

Phosphors and Quantum Dots for Solid State Lighting

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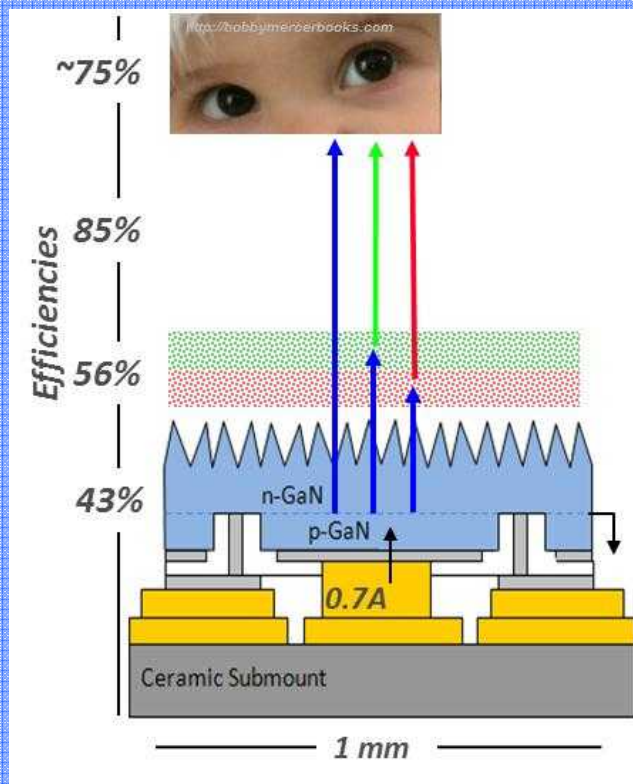
September 27, 2012



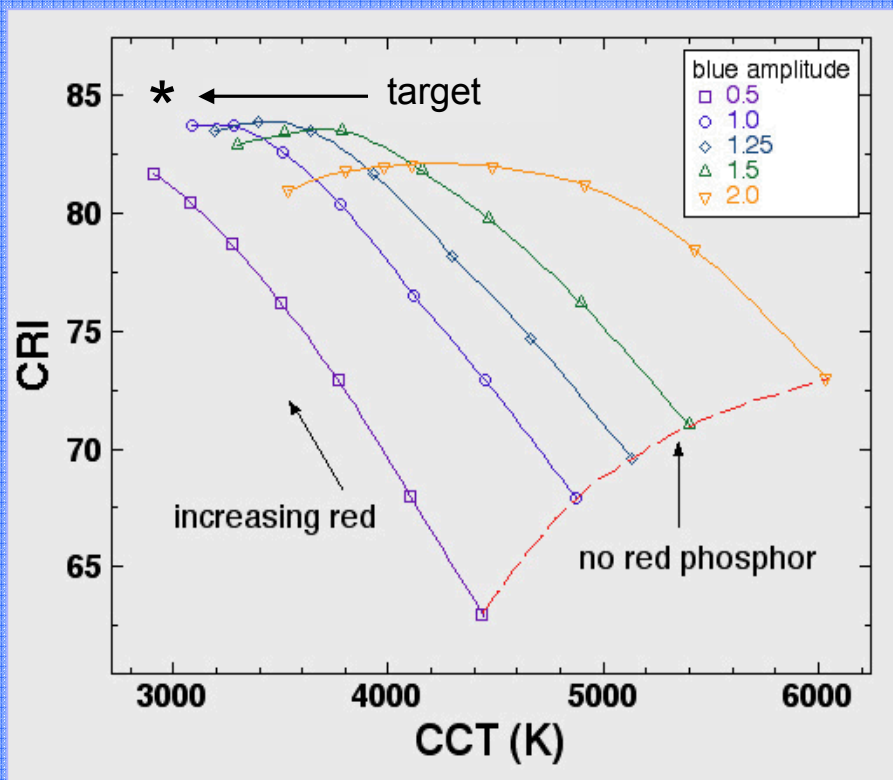
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Wavelength downconversion approach to solid-state lighting (SSL)



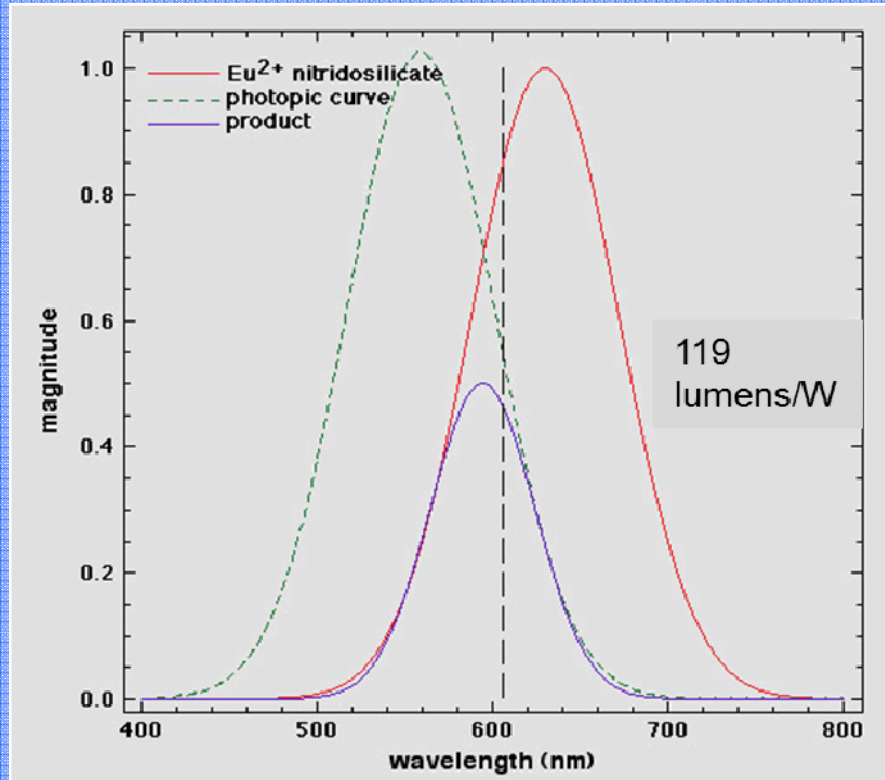
- Blue InGaN LEDs combined with green and red downconverters yields white light with improved CRI and CCT.



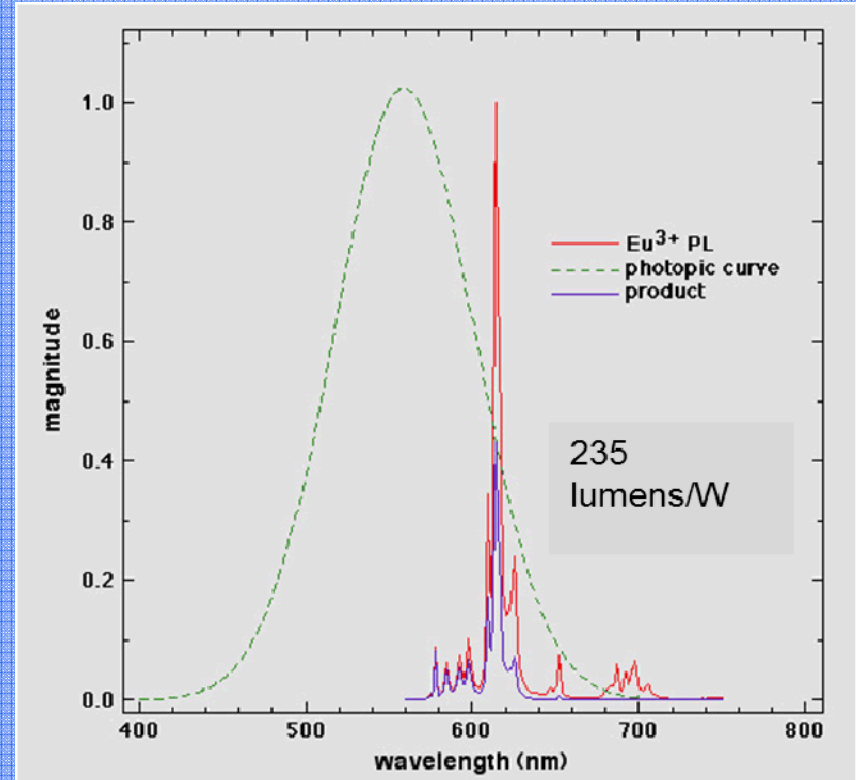
- Warm white target: CCT=3,000K; CRI=85*
- The red component is critical to reaching the target.

Narrowband red emitters are needed for warm white LEDs

Current red emitters (Eu^{2+}) are too broad, FWHM ~ 100 nm.



Eu^{3+} red emitters are narrow, FWHM ~ 5 nm.



- The luminous efficacy of radiation (LER) for red emission (>606 nm) can be significantly increased with narrow bandwidth emitters.

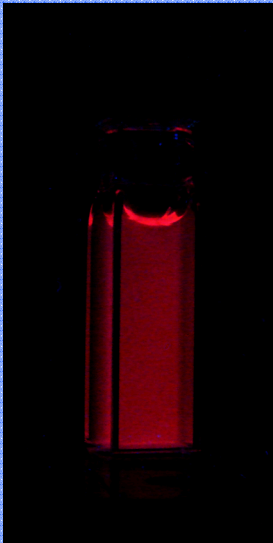
SSL downconverters must satisfy numerous criteria

- High quantum efficiency
- Low reflectance at excitation wavelength
- Low thermal quenching
- Photo-stability
- Thermal stability
- Chemical stability
- Saturation resistant
- High luminous efficacy of radiation
- Excitable with blue light
- No unwanted green/yellow absorbance
- Enables high CRI & low CCT
- Non-toxic
- Chemically inert
- Low light scattering

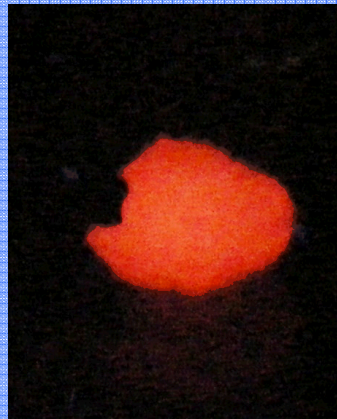
- **Currently there are no red emitters that satisfy all of these requirements.**

