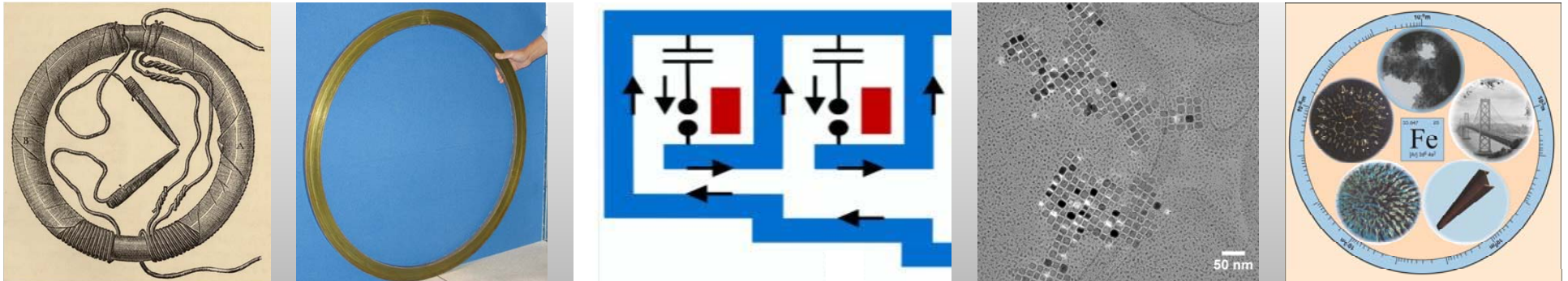


Exceptional service in the national interest



Nanocomposites for LTD Magnetic Cores

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Current State of the Art



- Wound composite with Metglas layers alternating with Mylar
- Parasitic current within an LTD cavity reduces the current delivered to the load. Presently, the magnitude of the parasitic current is $\sim 10\%$ of the load current.
- Present core materials have hysteresis, which means each core throughout the entire accelerator would need to be re-set after every shot using a low-amplitude long current pulse.
- At current manufacturing capacity it would take decades to fabricate the cores required for a 50-MA accelerator.

Materials for Magnetic Core Inductors



- Desired Properties of Magnetic Cores
 - High susceptibility
 - High saturation magnetization
 - Low/no hysteresis
- Current material of choice is wound amorphous iron alloy tape
 - Iron for high magnetization
 - Amorphous to decrease hysteresis
 - But, amorphous materials have depressed magnetization
- Iron nanoparticles provide an opportunity to improve all of the critical properties
 - Iron has the highest saturation magnetization of any element
 - Nanoparticles have enhanced susceptibility
 - Even crystalline iron nanoparticles can be hysteresis free
 - *Superparamagnetic nanoparticles have properties that are not possible in bulk materials*

<u>Saturation Magnetizations (Tesla)</u>	
Fe	2.15
SiFe alloy	1.97
Metglas 2605	1.56
SNL Iron nanoparticles	2.04

Enabling Science

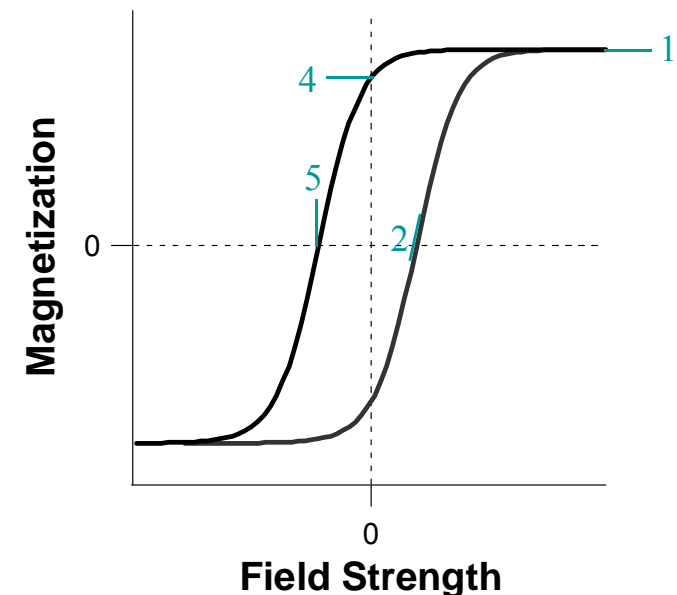
- Chemistry
 - Current approaches are not scalable
 - Developing new methods, one has been patented, but others can be once they are reduced to practice (eg continuous microwave synthesis)
 - Minimize cost through use on inexpensive reagents, processes.
 - Develop appropriate matrix with maximum loading and good mechanical properties—we have done this for high refractive index nanocomposites.
- Physics
 - Only the physics of ideal magnetic nanoparticles are well understood. Ideal implies isolated particles in non-interacting matrix
 - Real systems are non-ideal so we must understand:
 - Particle-matrix interactions (matrix can enhance or decrease magnetism)
 - Multi-particle dipolar interactions
 - Formation of structured, anisotropic materials

Project Plan

- Determine optimum size for the iron nanoparticles
 - We expect ~20 nm will be ideal
- Optimize particle loading
 - Loading too high may yield hysteresis
 - Low loading decreases magnetization
 - Develop a material with good flow properties before curing to allow molds to be poured. (Monson and Huber patent application relating to TiO₂)
- Create small scale test structures
 - We can make and test small parts in realistic configurations.
- Develop a detailed plan for scale up and tech transfer
 - A number of methods can be used for large scale synthesis of nanoparticles
 - Microwave synthesis
 - Continuous reaction (Huber patent recently issued)

Terminology in Magnetic Measurements

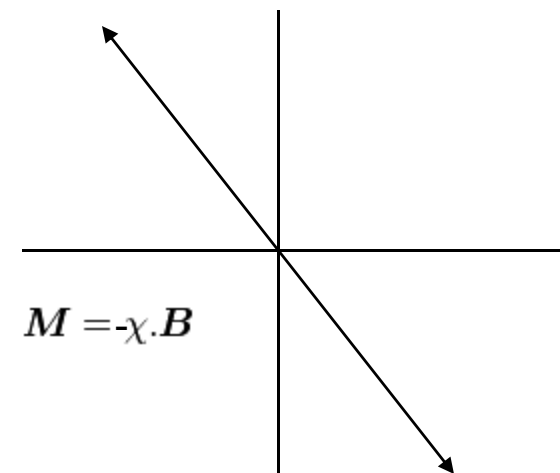
1. Saturation magnetization, maximum magnetization for a material (M_{sat} , M_s , σ_s)
2. Susceptibility, ease of magnetization, slope of magnetization curve (χ , χ_i)
3. Coercivity, amount of hysteresis.
4. Remanence, remaining magnetization when the external field is removed.
5. Coercive Field, the reverse field required to get to 0 magnetization. (H_c)



