

Microstructural Evolution of Nanocrystalline Nickel Thin Films due to High-Energy, Heavy-Ion Irradiation

SAND2012-6864C

S. Rajasekhara^{a,b}, P.J. Ferreira^b, K. Hattar^a

^a Sandia National Laboratories

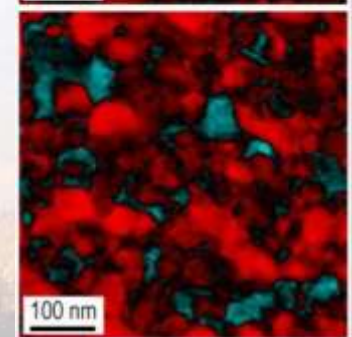
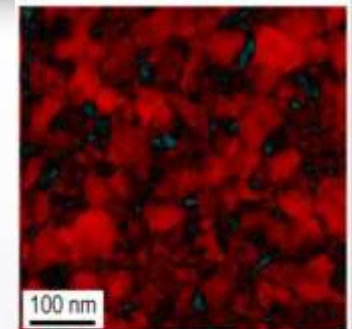
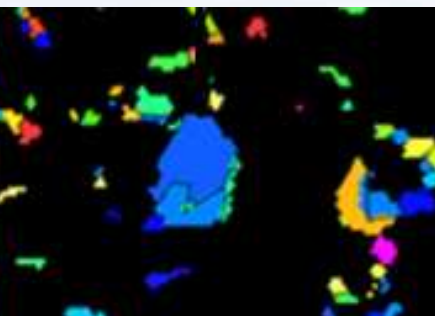
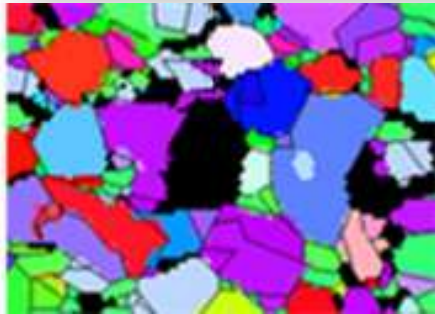
^a University of Texas – Austin

August 14, 2012



Outline

- Nanostructured metals provide the potential for radiation tolerant design
- Far-from-equilibrium nanograined metals can exhibit very unique and unexpected defect structures
- Thermal, mechanical, and radiation properties are probably related
- Precession microscopy provides a way to characterize grain orientation and grain boundary at the smallest scale
- *In situ* ion irradiation provides the potential to directly watch the evolution

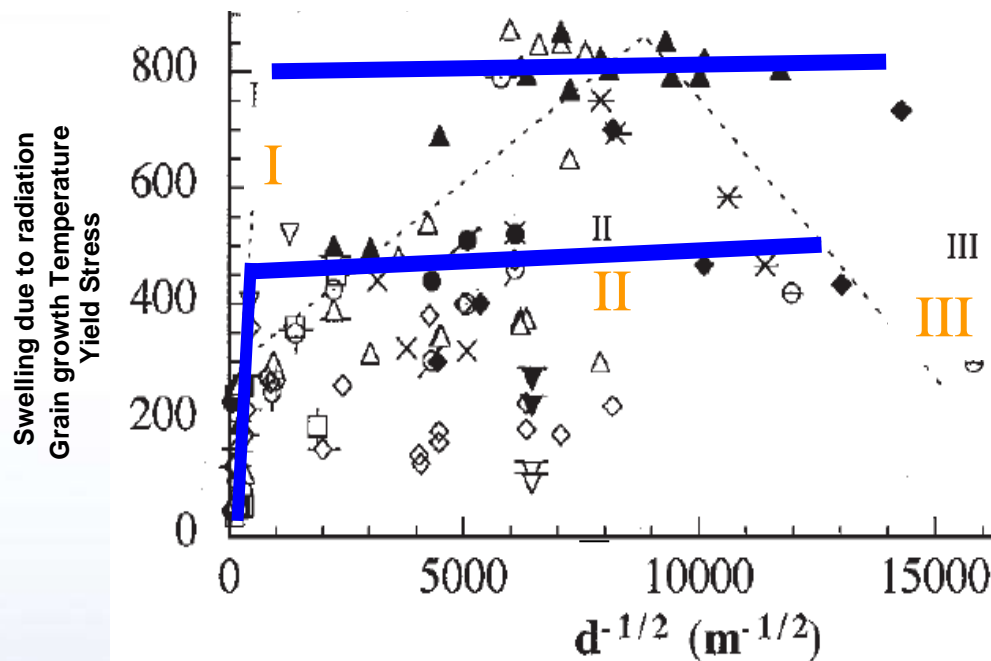


Sandia National Laboratories is a multi-program laboratory managed and operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000. This work was partially supported by DOE, Office of Science, Office of Basic Energy Sciences.



Sandia National Laboratories

Properties and Mechanisms Active in Nanograined Metals



Conrad, Metallurgical and Materials Transactions A: 2004. 35 p. 2681

Due to the variations in:

- Production methods
→ Range of microstructures
- Testing methods
→ Range of experimental uncertainty

Key overlooked factors

- Grain boundary type
- Relative grain orientation
- Grain boundary energy
- Etc.

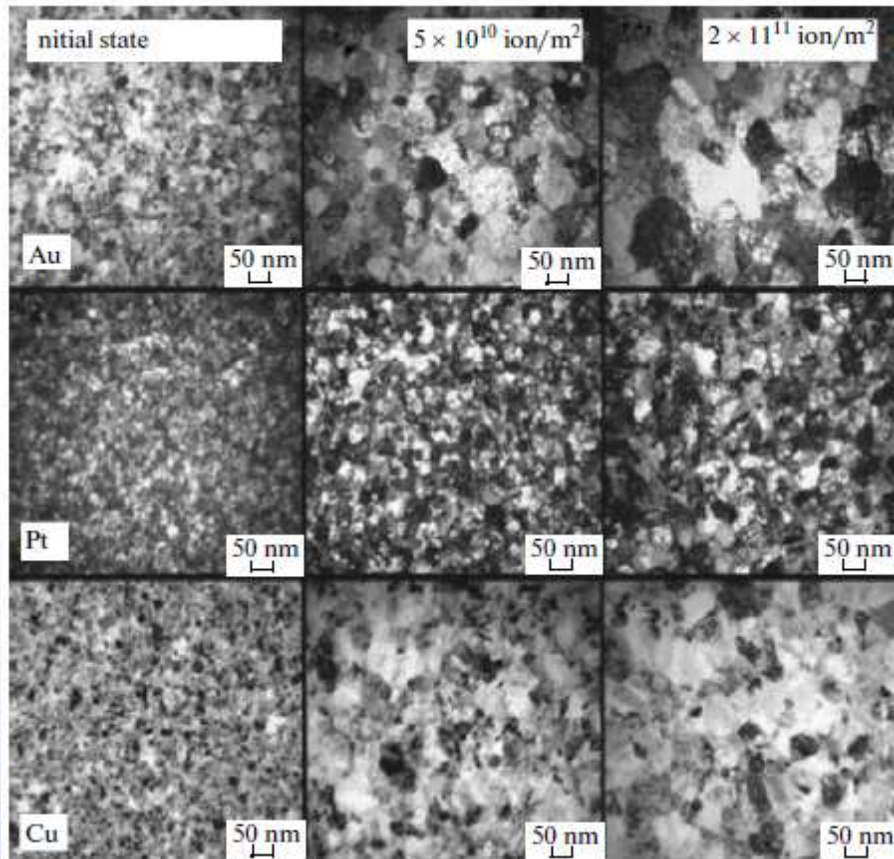
- 1) What are the thermal and mechanical properties and how do they effect the radiation tolerances?
- 2) Beyond just grain size, what is the microstructure?



Sandia National Laboratories

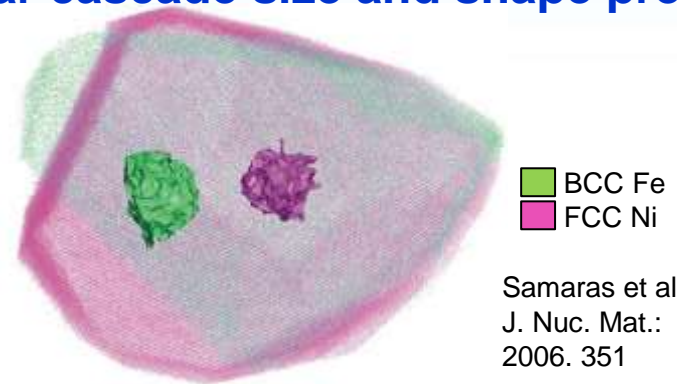
Radiation Tolerance from Nanograined Metals

