Bragg reflections broaden significantly beyond this value → difficult to distinguish from the pronounced diffuse features in the "background"
- Experimental maximum capacity is reached at ~2 I/Zn (110 wt.% I<sub>2</sub>) loadings





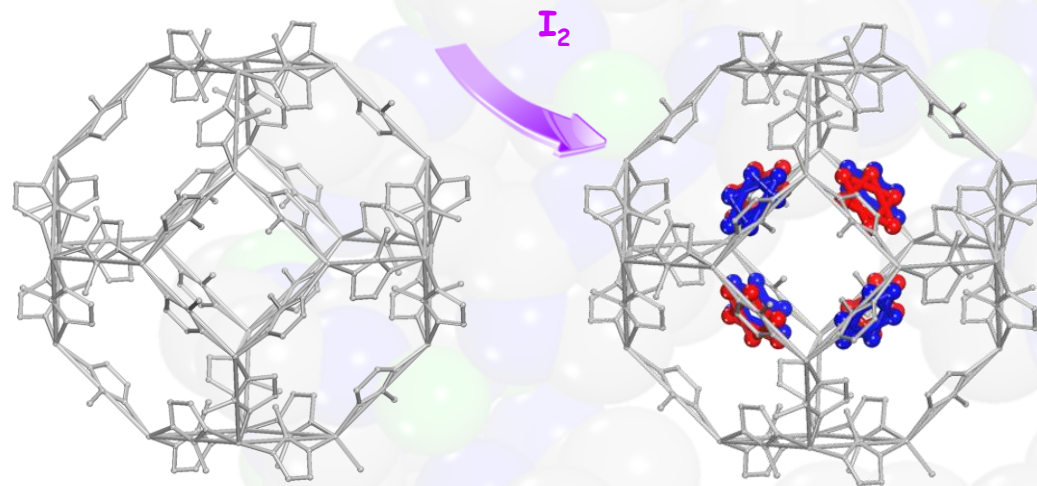
# Identifying the I<sub>2</sub> binding locations



Activated  $\beta$ -cage



# Identifying the I<sub>2</sub> binding locations



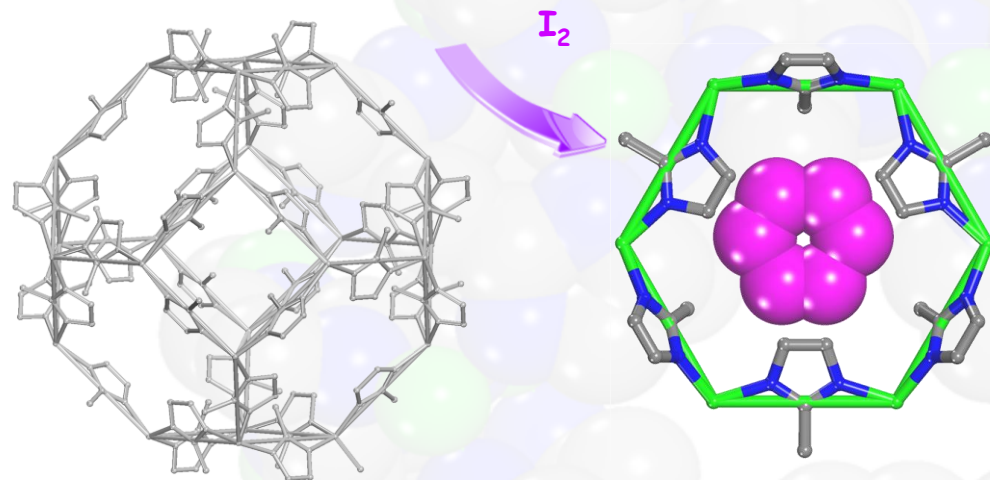
Activated  $\beta$ -cage

Dynamically disordered  
I<sub>2</sub> molecules





# Identifying the I<sub>2</sub> binding locations

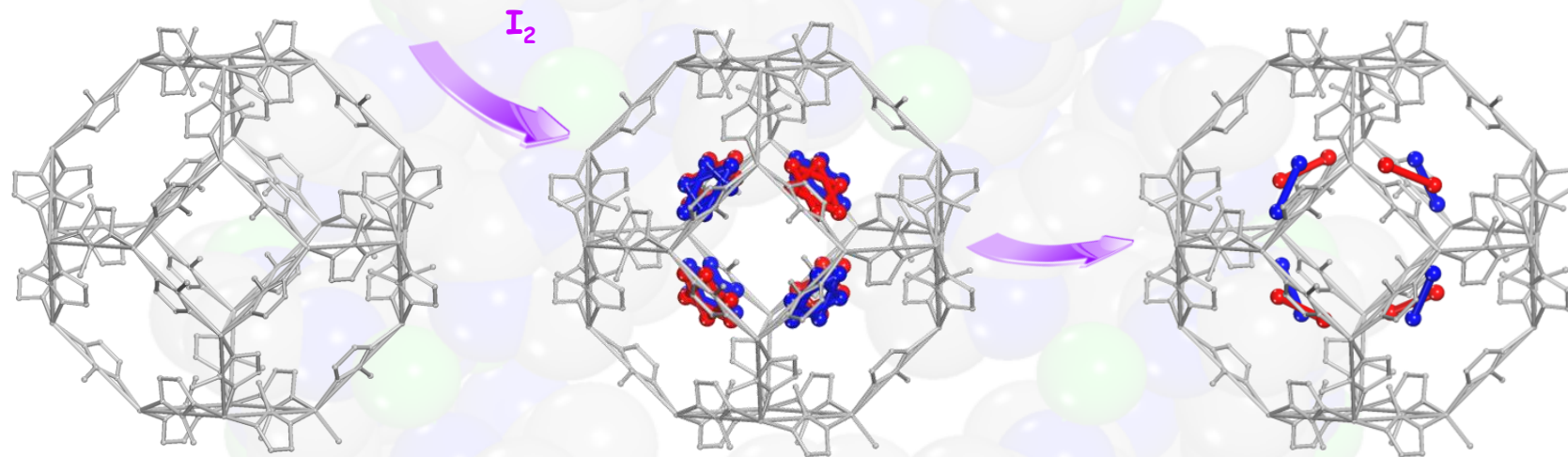


Activated  $\beta$ -cage

Dynamically disordered  
I<sub>2</sub> molecules



# Identifying the I<sub>2</sub> binding locations



Activated  $\beta$ -cage

Dynamically disordered  
I<sub>2</sub> molecules

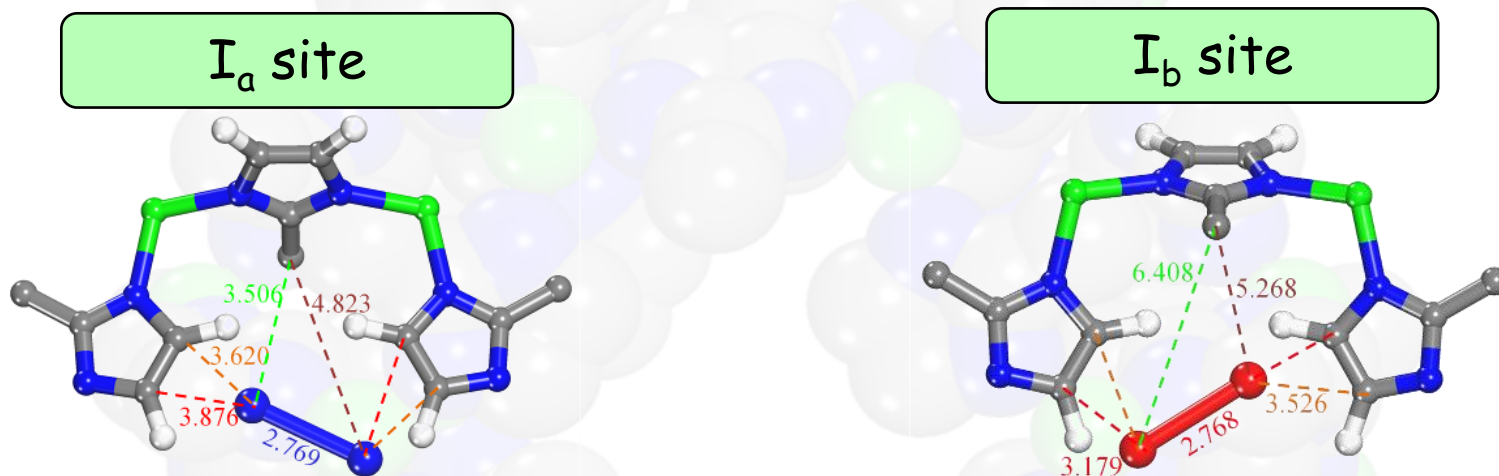
Refined I<sub>2</sub> sites:  
I<sub>a</sub> (blue) and I<sub>b</sub>  
(red)





# $I_2$ adsorption in ZIF-8 is due to favorable interactions with the 2-MeIM linker

Two distinct  $I_2$  binding sites:  $I_a$  and  $I_b$

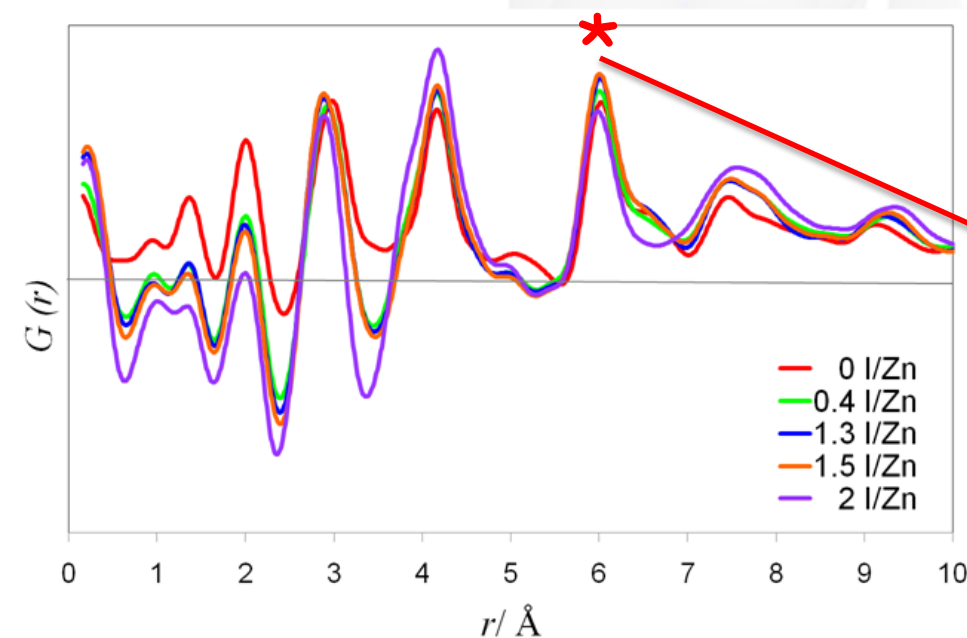


$I_2$  site occupancy and  $I_2 \cdots$ MeIM close contacts in  $I_2@ZIF-8$

$I_2$ site	Site occupancy		Contacts with MeIM	
	0.4 I/Zn	1.3 I/ Zn	C (CH <sub>3</sub> )	C(H=CH)
$I_a$	0.28	0.88	3.506 Å; 4.823 Å	3.620 Å; 3.876 Å
$I_b$	0.14	0.38	5.268 Å; 6.408 Å	3.179 Å; 3.526 Å



# Pair Distribution Function analysis: local structure probe

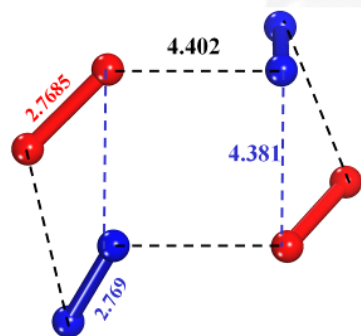
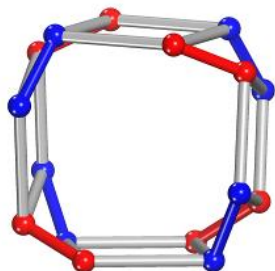


Short-range order and framework connectivity are maintained at all loading levels