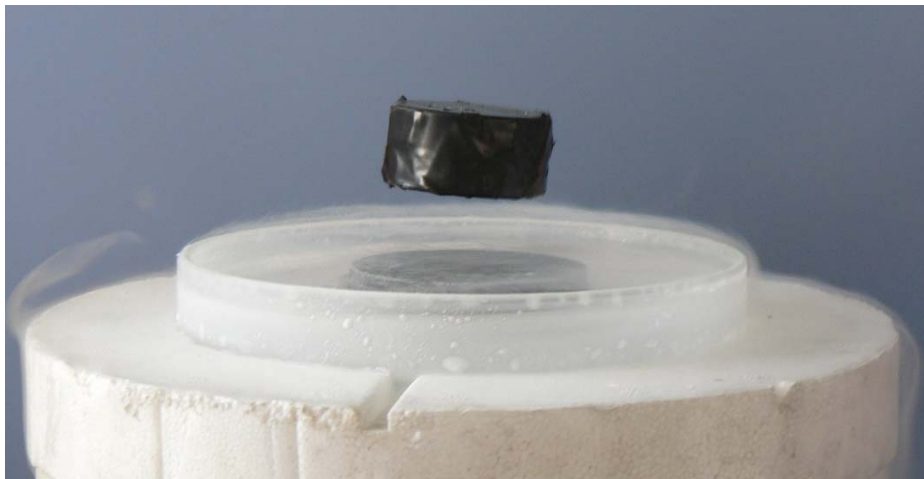
Diamagnetism

- Weakly repulsive force
- Caused by the repulsion between a magnetic field and moving electrons.
- Everything has diamagnetism, not everything is diamagnetic.
- Susceptibility of about -10^{-6}
- No saturation
- No coercivity
- No temperature dependence
- Examples: water, gold, element & compounds with full valence shells.
- Superconductors are “superdiamagnetic” with a susceptibility of -1

$M \propto B$
For all fields

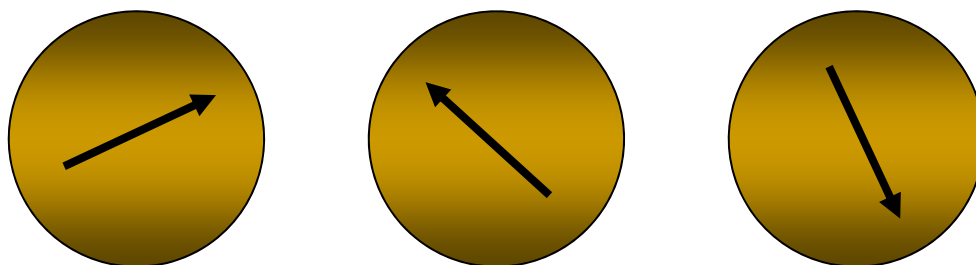


Diamagnetism Demonstrations

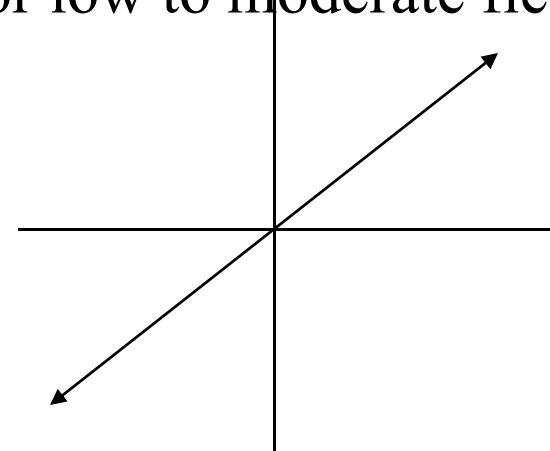


Paramagnetism

- Weakly attractive, susceptibility of $<10^{-3}$
- Caused by attraction between magnetic field and unpaired electrons.
- Materials are “paramagnetic” when their paramagnetism is greater than their diamagnetism.
- Temperature dependant, higher T, lower χ
- “Strongly” paramagnetic materials include most of the d- and f-block elements and ions



$M \propto B$
For low to moderate fields



Curie's law

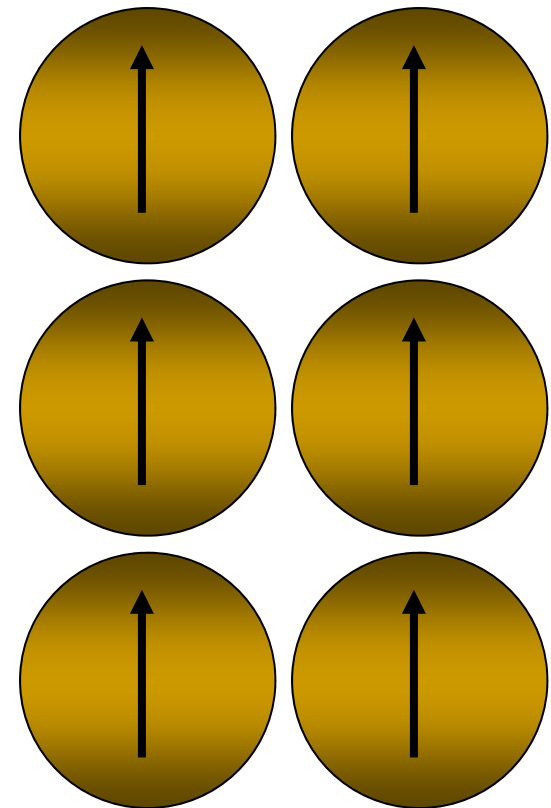
$$M = \chi \cdot B = C \cdot \frac{B}{T}$$

T in K

Material specific Curie constant. This relation holds only when far from saturation

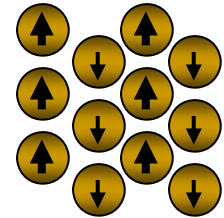
Ferromagnetism

- Paramagnetic atoms acting collectively with spins reinforcing
- Strongly attractive, susceptibility can be as high 10^6
- These are what we typically think of a magnetic: iron, cobalt, nickel, etc