Narrowband red-emitting nanophosphors made at SNL

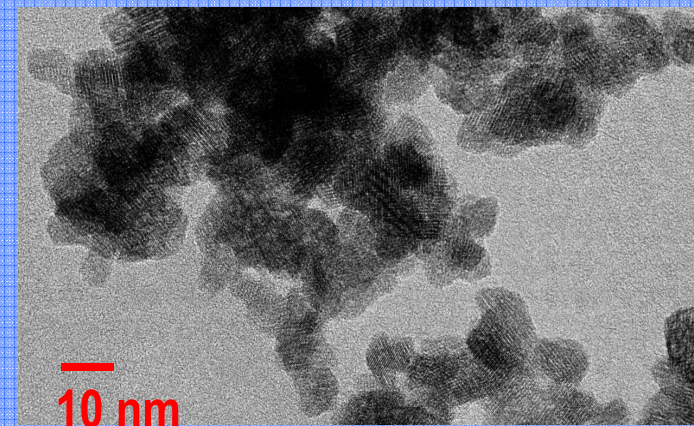
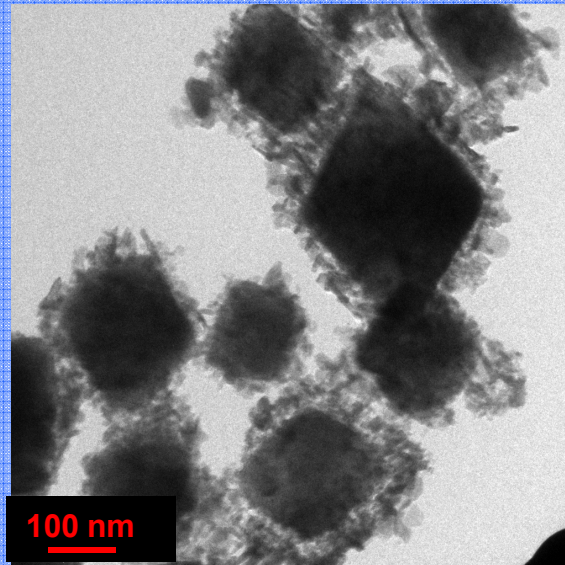
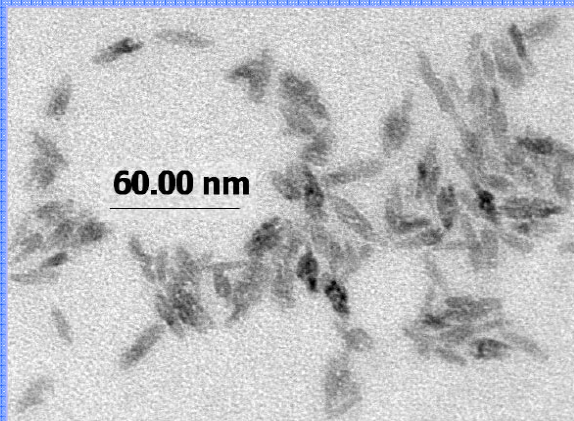
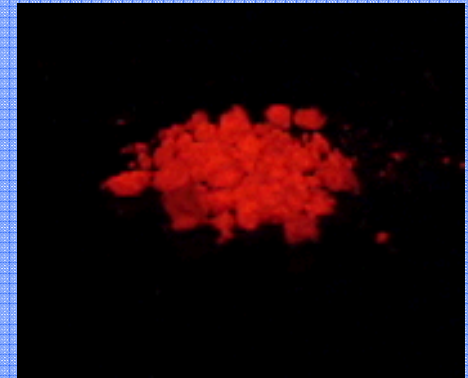
Zircon structure:
 $\text{YVO}_4:\text{Eu}^{3+}$



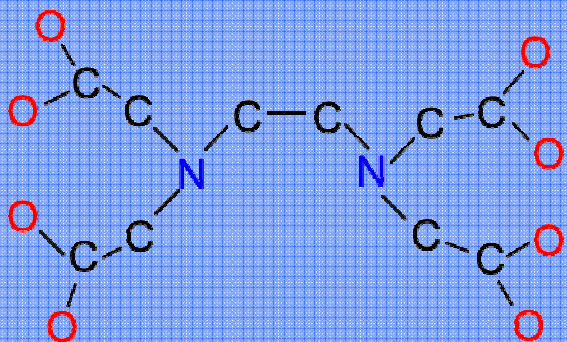
Pyrochlore:
 $\text{K}_{1-2x}\text{GdTa}_2\text{O}_{7-x}:\text{Eu}^{3+}$



Pyrochlore:
 $(\text{A},\text{RE})\text{Ta}_2\text{O}_7:\text{Eu}^{3+}$
 $\text{A} = \text{K}^+, \text{Rb}^+; \text{RE} = \text{Lu}, \text{Y}$



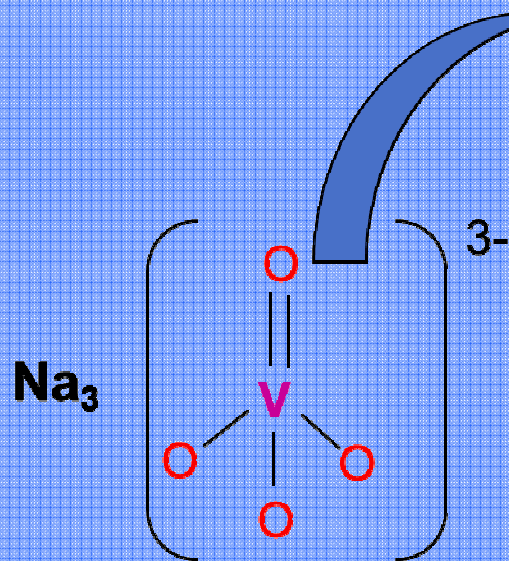
Synthesis of $\text{YVO}_4:\text{Eu}^{3+}$ nanoparticles



EDTA—complexes metals, caps nanoparticles

+

Metal salt (Sr, Ca, Y)
Dopant metal salt
(rare earth)