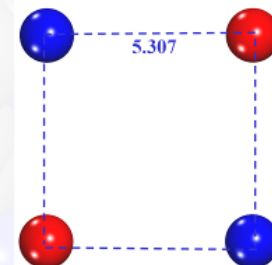
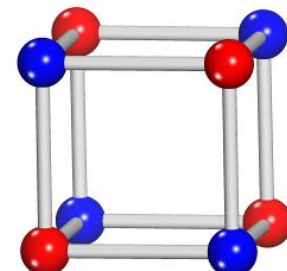
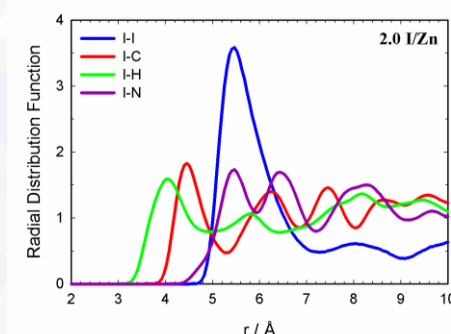
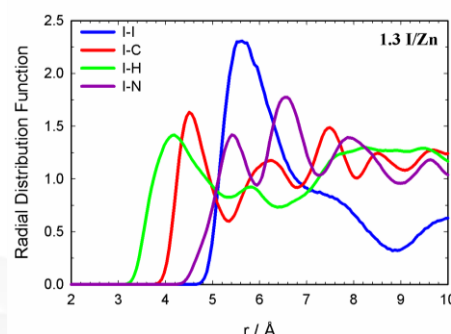
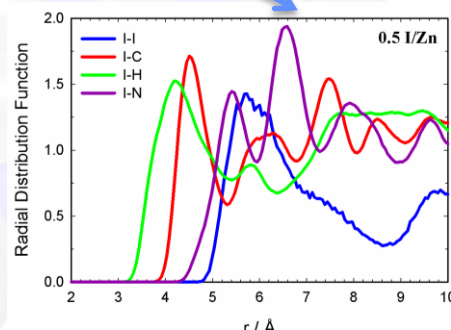
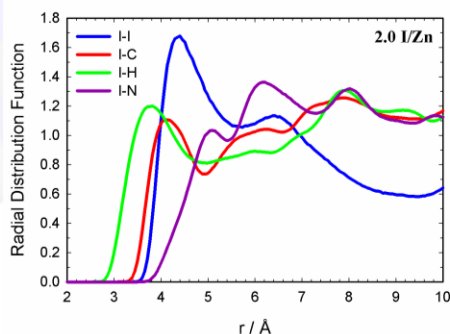
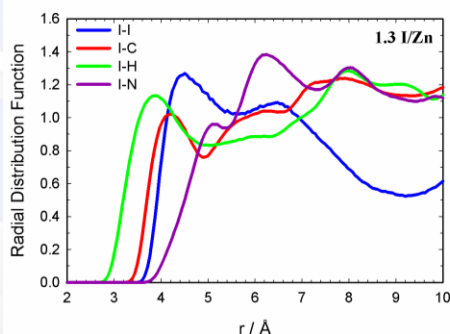
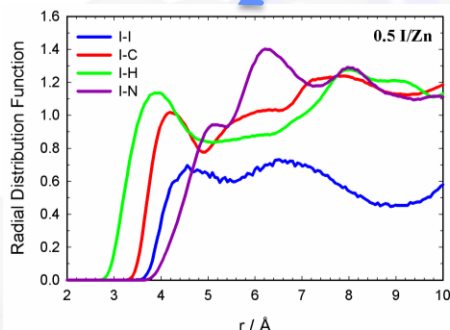
- The PDF method- a weighted histogram of the atomic distances, independent of sample crystallinity
- Below  $\sim 6 \text{ \AA}$ , the MOF cage features are retained in the PDF at all  $I_2$  loading levels
- The persistence of the peak at  $\sim 6 \text{ \AA}$ , corresponding to the Zn-(MeIM)-Zn' distance
- $>1.3 \text{ I/Zn}$  (70wt%) lose long range structural information though individual cages maintain crystalline integrity



# Radial Distribution Functions (RDFs) for diatomic and united-atom models



*Distorted octagonal prism formed by refined  $I_2$  molecules, and corresponding intermolecular interactions.*

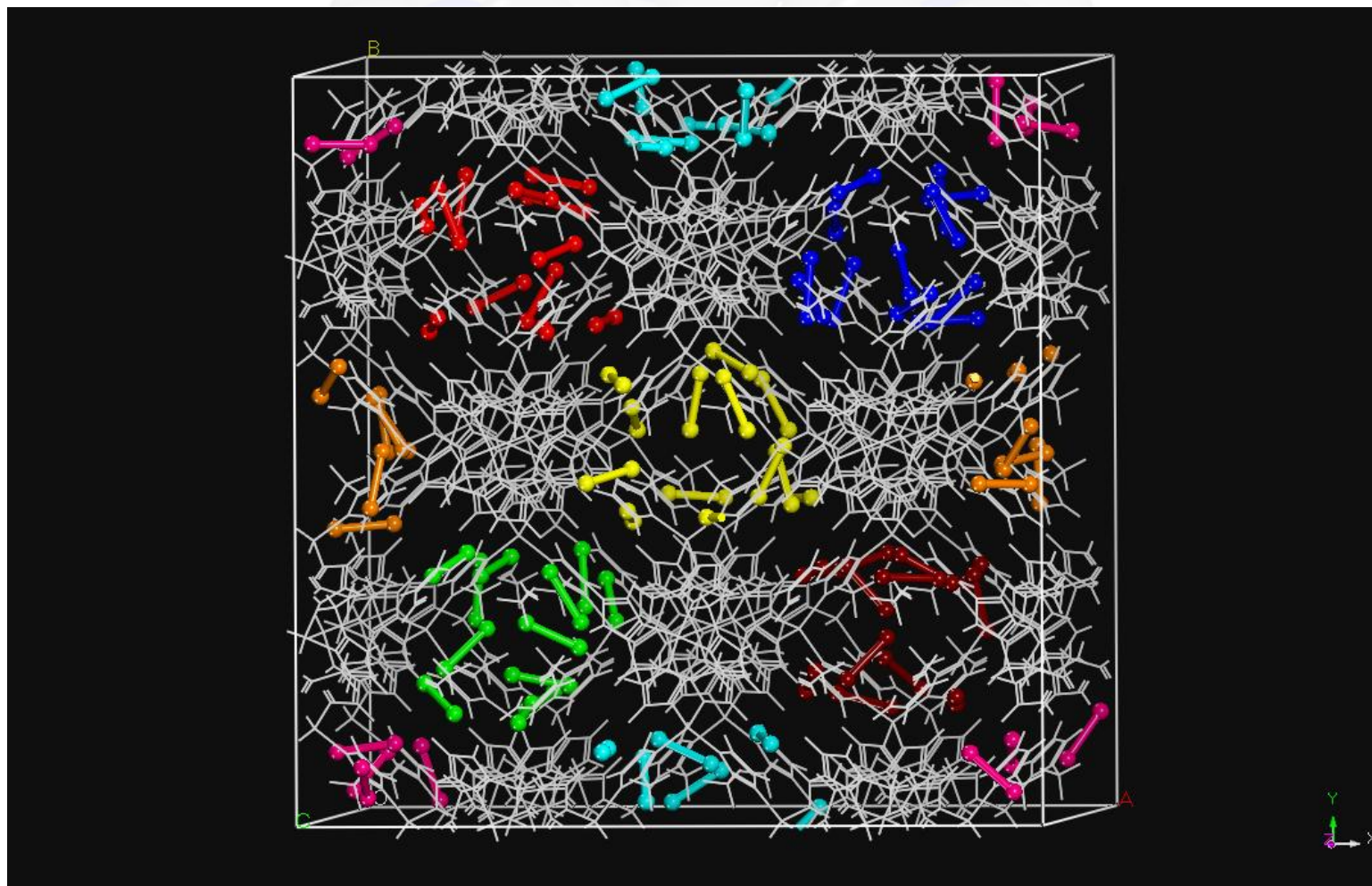


*Cubane-like cluster derived from a united-atom model for  $I_2$ , and representative contacts for this molecular arrangement.*

Good agreement between crystallography, PDF and modeling regarding nearest neighbors distances



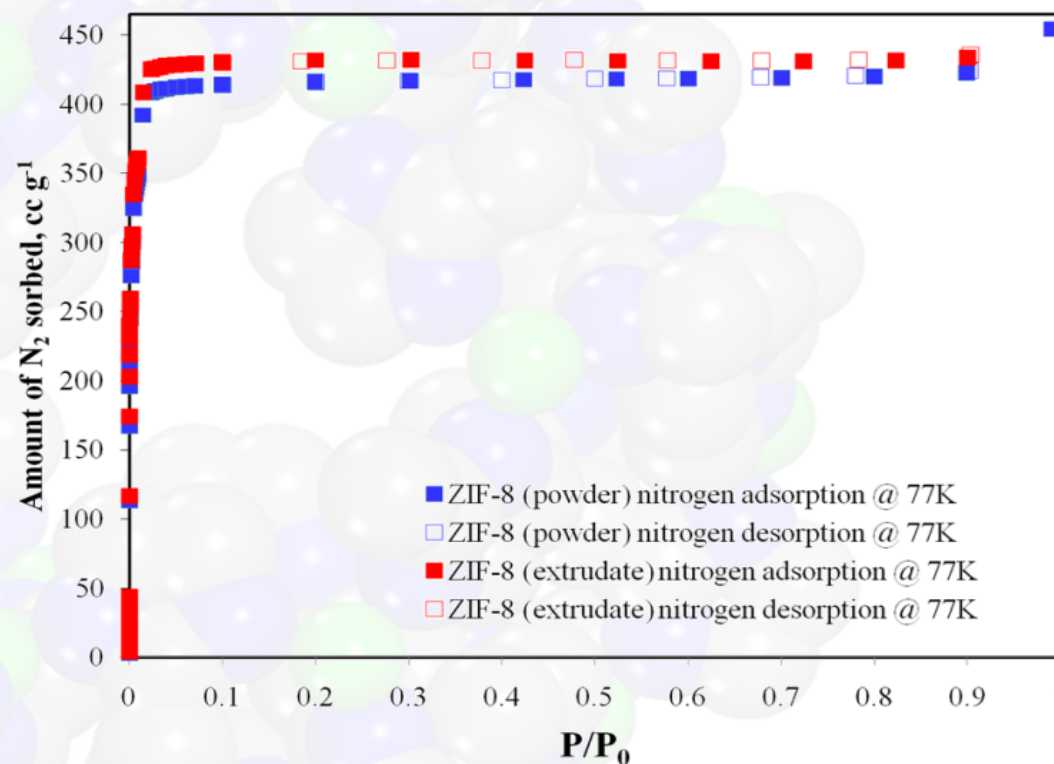
# Dynamics of $I_2$ within cages: $I_2$ mobility is restricted within individual cages







# ZIF-8 binder-free extrudates: on par performance with ZIF-8 powder

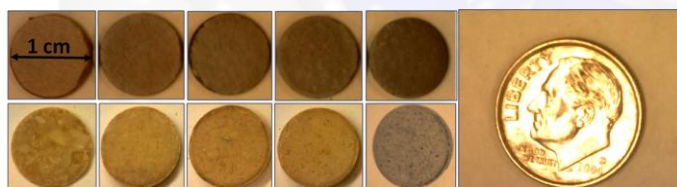
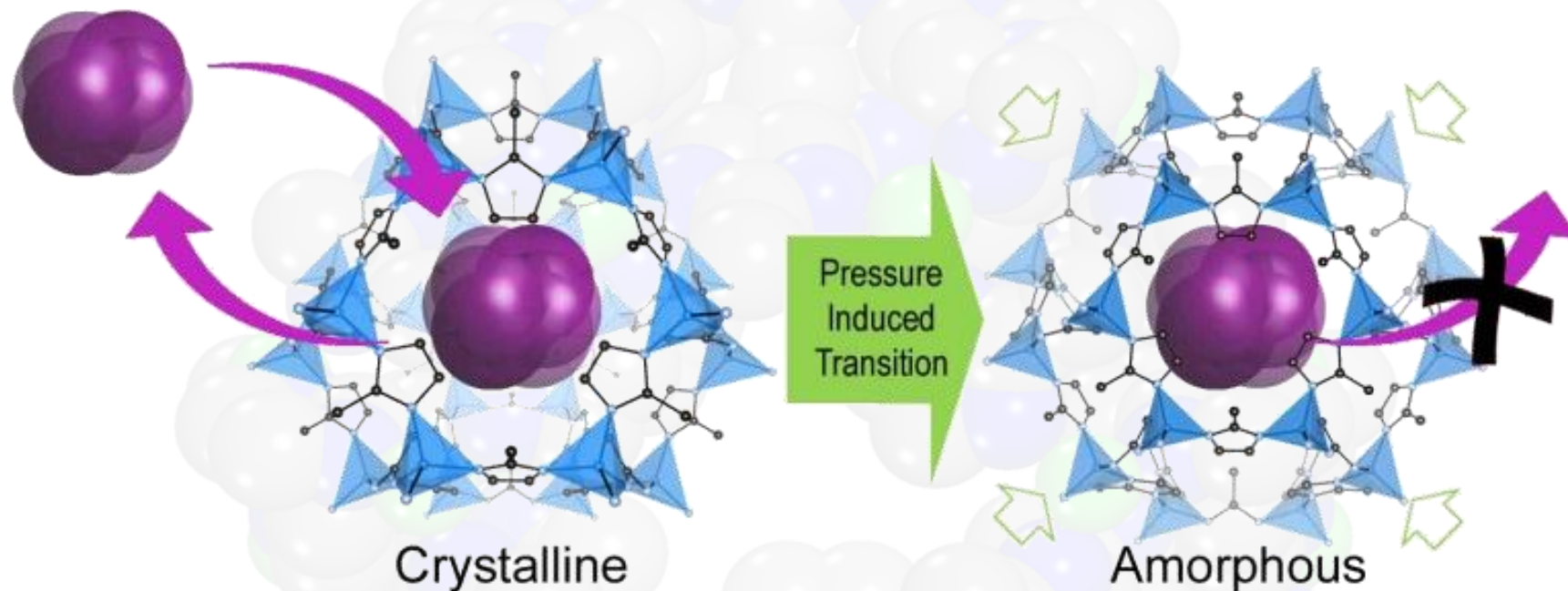