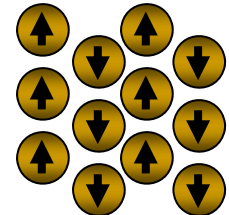


Other forms of collective magnetism

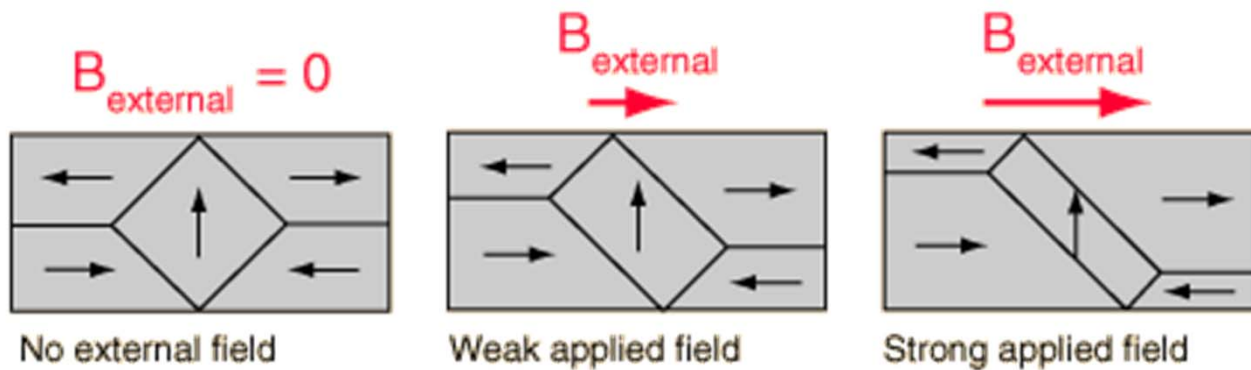
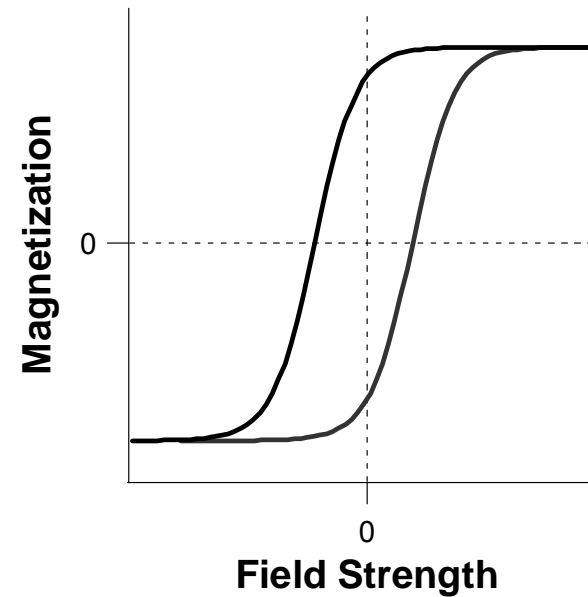
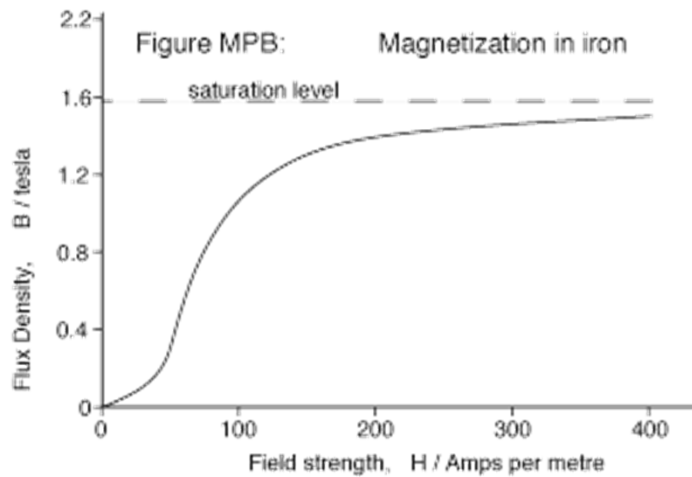
- Ferrimagnetic: magnetite and other ferrites, most magnetic metal oxides.
 - Spins opposed, but do not cancel out completely



- Antiferromagnetic: chromium
 - Spins opposed, cancel out completely
- Spin vortices and other unusual structures are present in rare earth elements.



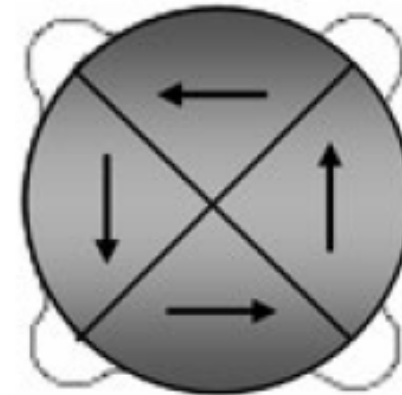
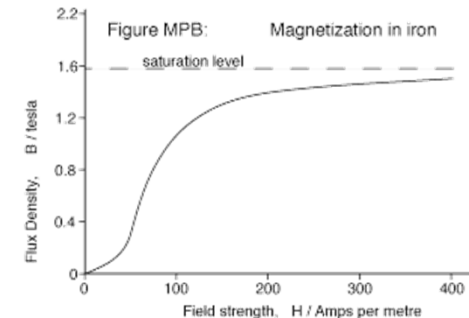
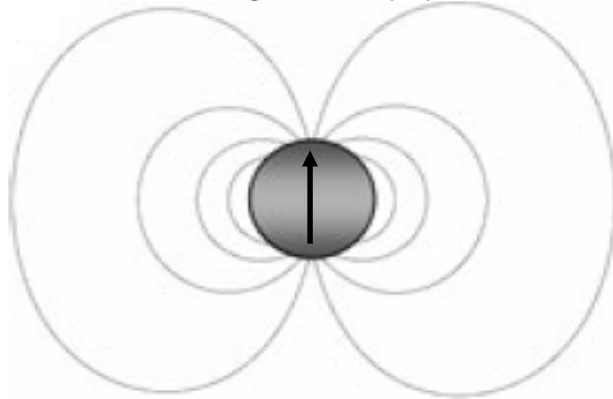
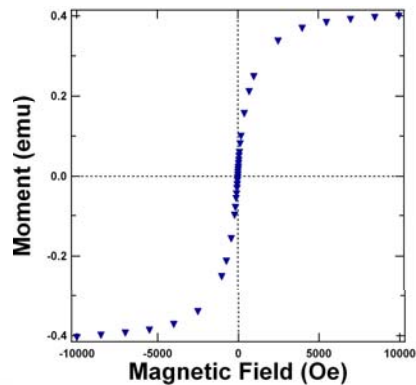
Magnetizing a Ferromagnet



The Other Form of Magnetism: Superparamagnetism

Superparamagnetic particles are single domain magnetic particles whose electron spins are aligned into a single giant moment.

Below a certain size (~40 nm diameter) the energy required to create a domain wall becomes larger than the decrease in energy the particle would experience through minimizing flux leakage.



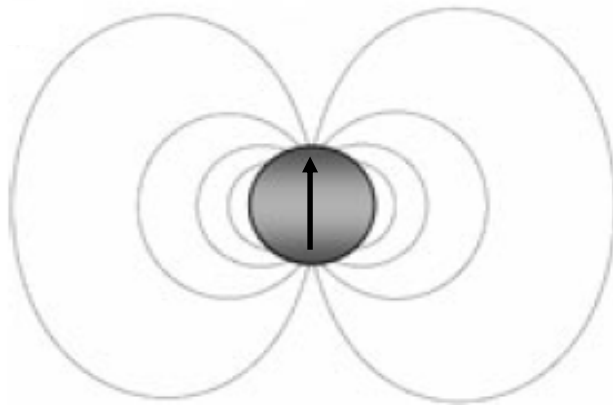
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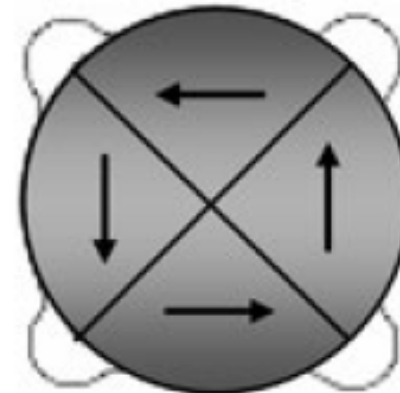
Single Domain:

- More flux leakage
- No energy spent creating domain walls



Multidomain:

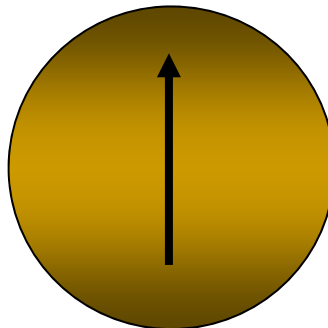
- Less flux leakage
- Energy required to create domain walls