Variation in radiation tolerances



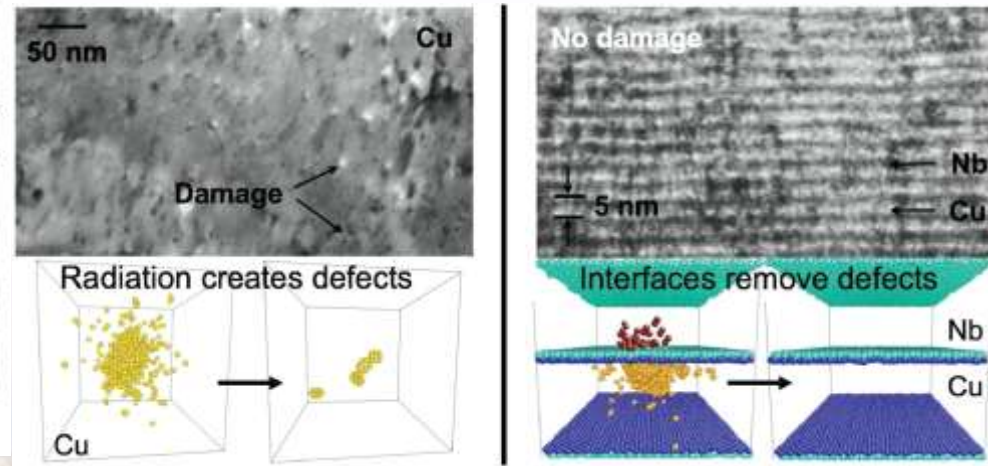
Kaomi et al., JAP: 2008. 104 073525

Similar cascade size and shape predicted



Samaras et al.,
J. Nuc. Mat.:
2006. 351

Nanolamellars are radiation tolerant



Demkowicz et al., MRS Bulletin: 2010. 35

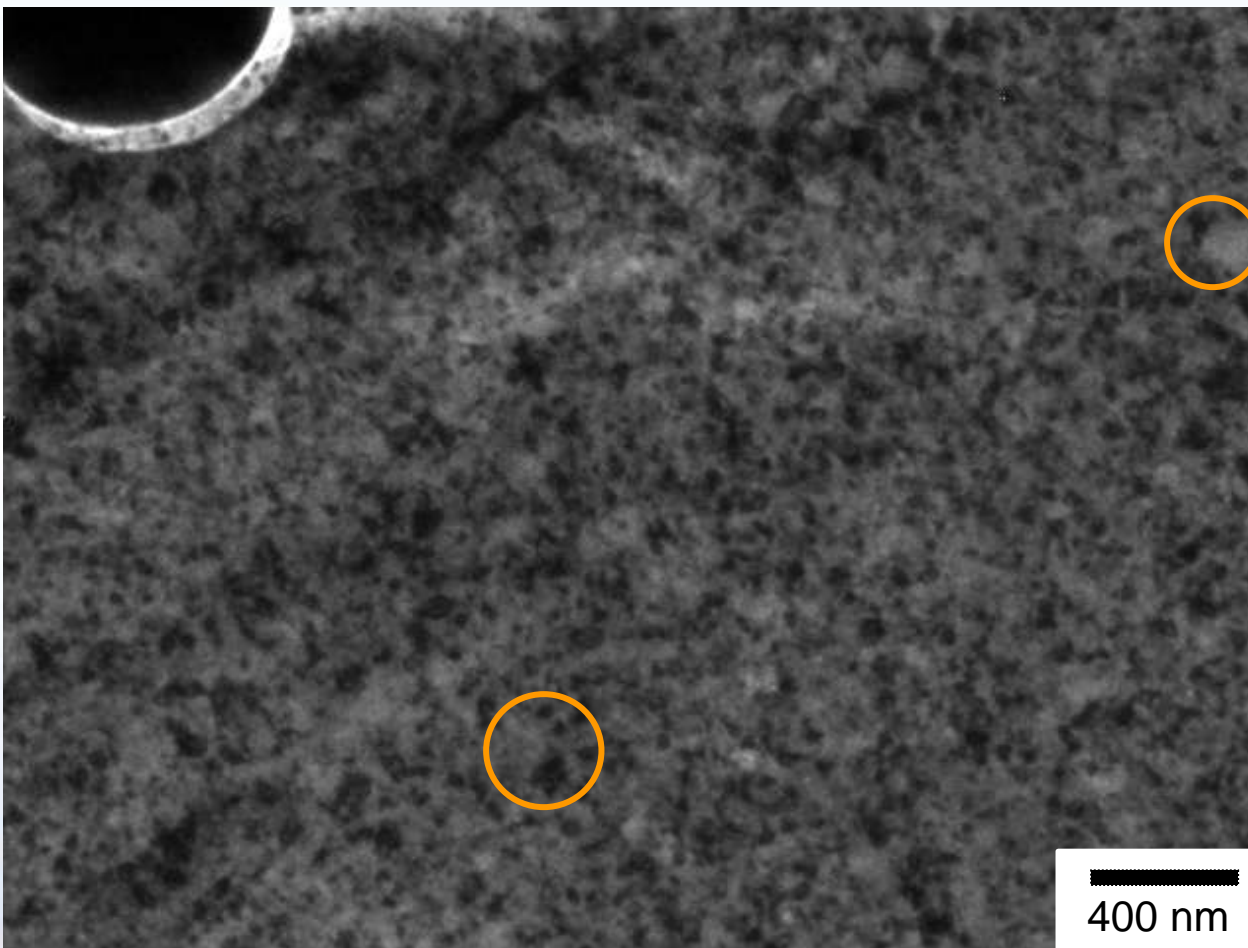
To a first order mean grain size comparison, these reports appear conflicting.

This may not be the case if initial microstructural details and associated properties are considered

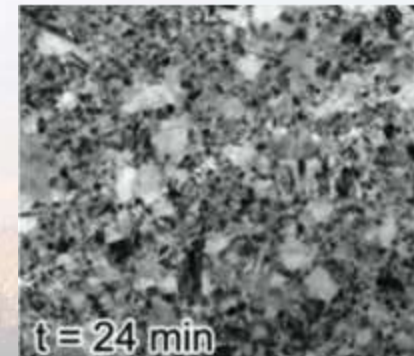
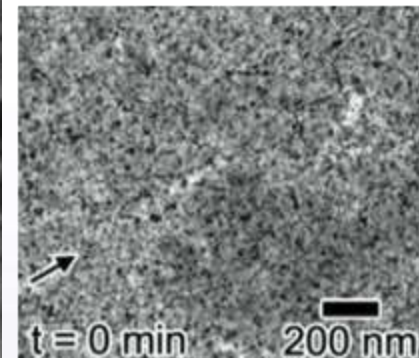
Thermal Stability of Nanocrystalline Materials

Abnormal Grain Growth is a function of:

- Time
 - 4 years of Aging
- Temperature
 - Surface Abnormalities
 - e⁻ beam
 - Ledges
- Film thickness
 - Splats (No effect)



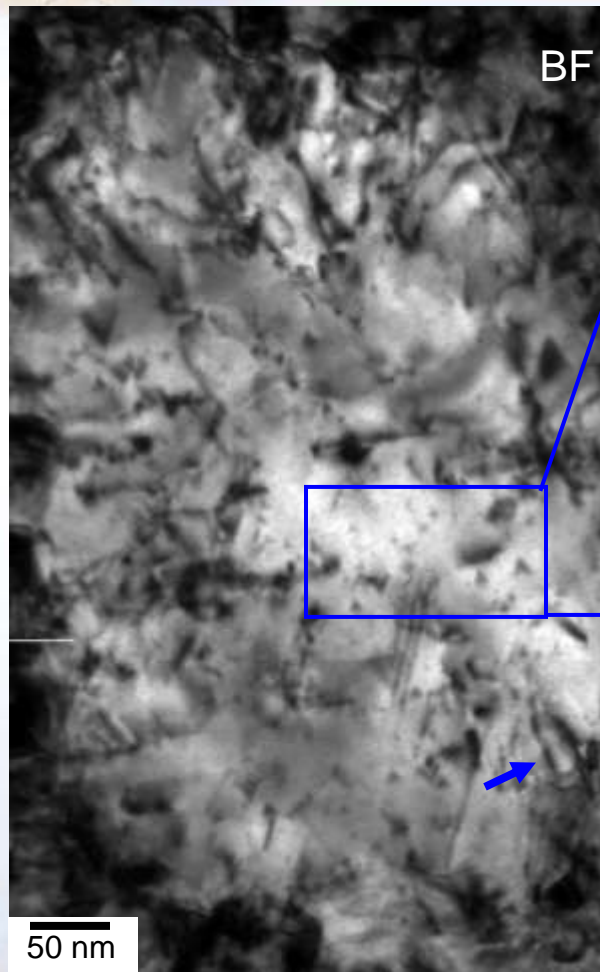
90 nm-thick PLD Ni during annealing at 350 °C



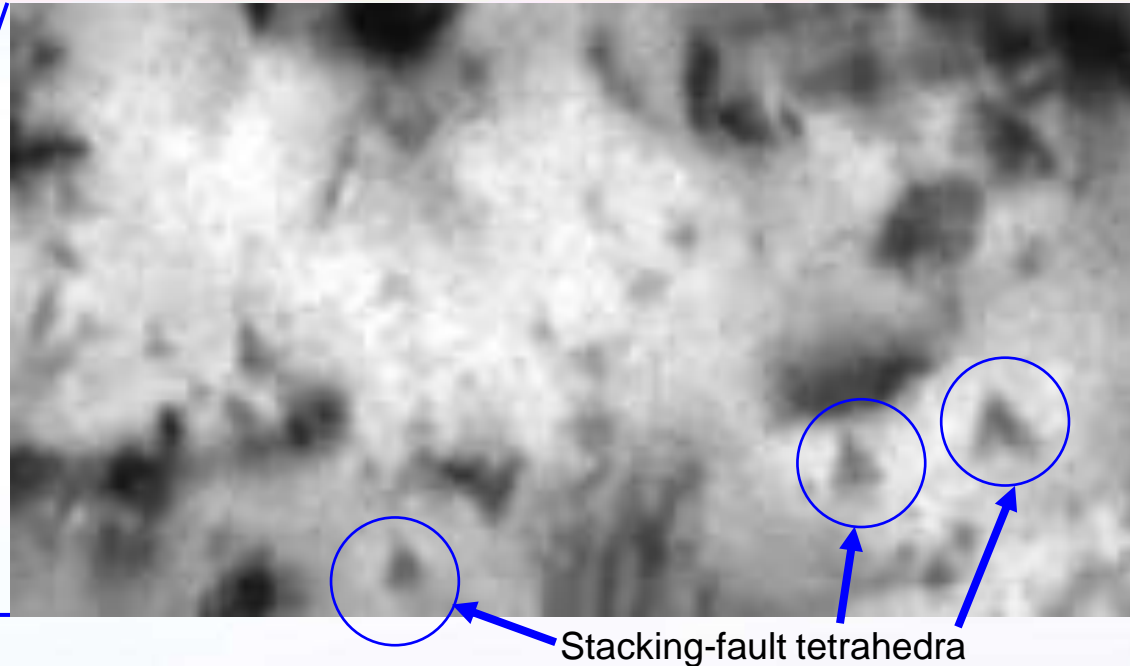
A few select grains grow at the expense of the remaining matrix



A Variety of Unexpected Defect Structures in Ni



230 nm thick PLD Ni Annealed
at 225 °C for 14 hrs.