I<sub>2</sub> loaded extrudates

*Technical advance for US Patent # 11971 submitted, 04/2011*



## IIb. $I_2$ @ZIF-8 pressure-induced amorphization for capture and temporary storage

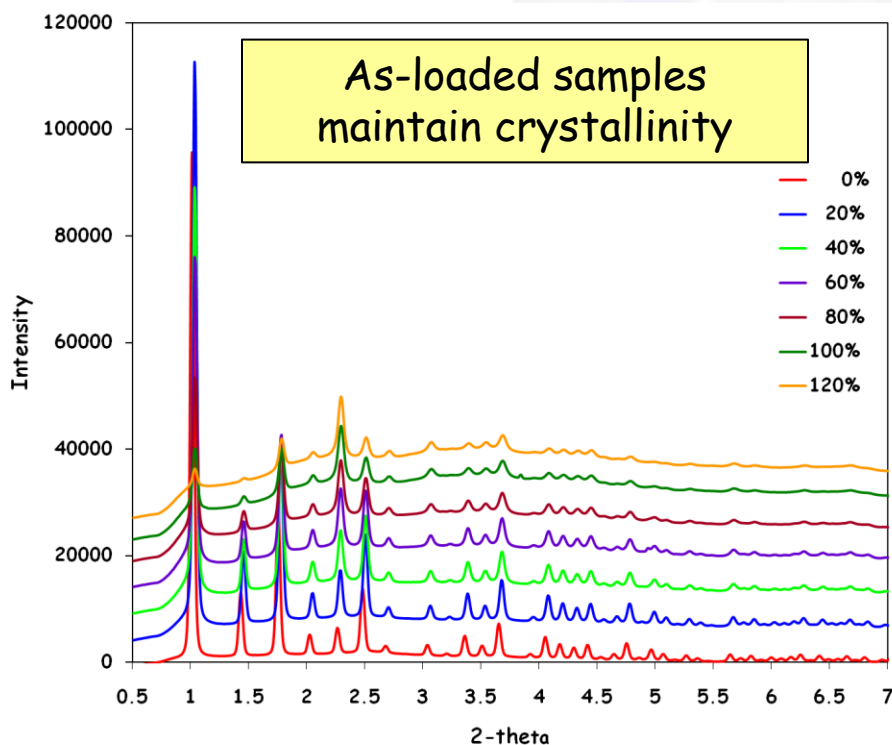


Secure consolidated interim storage  
before incorporation into a long term  
waste form

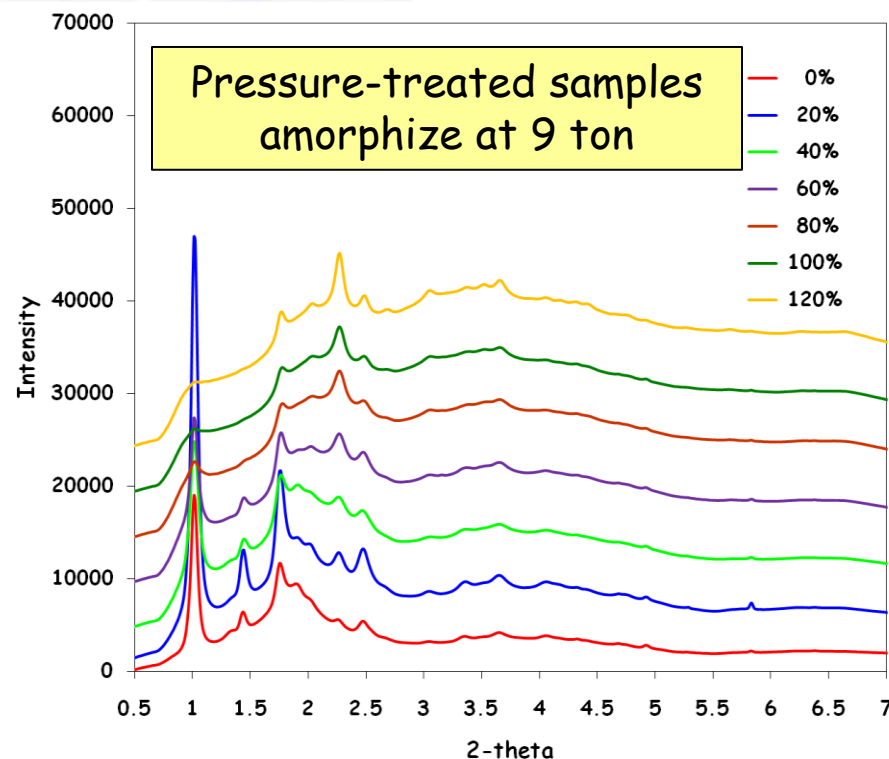
Crack free pellets of iodine loaded ZIF-8 powders were obtained by applying uniaxial mechanical pressure.



# High-resolution synchrotron-based XRD on a series of incrementally loaded $I_2@ZIF-8$



High-resolution synchrotron-based XRD of as-loaded 20, 40, 60, 80, 100 and 120 wt%  $I_2@ZIF-8$ .

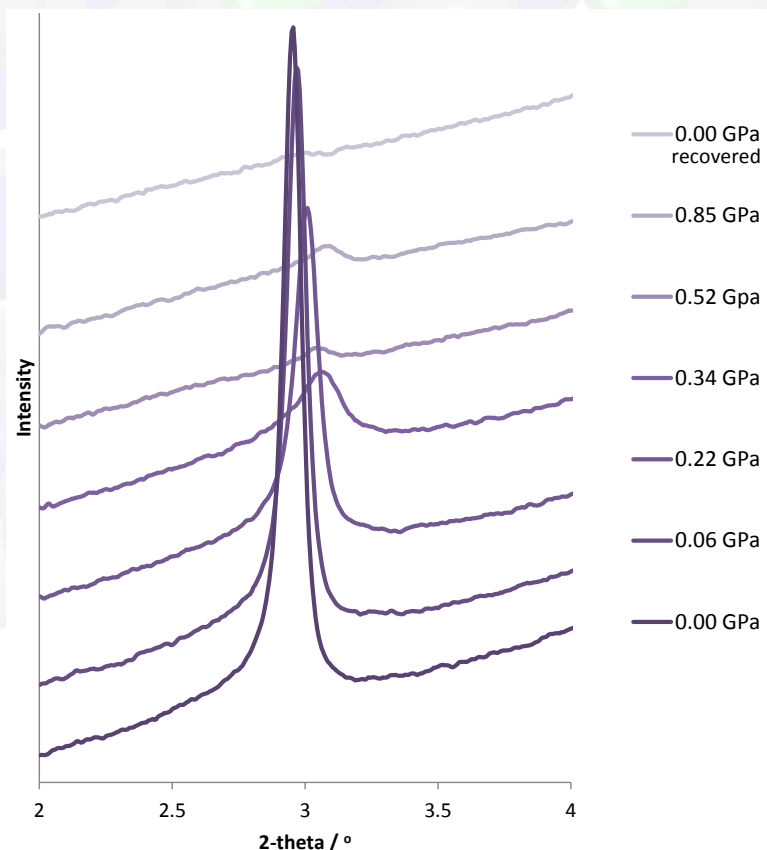


High-resolution synchrotron-based XRD of powders treated to 9 ton of 20, 40, 60, 80, 100 and 120 wt%  $I_2@ZIF-8$ .



# Variable pressure diffraction studies on I<sub>2</sub>@ZIF-8

The pressure dependence of I<sub>2</sub>-containing ZIF-8 highlighting the 110 reflection. Above 0.34 GPa, the crystalline diffraction features are largely eliminated and the framework remains amorphous upon recovery to ambient pressure.



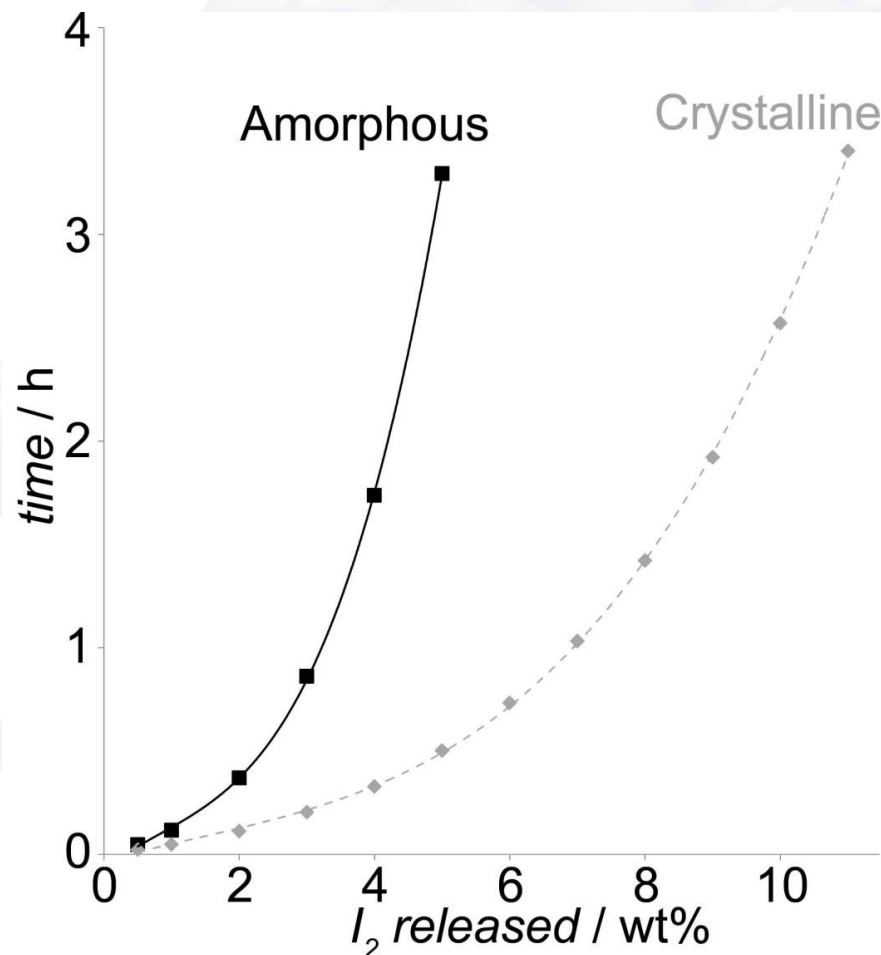
- *In-situ* X-ray diffraction were collected at the 1-BM beamline at the APS/ANL.

- The pressure-dependent structure of ZIF-8 loaded with 40wt% I<sub>2</sub>, was probed using synchrotron-based powder diffraction for the sample within a diamond anvil cell pressure apparatus.





# Enhanced guest retention through amorphization



The kinetics of  $I_2$  release from the crystalline and amorphized ZIF-8 based on isothermal TGA data collected at 200°C, 4 hours

Interim waste form: desorption kinetics are retarded in the amorphized material

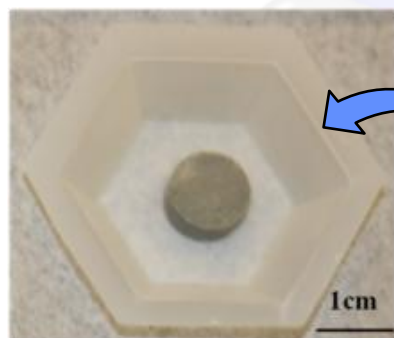


# I<sub>2</sub>@ZIF-8 successfully encapsulated in stable long-term waste forms

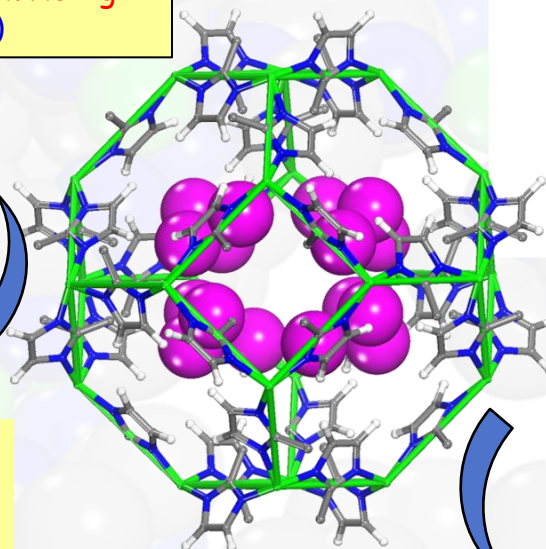
TA #SD11720, Provisional Patent 2012

## Incorporation into Low Temperature Glass Composite Materials (GCM)

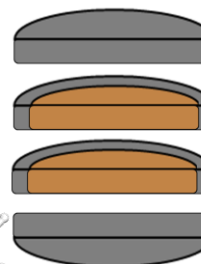
80 wt.% glass, 10 wt.% I<sub>2</sub>@MOF, 10 wt.% Ag  
(non-optimized compositions)



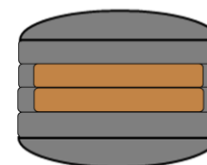
GCM reveal excellent thermal & chemical stability and are appropriate for long term storage



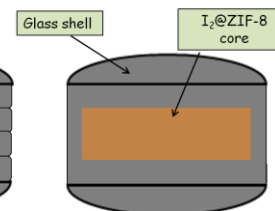
Cross-section of core-shell assembly



Pellets stacking



Core-shell sintering



Very high capacity of consolidated radiological waste



## Long-term core-shell waste form

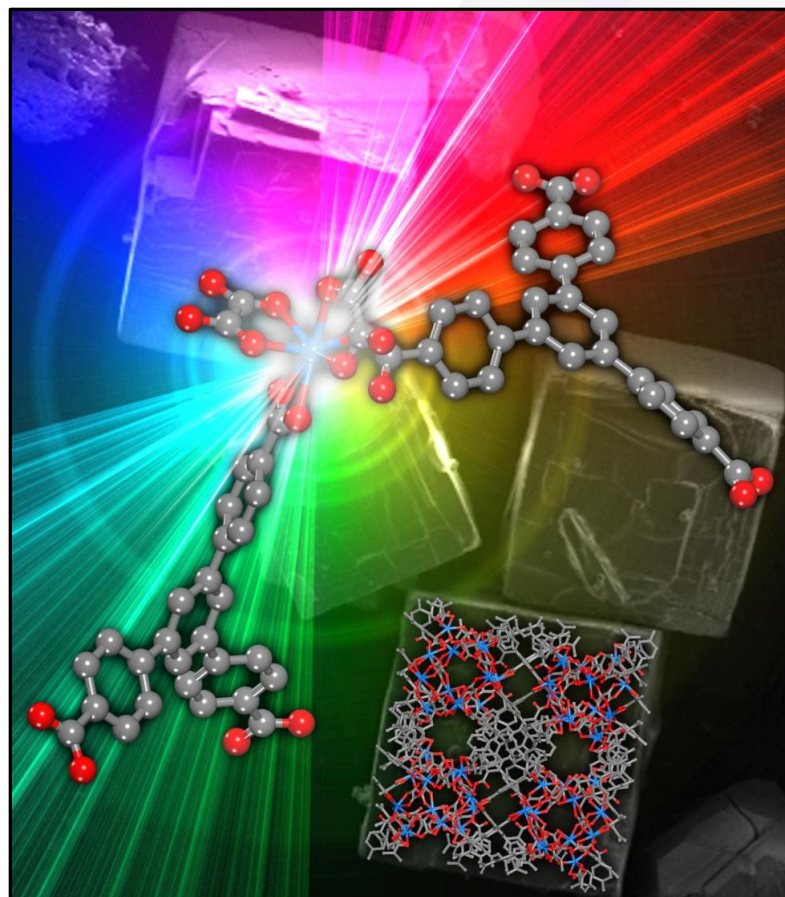
The core consists of amorphized I<sub>2</sub>@ZIF-8 pellets, and the barrier shell is made of the low-temperature sintering glass EG2922.

