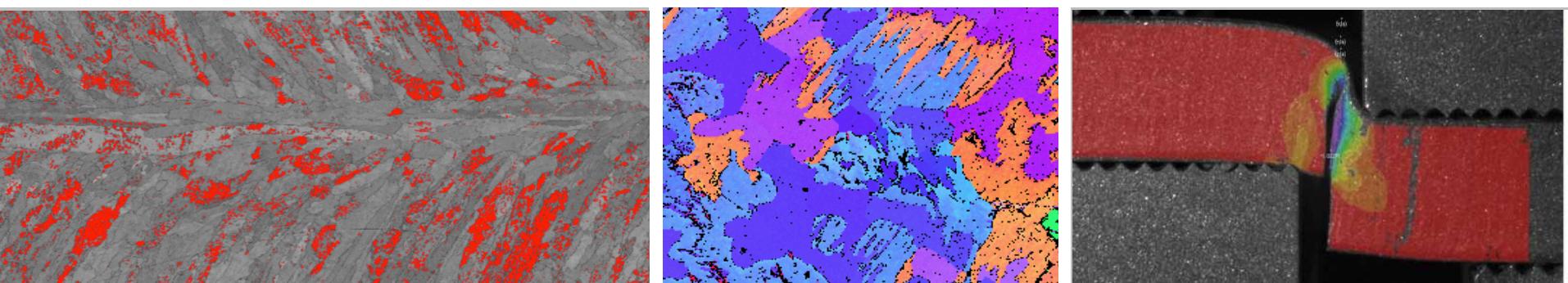


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Deformation-Induced Martensite Formation in Austenitic Stainless Steel Welds

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Sandia National Laboratories, Albuquerque NM

JOWOG – Joining Subgroup

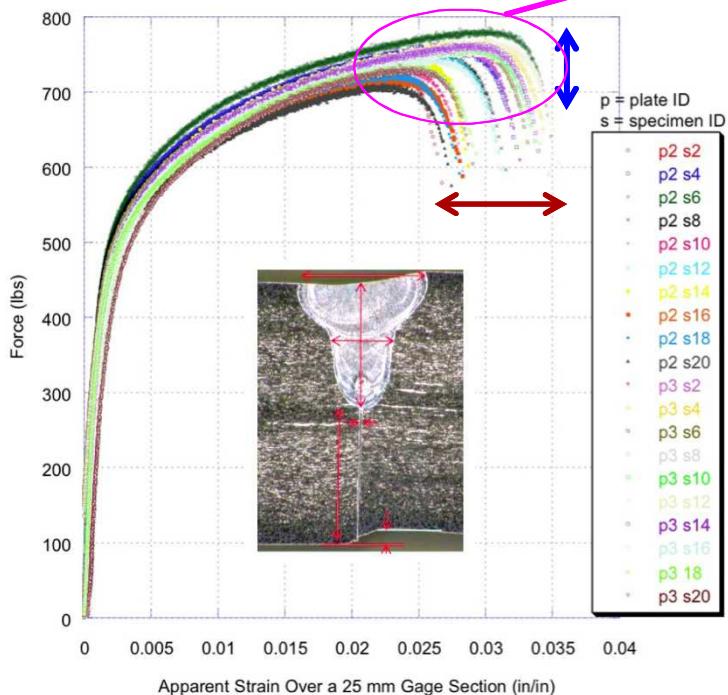
September 24, 2013



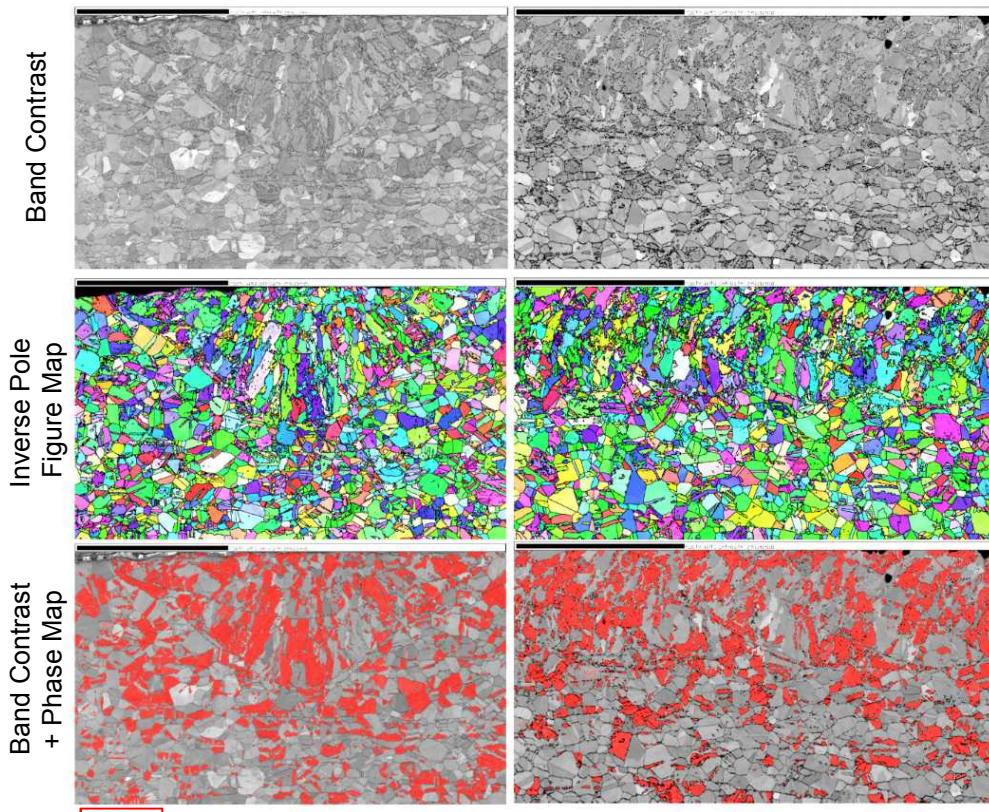
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Understanding Austenitic Stainless Steel Weld Mechanical Behavior Variation Requires Accurate Knowledge of Phase Distribution

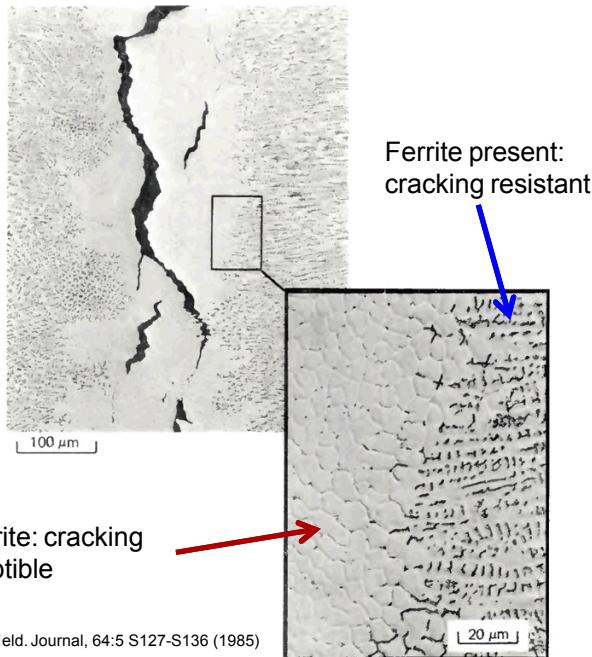
- 60-70% of mechanical behavior variation attributed to non-metallurgical factors (e.g., weld shape, joint geo., etc.)