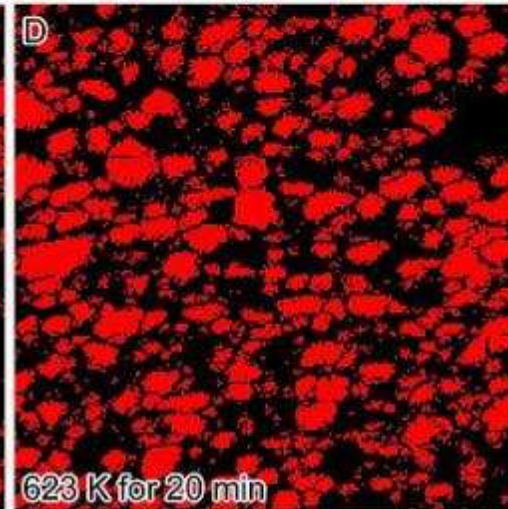
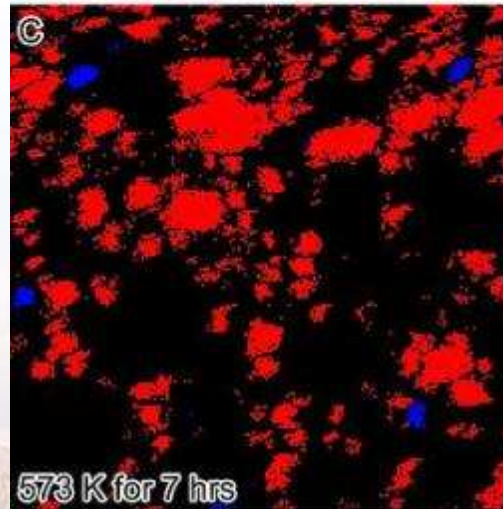
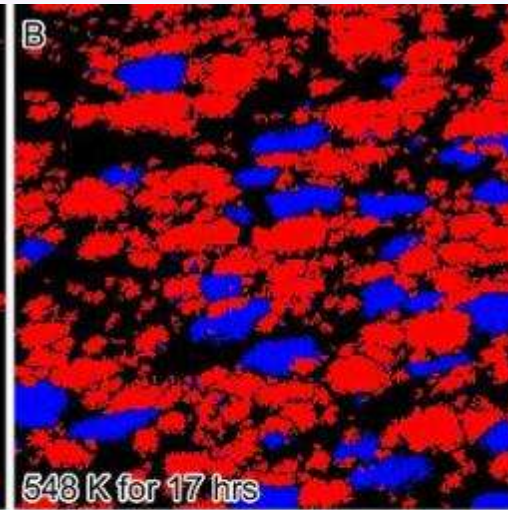
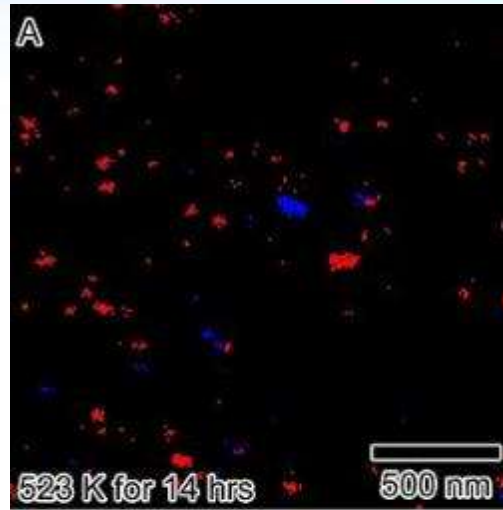
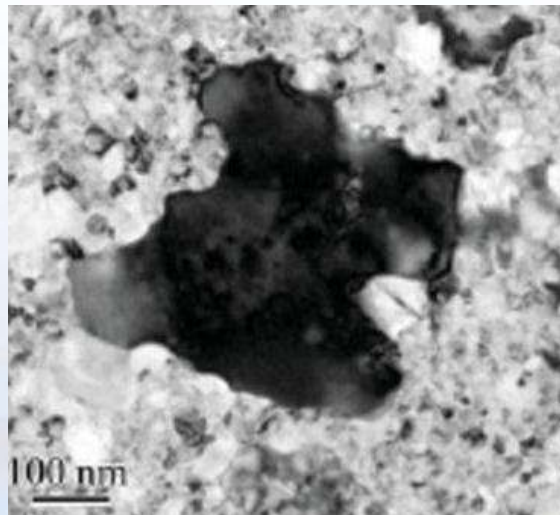
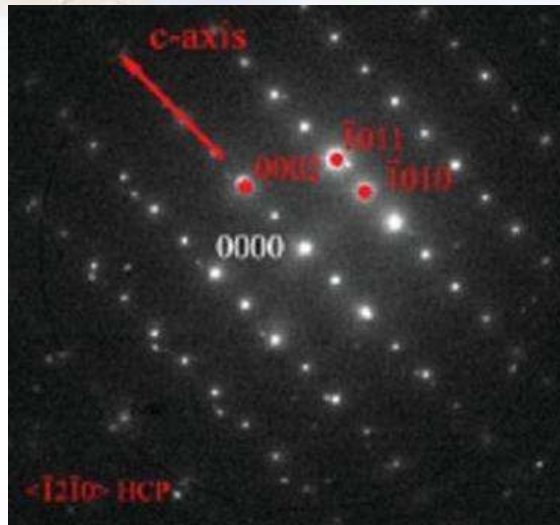


Multitude of defects in annealed PLD Ni

- SFT **at temperature**
- Stable microstructure for over 15 months
- SFT **not** due to irradiation, quenching, high strain rate
- SFT are theorized to be formed by rapid grain growth through the high free-volume at the initial grain boundaries



Thermal Stability of Nanocrystalline Materials: Evidence of *HCP* Phase Grains



■ FCC phase
■ HCP phase

However,
Spatial
resolution
limits
detailed
analysis of
the *HCP*
phase
evolution



L. N. Brewer, D. M. Follstaedt, K. Hattar, J. A. Knapp, M. A. Rodriguez, I. M. Robertson, Adv. Mater. 22 (2010), 1161

EBSD and SAD confirm the presence of
HCP phase in some abnormal grains



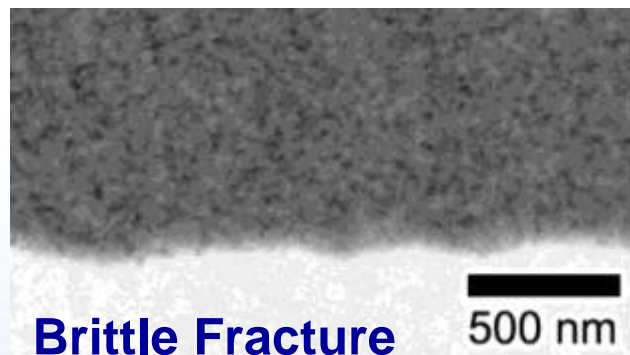
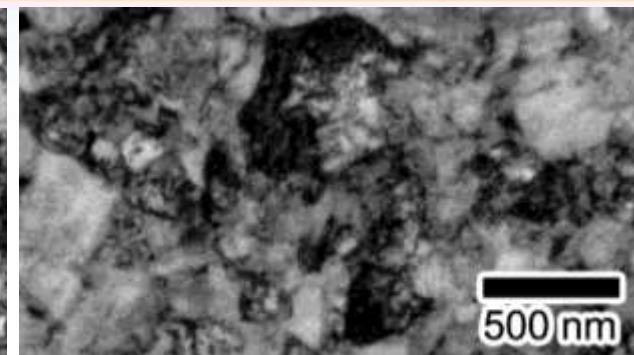
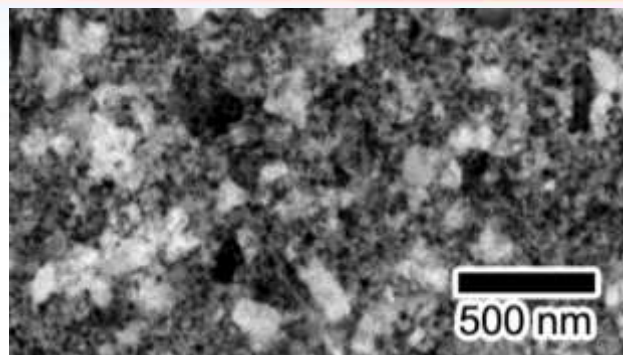
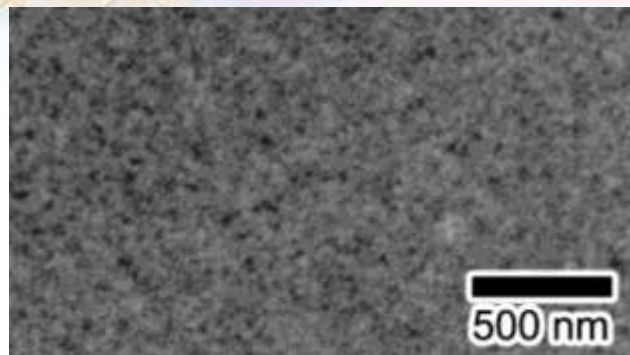
Sandia National Laboratories