Sava, D.F. et. al *Ind. Eng. Chem. Res.* **2012**, 51 (2), 614-620. Invited paper for the thematic Nuclear Energy special issue

Glass	ZnO		Bi <sub>2</sub> O <sub>3</sub>		Al <sub>2</sub> O <sub>3</sub>		SiO <sub>2</sub>	
	mole %	wt. %	mole %	wt. %	mole %	wt. %	mole %	wt. %
EG 2922	14.2	7.8	20.2	63.4	7.8	5.4	57.8	23.4



### III. Novel MOFs with Tunable Color Properties

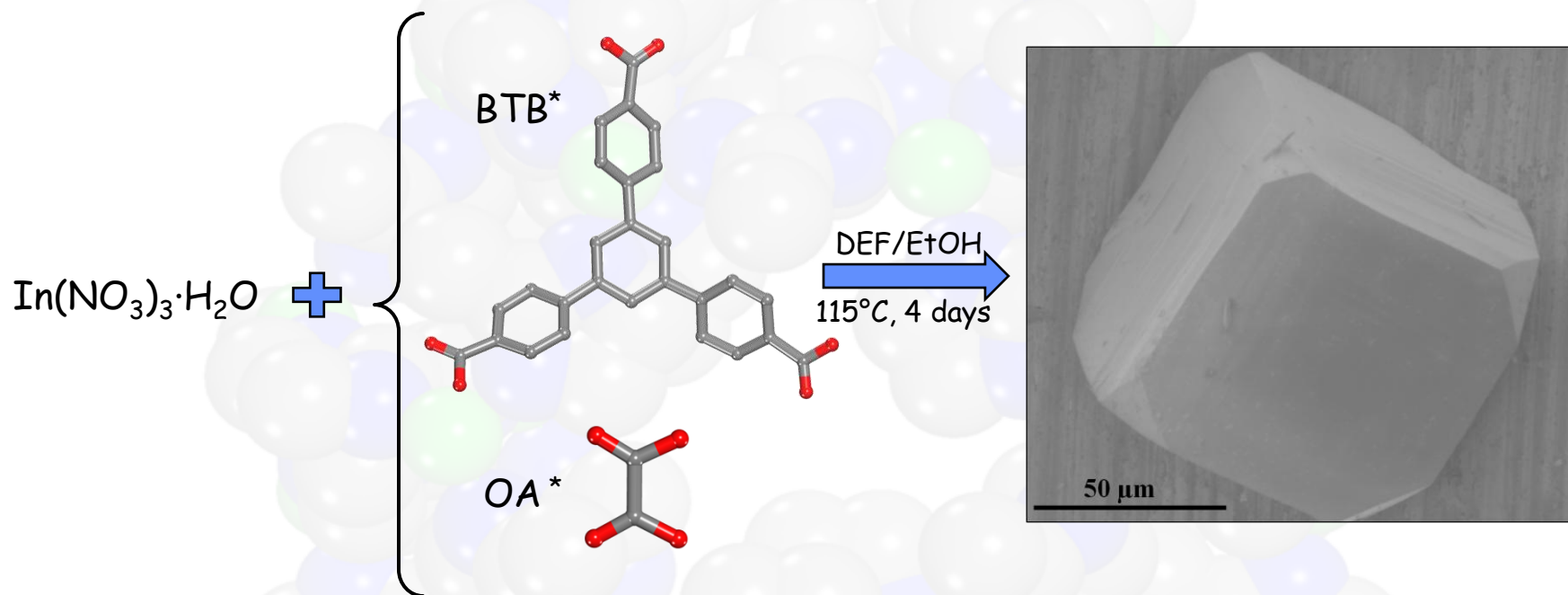


- White light from a *single phosphor* - alternatives are sought to existing color mixing approaches
- Current white LEDs for SSL: InGaN LEDs excite a yellow-emitting YAG:Ce phosphor - *cool white light made warmer* by incorporating a red-emitting phosphor
- Warm white LEDs: near-UV InGaN LEDs to excite blends of red-, green-, and blue-emitting phosphors → the additional down-conversion step (near-UV to blue) *significantly lowers the conversion efficiency of the device.*





# SMOF-1 (Sandia Metal-Organic Framework-1): a tunable novel In-BTB framework



✓ The first In-BTB reported net

✓ The first oxalic acid-BTB system explored to date

Cubic, 3-periodic

$a = 33.975(3) \text{ \AA}$ , Ia-3  
 $V = 39,217(10) \text{ \AA}^3$

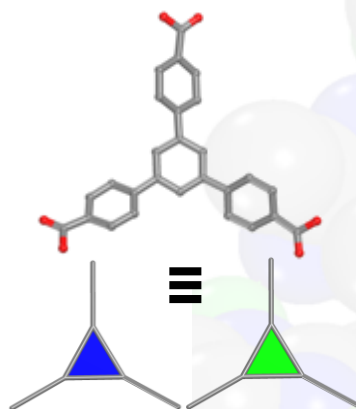
\* BTB (4,4',4''-benzene-1,3,5-triyl-tri-benzoic acid)

\* OA (oxalic acid)

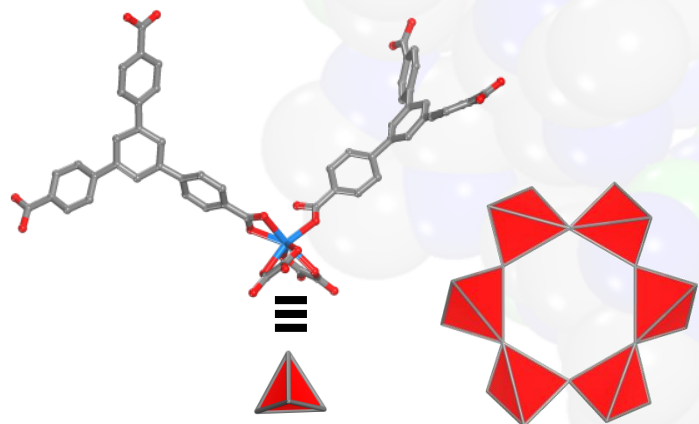




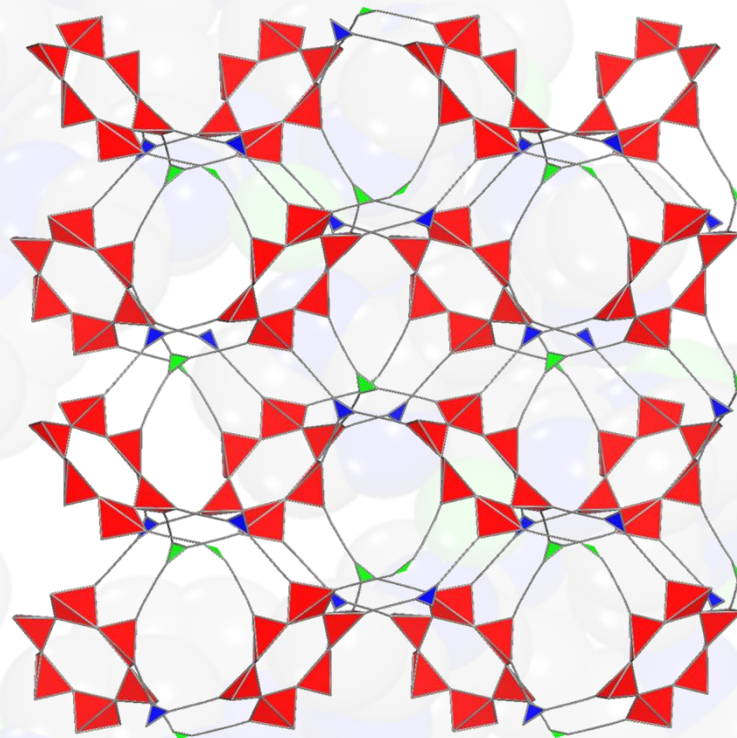
# Topological evaluation: unprecedented (3,3,4) trinodal net



Two topologically distinct  
3-connected nodes



4-connected node



Coordination sequences:

V1: 3 9 22 36 58 88 114 151 196 234

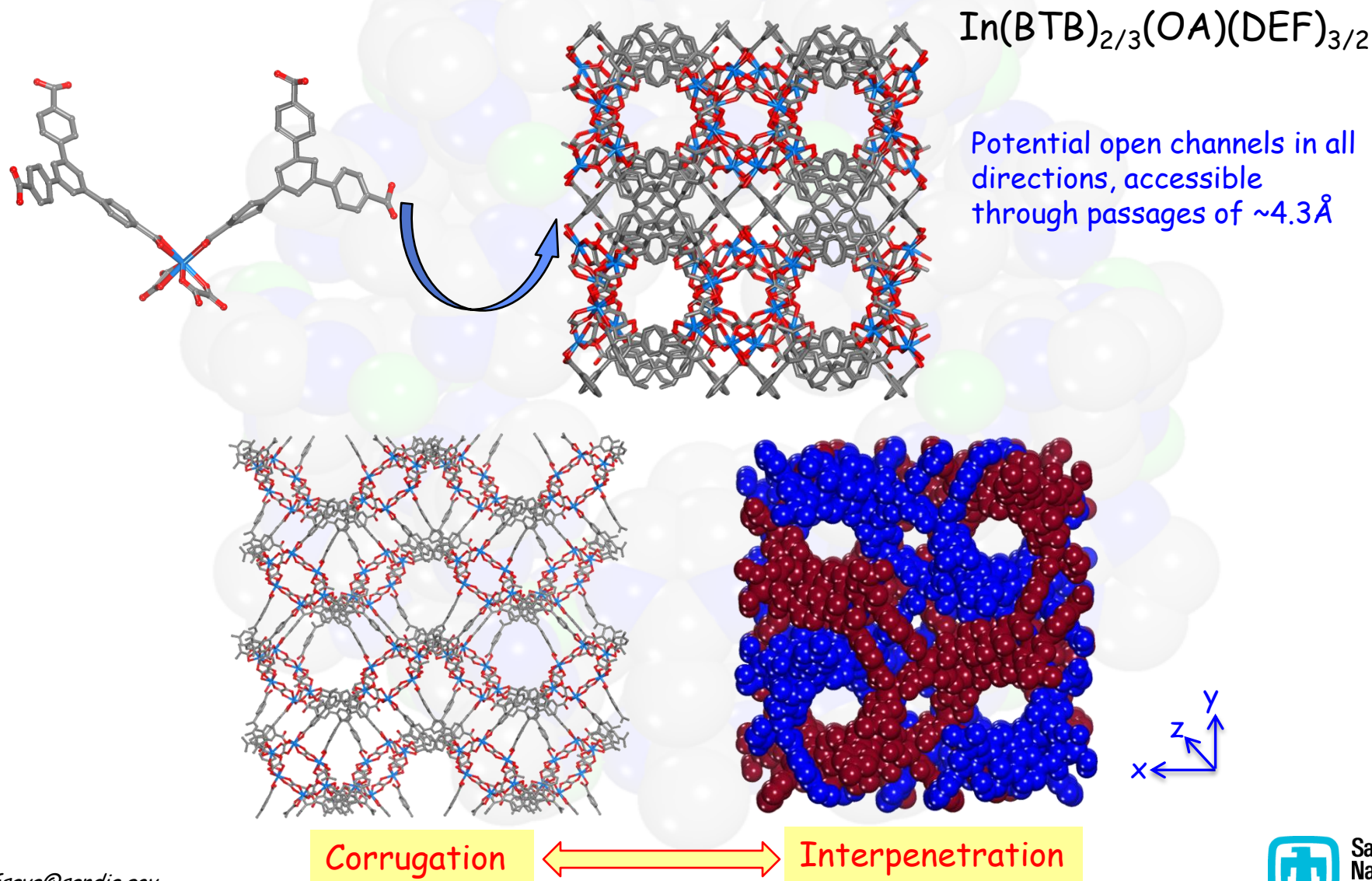
V2: 3 9 22 39 54 79 119 151 186 240

V3: 4 10 21 39 60 85 117 154 195 242

The short Schläfli (point) symbol:  $\{6^3.8^3\}3\{6^3\}2$



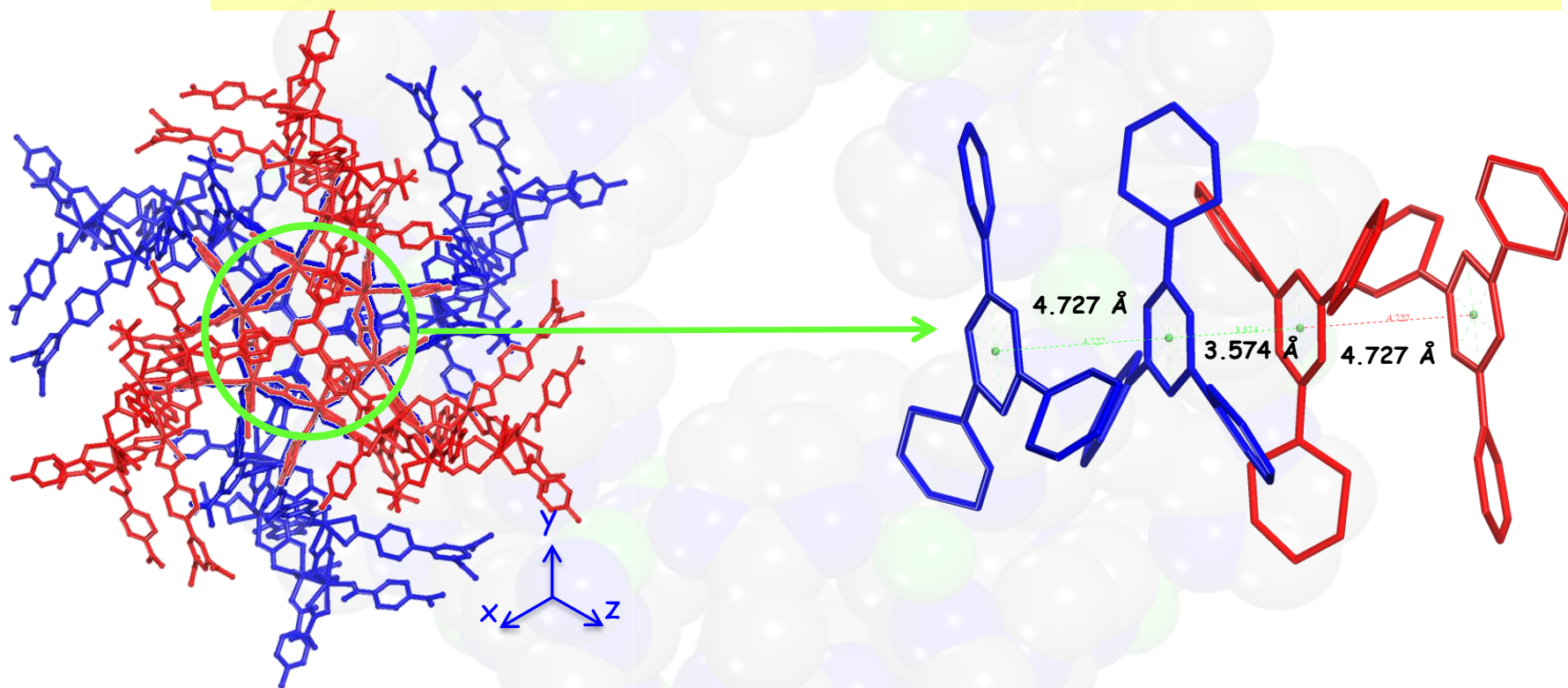
# Complementary structural features direct enhanced linker interactions





# Photoluminescence originates from the unique arrangement of the conjugated aromatic rings

The structure-function relationship in SMOF-1 is driven by two complementary unique structural features: corrugation and interpenetration



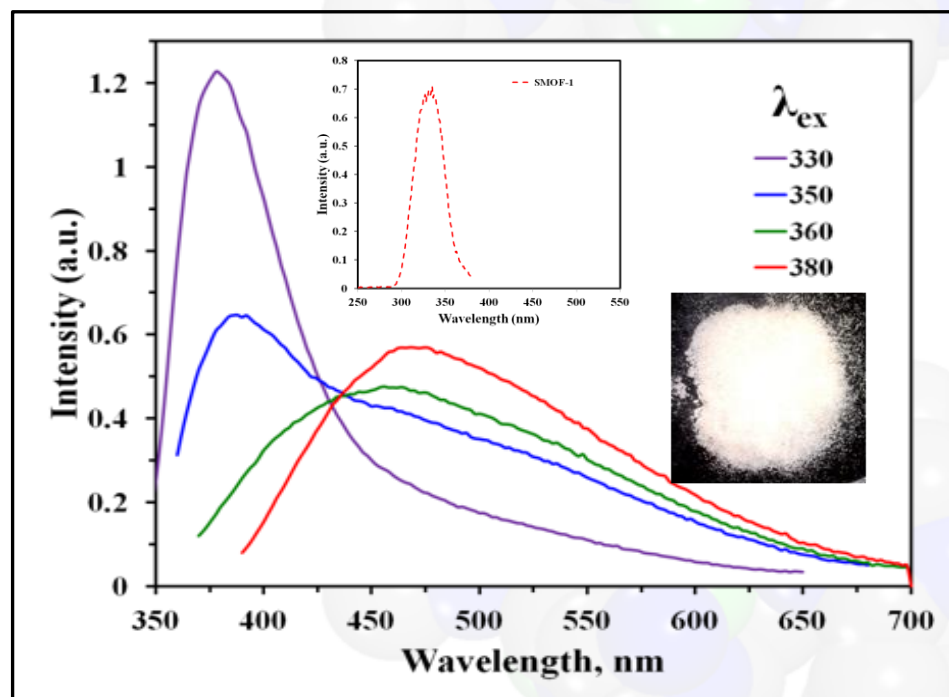
Unique arrangement of BTB linkers in SMOF-1 results in a cascade of  $\pi$ --- $\pi^*$  aromatic interactions





# SMOF-1 framework: direct broadband white-light emitter

White-light emission in SMOF-1 is due to a combination of  $\pi\text{---}\pi^*$  aromatic interactions and ligand to metal charge transfer (LMCT)



Excitation and emission spectra in SMOF-1

## Color properties in SMOF-1

$\lambda_{\text{ex}}$	CRI*	CCT* (K)	x	y
330	77.4	34463	0.209	0.193
350	84.5	22413	0.241	0.268
360	85.1	33290	0.234	0.275
380	81.1	21642	0.235	0.387

✓ CRI values fall within intended ranges (81-85)

- High CCT (21642-33290 K)

Department of Energy: Solid-State Lighting  
<http://www1.eere.energy.gov/buildings/ssl/>

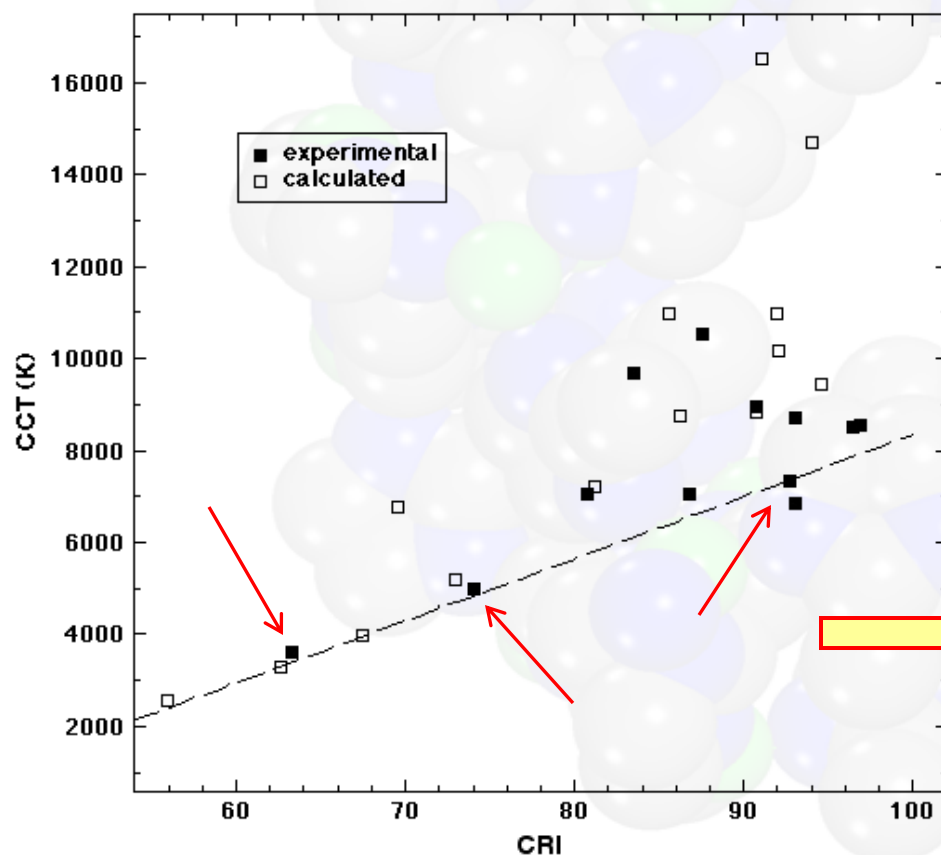
\*CRI (Color Rendering Index) ~ 90  
\*CCT (Correlated Color Temperature) ~ 3200 K





# Optimized color properties via simulated spectra

Calculated and experimental CCTs and CRIs for 2.5, 5, and 10% Eu-doped SMOF-1



➤ *Simulated spectra* were generated by summing an SMOF-1 and  $\text{Eu}^{3+}$  spectra, at excitation wavelengths of 350, 360, and 380 nm, respectively.

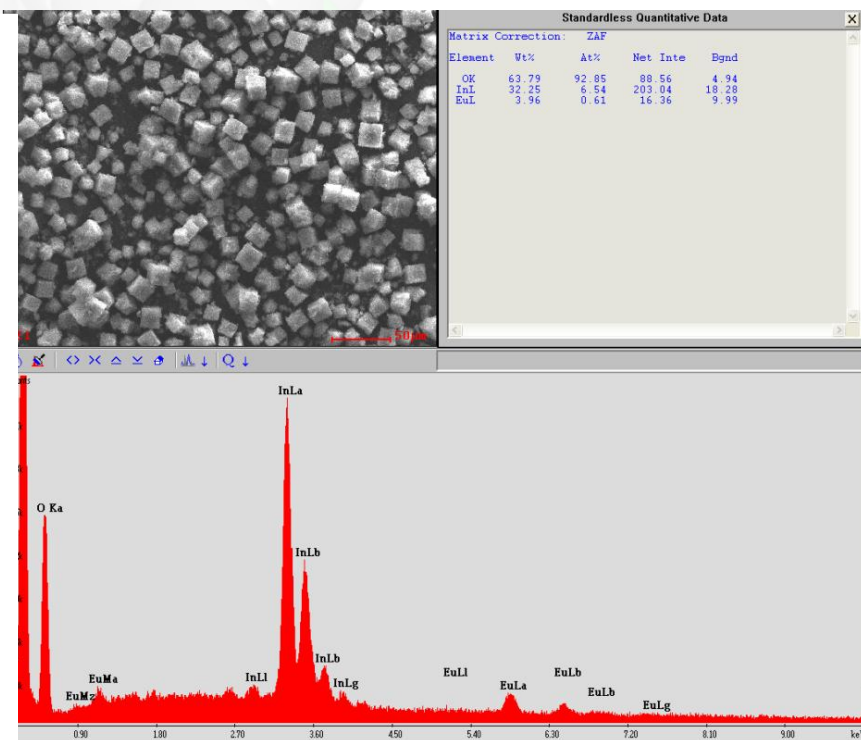
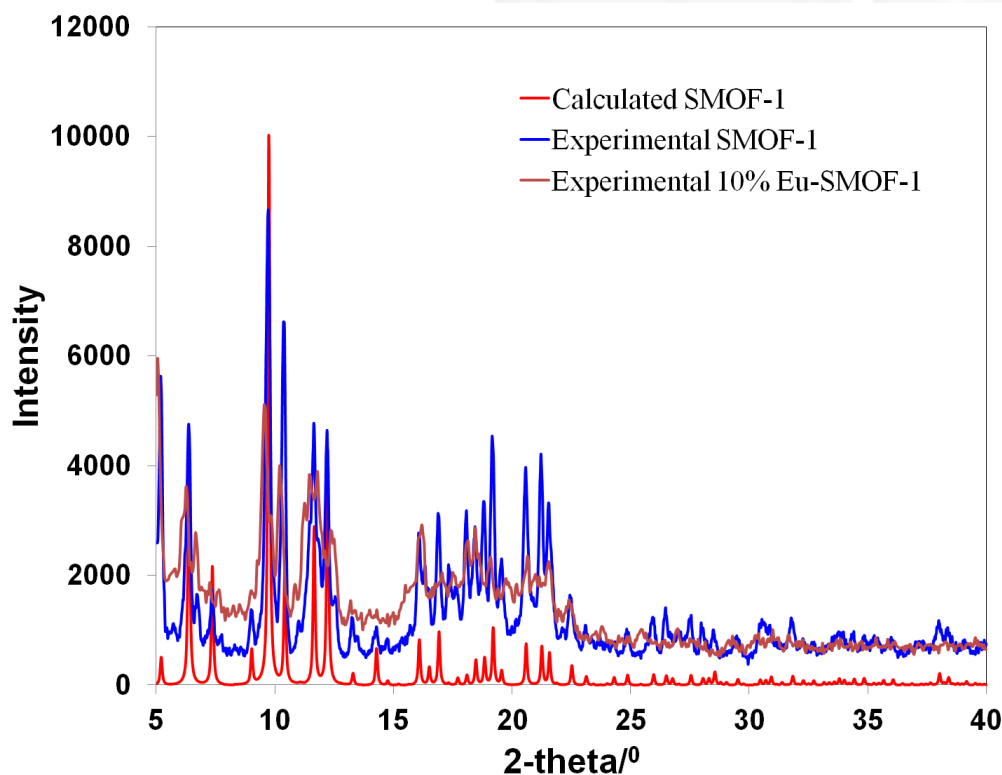
➤ The best values of CRI and CCT fall *along or below* the dashed line in the plot