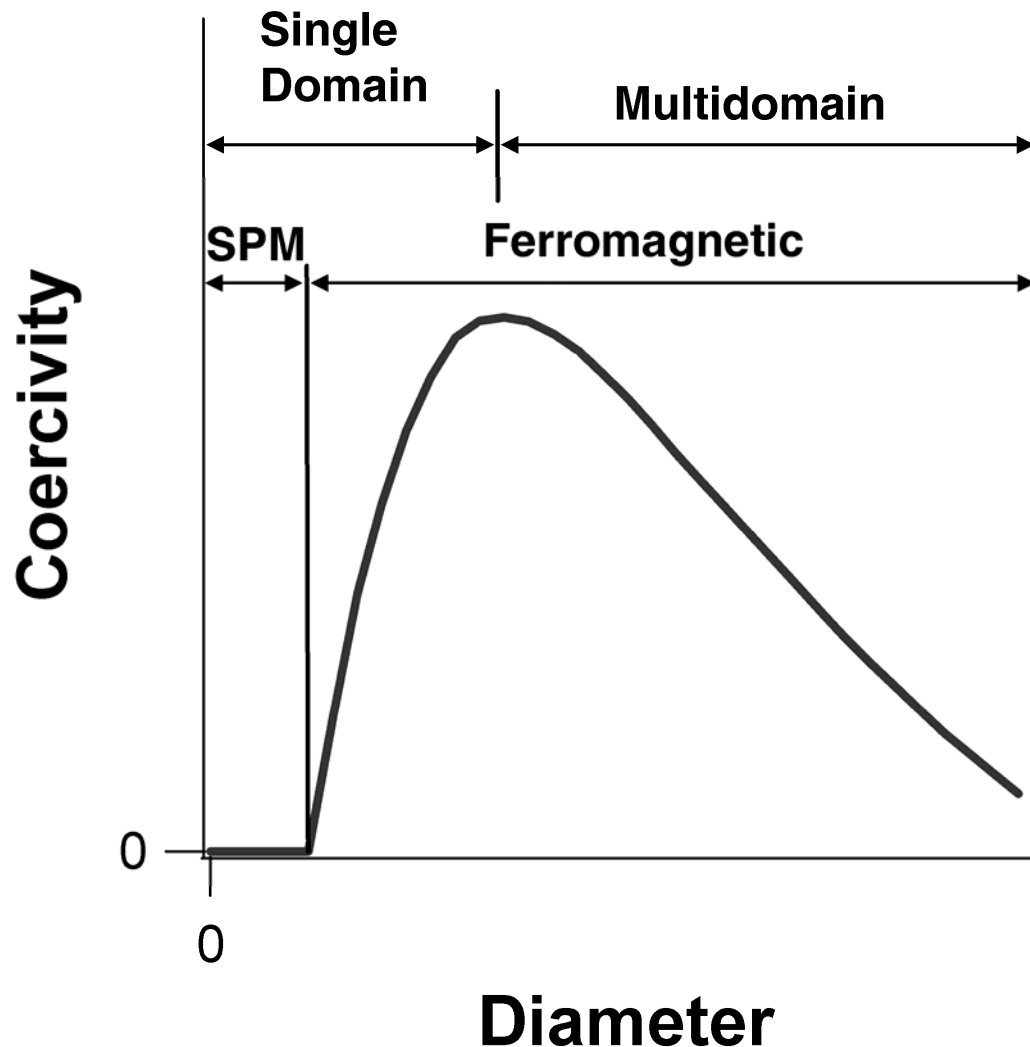
Blocking Temperature

Blocking Temperature (T_B): Point where a particle's dipole can reorient in the timescale of an experiment (100 s).

- T_B is directly dependent upon the product of the magnetocrystalline anisotropy and particle volume.
- Above the T_B particles are superparamagnetic (no hysteresis).
- Below T_B they are ferromagnetic (hysteresis, coercivity).
- Particle susceptibility (χ) has a maximum at T_B
- χ values much larger than multiple-domain particles ($\chi \leq 3$ for multiple-domain particles)

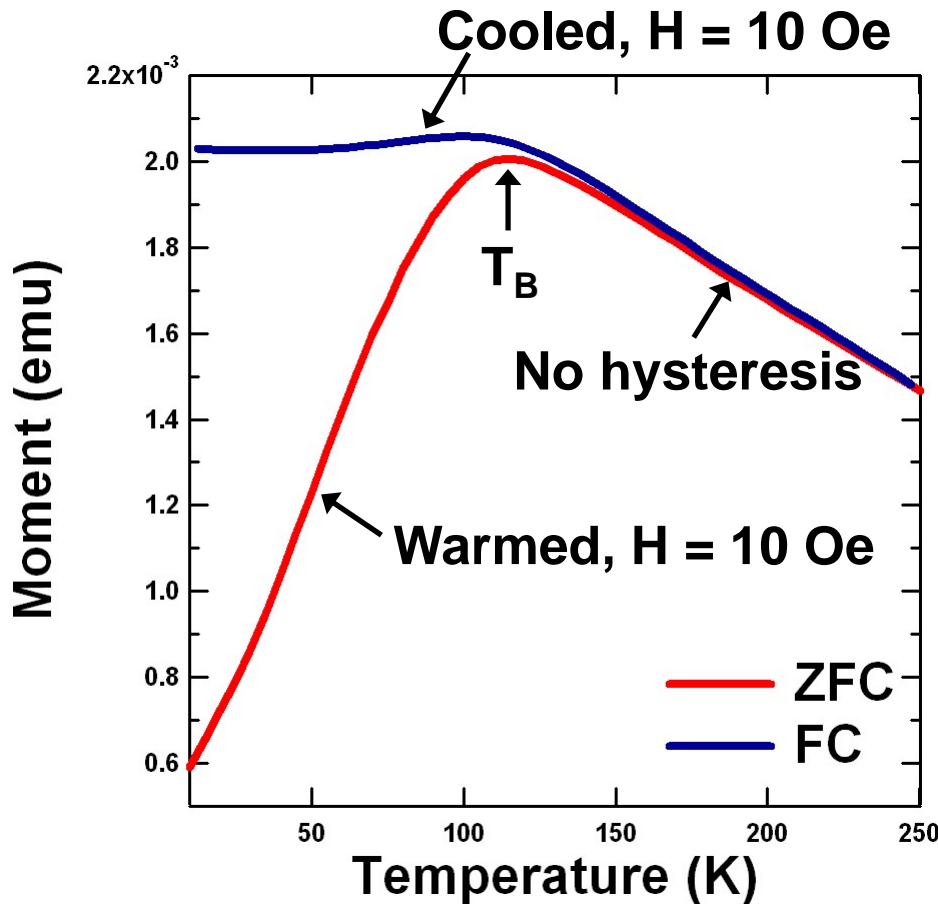


Size Dependent Behavior



- Plot is for a fixed Temperature
- Behavior is for an ensemble of magnetic nanoparticles
- Particles cease to be superparamagnetic above ~ 40 nm in diameter
- Gradual approach to bulk behavior

SQUID Magnetometry – Thermal Response



- ZFC: Zero field cooled
 - Cooled to low T while H = 0
 - Small field (H = 10 Oe) applied
 - Data collected as sample warms
- FC: Field cooled
 - Data collected while cooled to starting point
- Particles dispersed in frozen solvent, measuring Néel relaxation only (internal rotation of moment)
- T_B is maximum in ZFC data
- Double diameter, increase T_B by 8