Failure Analysis of Strained PLD Ni

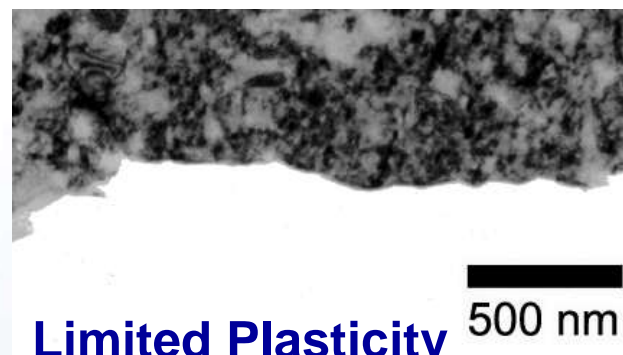
Nanograined

Bimodal

Ultra-fine grained



Brittle Fracture



Limited Plasticity



Shear Failure

- No observation of global plasticity

- Dislocation pile-up
- Local shear

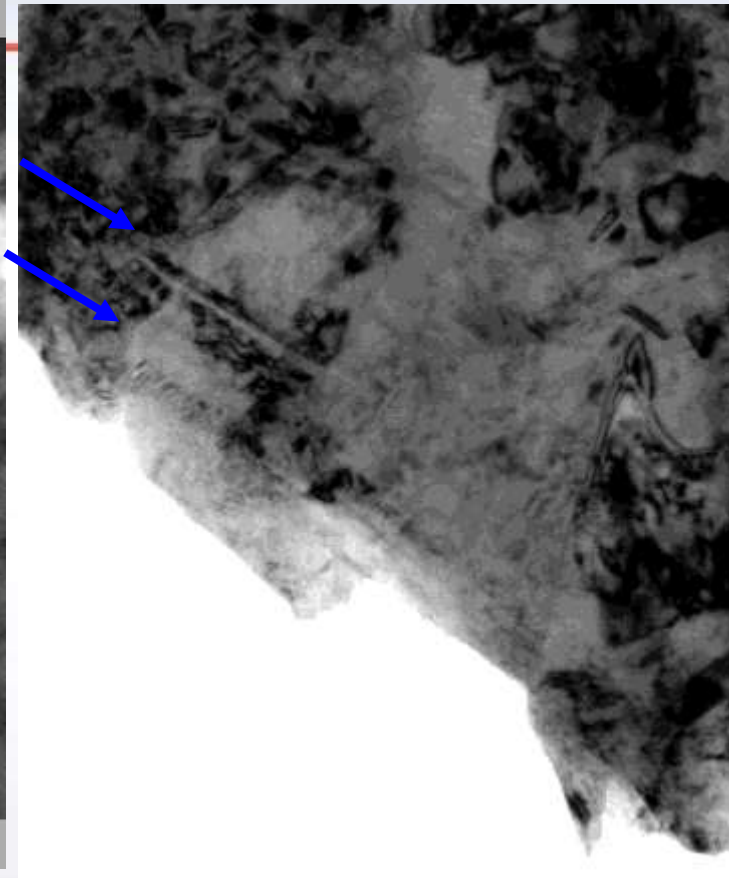
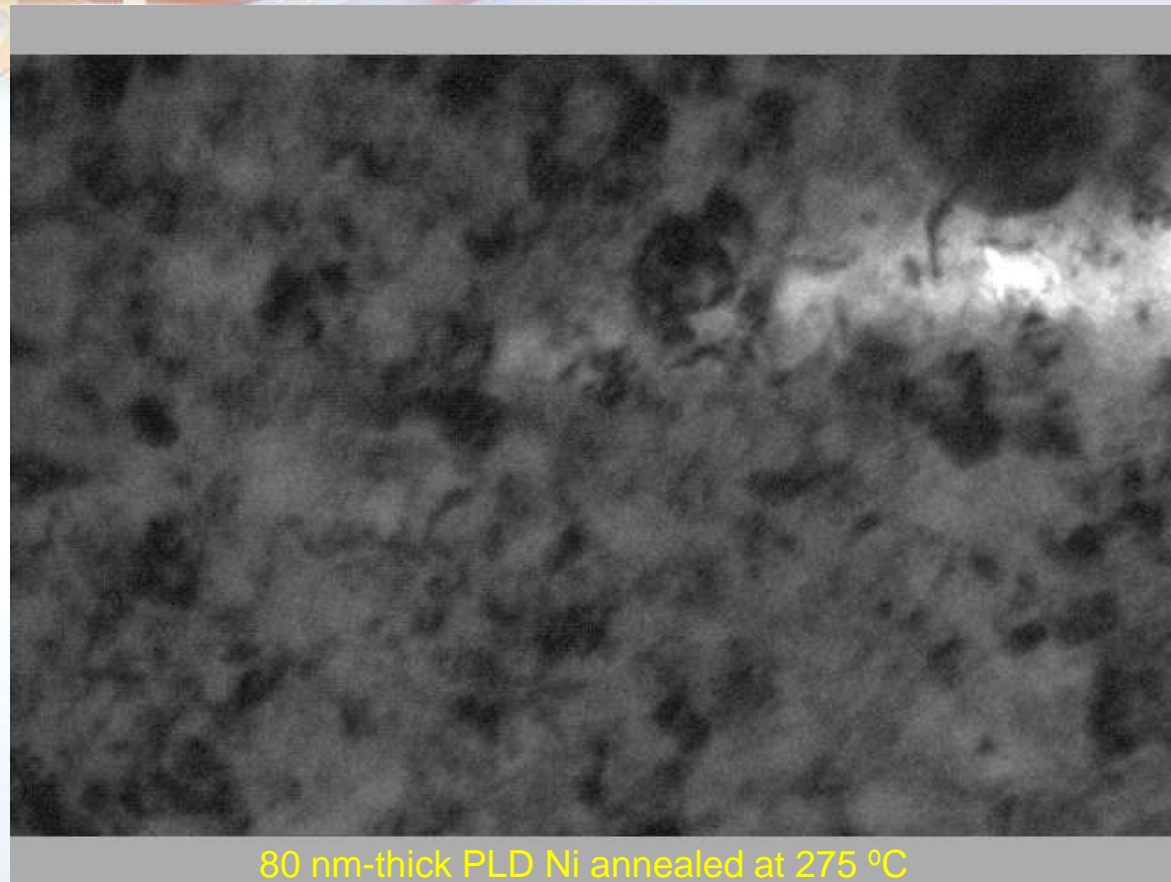
- Shear teeth
- Dislocation structure

Fracture surfaces provide insight to deformation processes



Sandia National Laboratories

Deformation and Failure in Bimodal PLD Ni



Throughout the film

- Elastic strain
- Limited dislocation slip

In the plastic zone

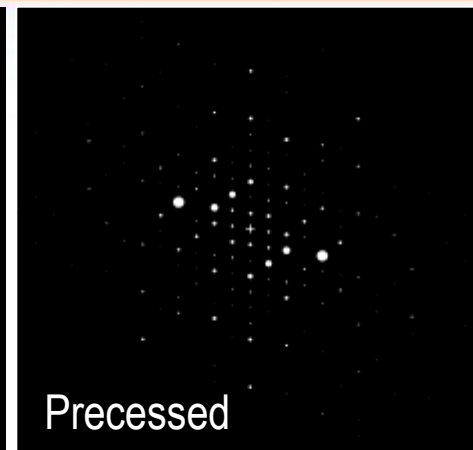
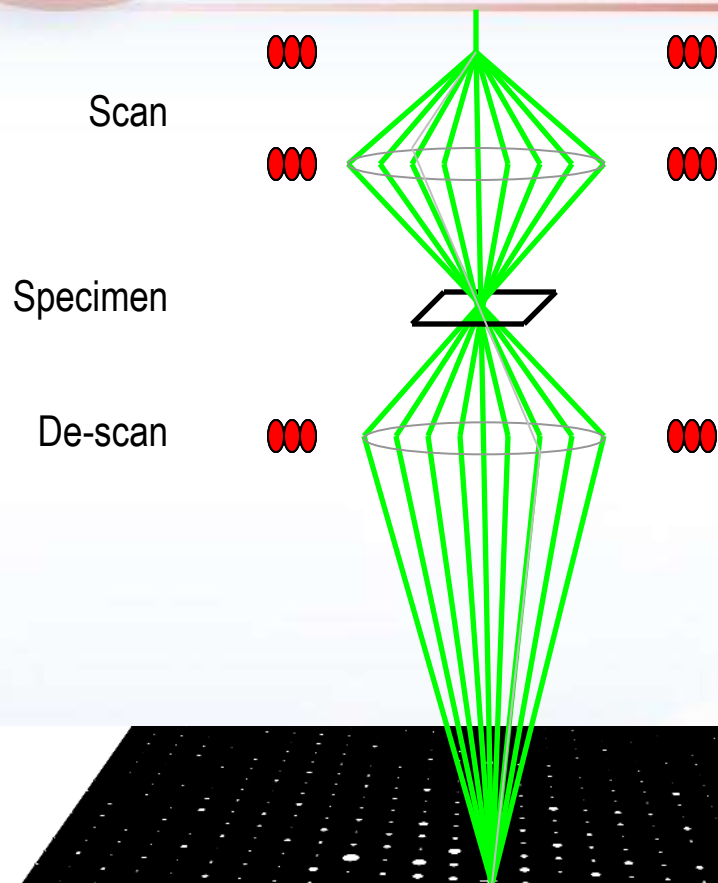
- Extensive dislocations slip
- Twinning

At Crack Tip

- Necking
- Grain agglomeration

We have some insight into the unique thermal and mechanical mechanisms and properties. What is the initial nanostructure that causes this?