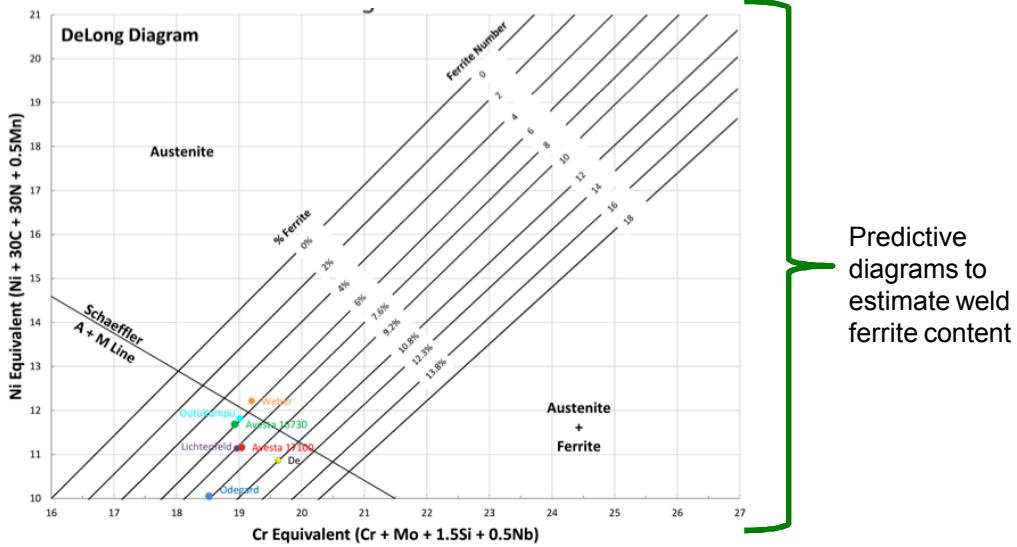
- Characterization of property variation due to metallurgical factors requires accurate characterization of phase distribution



Ferrite Plays Important Role in Austenitic Stainless Steel Welds

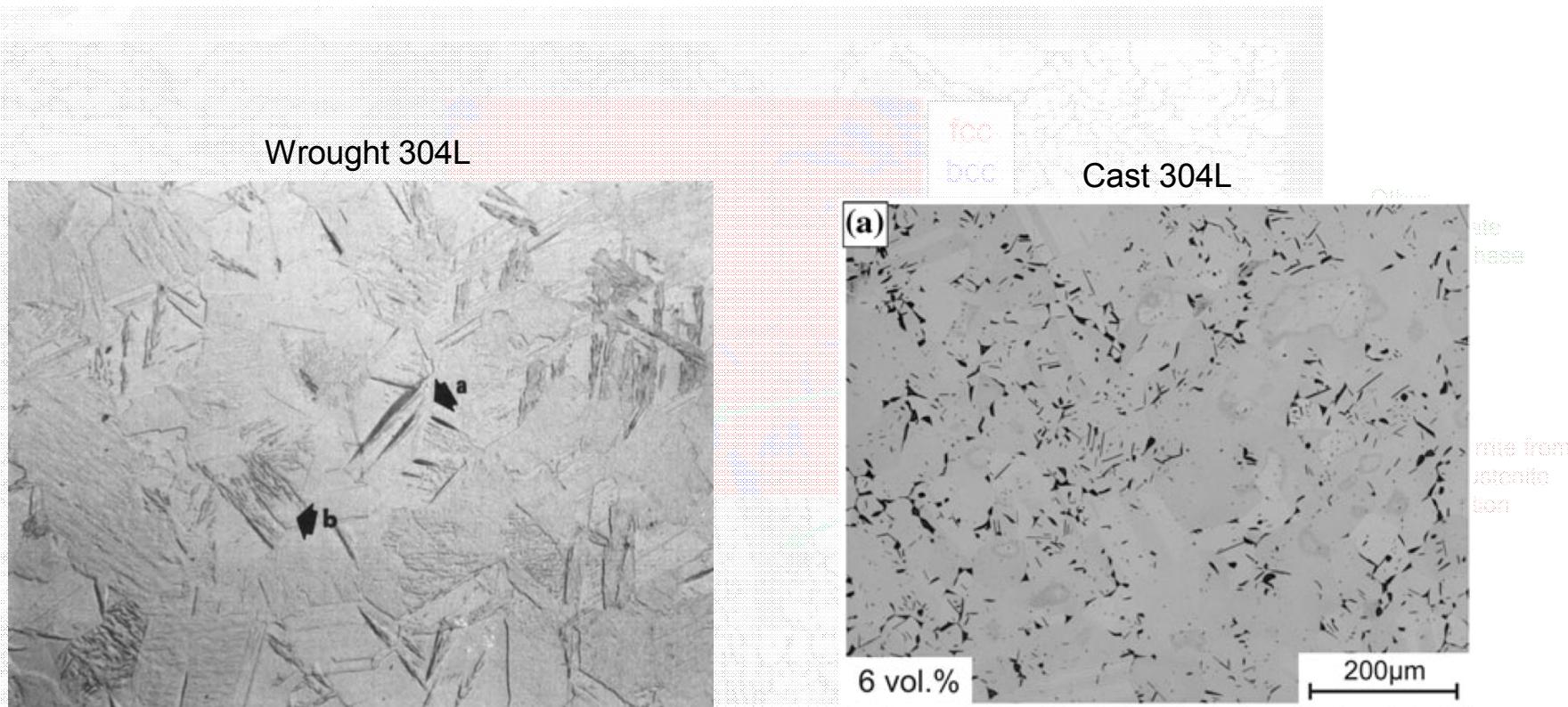


Lippold, J.C., Weld. Journal, 64:5 S127-S136 (1985)



- Ferrite in a weld influences:
 - Solidification cracking susceptibility
 - Mechanical response
 - Environmental cracking susceptibility
 - Cryogenic & elevated temperature properties
- Accurate metallographic determination of weld ferrite content of considerable interest

Metallographic Preparation for EBSD Phase Distribution Requires Additional Scrutiny