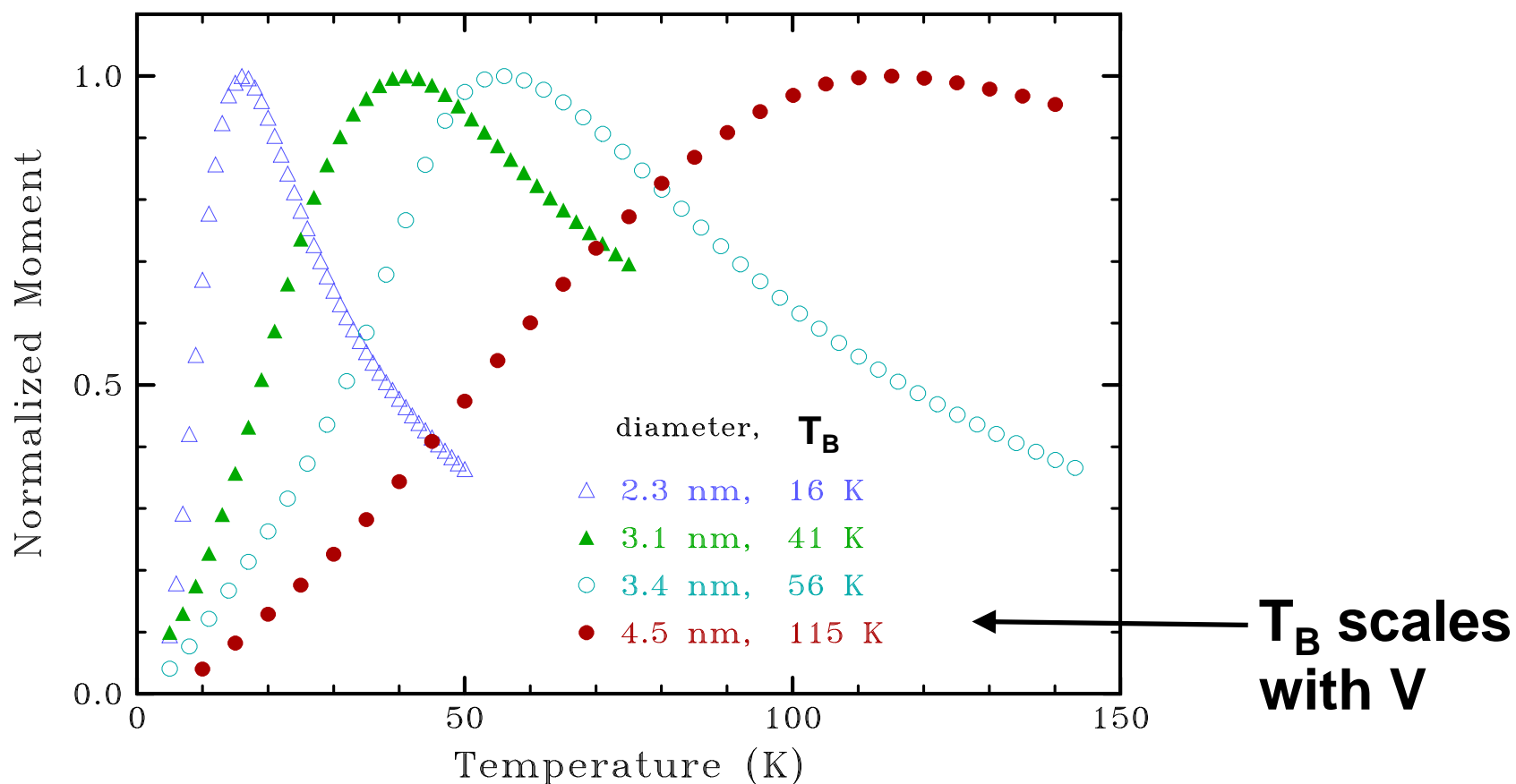
$$T_B = \frac{KV}{30k_B} \quad T_B \propto r^3$$

K = anisotropy

V = particle volume

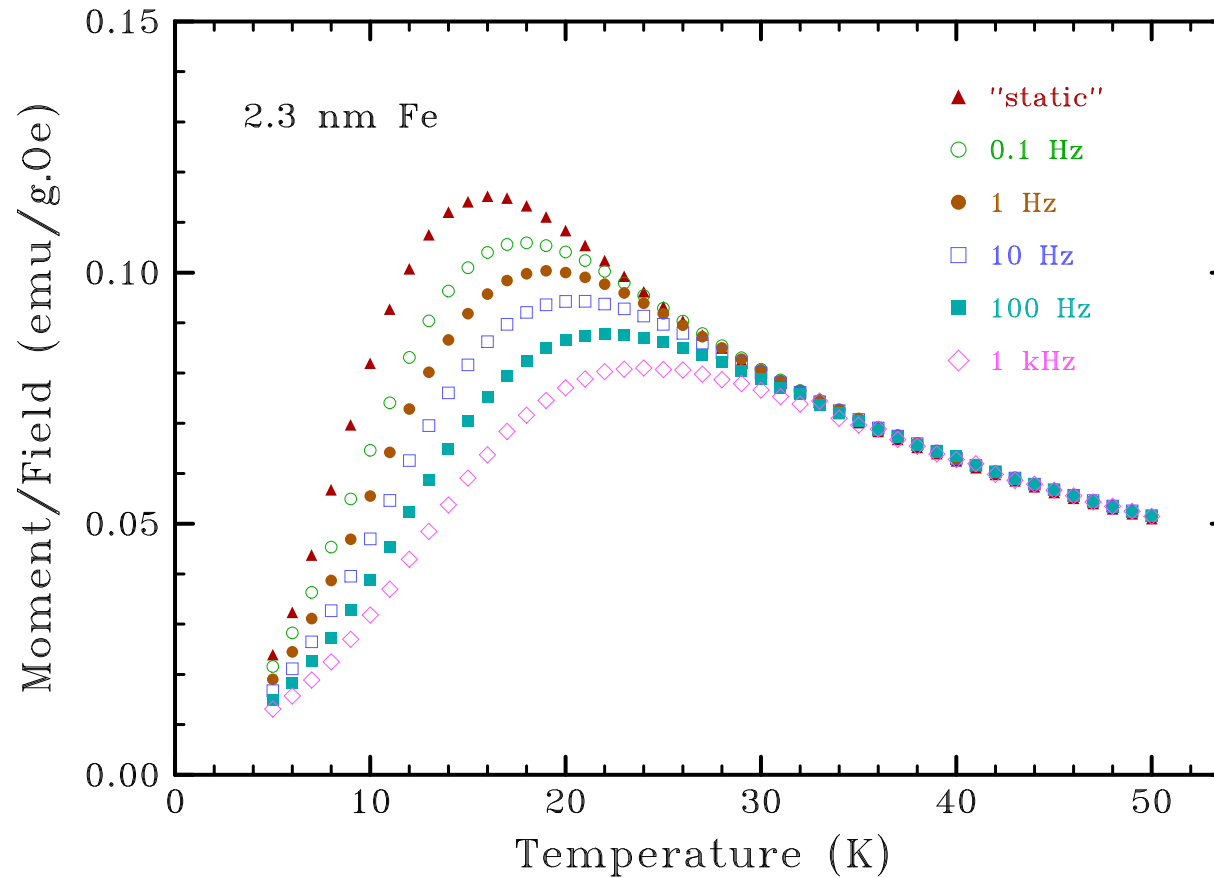
k_B = 1.3807×10⁻¹⁶ ergs/K

T_B vs. Fe Nanoparticle Diameter



- All curves ZFC
- Broad peaks for larger particles expected
- Plotting σ vs. T/T_B peaks would superimpose