Precession Electron Diffraction Microscopy



Advantages:

- < 10 nm spatial resolution
- Near kinematical electron diffraction
- Symmetry ambiguities are resolved
- Fast and automated acquisition
 - ~200 grains in 15 min.

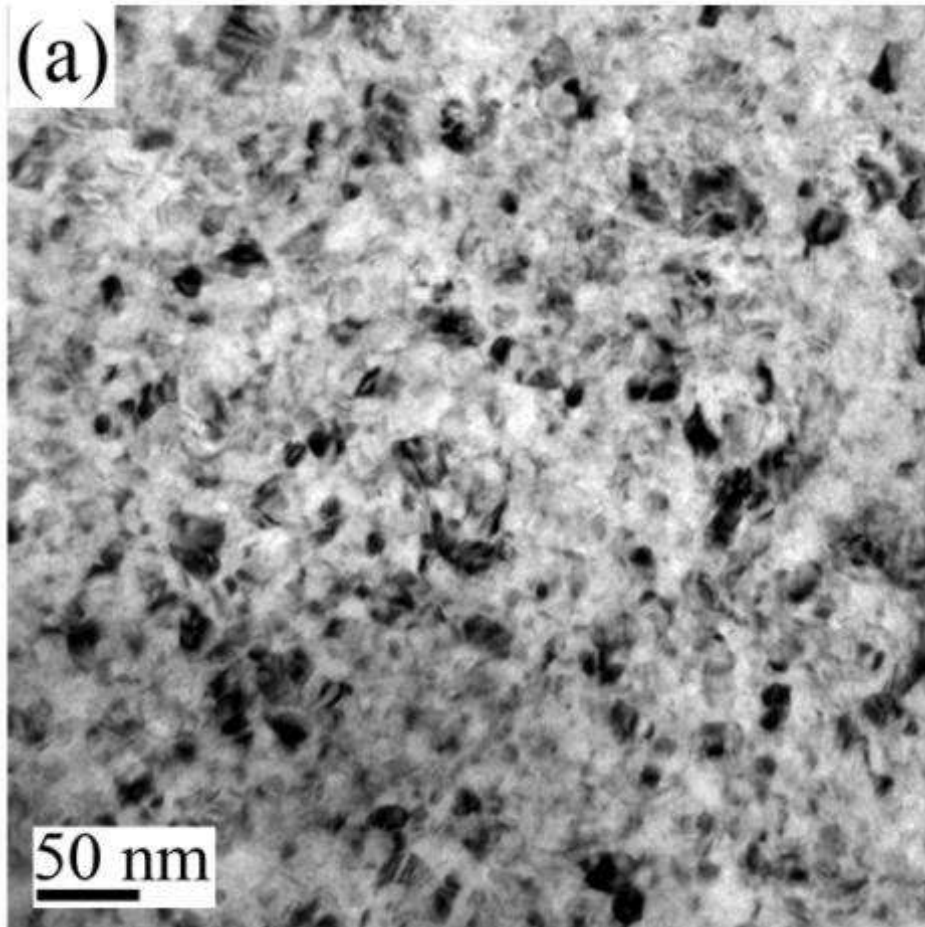
(Diffracted
amplitudes)



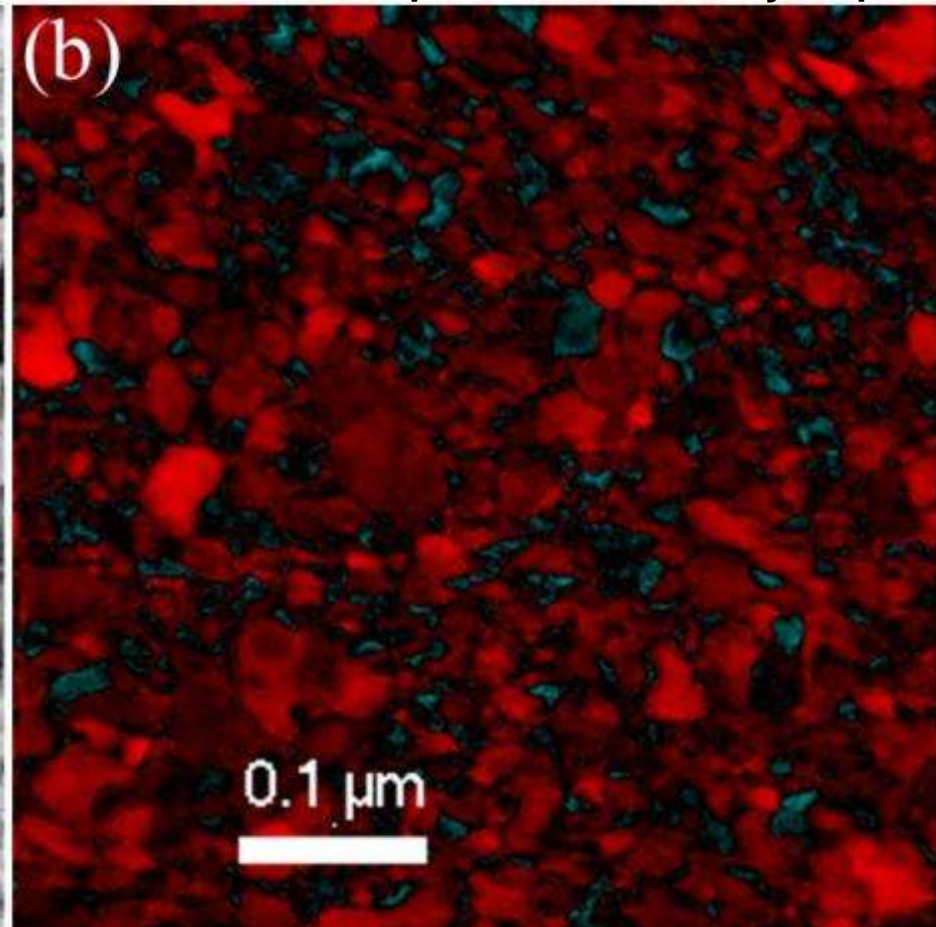
Sandia National Laboratories

Phase Determination in 50 nm As-deposited Ni Film

BF – TEM



Re-constructed phase and reliability map



- 1,124 HCP phase grains (in $1.5 \mu\text{m}^2$)
- Mean HCP grain size : $8.1 \pm 0.3 \text{ nm}$
- Mean HCP phase percentage: 6.0%

Clear observation of morphology
in nanocrystalline films

■ FCC phase
■ HCP phase

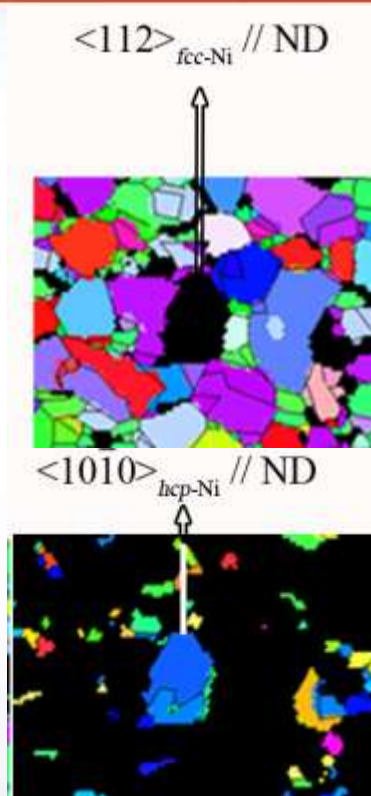
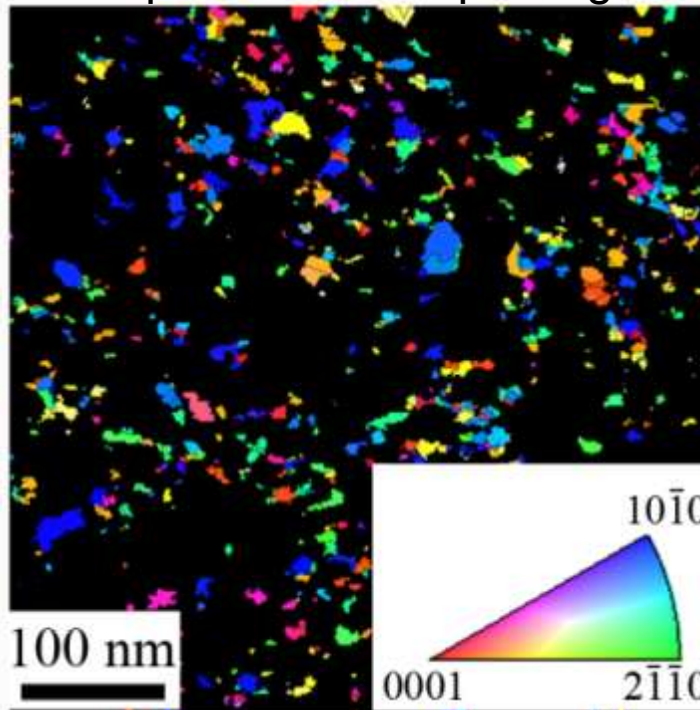
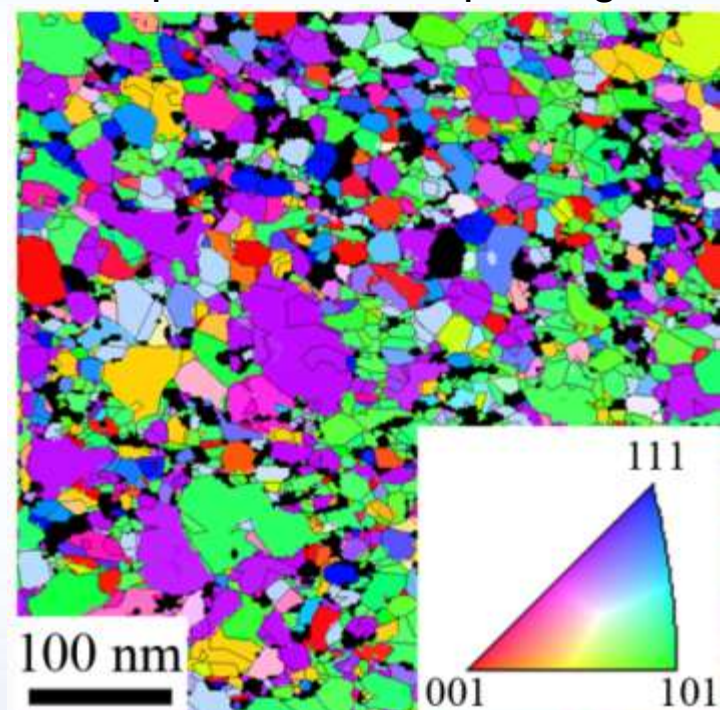