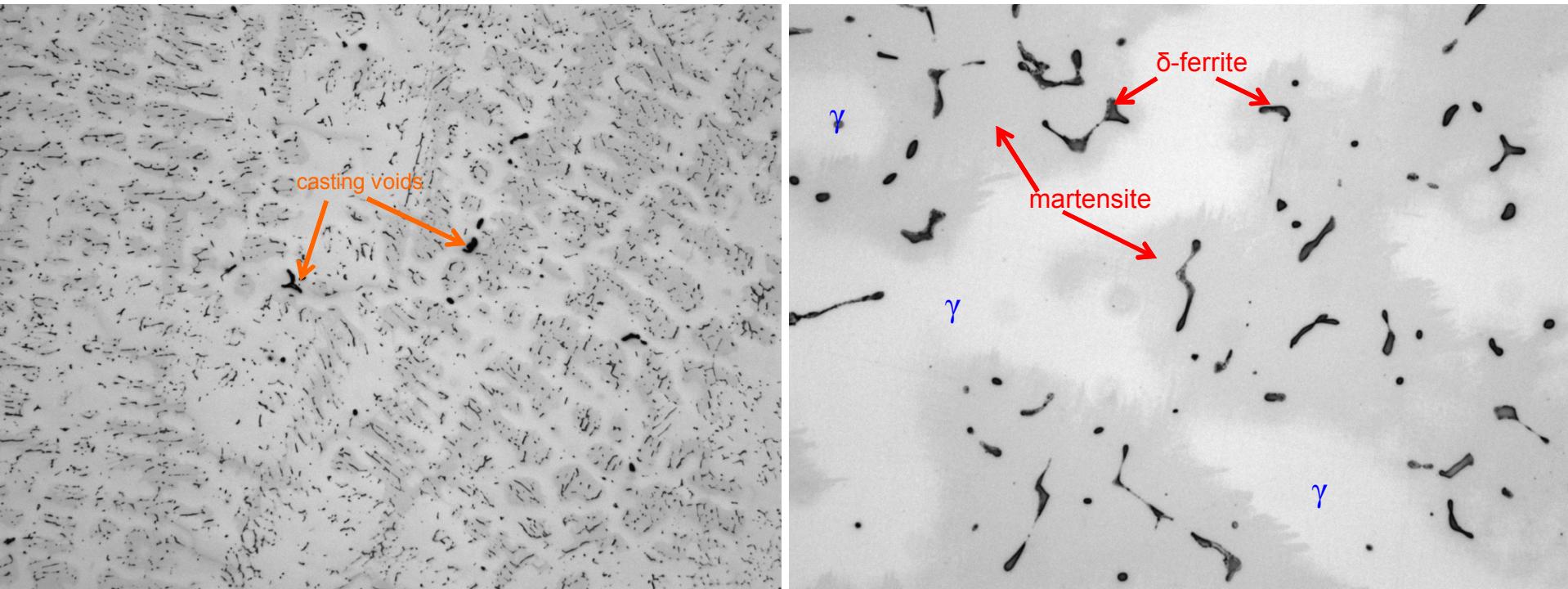


Odegard, B.C. Metallography, 7, 129-135 (1974)

Weber, S. et al. J. Mat Sci. 47:16, 6095-107 (2012)

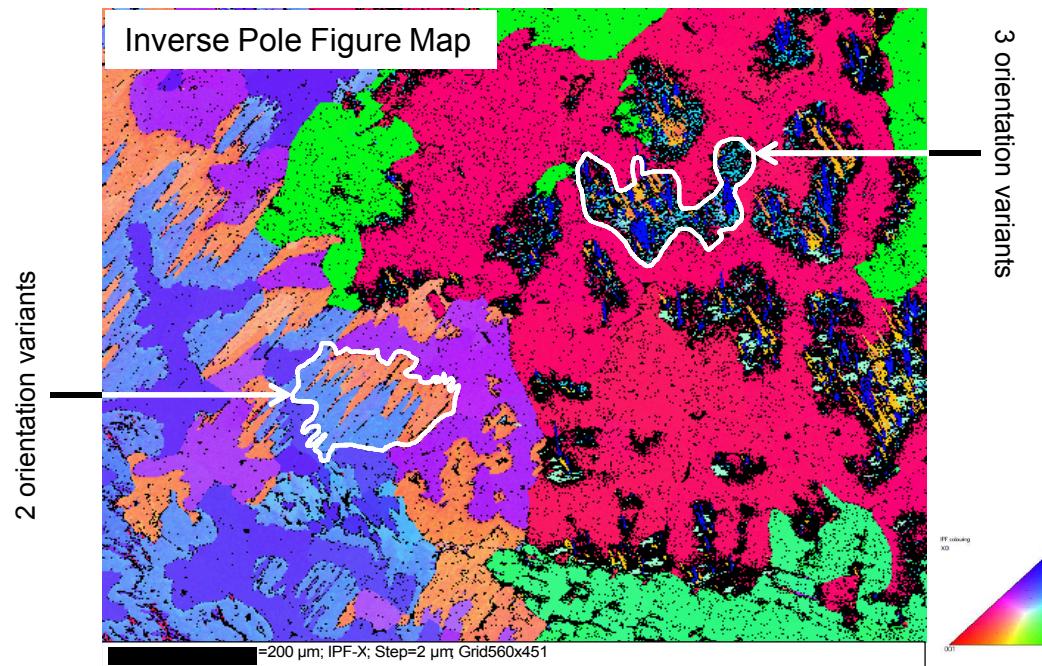
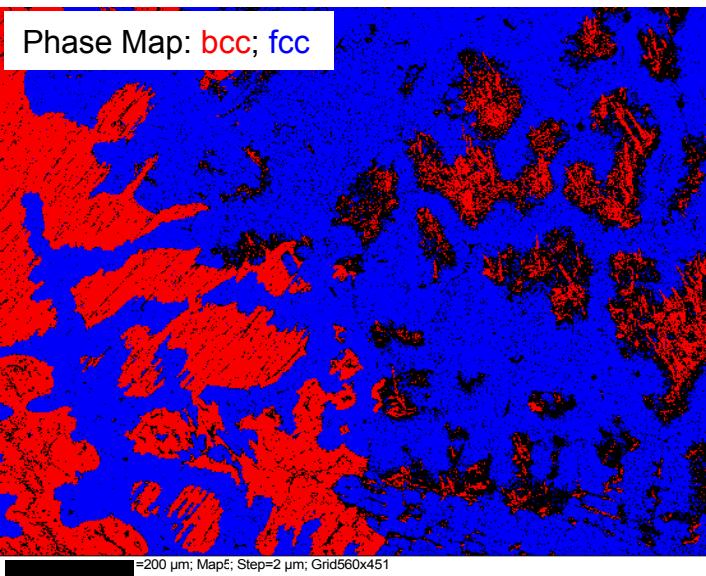
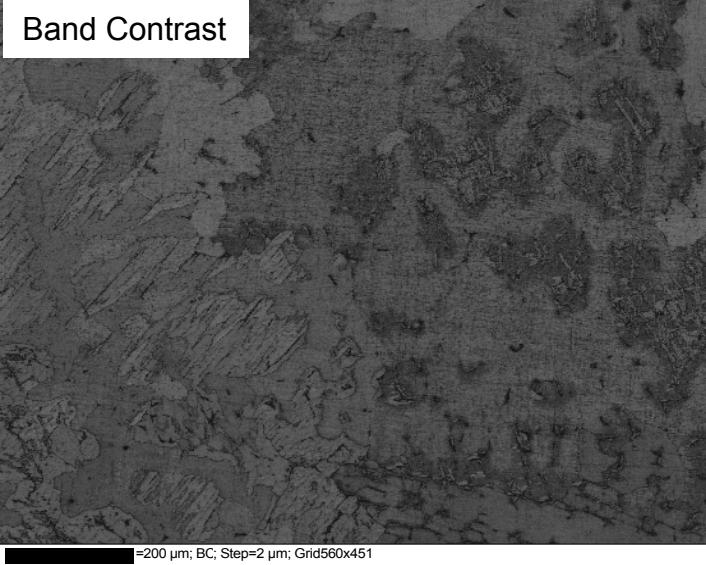
Mechanical Polishing-Induced Martensite Example: Cast SS 304

- Deformation during sample preparation leads to martensite formation preferentially around residual δ -ferrite in dendrite cores



- Sample prepared using conventional mechanical polishing and grinding
 - Lightly etched (electrolytic NaOH) to stain δ -ferrite