Sodium vanadate

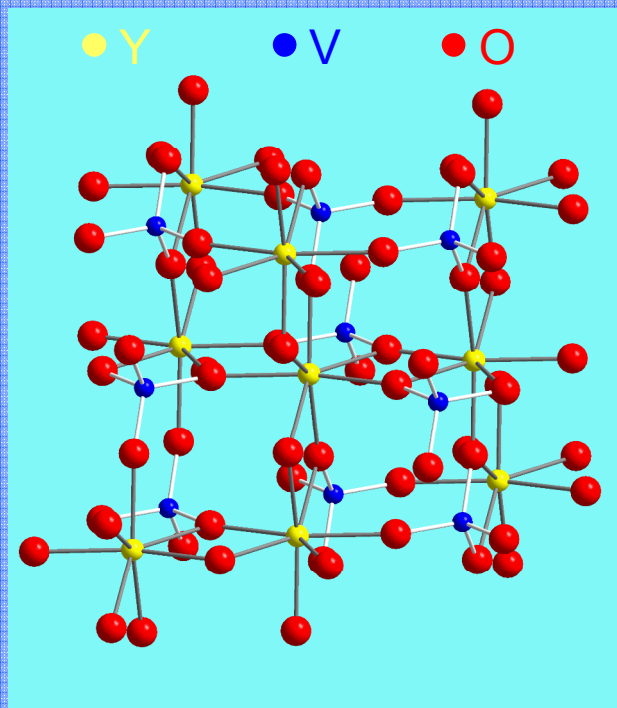
Dissolve in water

Dissolve in water

pH~9

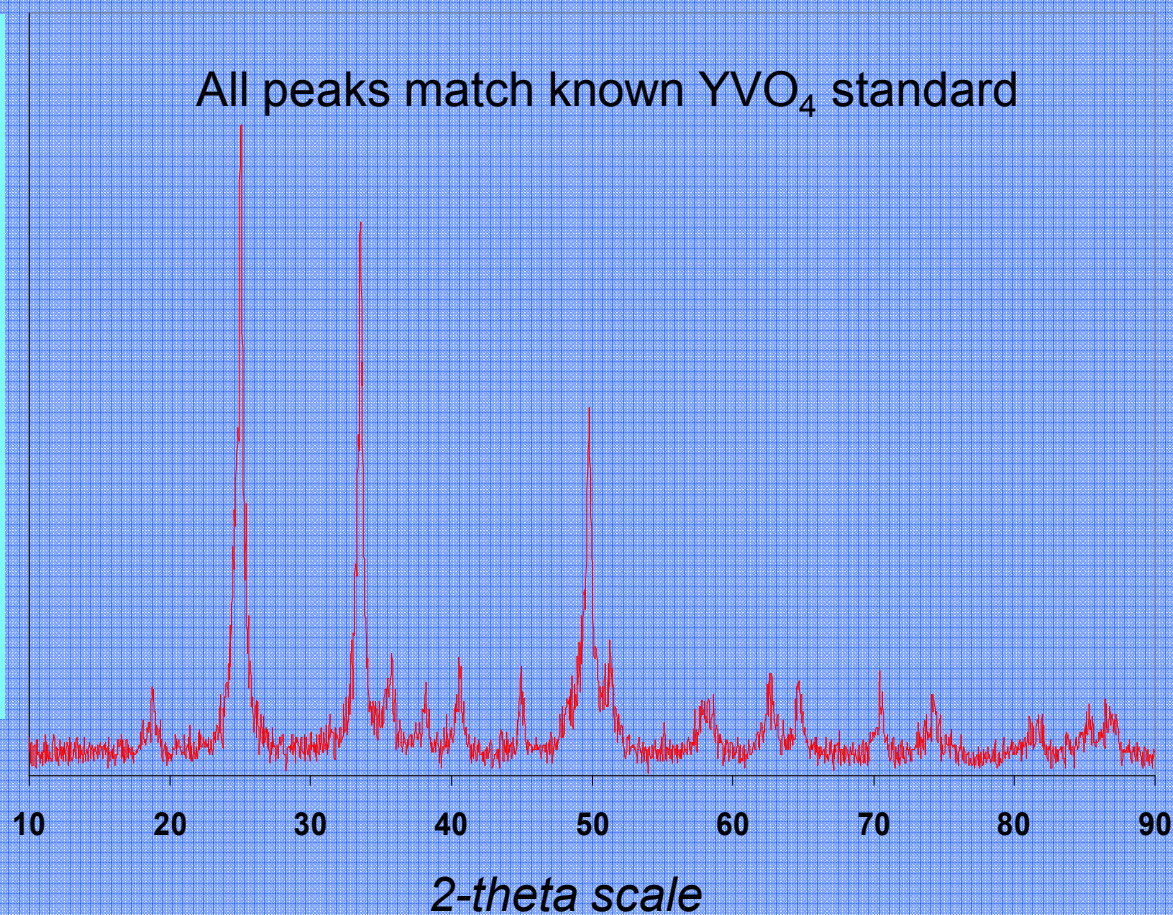
Nano-YVO₄ structure characterization

YVO₄

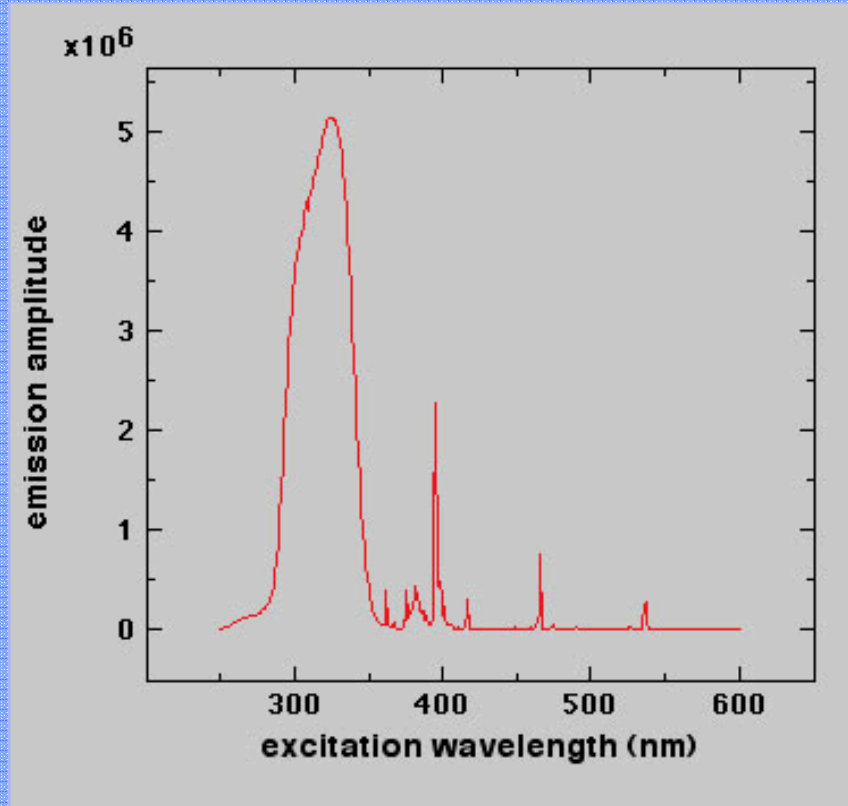


Zircon-type structure
Regular VO₄ sites
8-coordinate Y site

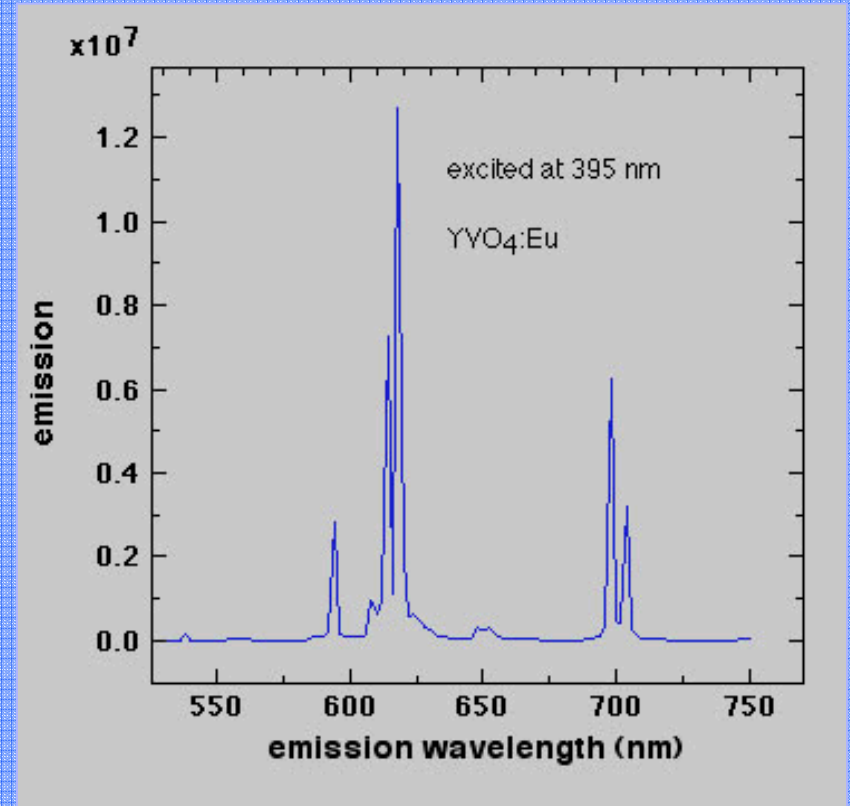
Nano-YVO₄



Nano-YVO₄:Eu³⁺ has bulk PL properties

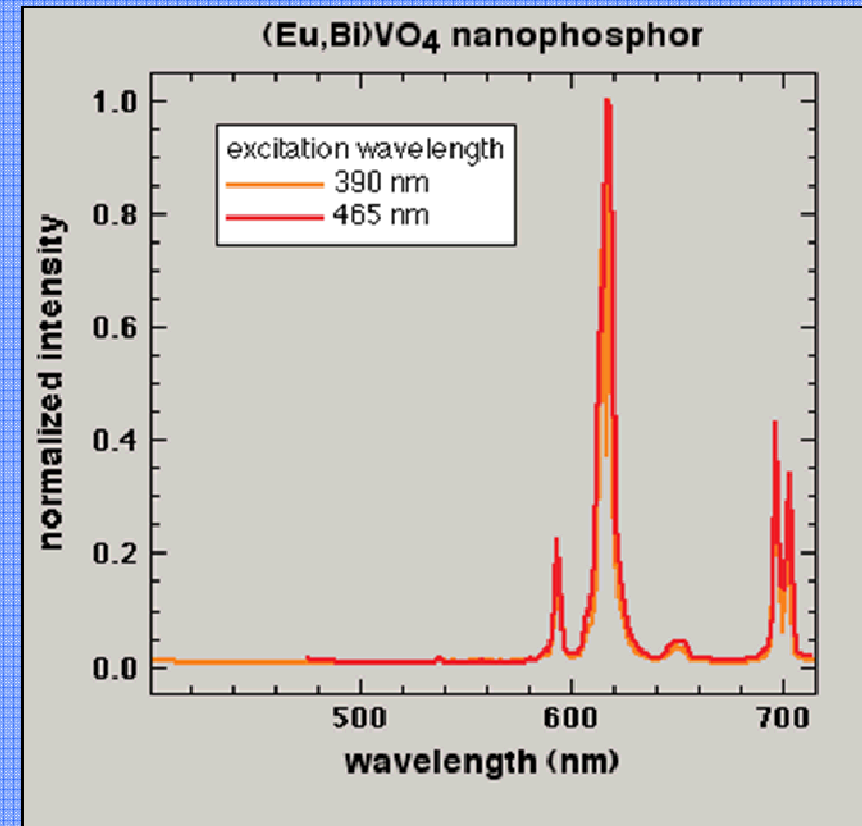
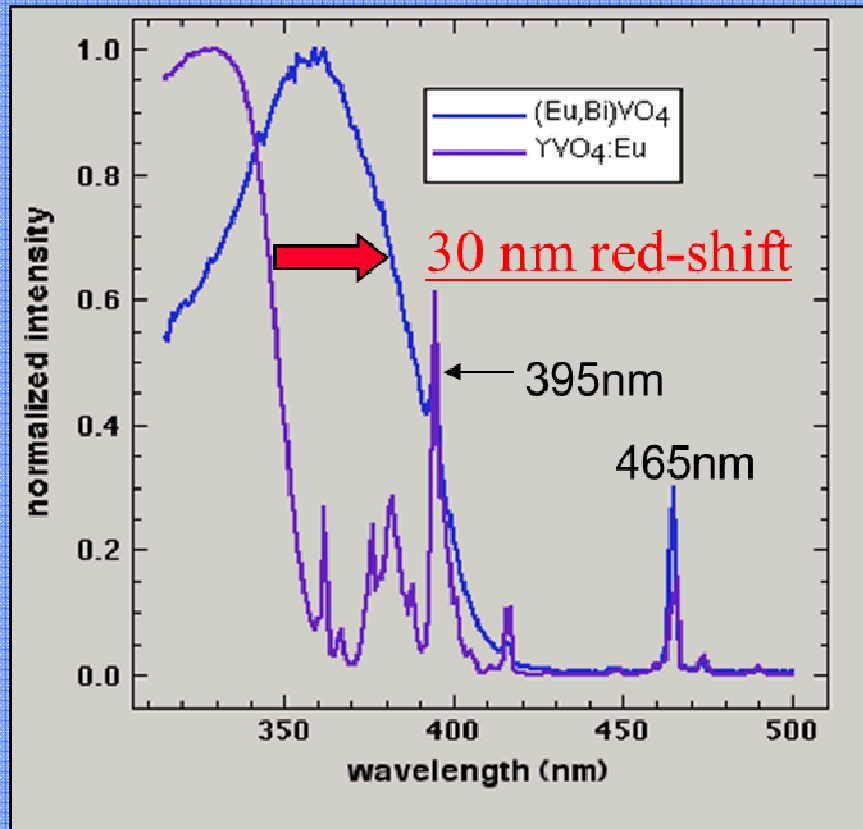


- Absorption at 395 and 465 nm is ideal for InGaN LED excitation.
- Strong UV absorption.



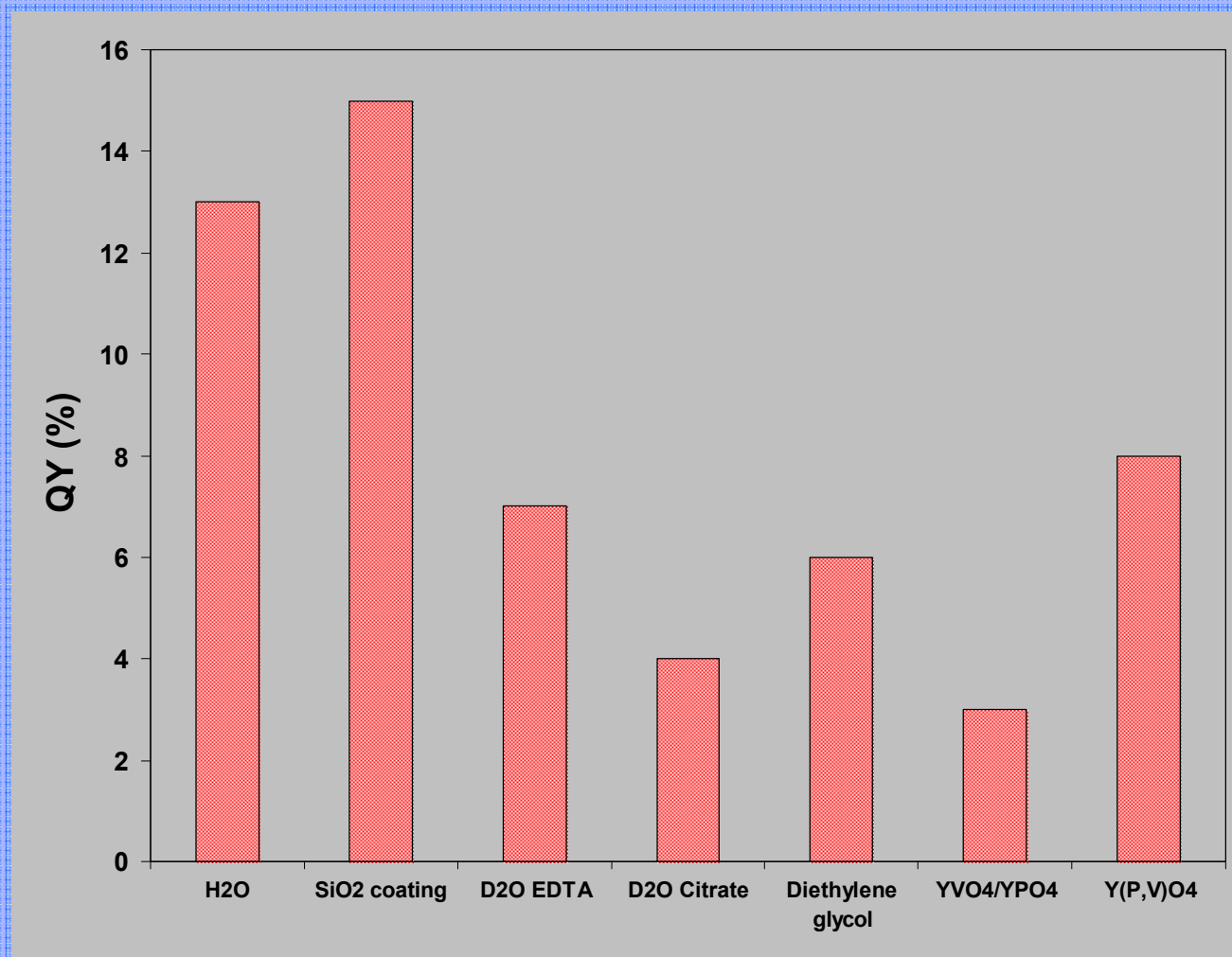
- Eu³⁺ emits at ~619 nm, as in bulk YVO₄.
- FWHM ~4 nm.