Sandia National Laboratories

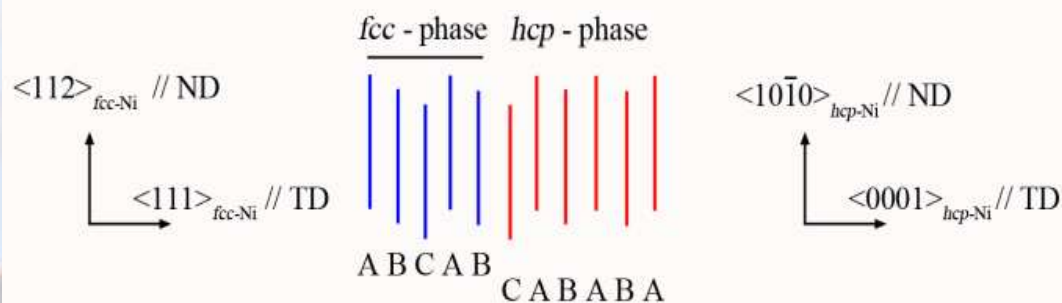
FCC and HCP Texture Determination in 50 nm As-deposited Ni Film

FCC phase inverse pole figure

HCP phase inverse pole figure



Texture maps at the nanoscale obtained from a TEM

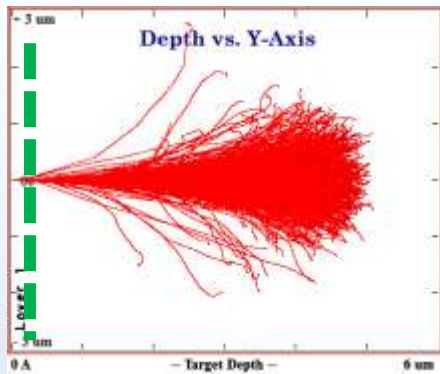
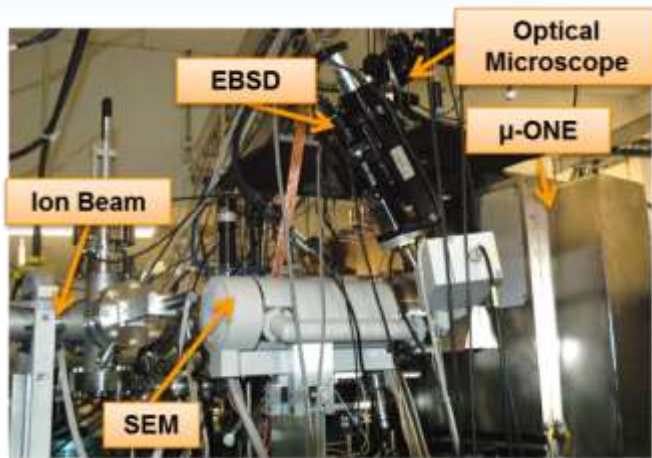


$\langle 112 \rangle_{\text{FCC-Ni}} // \text{ND}$ results in an in-plane
 $\langle 111 \rangle_{\text{FCC-Ni}} // \text{TD}$

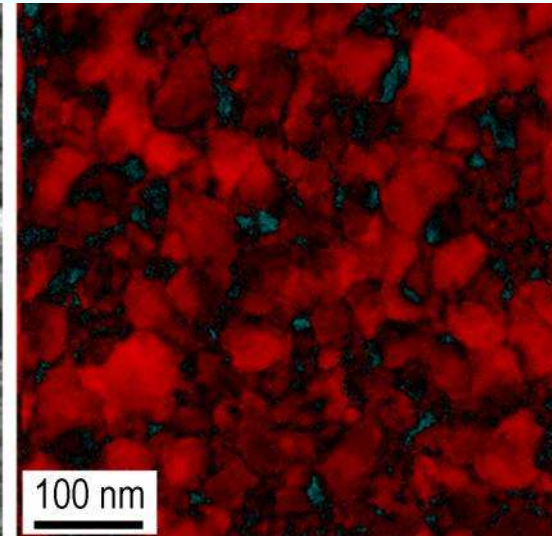
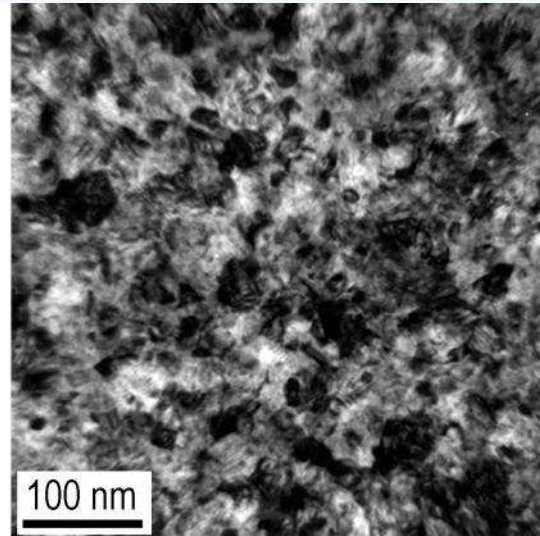
High energy PLD may introduce stacking
faults leading to a $\langle 0001 \rangle_{\text{HCP-Ni}} // \text{TD}$



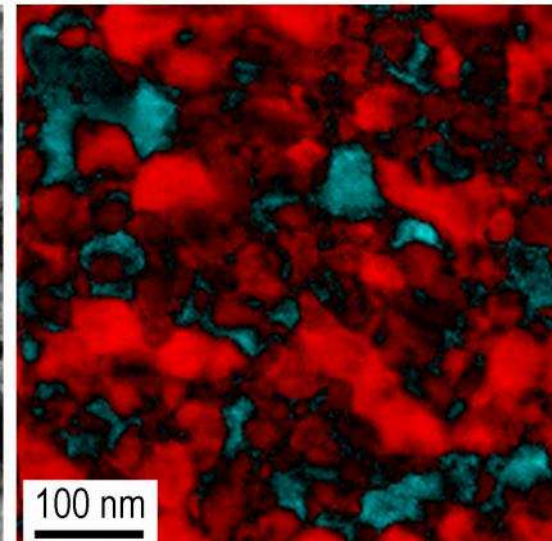
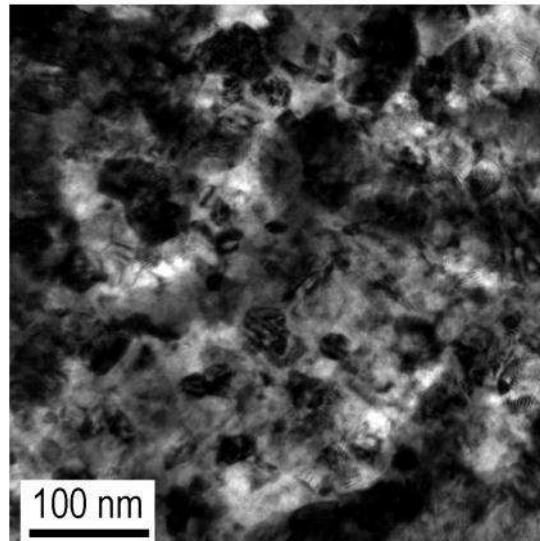
FCC and HCP Phase Evolution after 35 MeV Ni Irradiation



As-deposited



35 MeV Ni ~ 10 dpa

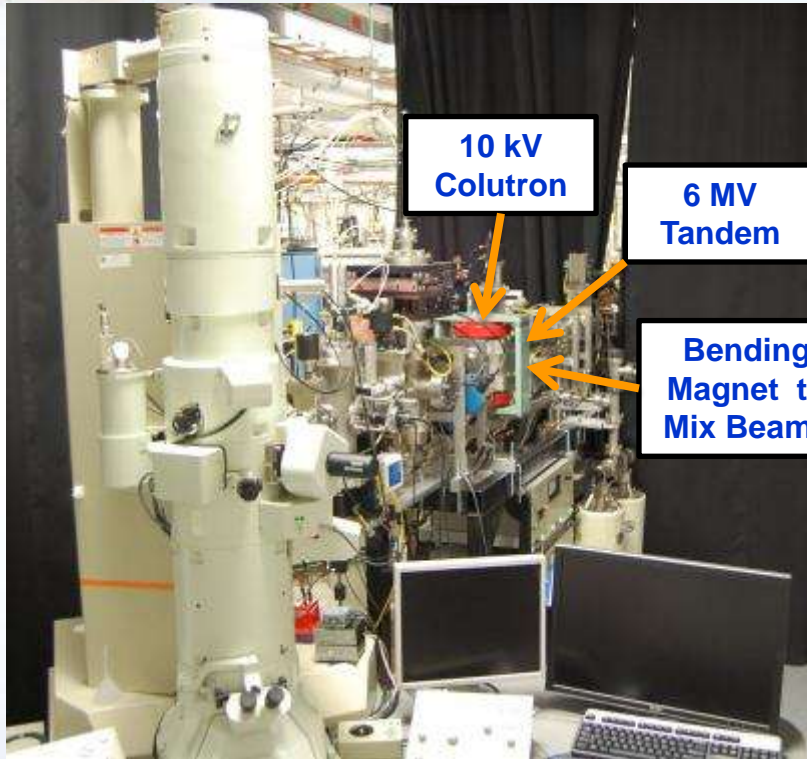


Despite the minimal interaction predicted in 100 nm film, grain growth was observed and increased *HCP* phase resulted