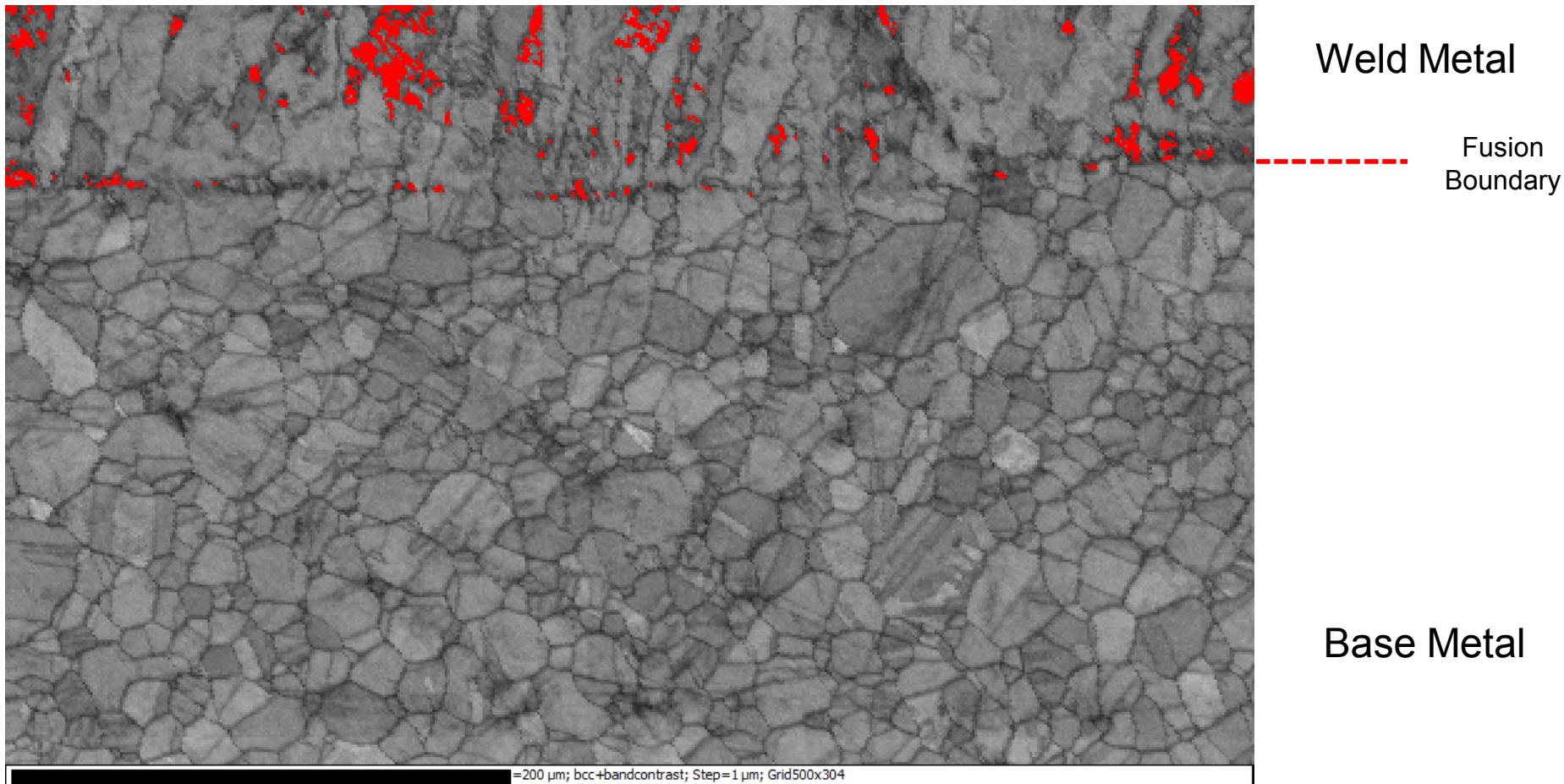
EBSD of Mechanically Polished Cast SS 304



- Intermediate gray phase observed in optical micrographs indexed as bcc phase
- EBSD measurements of mechanically polished cast 304 produces erroneous measurements of residual δ -ferrite

Fusion Boundary: Laser Weld – Bead on Plate

- Increased propensity for polishing-induced martensite in weld metal microstructure



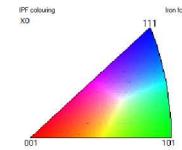
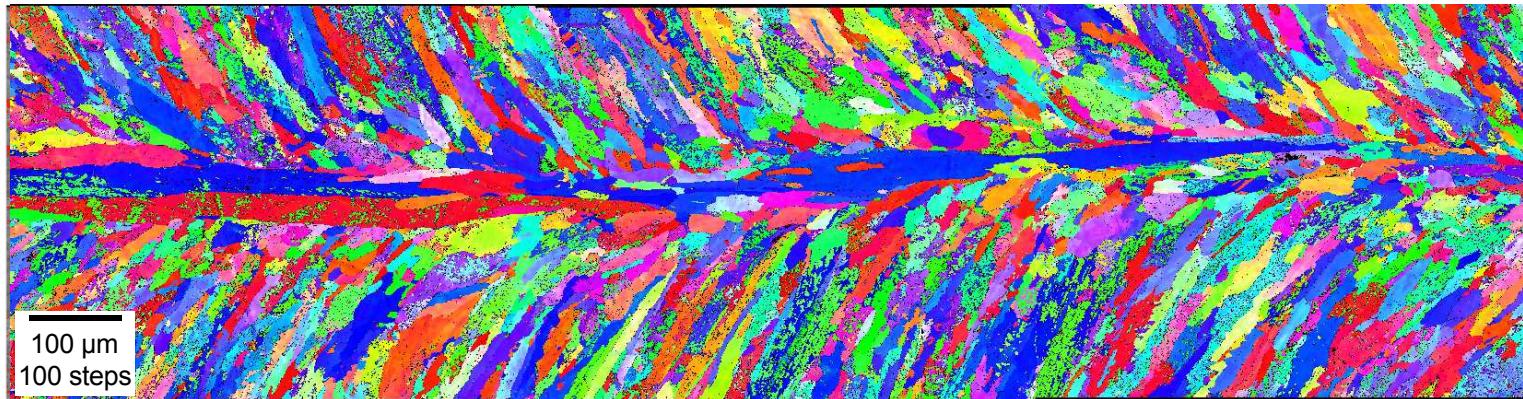
Band Contrast + Phase Map: **bcc**

Electropolishing Eliminates Ambiguity in EBSD Ferrite Determination

Band Contrast + Phase Map: **bcc**

Inverse Pole Figure Map

Mechanically Polished



Electropolished



80 vol% H_3PO_4 + 20 vol% n-Butanol @ 70°C, 1 A/cm²

Welding Direction 

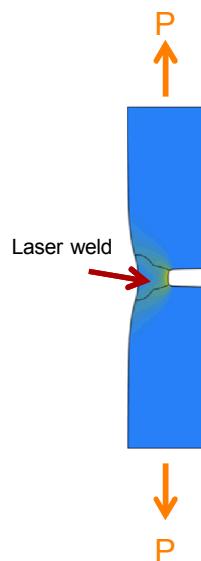
Continuous Wave Nd:YAG Laser Weld on 304L