Co-doping with bismuth yields efficient excitation



- Bi³⁺ transfers energy to Eu³⁺.
- Enhanced 395 & 465nm absorptions.
- Red-shifted UV absorption.
- Narrowband red emission is preserved.

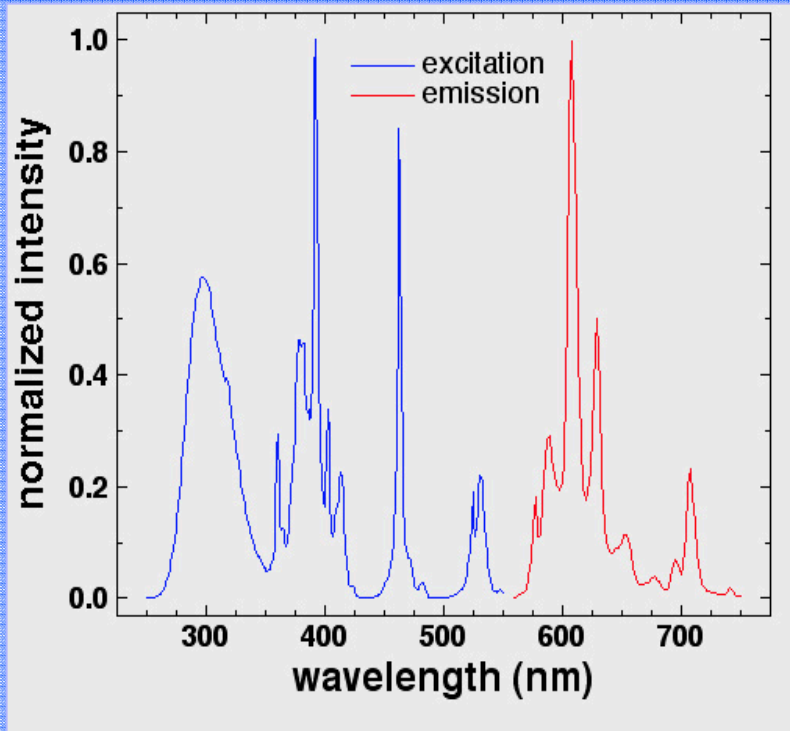
The QY of nano-YVO₄:Eu³⁺ is affected by synthesis variations



- We increased the QY to 15% (near-UV excitation) with SiO₂ coatings.
- QY's of 15-25% reported for these materials under UV excitation.
- The QY is currently too low for SSL applications.

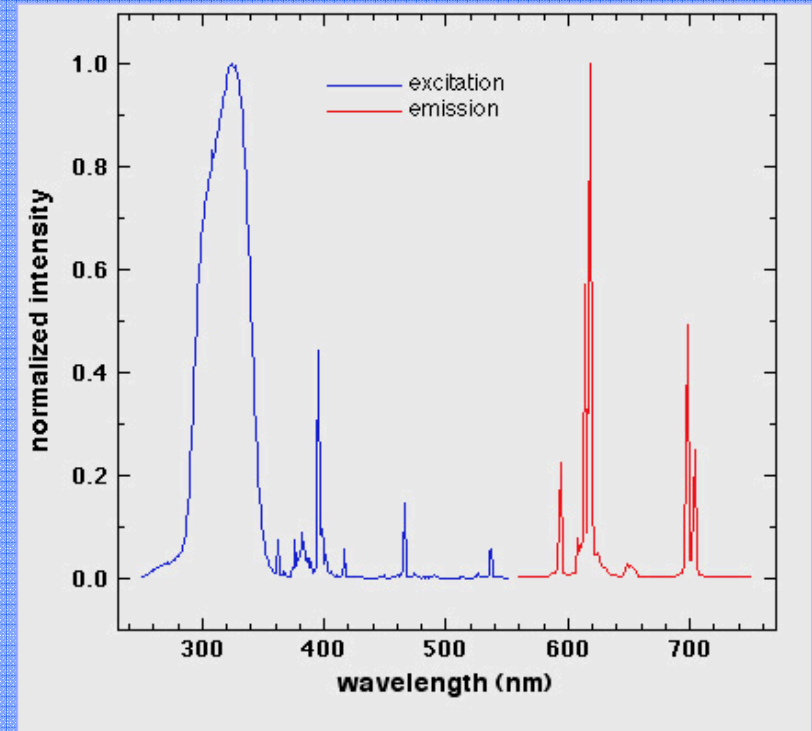
Pyrochlore tantalates have advantages over YVO_4

Pyrochlore: $\text{K,YTa}_2\text{O}_7:\text{Eu}^{3+}$



- Weak UV absorption.
- Stronger blue absorption and less deep red emission than YVO_4 .

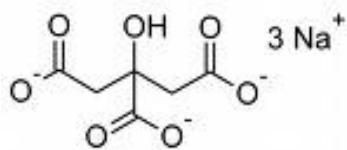
Zircon structure: $\text{YVO}_4:\text{Eu}^{3+}$



- Strong UV absorption.
- Deep red emission >700 nm.

Hydrothermal synthesis of pyrochlore tantalate nanophosphors

Synthesis: RE(citrate)



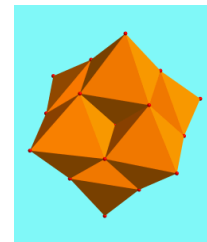
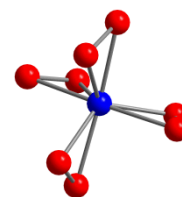
precip

$\text{OH}^-/\text{pH}-13.5$



dissolve

Anionic Ta precursors



$[\text{Ta}(\text{O}_2)_4]^{3-}$, $[\text{Ta}_6\text{O}_{19}]^{8-}$

RE=La, Gd, Lu, Y, Eu

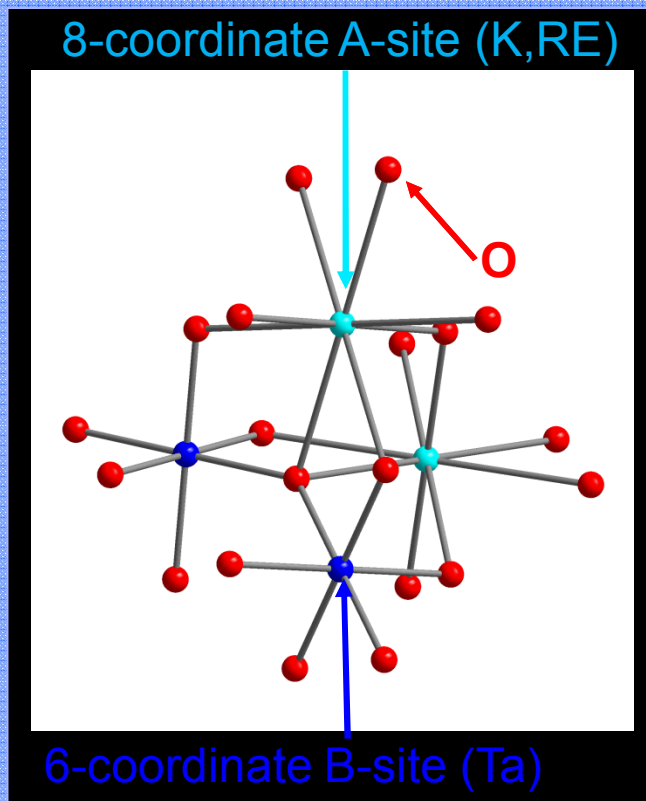
Hydrothermal
treatment
at 220°C



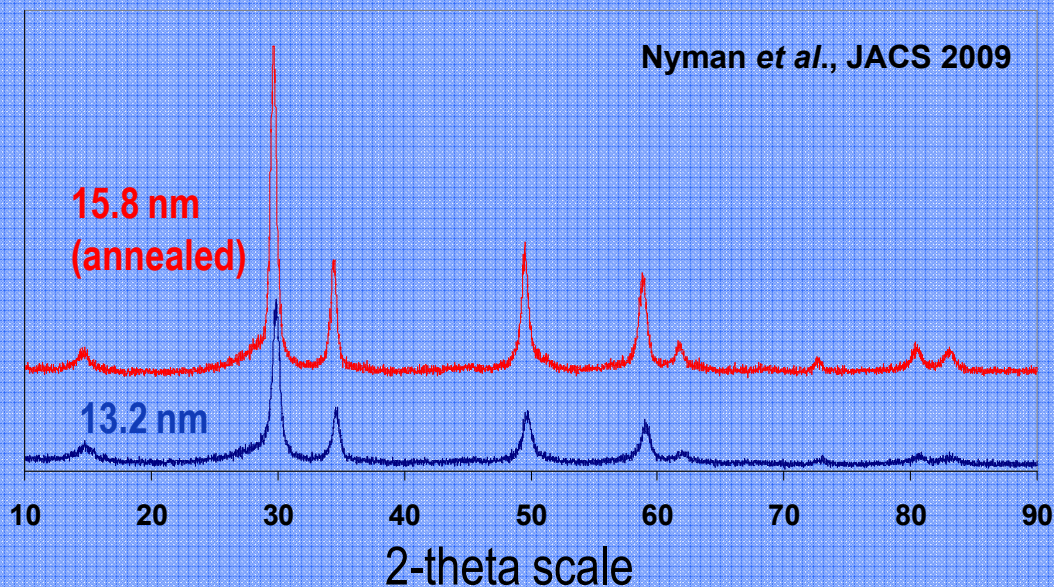
Pyrochlores: $(\text{RE},\text{A})\text{Ta}_2\text{O}_7$; A=alkali
 $(\text{RE},\text{A})_{1-2x}\text{Ta}_2\text{O}_{7-x}$