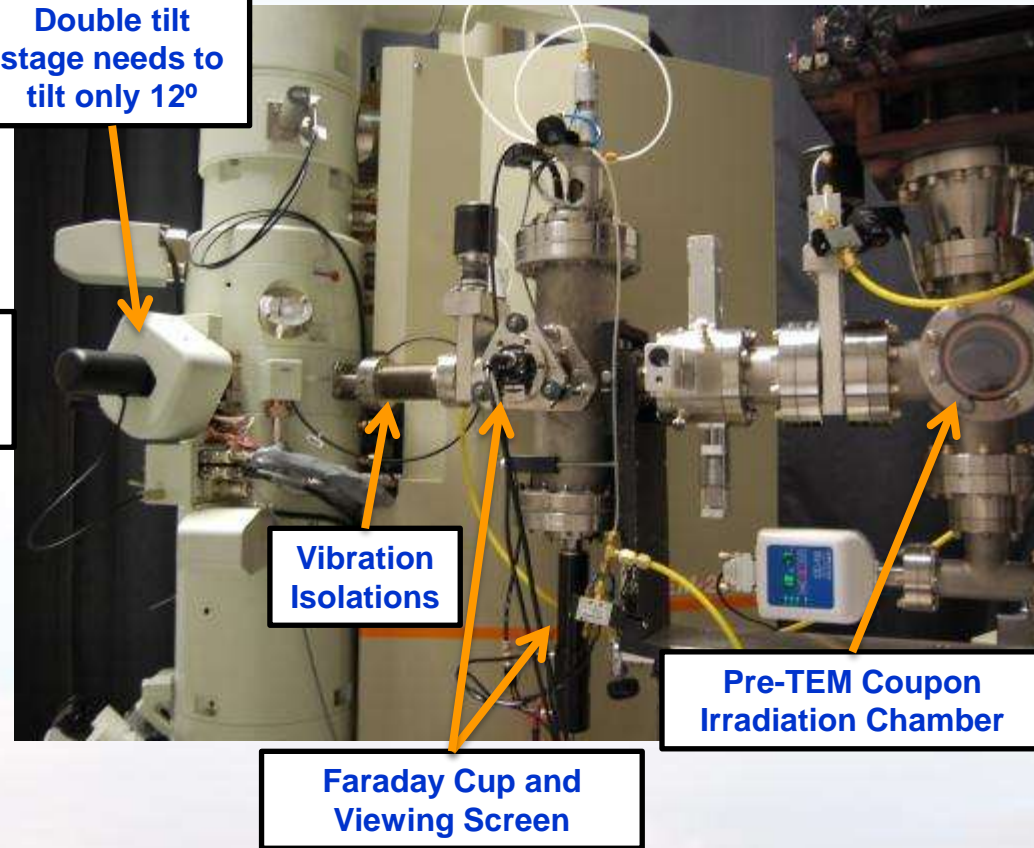
■ FCC phase ■ HCP phase

In situ TEM Beamline

Collaborators: D. Buller and J.A. Scott



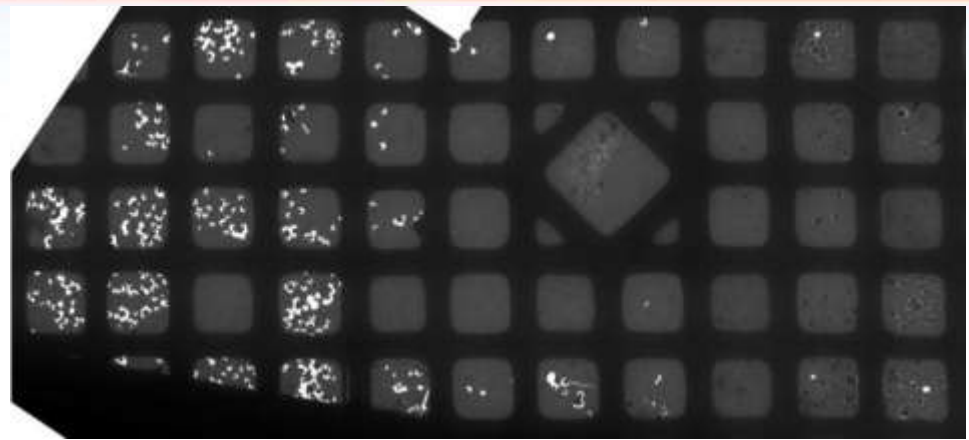
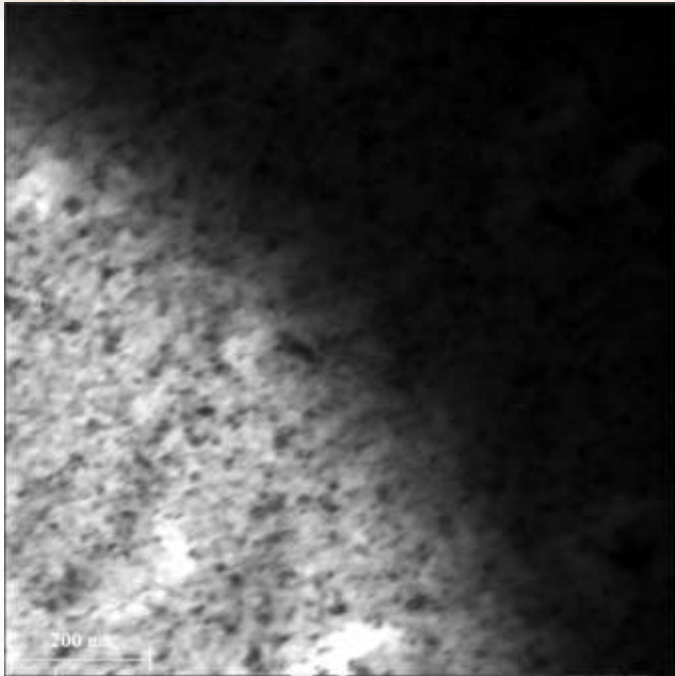
Double tilt stage needs to tilt only 12°



Beam burn from
14 MeV Si

Tandem beamline into the TEM is completed and operated regularly
Colutron beamline is assembled, under vacuum, and baked out
We hope to have concurrent heavy and light ion irradiation facility
operational in 2012

Initial *In situ* Ion Irradiation Results of PLD Ni



- 80nm PLD Ni deposited on salt and transferred to Cu grid
- Annealed 2 hrs at 250 °C
- 60 nA 5 MeV Si³⁺ beam with $d_{\text{beam}} \sim 3\text{mm}$
- Sample irradiated 45° normal to ion beam

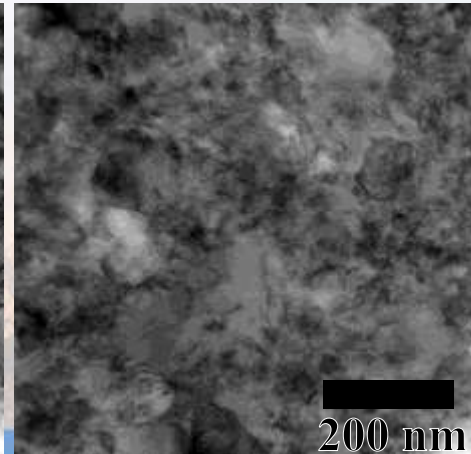
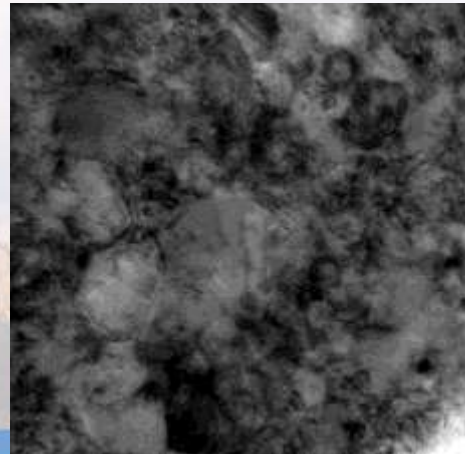
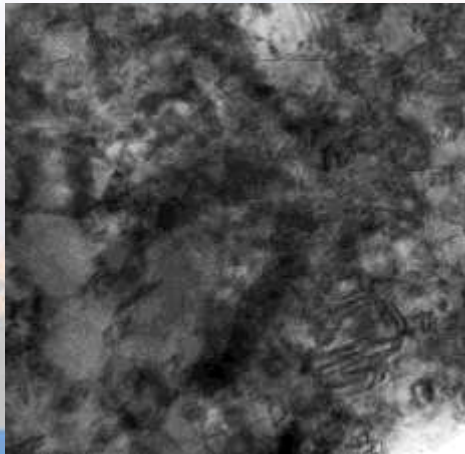
High energy heavy ion beams alter the electron beam when not grounded

Some structural changes observed during *in situ* TEM

0 dpa

5 min (~.02 dpa)

10 min (~.04 dpa)



Collaboration with: A. Kinghorn and B. Yates

Conclusions

- Unique structures result from nanograined processing
- Thermal, mechanical, and radiation stability of nanograined metals are probably intertwined
- Precession microscopy provides a unique tool to study the grain orientations and boundary relationships