No Microstructural Instability with Room Temperature Tensile Deformation of 304L

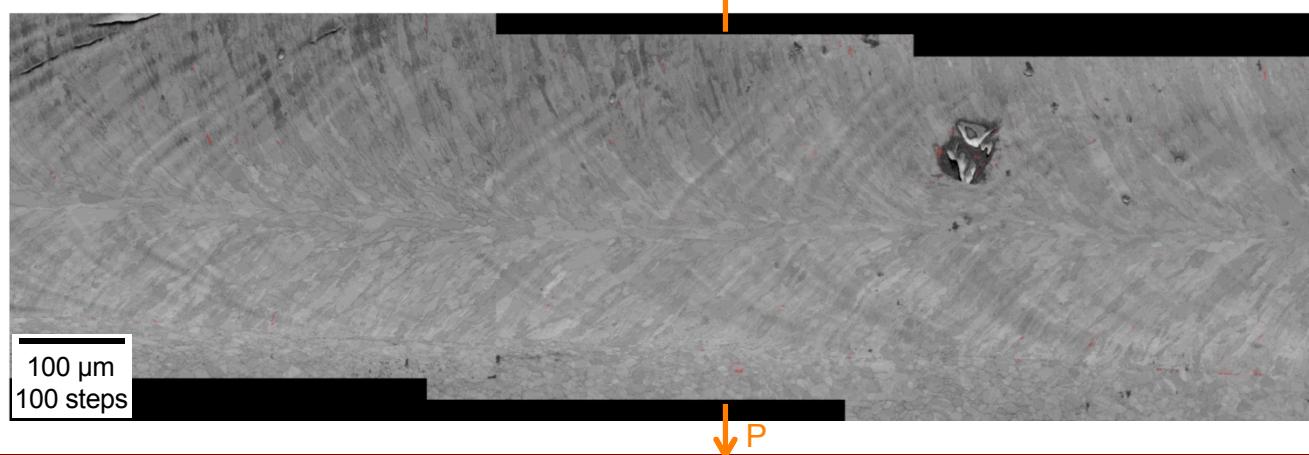
- No appreciable formation of bcc-indexed phase in laser weld when sample loaded in tension

Band Contrast + Phase Map: **bcc**

Unstrained
(electropolished)

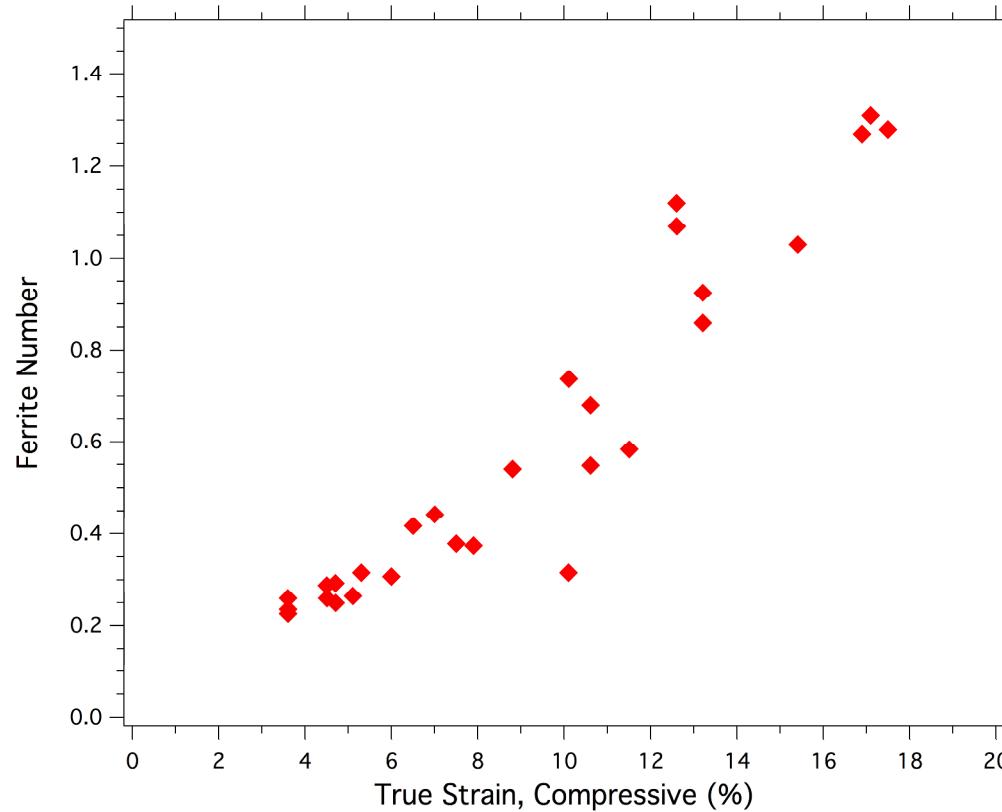


Strained
(electropolished)



No Microstructural Instability with Room Temperature Compressive Deformation of 304L

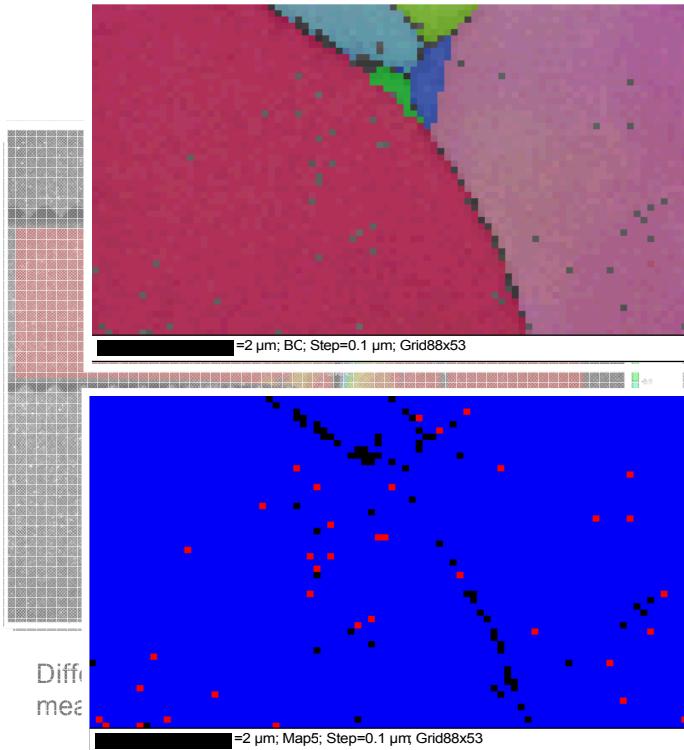
- No significant increase in measured 'ferrite' content with total compressive strain up to 20%



- Material: Avesta 16730 304L
- Material cold rolled up to 20% strain with bcc proportion measured magnetically

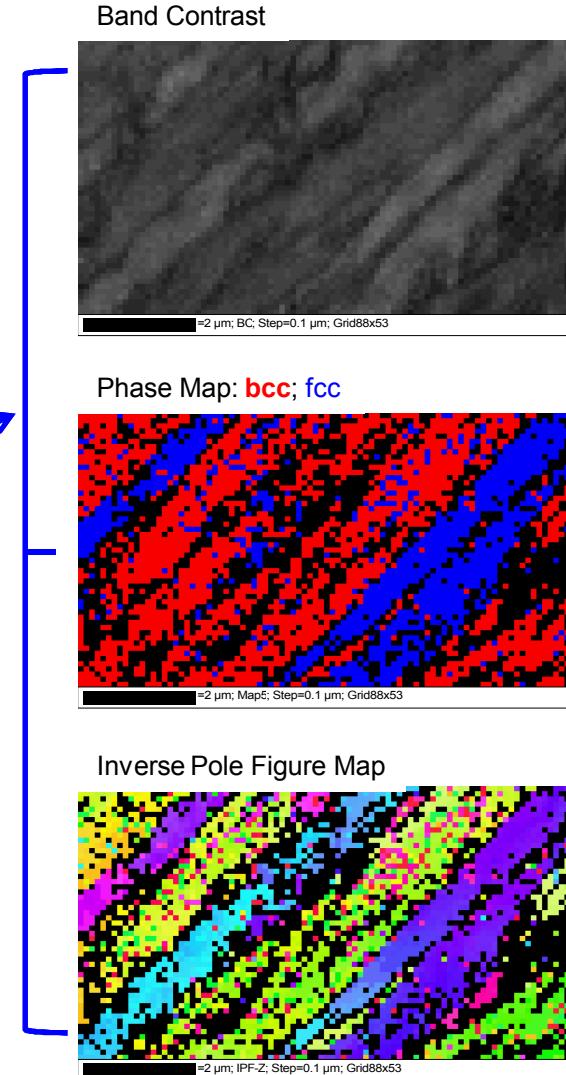
EBSD suggests FCC to BCC Phase Change with Shear Loading