AC Susceptibility



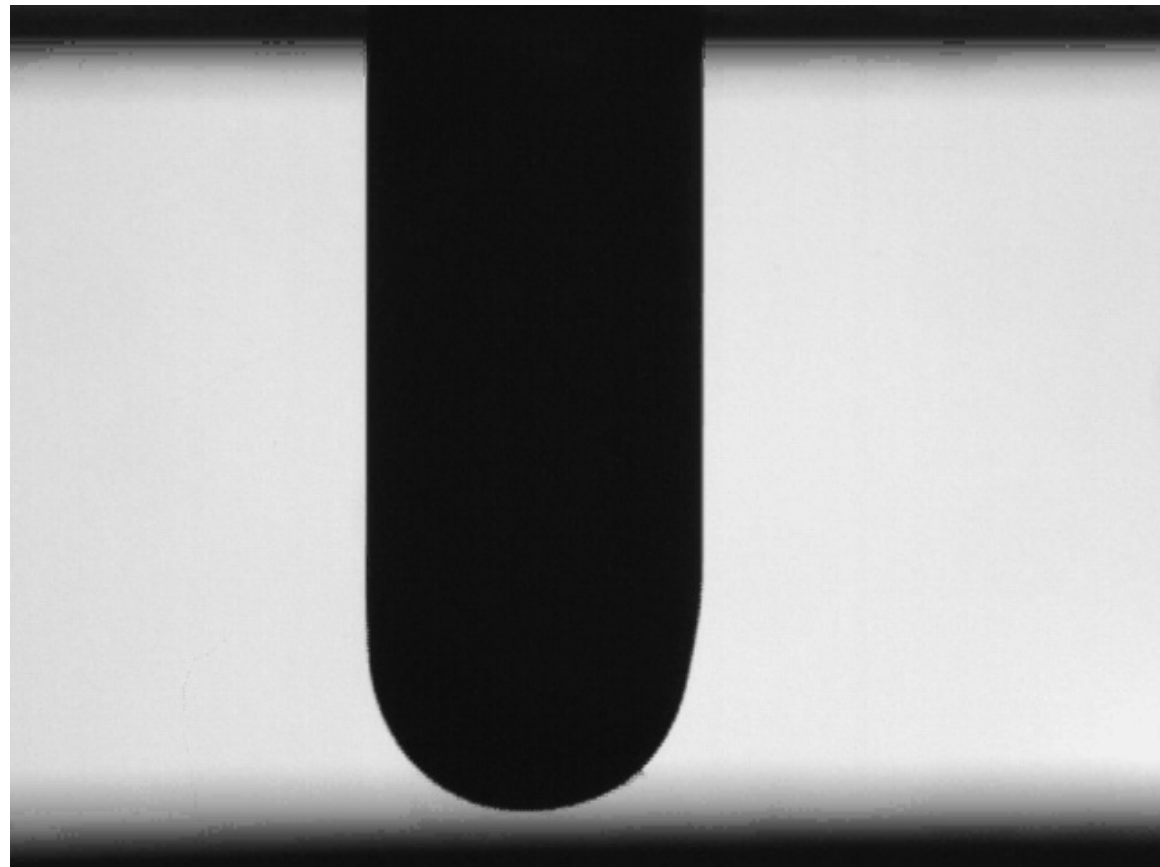
- Spin reorientation time is thermally activated
- AC measurements determine T_B vs. frequency
- "static" measurement time ~ 100 s (0.01 Hz)

Magnetically Structured Composites

Horizontal Sheets



- 3 nested Helmholtz coils allow the application of magnetic fields in x, y, and z axes
- Rotating fields created by applying 2 AC fields with 90° phase lag
- Sample held in a 5 mm glass tube
- 100 G rotating field



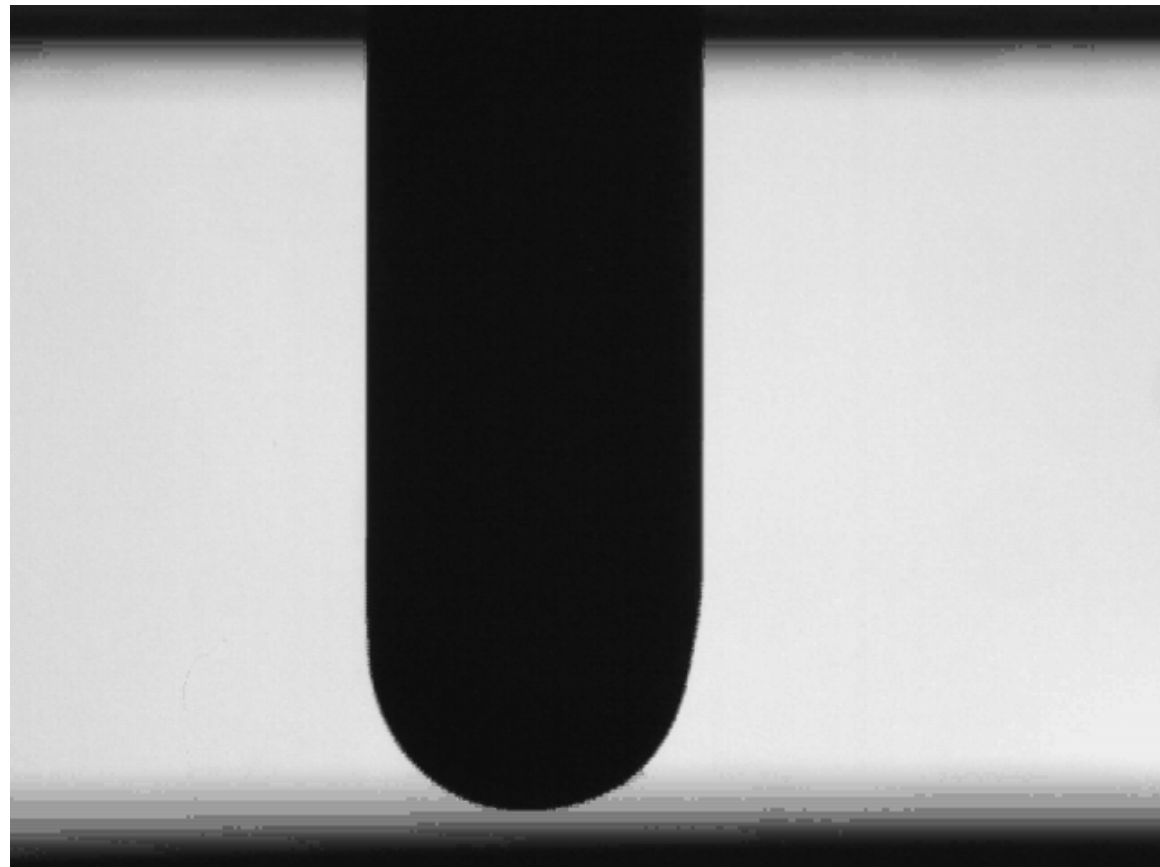
With J.E. Martin

Magnetically Structured Composites

Vertical Sheets



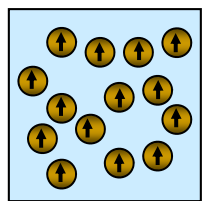
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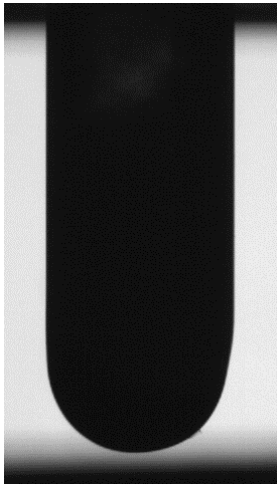
Effect of Structuring on Magnetic properties