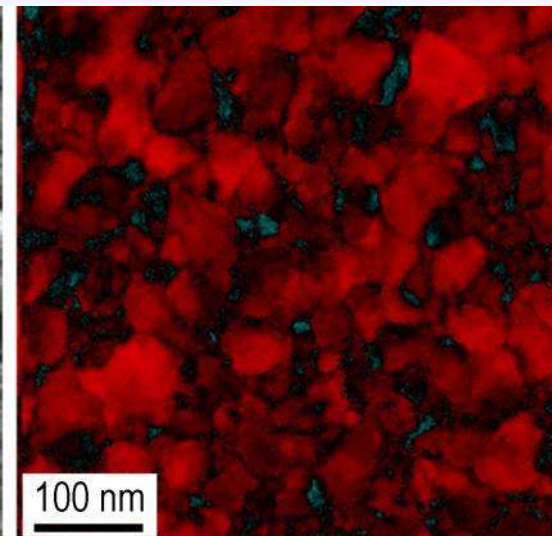
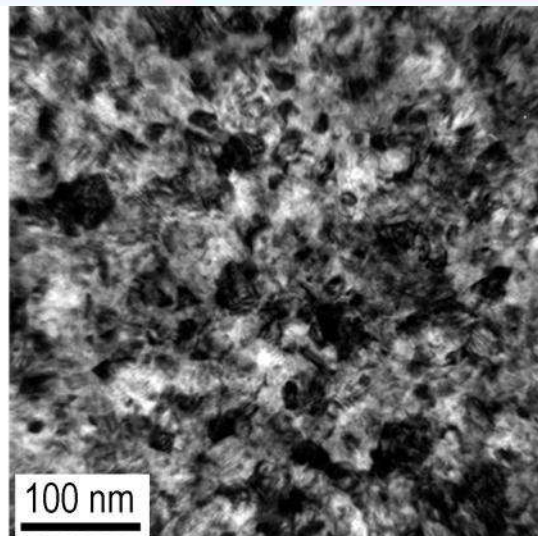
Ion Irradiation

+

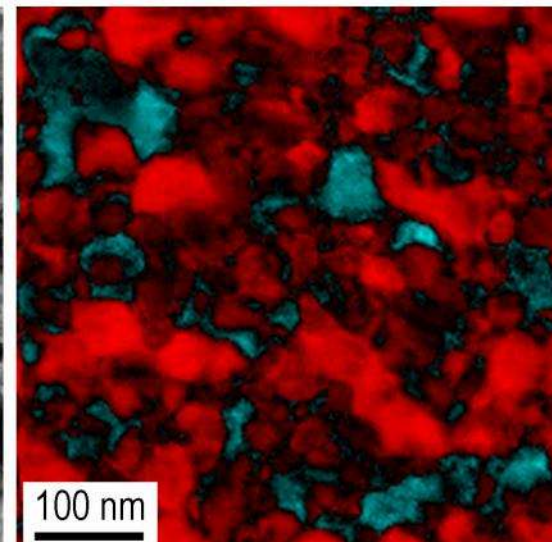
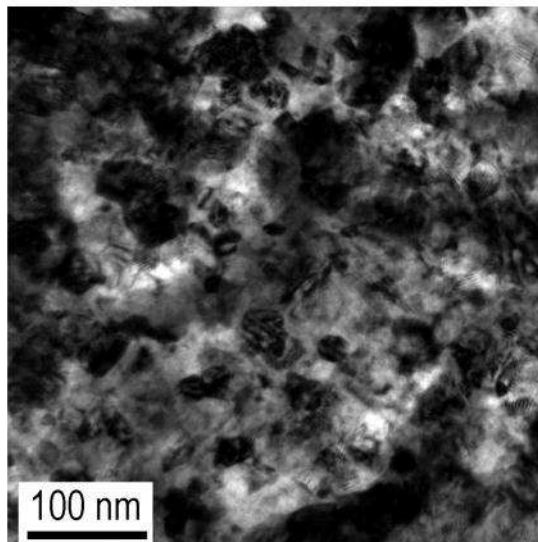
Precession Microscopy

Greater insight into
structural evolution due
to radiation damage

As-deposited



35 MeV Ni ~ 10 dpa



FCC phase



HCP phase



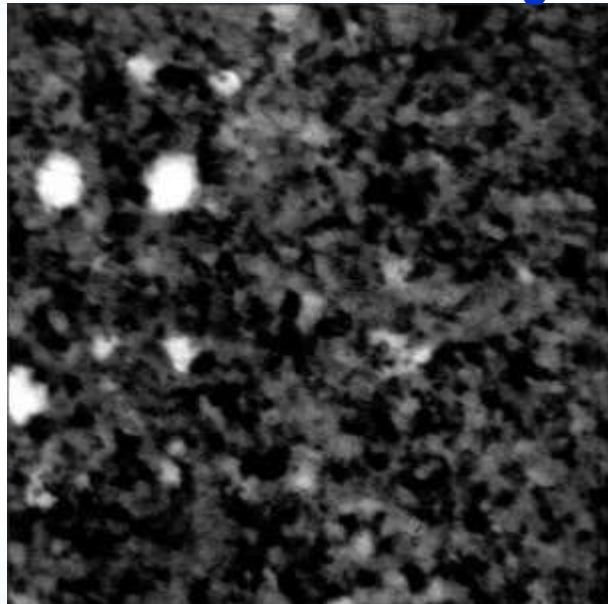
Sandia National Laboratories



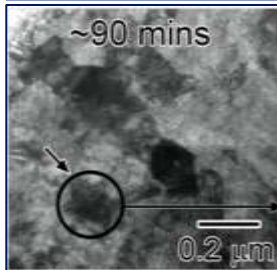
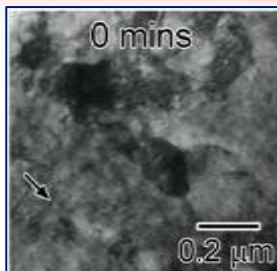
Sandia National Laboratories is a multi-program laboratory managed and operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000. This work was partially supported by DOE, Office of Science, Office of Basic Energy Sciences.

Interesting New *In situ* TEM Directions...

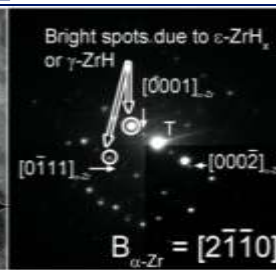
Acetic acid flow over ng-Fe



Collaboration with: D. Gross,
J. Kacher, and I.M. Robertson



Hydride formation in Zirlo exposed to pure H₂



Collaboration with: S.
Rajasekhara and B.G. Clark

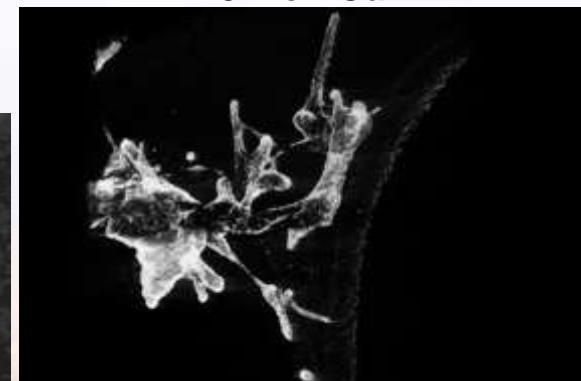
Zircaloy heating in air

Irradiation stability of scintillators

Initial CdWO₄ model



Irradiation with 20 nA,
3 MeV Cu⁺



Collaboration with: S. Hoppe, B.A.
Hernandez-Sanchez

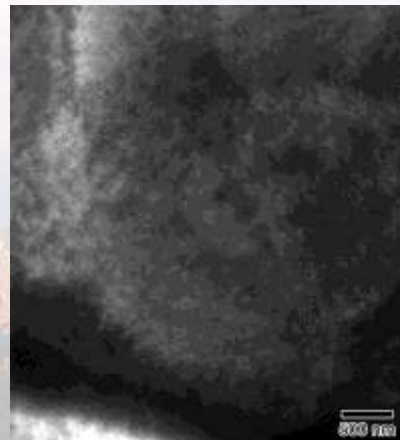
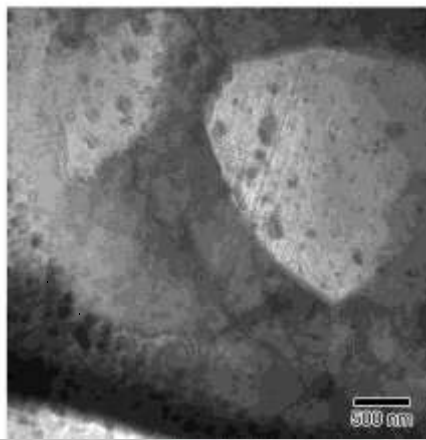
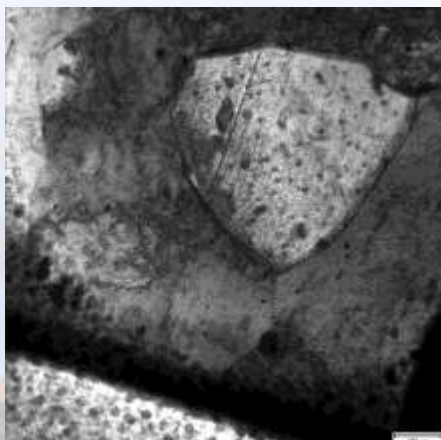


Sandia National Laboratories

Vacuum & Single Window

~1 atm & Two Windows

~1 atm & ~1200 C



Collaboration with: S. Rajasekhara and B.G. Clark