By increasing the  $\text{Eu}^{3+}$  concentration to 10%, the CRI and CCT shift closer to the set target of CRI~90 and CCT~3200K



# Successful in-framework Eu co-doping at 2.5, 5, and 10%

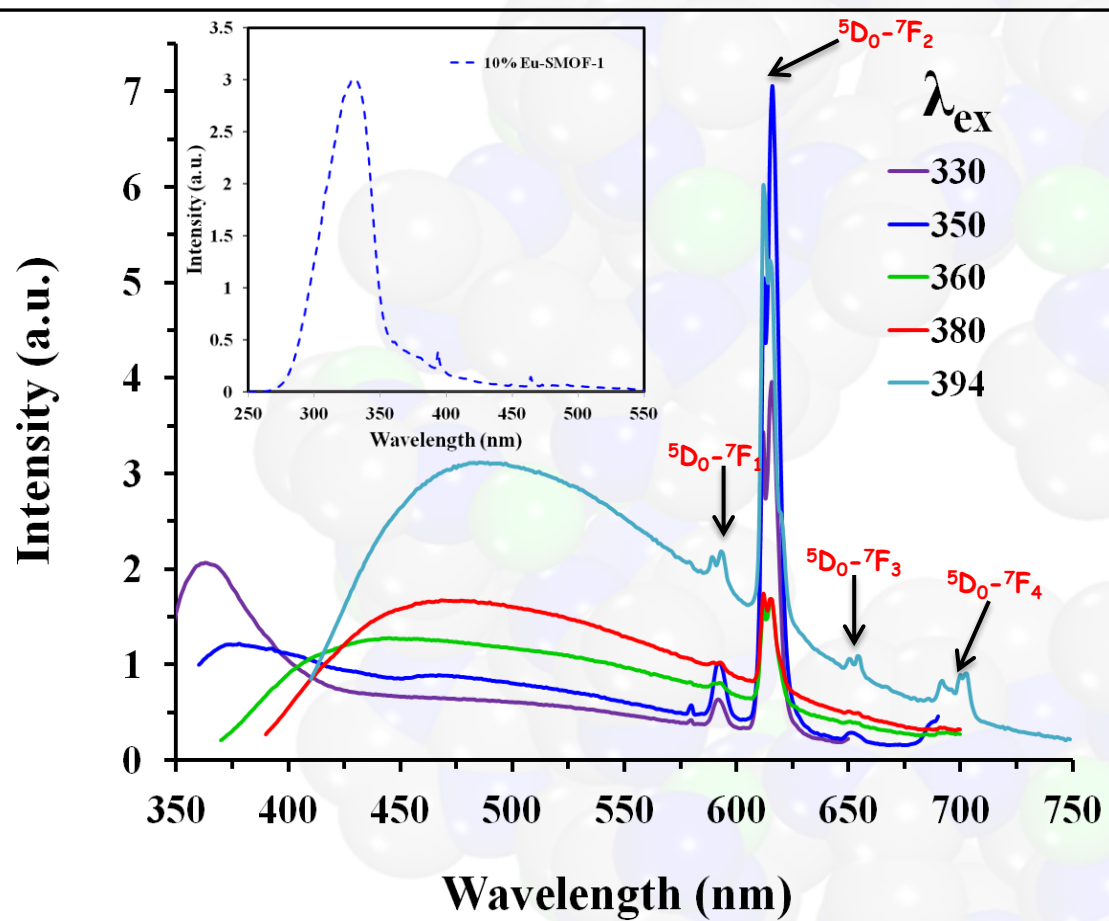
## SEM-EDS



Unit cell refinement of the 10% Eu-doped SMOF-1 sample reveals *enlarged unit cell parameters*  $a=34.57(6) \text{ \AA}$ , compared to  $a=33.975(3) \text{ \AA}$ .



# Enhanced system tunability: improved color properties with in framework 10% Eu co-doping



## Color properties in 10% Eu-doped SMOF-1

$\lambda_{ex}$	CRI	CCT (K)	x	y
350	63	3606	0.369	0.301
360	81	7068	0.309	0.298
380	93	8695	0.285	0.309
394	93	6839	0.304	0.343

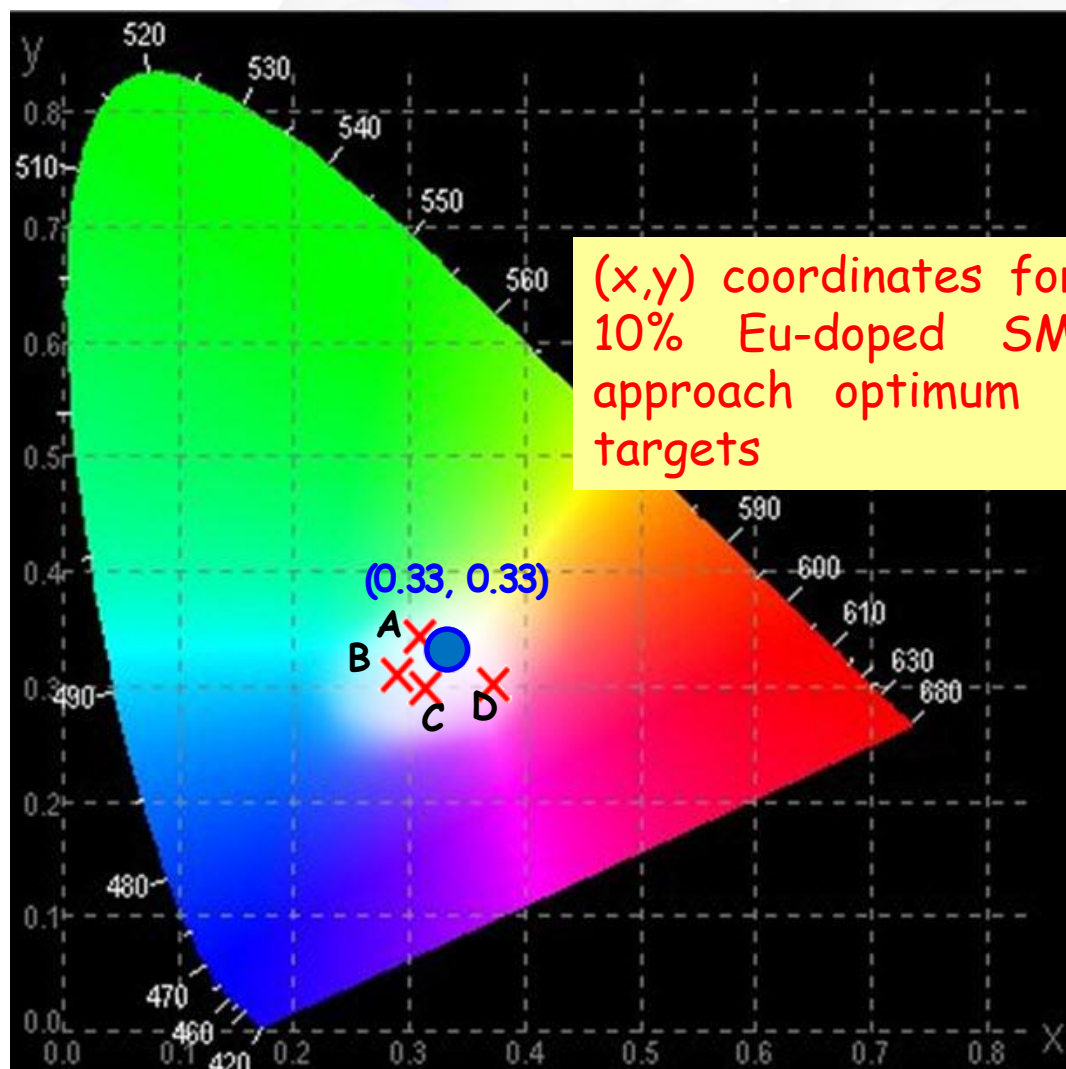
✓ CCT values are significantly improved

Narrowband emission peaks observed from the  $\text{Eu}^{3+}$  component, attributed to the parity forbidden  $^5D \rightarrow ^7F$  transitions

Absolute QY  $\sim 4.3\%$  at 330nm



# CIE\* optimum white-light chromaticity coordinates: (0.33, 0.33)



\*CIE: Commission  
Internationale de l'Eclairage

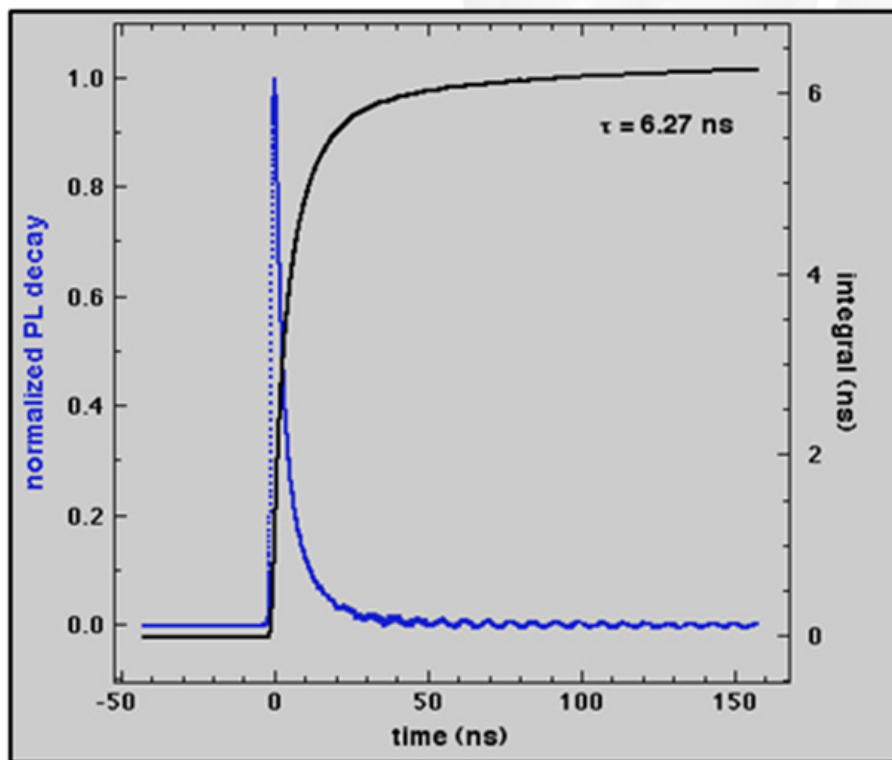
(x,y) coordinates for the  
10% Eu-doped SMOF-1  
approach optimum CIE  
targets

A-D: (x, y) chromaticity coordinates  
for  $\lambda_{ex}$  = 394, 380, 360, 350 nm.