Dispersed particles

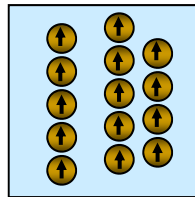


$\uparrow H$

Local field
is equal to
applied
field

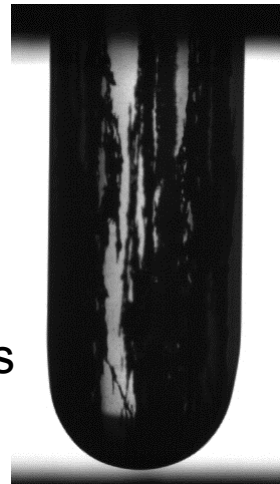


Vertical Structures

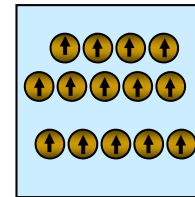


$\uparrow H$

Local fields
are
reinforced

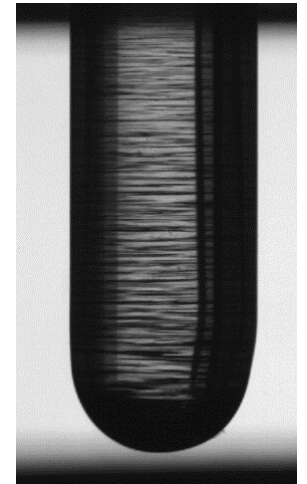


Horizontal Structures



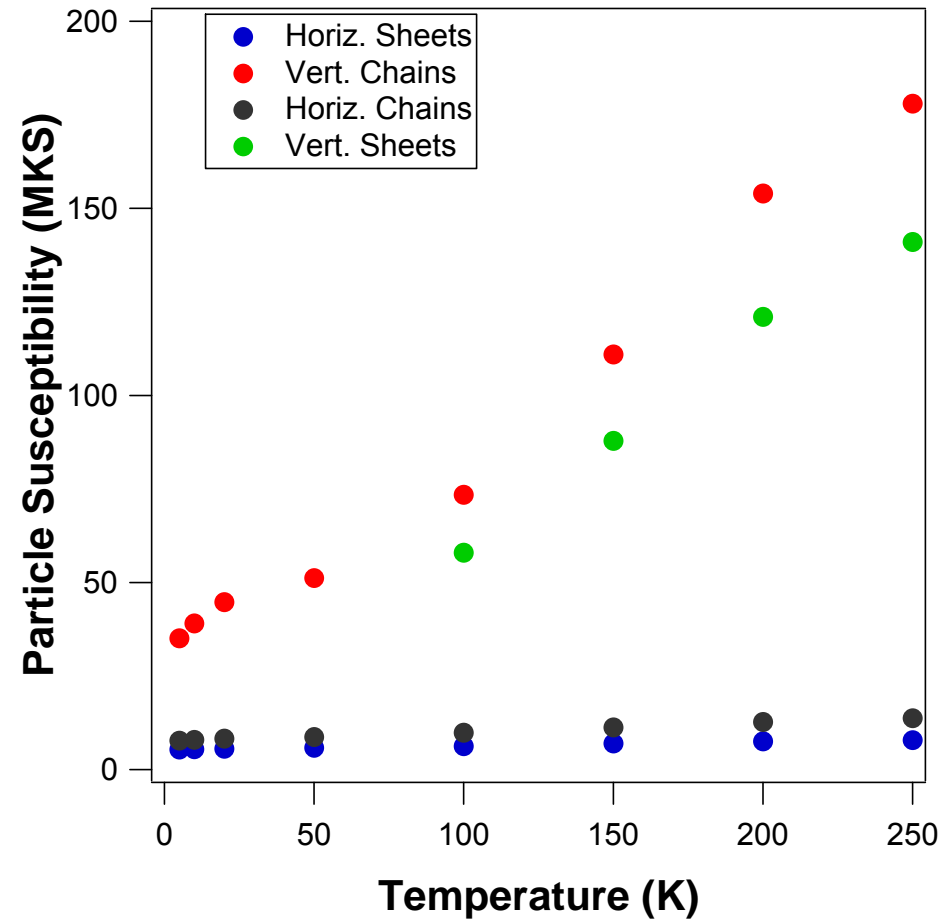
$\uparrow H$

Local fields
are
weakened



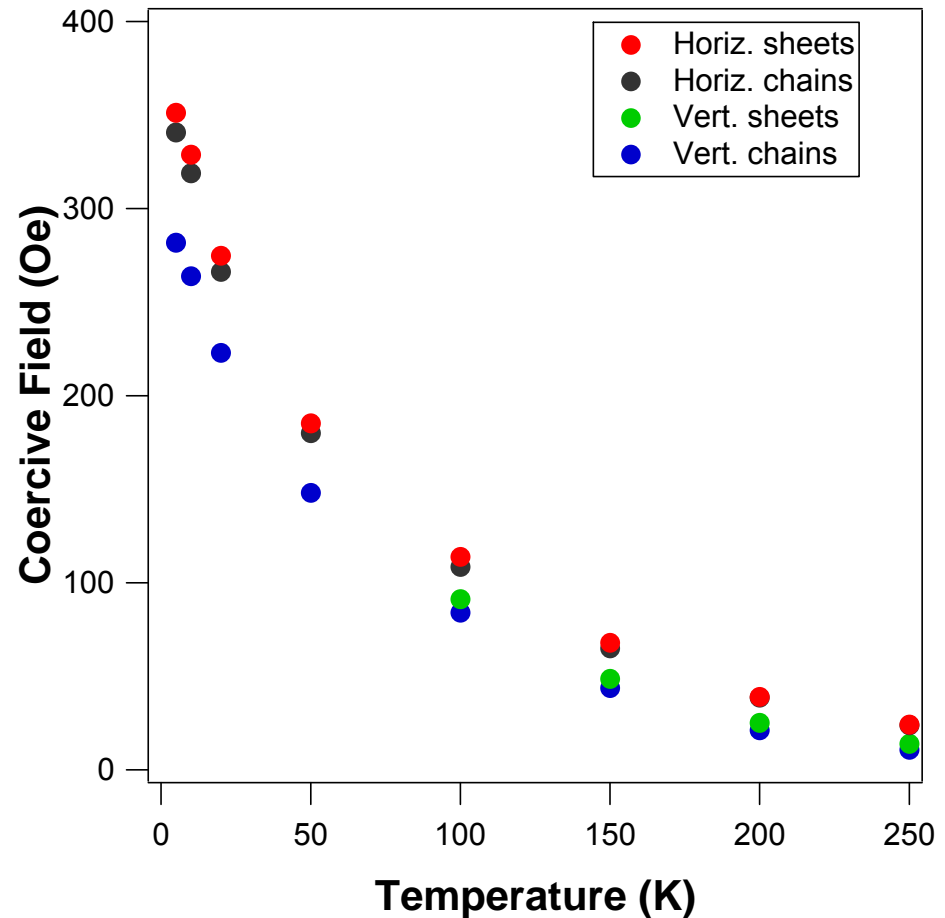
Susceptibility Enhancements in Structured Composites