Pyrochlore nanophosphor characterization

Pyrochlore structure: AB_2O_7



$(K,RE)Ta_2O_7:Eu$; RE = Lu or Y



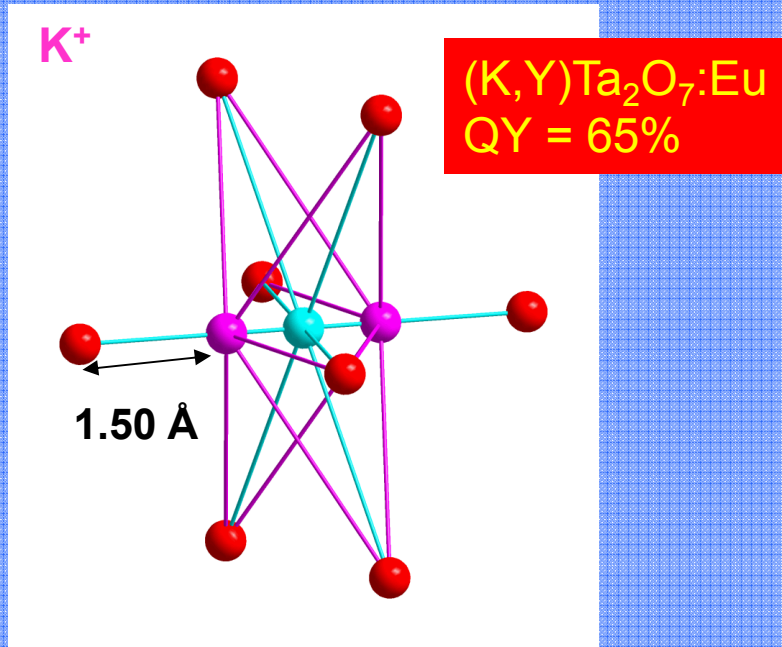
RE	r (Å)	QY(%)	τ (ms)
Lu	0.85	67	1.6
Y	0.893	65	1.5

- The Lu and Y pyrochlores have comparable QYs and lifetimes under blue excitation.

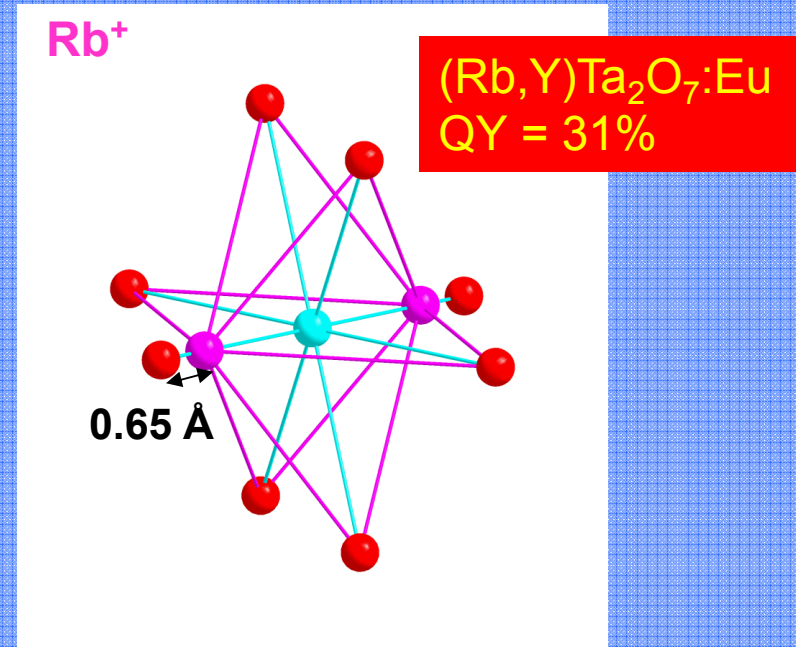
Pyrochlore tantalates offer flexibility in lattice structure.

Alkali substitutions can alter the symmetry of the A site and the alkali-O bond lengths.

● = O ● = Alkali ● = Ta



- 7-coordinated A site



- 8-coordinated A site

- The QY increases with increasing A-site distortion.

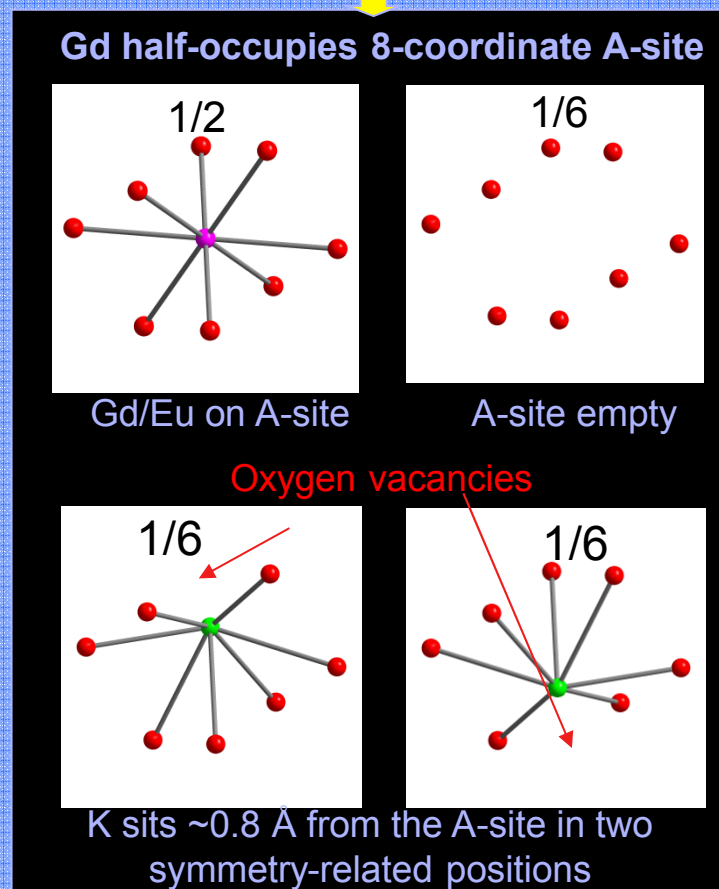
The addition of Gd increases the desired lattice distortions



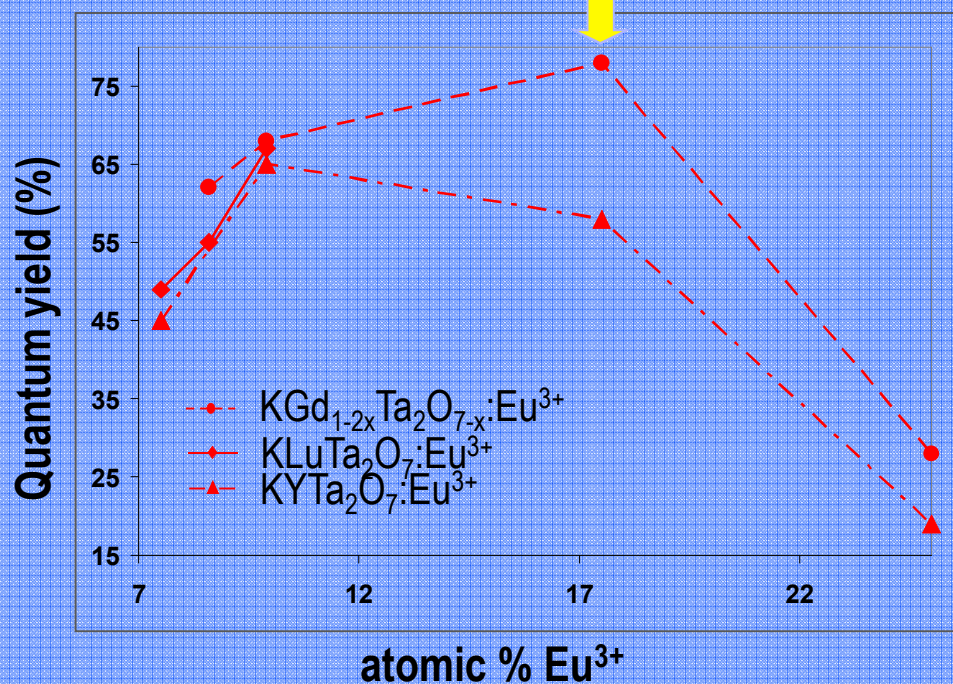
1. Gd creates oxygen vacancies in pyrochlores.