- Shear strain levels of ~50% resulted in deformed region with significant fraction of bcc-indexed phase



Away from fracture surface, scan area shows predominantly austenite

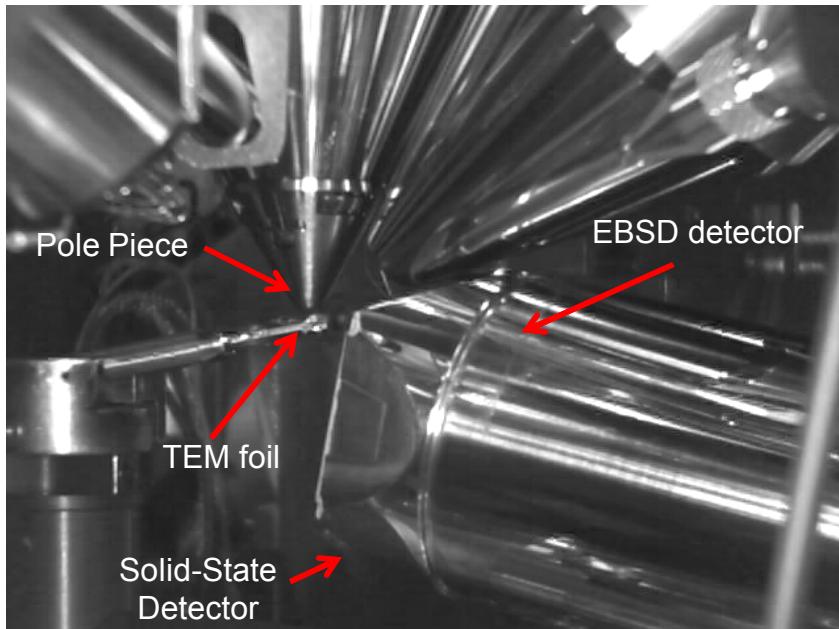
Sample electropolished after shear testing



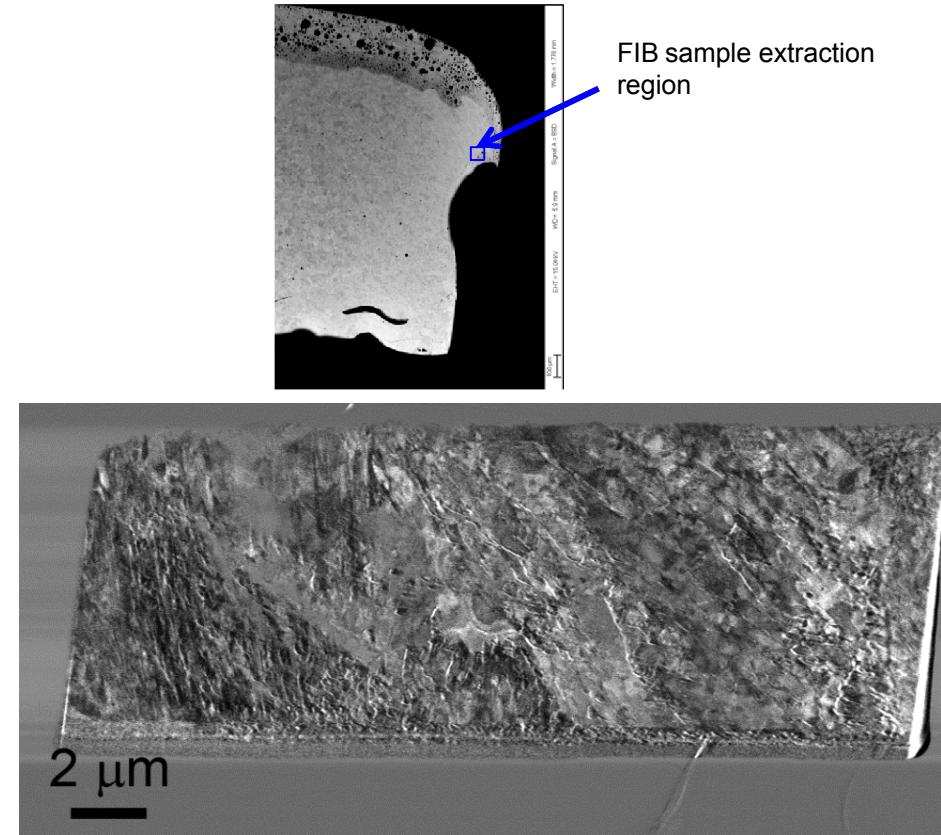
- EBSD problematic due to highly deformed shear region producing diffuse diffraction patterns

Transmission Kikuchi Diffraction using FIB-prepared specimens

- Recently-developed Transmission Kikuchi Diffraction (TKD) used to enable phase and orientation mapping of highly deformed, fine-scale bulk samples
- Spatial resolution significantly $< 10\text{nm}$