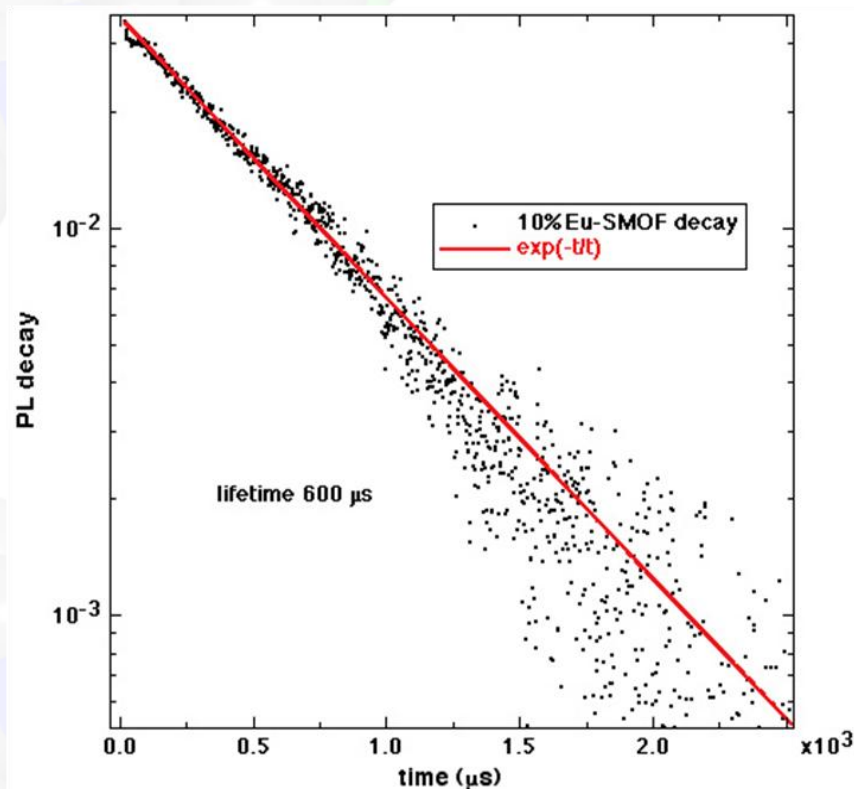




# Framework tunability is reflected in the PL lifetime



SMOF-1 harmonic  
average lifetime 6.27 ns

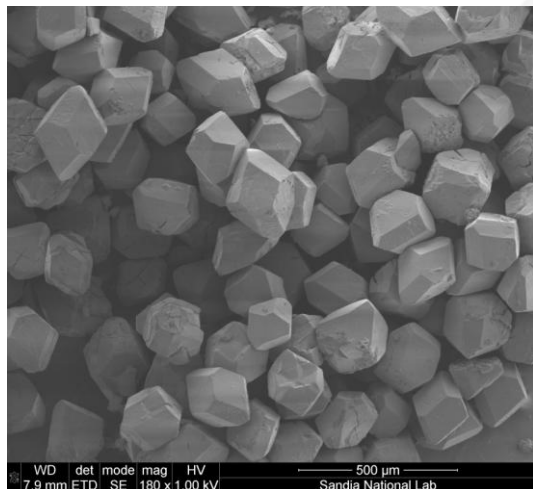


10% doped Eu-SMOF-1 harmonic  
average lifetime 600  $\mu$ s

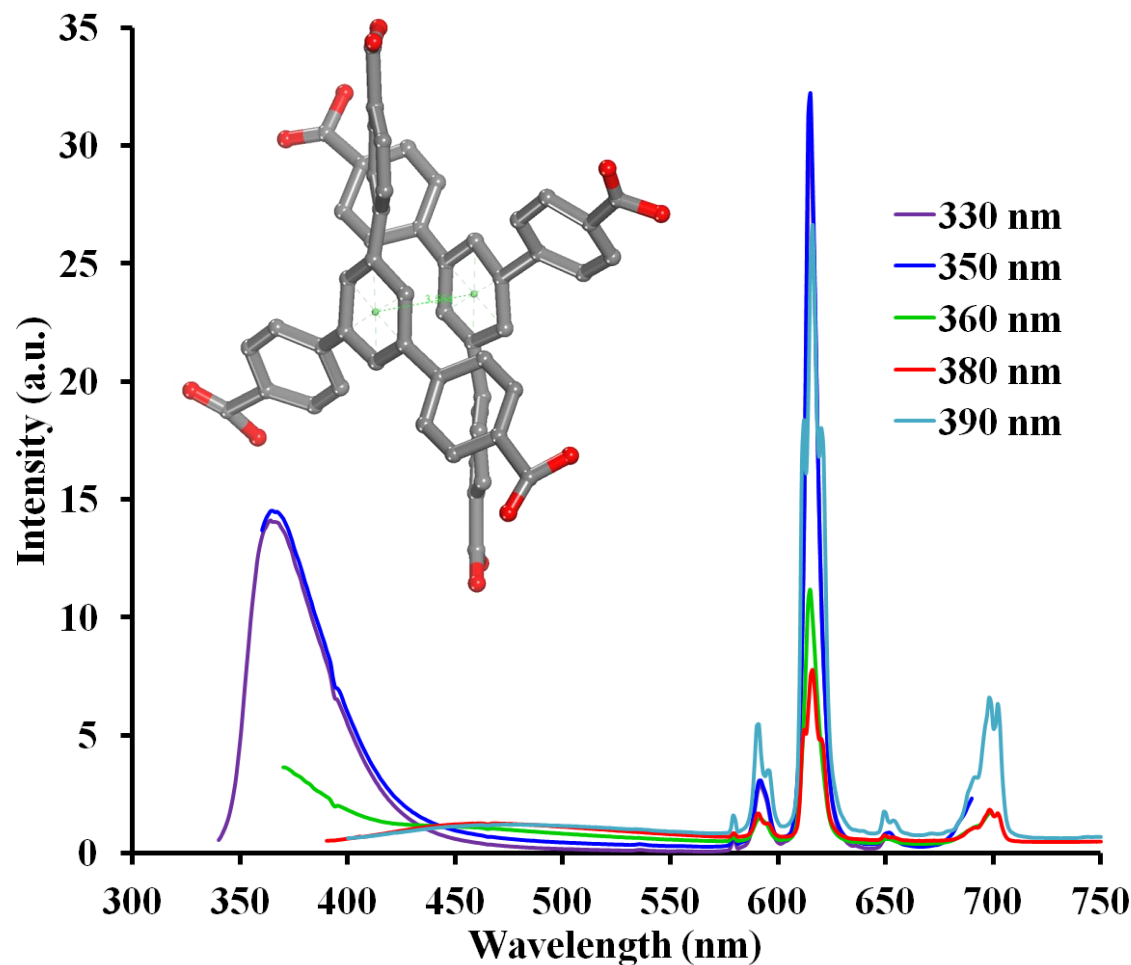


## On-going studies

# SMOF-2: a novel In-30% Eu- BTB framework red emitter



$a = b = 43.671 \text{ \AA}$   
 $c = 41.867 \text{ \AA}$   
 $\alpha = \beta = 90$   
 $\gamma = 120$   
Volume =  $69149.5 \text{ \AA}^3$



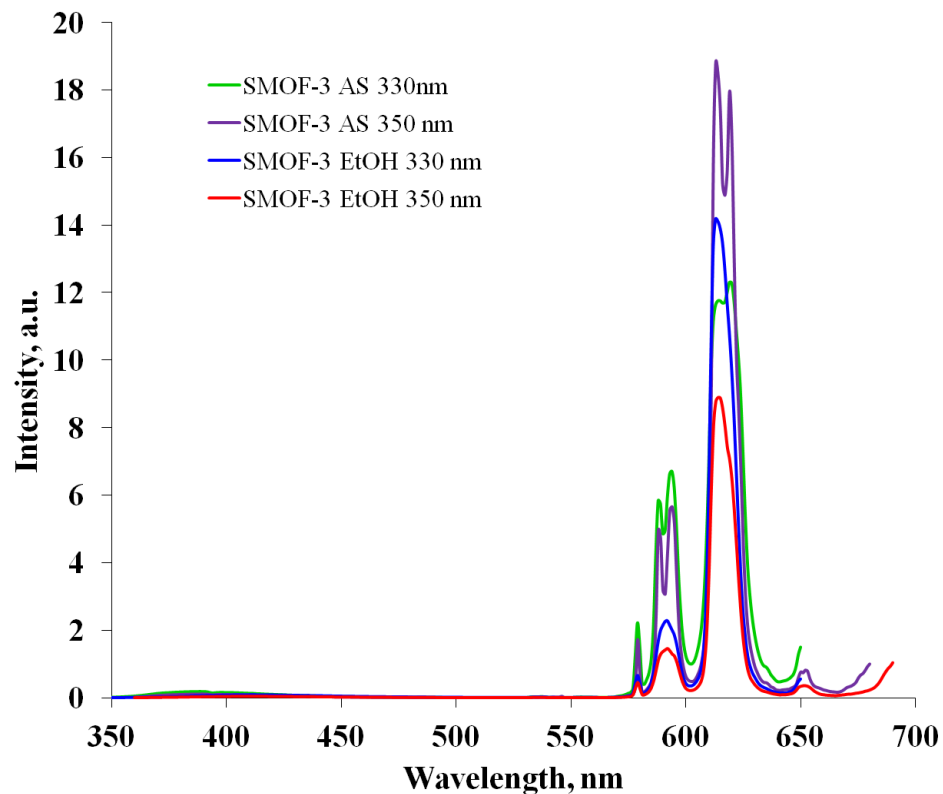
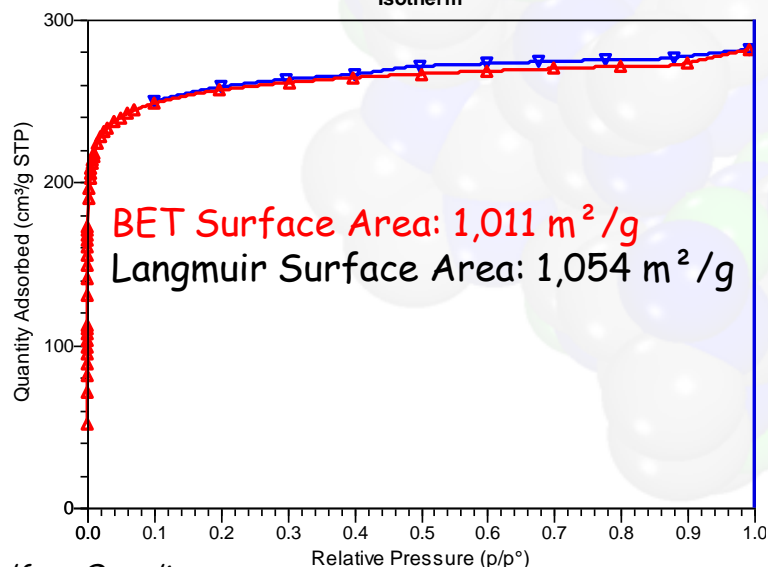
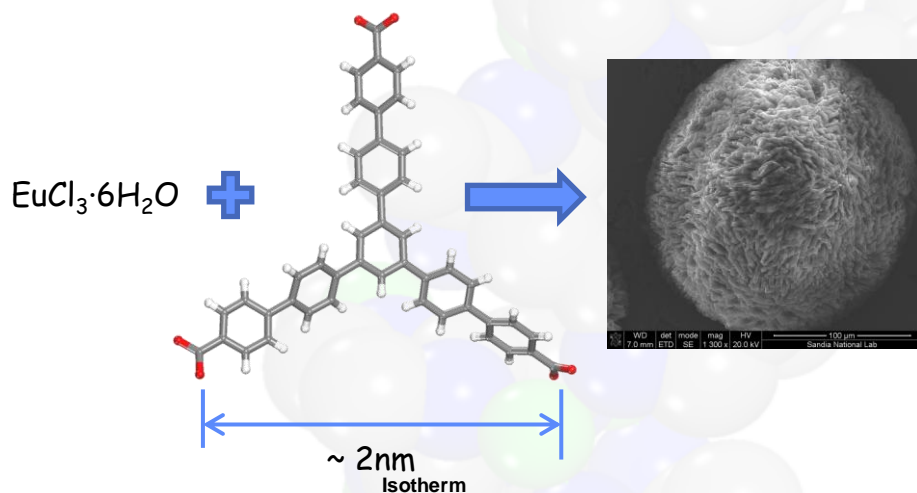
Absolute QY increases to  $\sim 11.2 \%$



# SMOF-3: a novel Eu-TCBB framework red emitter

## SMOF-3:

> 115 wt% I<sub>2</sub> sorption ability (non-optimized)



Absolute QY ~ 21 %



# Summary and On-going studies

- MOFs modularity and tunability allows for a variety of energy related applications
- Introduced MOFs in nuclear waste remediation and demonstrated that ZIF-8 is an appropriate candidate for  $I_2$  *capture and interim storage*
- $I_2$  is retained in ZIF-8 under amorphizing conditions providing for *consolidated secure interim storage* before incorporation into a long term waste form, ensuring non-contamination of the environment
- Developed a prototype novel MOF material featuring intrinsic broadband direct white light emission and derived a tunable platform which allows for the modifications of associated color properties
- Incorporation of MD simulations and GCMC modeling with structure analysis to study
  - competitive gas sorption from industrially relevant complex streams
  - establish binding locations, mechanisms of sorption and transport in HKUST-1 vs. ZIF-8
- On-going studies into novel MOFs , as well as pre- or post-synthesis framework functionalization for enhanced absolute quantum yields