2. These vacancies create lattice distortions.

4. The QY of the Gd pyrochlore is higher than that of the Lu & Y pyrochlores.



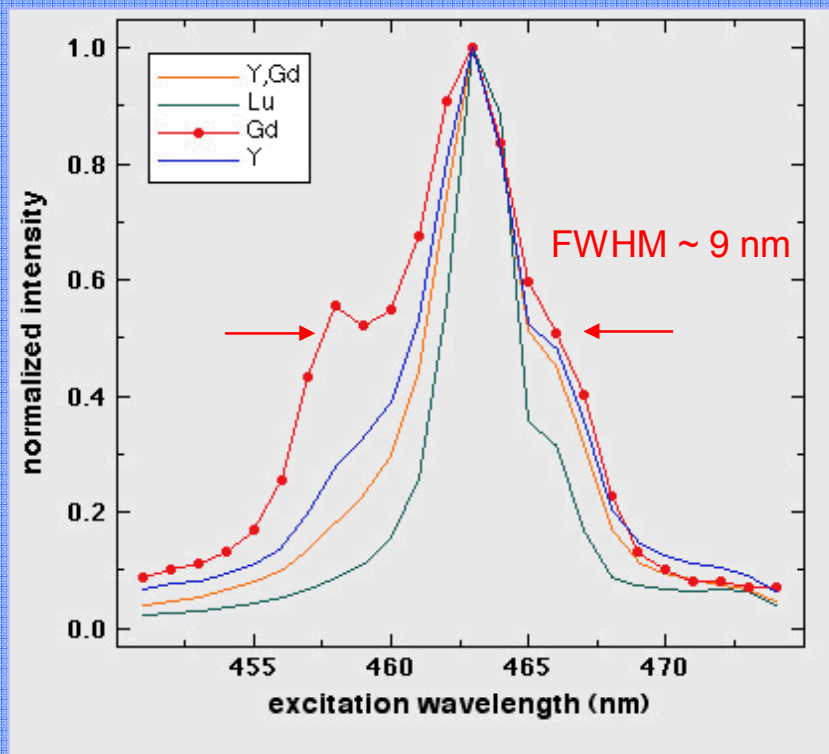
QY = 78%



3. These lattice distortions affect the Eu emission.

Nyman *et al.*, JACS 2009

The blue absorption linewidth is broadest in the Gd pyrochlores

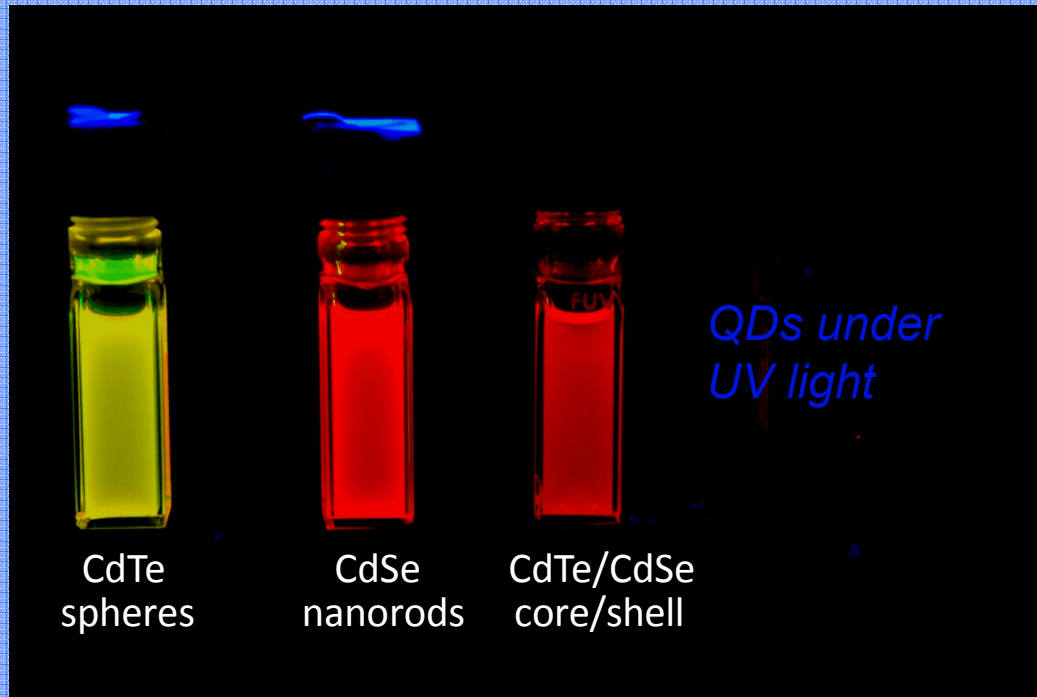


- Broad blue absorption enables excitation with a range of LED wavelengths.
- A transparent, ~1 mm thick material would enable enough blue absorption.

Summary of Eu³⁺-doped nanophosphors

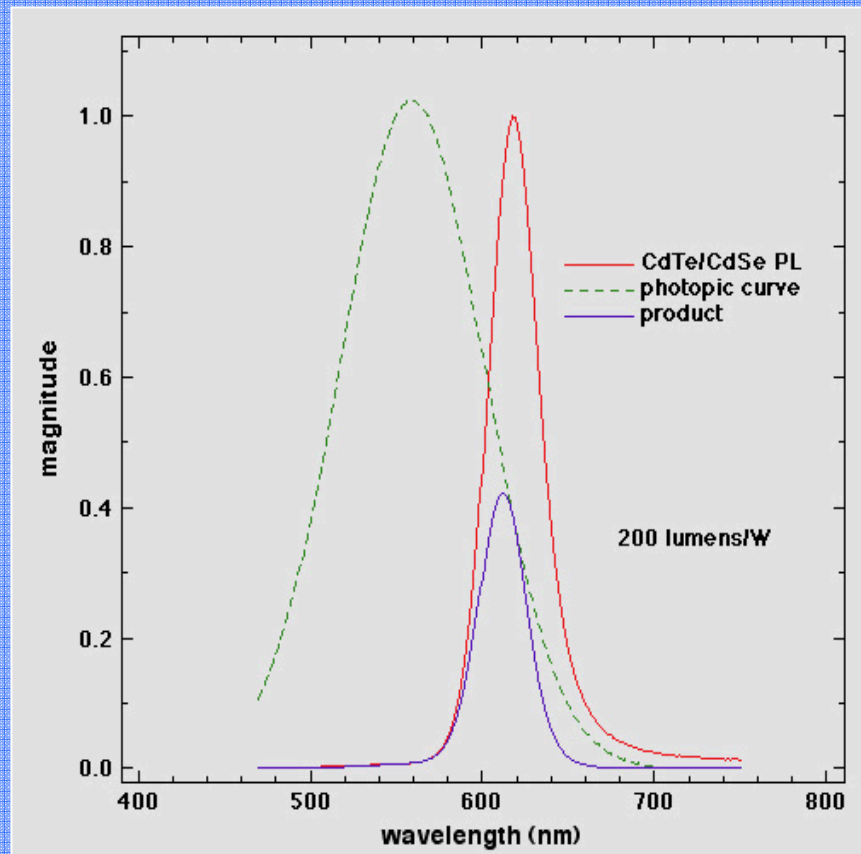
- Narrow band red emission at ~610 nm is ideal for warm white lighting.
- QY as high as 78%.
- Only 3% thermal quenching at 130°C.
- Narrow blue excitation linewidth (~4-9 nm) with a small cross section due to the low oscillator strength of Eu³⁺.
- Developing these materials into transparent wafers for SSL.

CdTe-based quantum dots



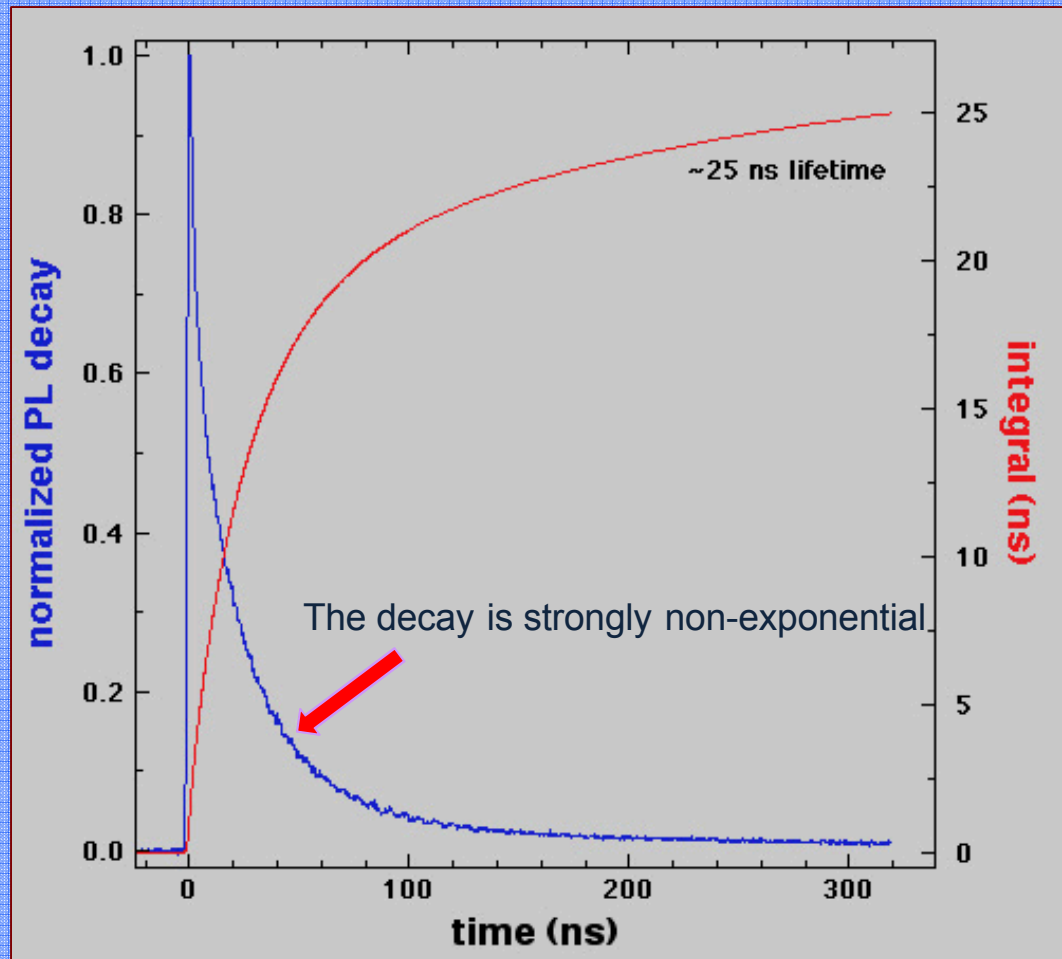
CdTe-based quantum dots can have narrow red emission

- CdTe/CdSe core/shell QD emission (FWHM~33 nm) gives a high LER (200 lm/W_{emitted}).



- The red emission of these materials is 2.5X narrower than that of the nitridosilicates used in warm white LEDs.

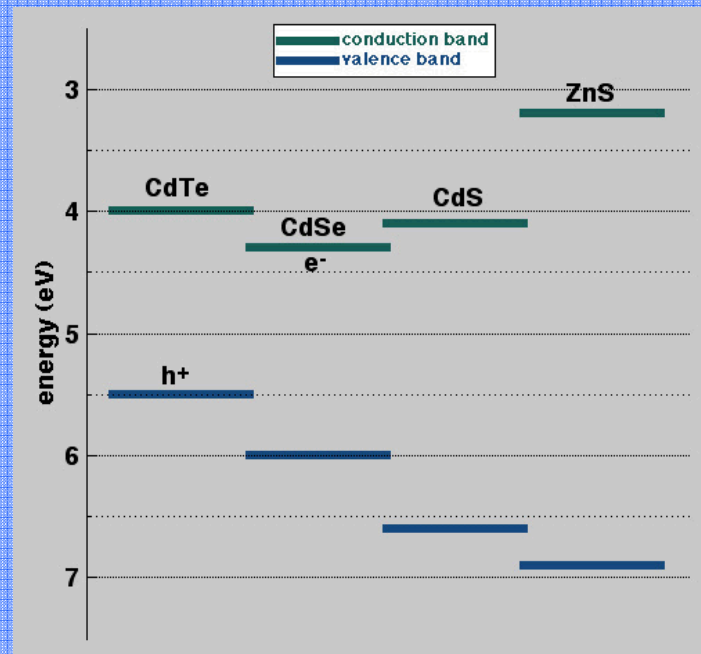
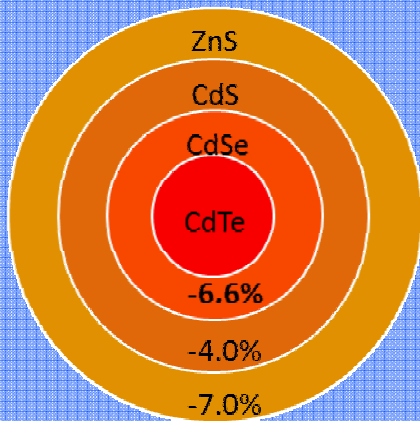
The fast PL decay of CdTe QDs indicates high oscillator strength



- The absorption cross section is consequently very large.

Our goal is a photostable strain-graded heterostructure

Type II core/shell QD



- h^+ is confined to the core
- e^- is in the CdSe shell.