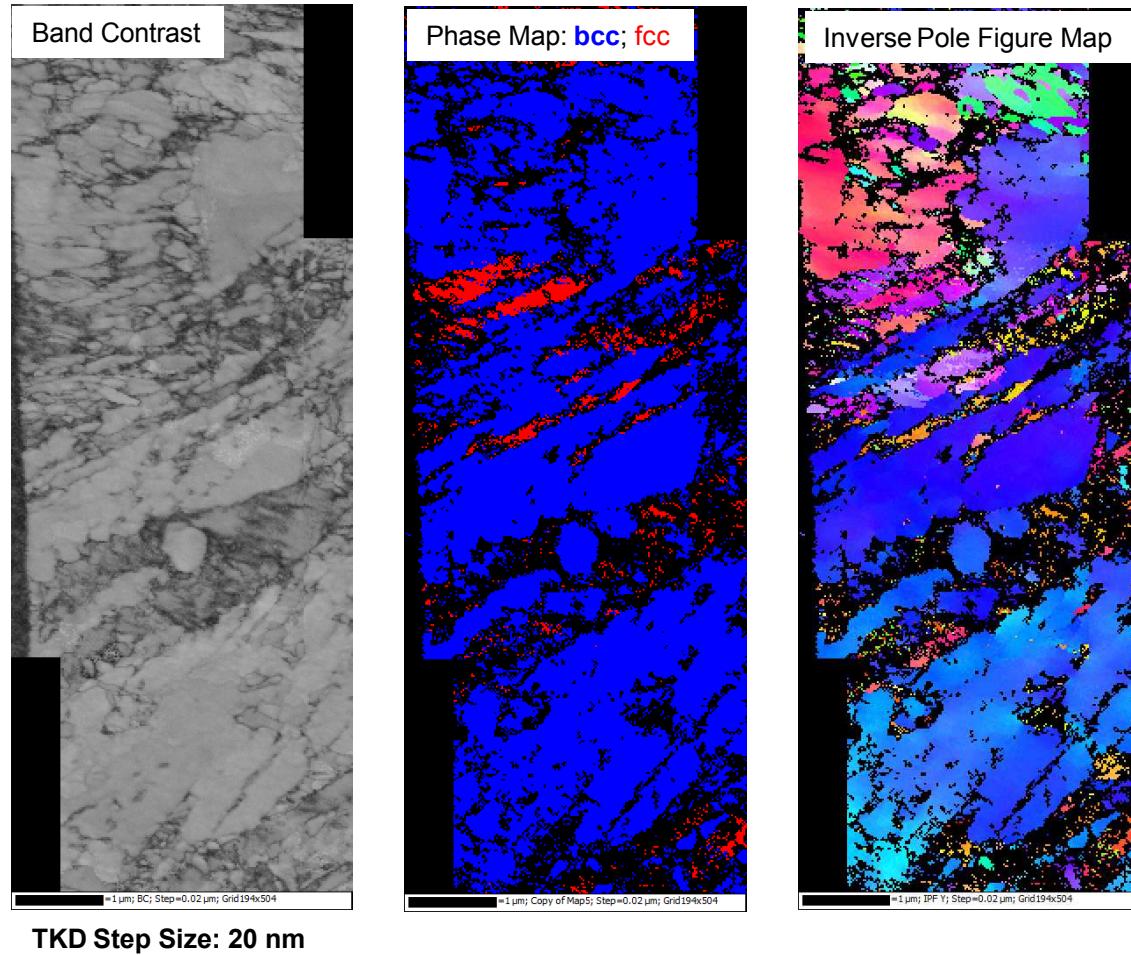
TKD setup in dual-beam FIB



STEM image of FIB prepared 304L shear sample generated with SEM operated at 30 kV in transmission mode

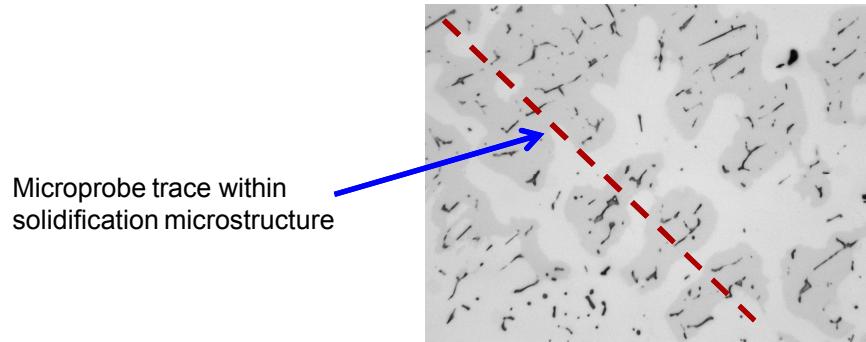
FCC to BCC Phase Change Observed in Shear Loading

- Examined shear region predominantly indexed bcc
- Controlled shear loading test demonstrates austenite instability for a commercial 304L composition at room temperature
- A shear component to the deformation during polishing likely contributes to austenite instability
 - Can produce misleading information regarding δ -ferrite present in microstructure

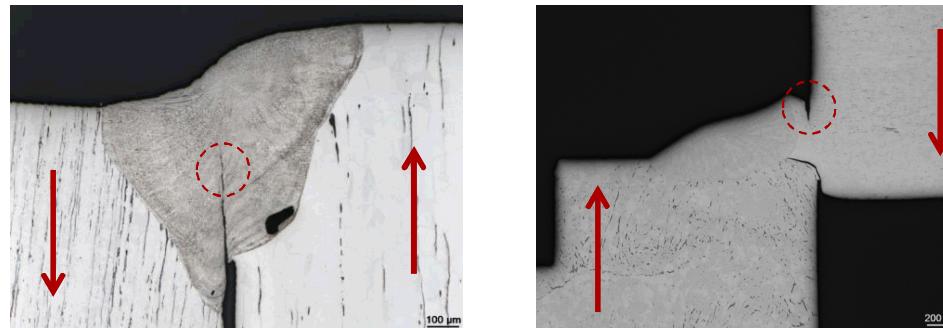


Ongoing work

- Determine compositional dependence of austenite transformation



- Further characterization of weld shear-type failure



- Continuing assessment of martensite transformation kinetics

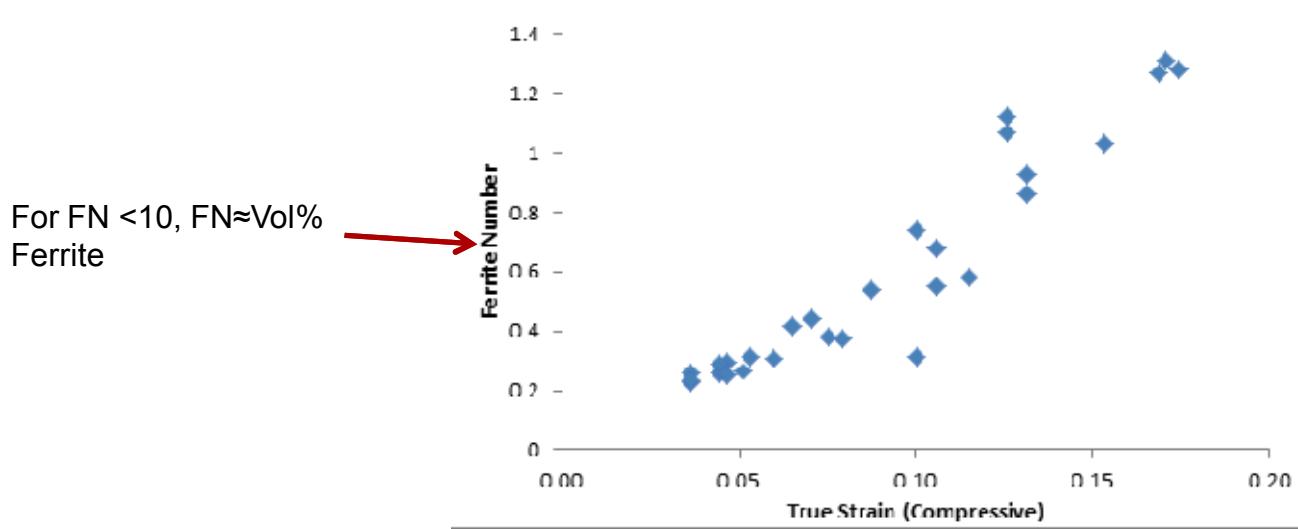
Conclusions

- Polishing-induced martensite in austenitic stainless steel welds can result in misleading ferrite content measurements
- For unambiguous determination of ferrite content via EBSD, electropolishing is required
- Global tensile or compressive loading does not appear to induce martensite at room temperature
- Deformation via shear loading results in martensite formation
 - Loading condition likely analogous to loading during mechanical metallographic preparation