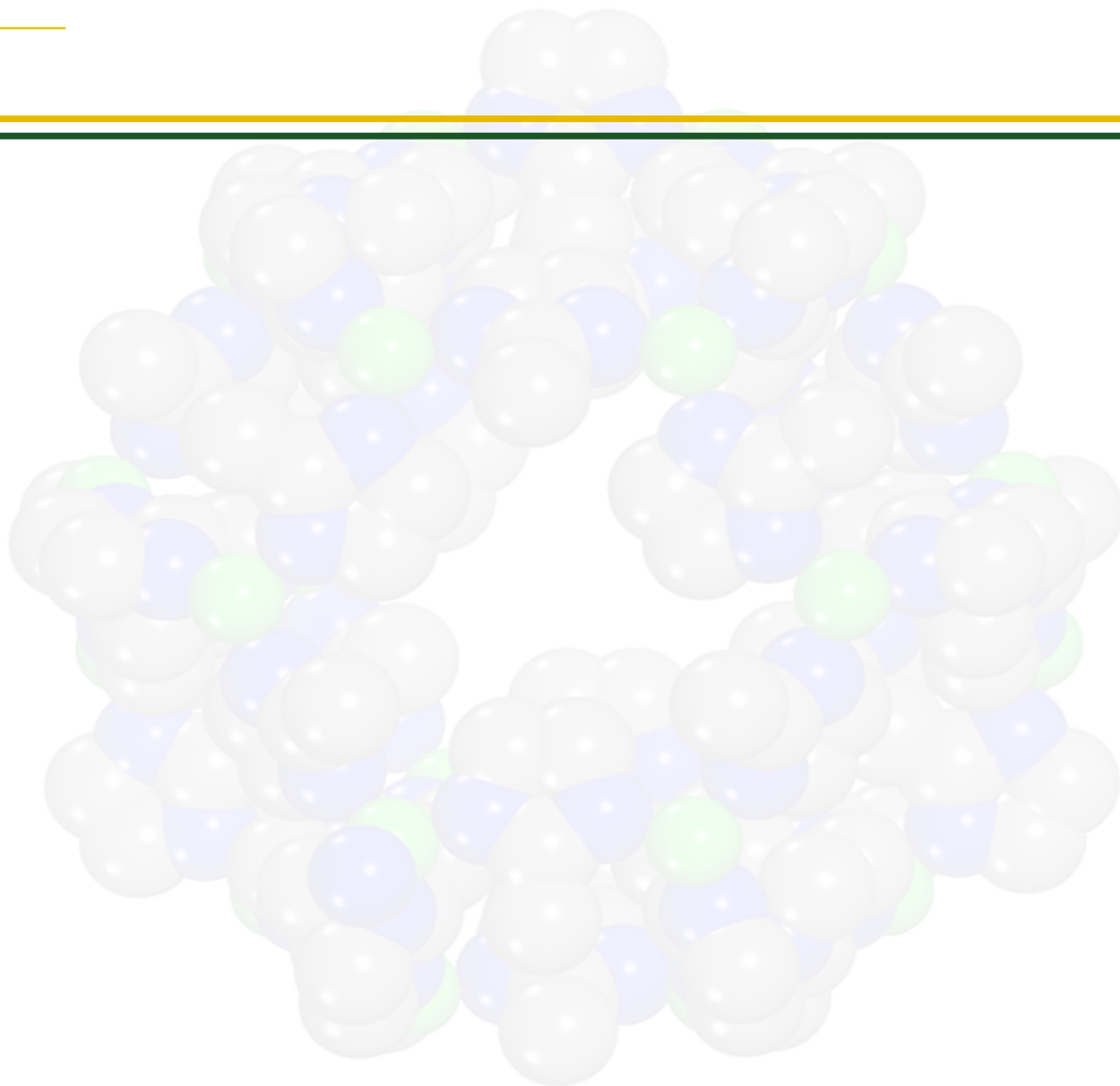




U.S. DEPARTMENT OF  
**ENERGY**

Nuclear Energy

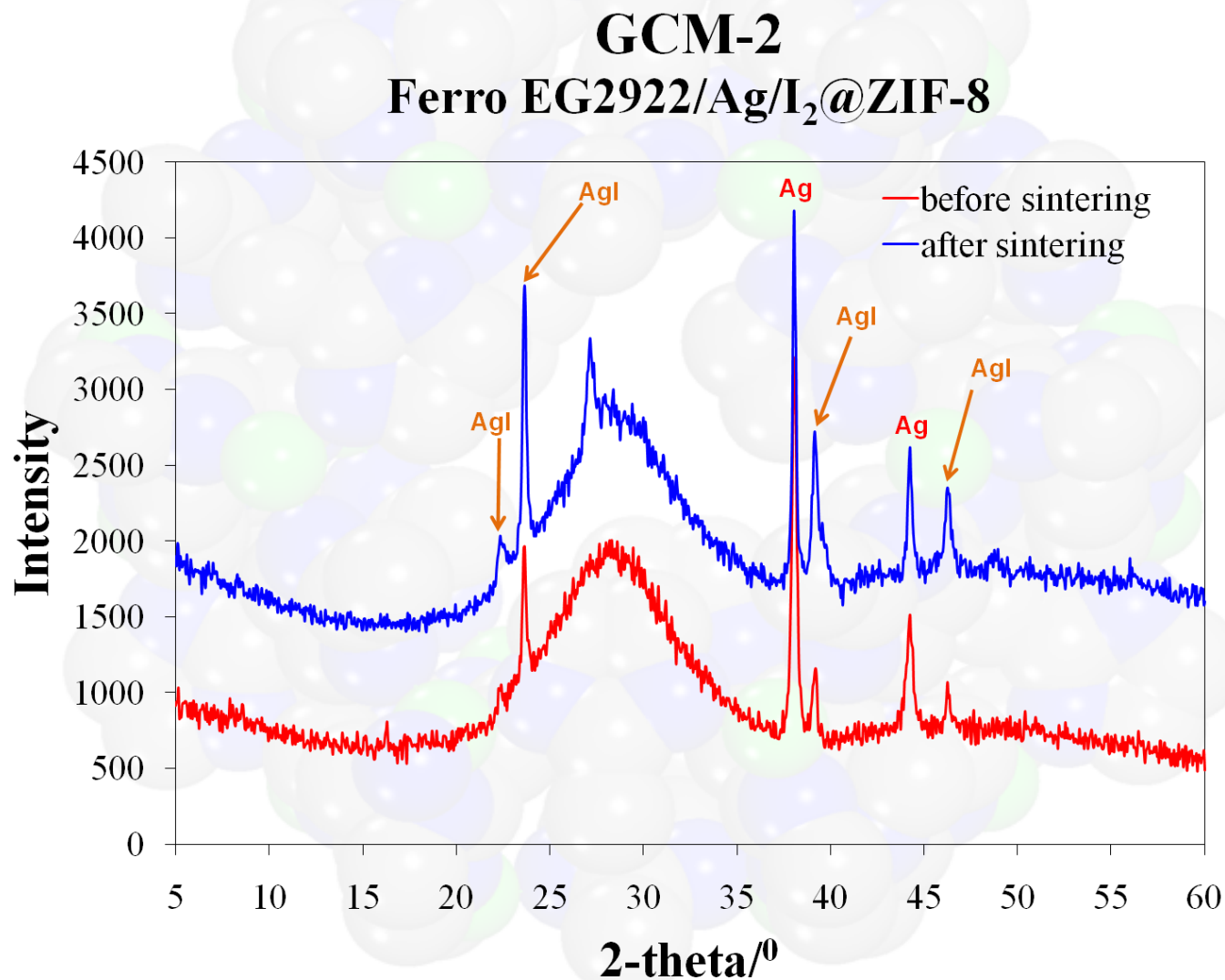
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Sandia  
National  
Laboratories



# Powder X-ray diffraction pattern for GCM-2 before and after thermal treatment



The figure displays X-ray diffraction (XRD) patterns for BiOI and Bi<sub>12</sub>SiO<sub>20</sub> samples before and after sintering. The x-axis represents the diffraction angle 2-theta in degrees (°), ranging from 5 to 60. The y-axis represents the XRD intensity, ranging from 0 to 8000. The legend indicates that the red line represents the sample 'before sintering' and the blue line represents the sample 'after sintering'. Orange diamonds represent BiOI and yellow triangles represent Bi<sub>12</sub>SiO<sub>20</sub>. The BiOI peaks are marked with orange diamonds and arrows, while the Bi<sub>12</sub>SiO<sub>20</sub> peaks are marked with yellow triangles and arrows. The 'after sintering' patterns show significantly higher intensity peaks compared to the 'before sintering' patterns, indicating improved crystallinity.



# Bi-Zn oxide low-temperature sintering glasses



80 wt.% glass, 10 wt.%  
I<sub>2</sub>@MOF, 10 wt% Ag  
(non-optimized compositions)

Glass characteristics	*EG2998	*EG2922
Composition	Bi-Zn-B	Bi-Zn-Si
Sintering temperature	500°C	525°C-550°C
Crystallinity	Crystallizing	Vitreous
Density	5.65 g cc <sup>-1</sup>	5.8 g cc <sup>-1</sup>

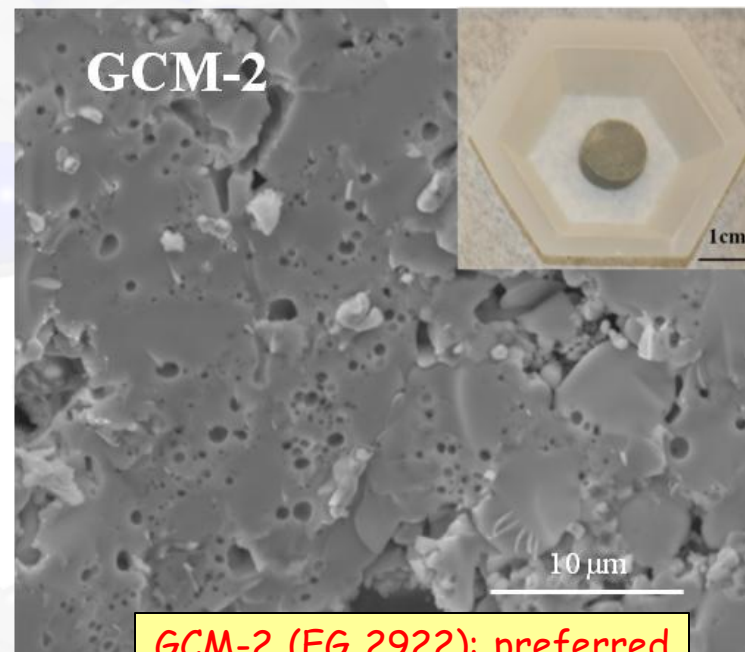
Glass	ZnO		Bi <sub>2</sub> O <sub>3</sub>		Al <sub>2</sub> O <sub>3</sub>		B <sub>2</sub> O <sub>3</sub>		SiO <sub>2</sub>	
	mole %	wt. %	mole %	wt. %	mole %	wt. %	mole %	wt. %	mole %	wt. %
EG 2922	14.2	7.8	20.2	63.4	7.8	5.4			57.8	23.4
EG 2998	49.7	26.9	18.9	58.6			31.3	14.5		







# $I_2$ @ZIF-8 successfully encapsulated in stable long-term waste forms

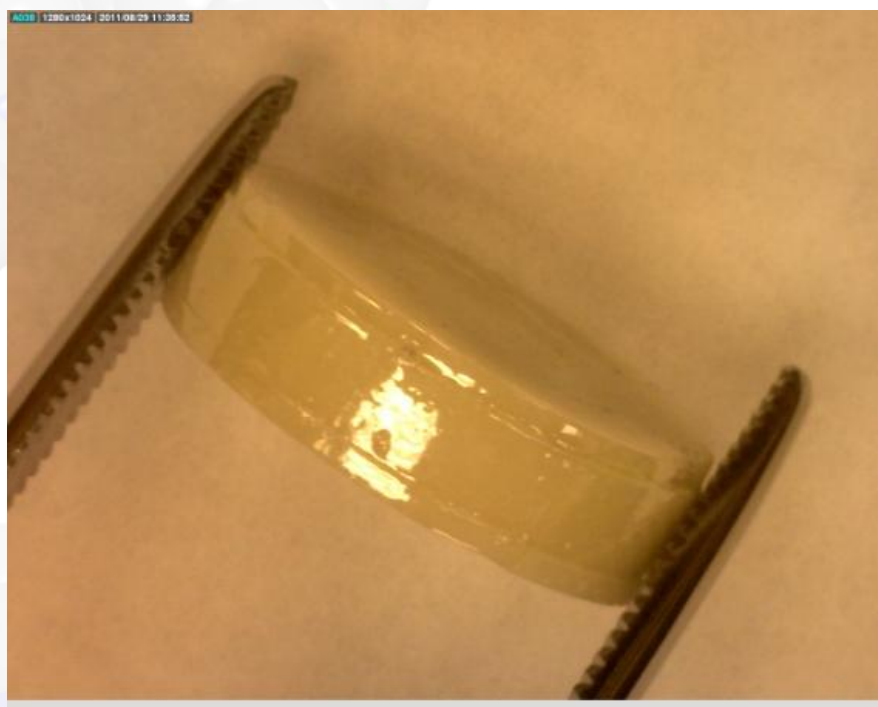


GCM-2 (EG 2922): preferred waste form based on higher leach durability studies

GCM reveal excellent thermal & chemical stability and are appropriate for long term storage: little to no  $I_2$  is lost during processing or after undergoing leach durability studies



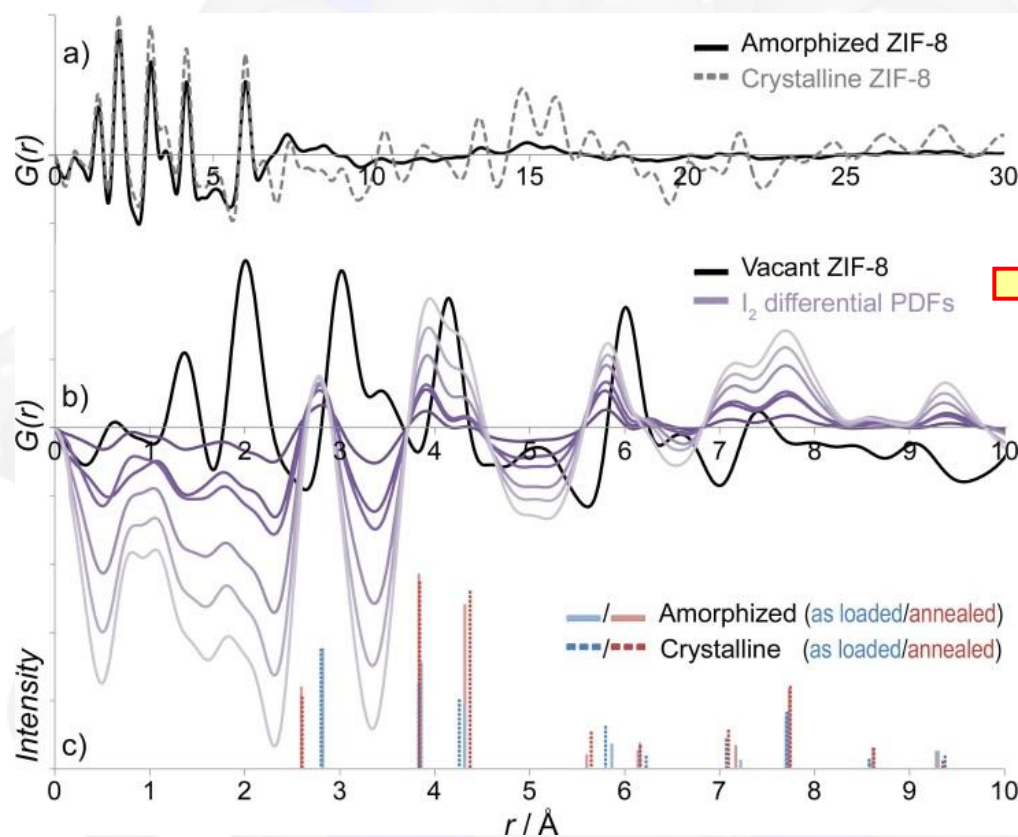
# Successful sintering of core-shell structures containing amorphized $I_2$ @ZIF-8 pellets



Compact monoliths comprise high-capacity consolidated radiological waste content for extended storage (1 pellet); scale up to date: 5 pellets per shell



# The short-range I $\cdots$ I and I $\cdots$ framework interactions remain unchanged upon amorphization



The framework retains  $I_2$  after pressure treatment

High resolution synchrotron PDFs and differential PDFs collected at APS/ANL for the crystalline and amorphous materials