Type I (mismatch)	Type II (mismatch)
CdSe/CdS (-4%)	CdTe/CdSe (-6.6%)
CdSe/ZnS (-11%)	ZnTe/ZnSe (-8.9%)
CdSe/ZnSe (-6.3%)	ZnTe/CdSe (-0.3%)
CdTe/ZnSe (-13.9%)	ZnTe/CdTe (+5.0%)
CdS/ZnS (-7.0%)	

- Each shell adds a compressive strain to the underlying heterostructure.

Synthesis of CdTe cores

CdO + Oleic Acid + ODE

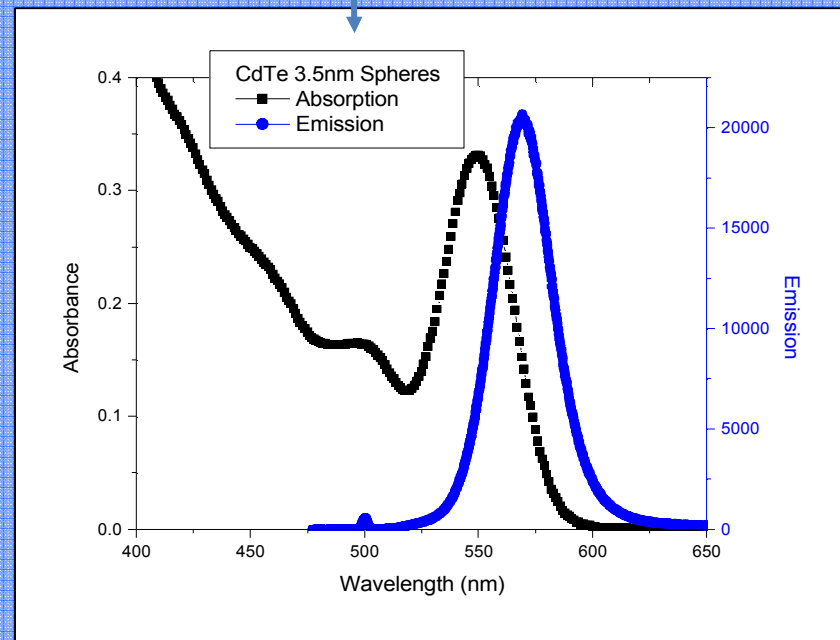
Heat to 280 °C

Clear Cd Precursor

TOP/Te + ODPA + ODE

React at 260 °C

- Yields ~3.5 nm spherical zinc-blende cores.
- Emission FWHM of 25-30 nm

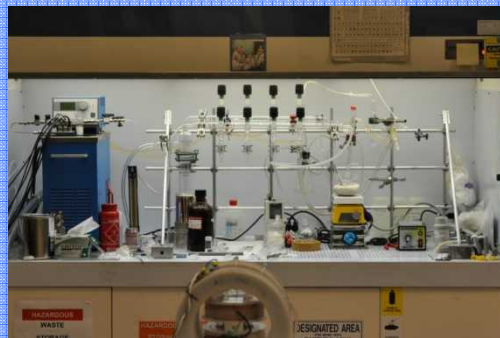


Strong self-absorption



Precursor synthesis

Precursor and QD synthesis



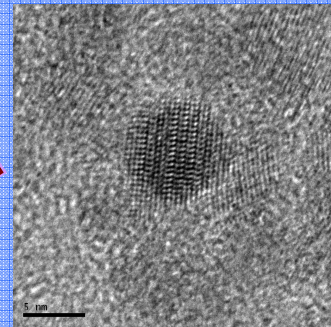
Selective Ion Layer Absorption and Reaction for CdSe shells

Se precursor: TOP/Se + ODE

Cadmium Oleate: CdO + Oleic Acid + ODE

Multiple injections
at 240-260 C

CdTe spherical cores

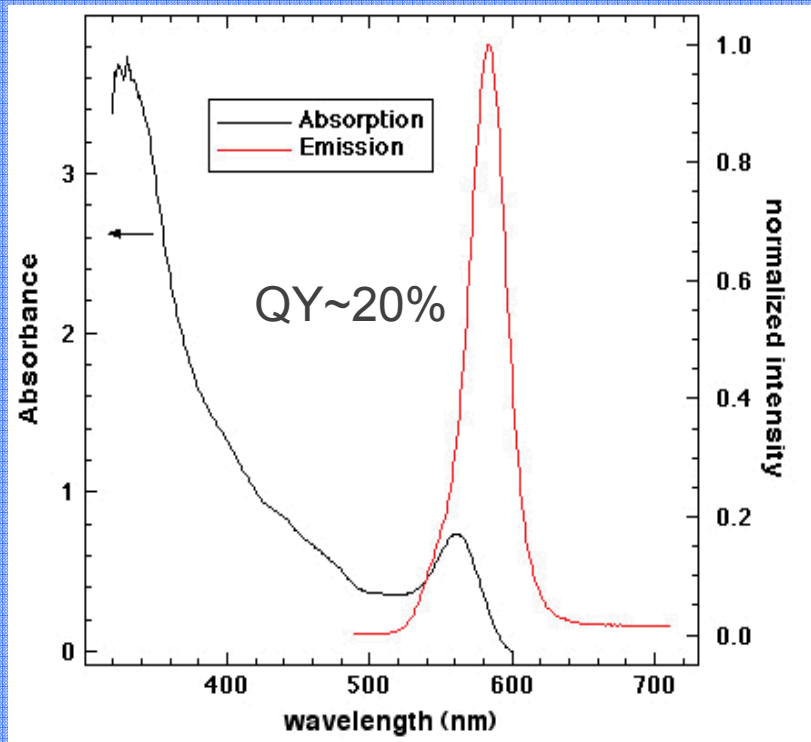


CdTe cores coated with 10 layers of CdSe

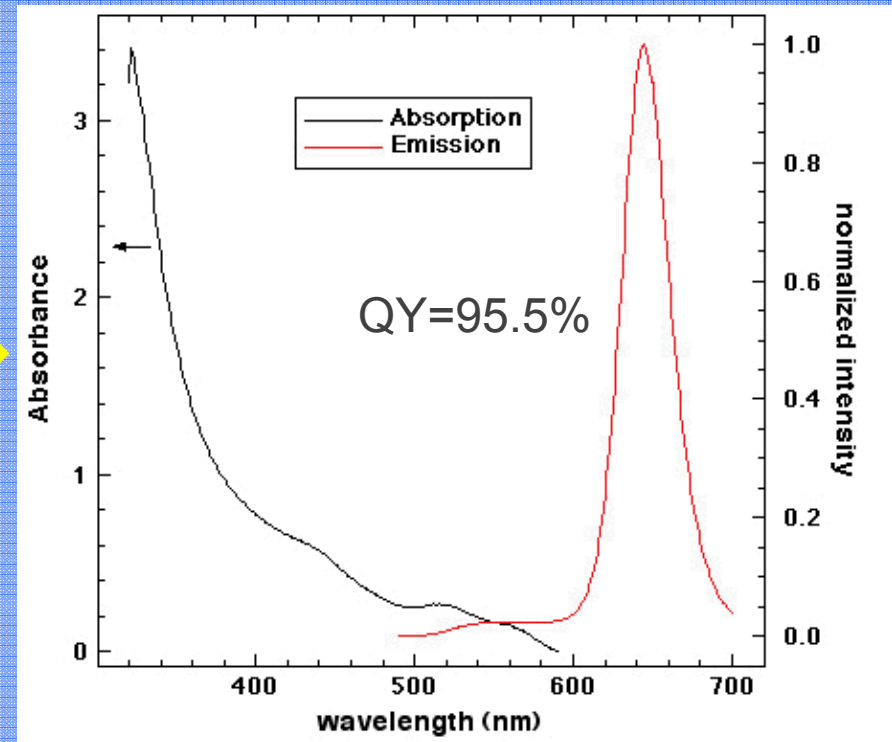
- Oleic Acid is a weakly binding ligand, so growth occurs on all facets to yield spherical zinc-blende shells.

CdSe shells reduce the self absorption

Yellow-emitting CdTe QDs

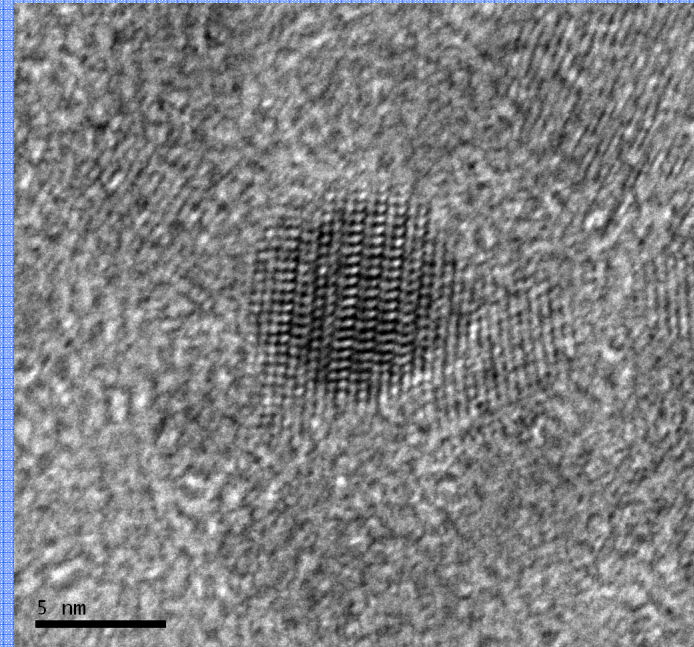
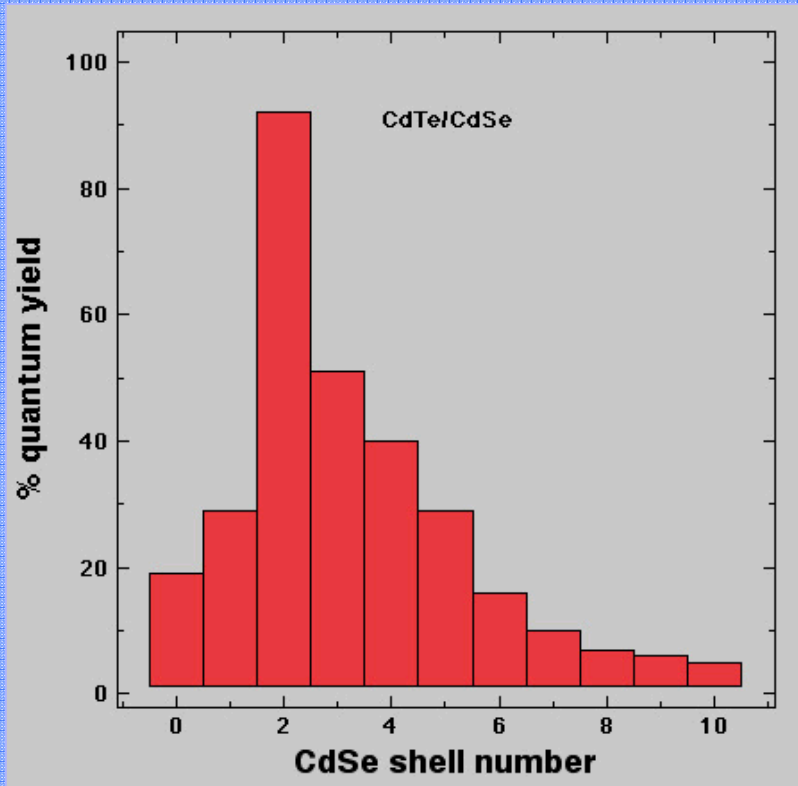