- Structured with 150 G AC fields
- χ_p as high as 180 (at 250 K) should continue to increase with increasing T
- χ_p limited to 3 for micron scale particles
- Highly anisotropic composites



Coercivity of Structured Composites

- Enhanced susceptibility composites show reduced coercivity
- Cooperative flipping of spins



Optimizing Size

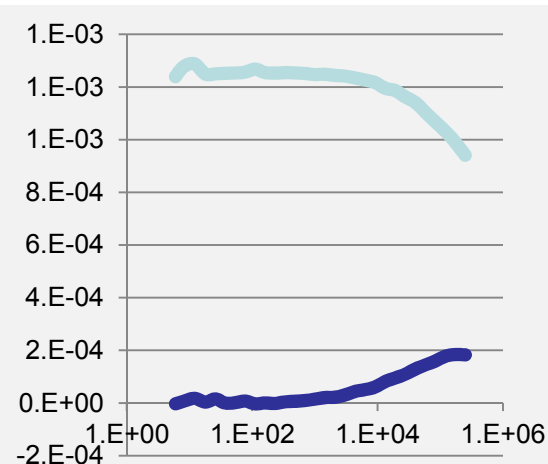
Scalable synthesis

- Microwave assisted thermolysis of $\text{Fe}(\text{CO})_5$ in octadecene.
- Reactions can be completed in 5 min.
- Opens the possibility of a flow reactor for production quantities.
- Surfactant used: oleyl amine.

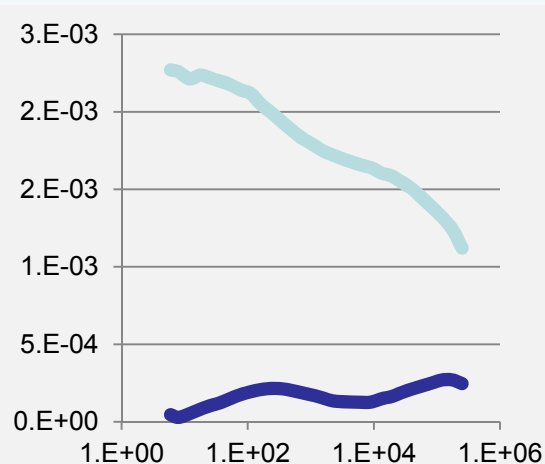
Complex susceptibility vs frequency tells us the maximum size to use

- Real Susceptibility (in phase)
- Imaginary Susceptibility (out of phase or loss)

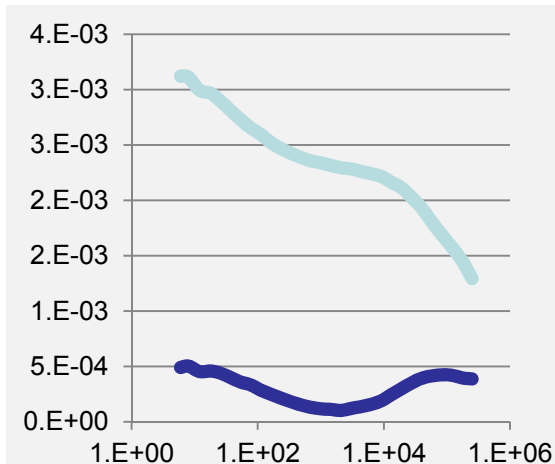
15 nm Fe particles



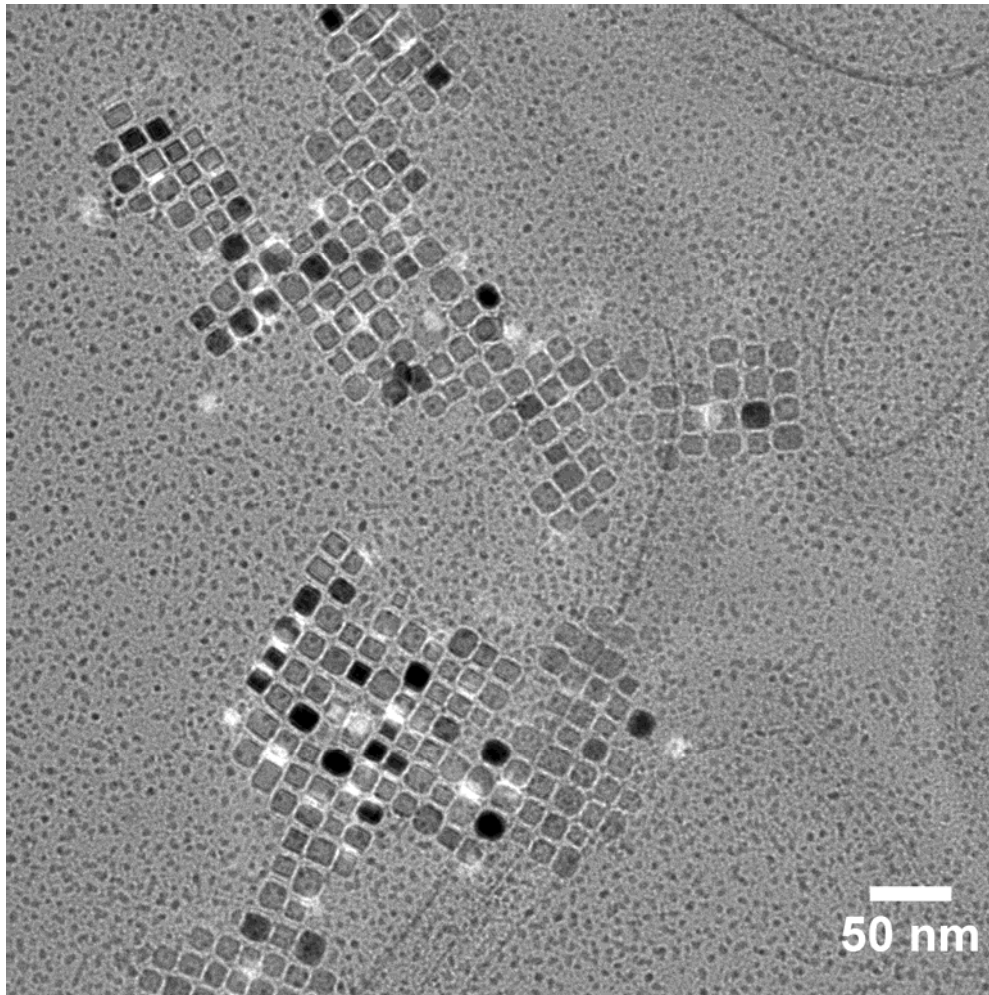
20 nm Fe particles



25 nm Fe particles



Cubes on a Simple Cubic Lattice



- Thermolysis of $\text{Fe}_3(\text{CO})_{12}$ in octadecene
- Antiferromagnetic ordering between particles?
- Main focus is on avoiding coupling between particles, but it could be beneficial in some cases

Project Status



- Optimizing particles
- Beginning initial experiments with composites
- Small scale test parts within 6 months
 - Allows optimization
- Within 2-3 years we will have full scale part available for testing in real circuit.

- Manufacturing
 - Lab scale: pouring molds
 - Scale up: Injection molding, reactive injection molding