Red-emitting CdTe/CdSe core shell QDs



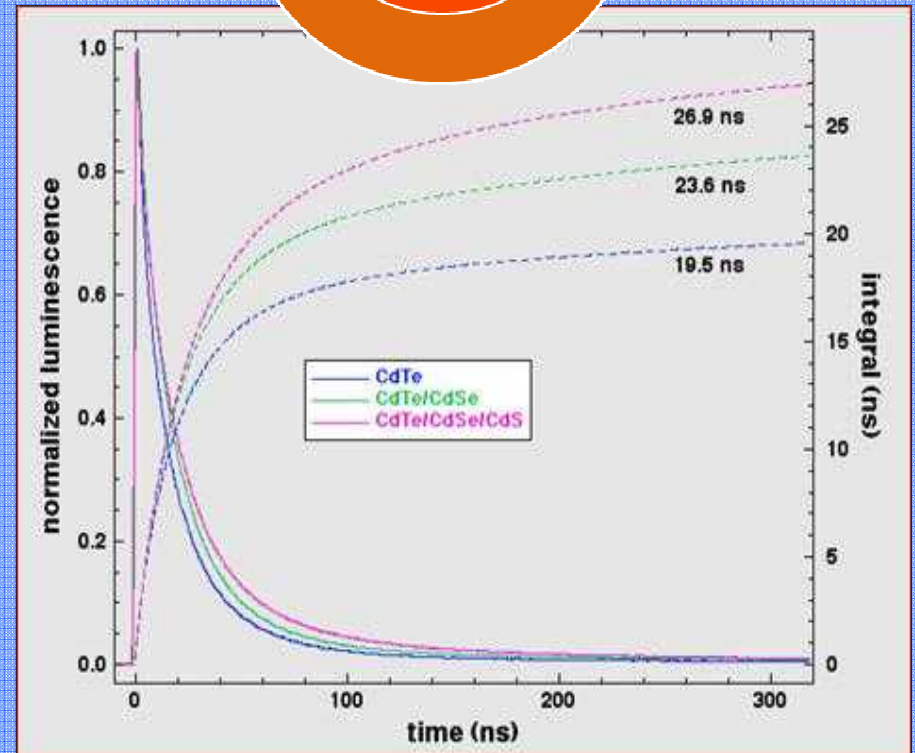
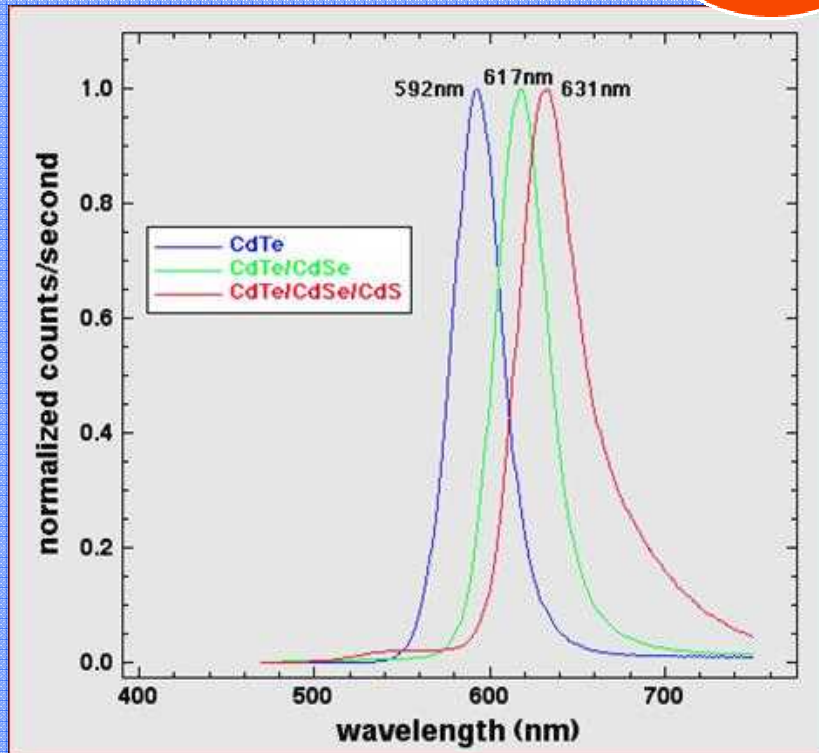
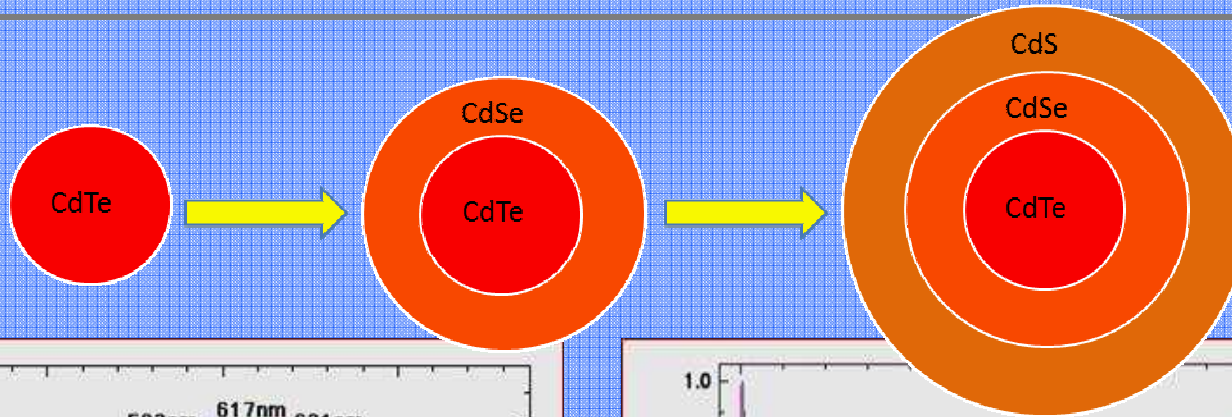
- The emission/absorption overlap can be reduced by coating with CdSe shells to create a Type II heterostructure.

The QY is a strong function of the CdSe shell thickness



- The QY is optimal at 2 layers of CdSe.
- We can now routinely get QYs of 70% through the 5th shell.

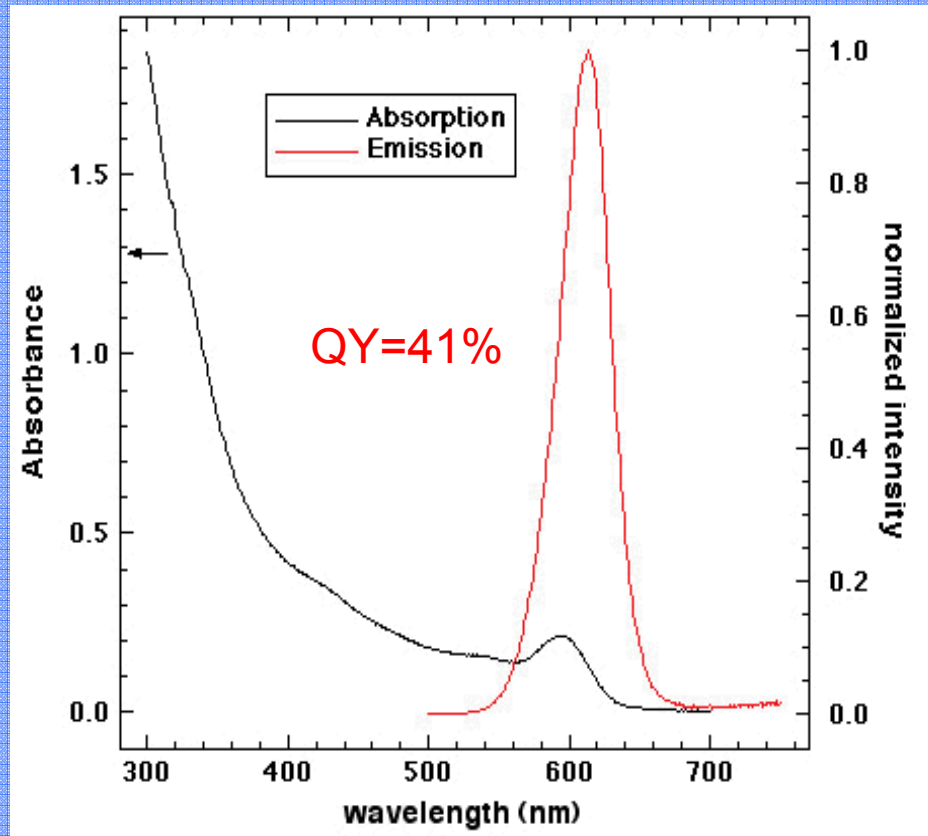
Effect of shell on PL emission and lifetime



- The increased lifetime is due to a reduction in the electron/hole wavefunction overlap.

We have synthesized CdTe/2CdSe/CdS/5ZnS QDs

- Even with all of these shells, the emission is ideally located for SSL.



LER=227 lm/W_{emitted}

- The QY is expected to increase if an intermediate alloy layer is formed between the core and shell interfaces to create a graded composition that would reduce the compressive strain.

The ZnS shells improve the thermal stability

$$TQ = [1 - (QY_{100^{\circ}C} / QY_{25^{\circ}C})] \times 100$$

Sample	TQ @ 100°C	Recovery
CdTe	73%	90%
CdTe/2CdSe	39%	92%
CdTe/2CdSe/CdS	38%	92%
CdTe/2CdSe/CdS/5ZnS	39%	100%
CdSe/ZnSe	87%	67%

- Future work will focus on the mechanisms of thermal quenching.

Summary of CdTe – based quantum dots

- SILAR has been used to create a variety of spherical QD heterostructures:
 - CdTe/CdSe
 - CdTe/CdSe/CdS
 - CdTe/CdSe/CdS/ZnS
 - Shell thickness dependent optical properties and recombination dynamics were observed in all cases.
- QYs as high as 95% were observed and narrow emission peaks can be positioned to be of use for SSL if the CdTe cores are ~3.5 nm.
- Luminous efficacy of radiation as high as $227 \text{ lm/W}_{\text{emitted}}$.
- ZnS shell enables reversible thermal quenching at 100°C.
- Future work will focus on the reduction of lattice mismatch strain through alloying.

Acknowledgments

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 - Professor David Kelley, UC Merced
 - Dr. Xichen Cai, UC Merced