Extra Slides

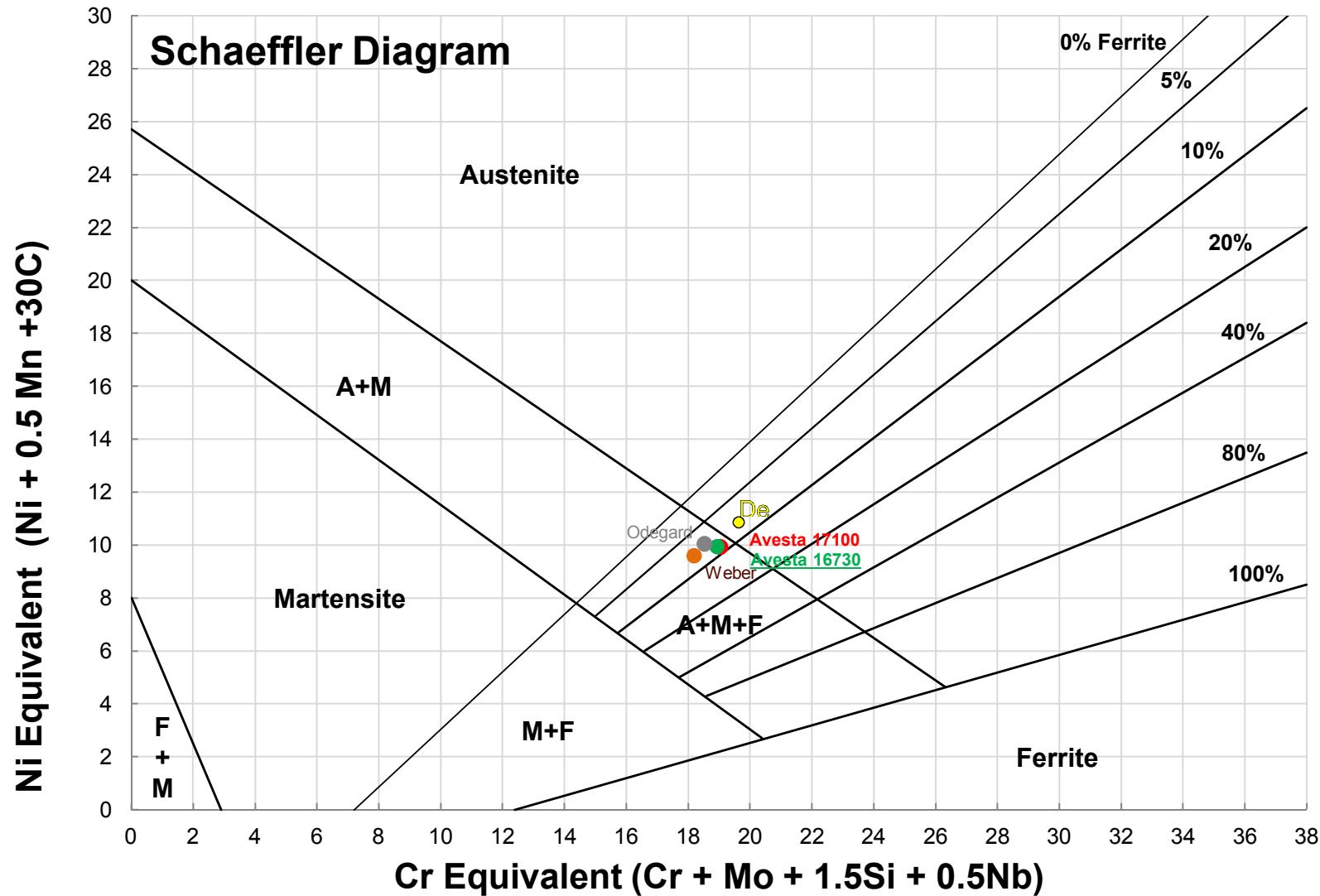
Deformation Induced Martensite Formation: Room Temperature Compressive Loading

- No significant increase in measured 'ferrite' content with total compressive strain up to 20%

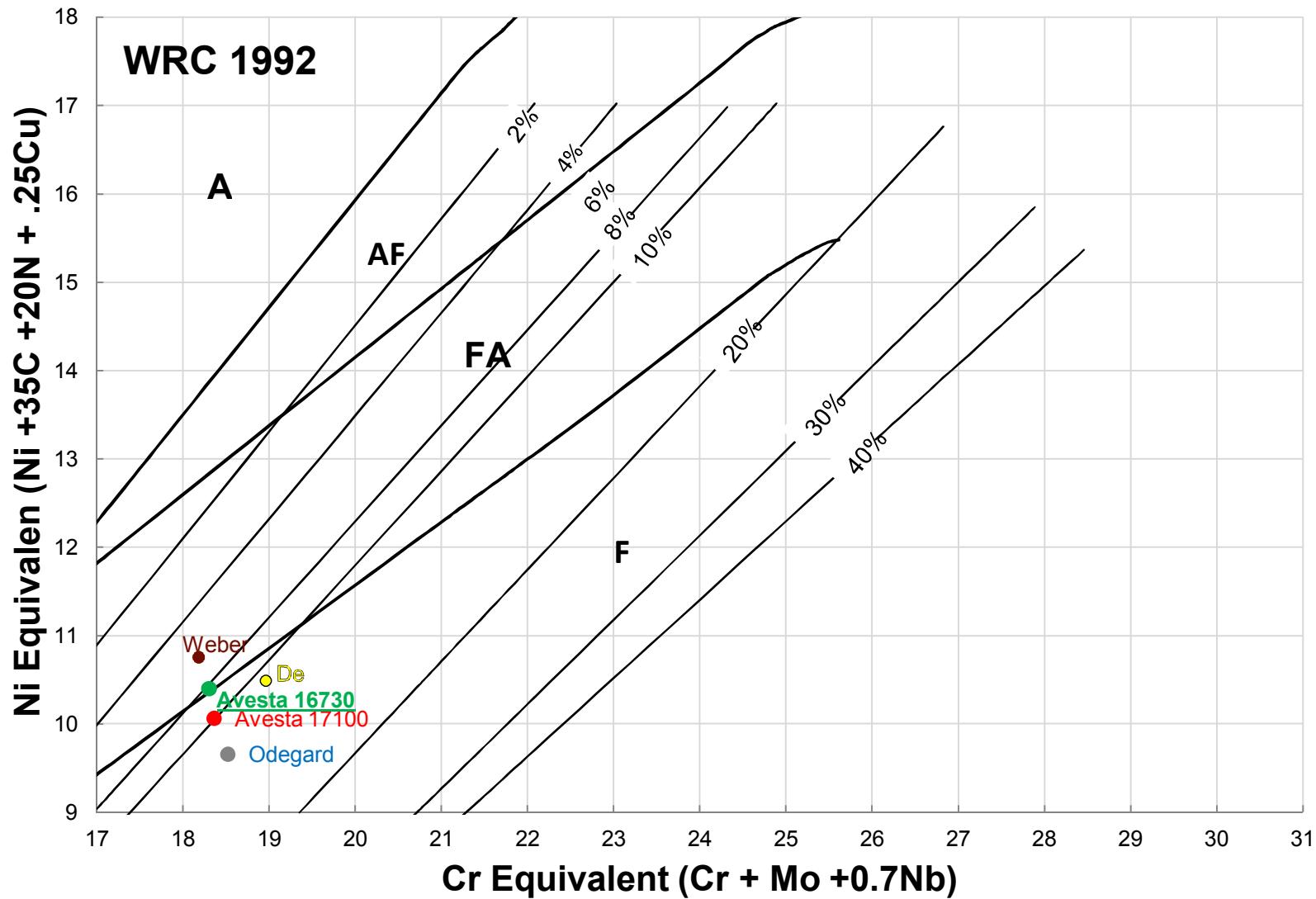


- Material: Avesta 16730 304L
- Material cold rolled up to 20% strain with bcc proportion measured magnetically

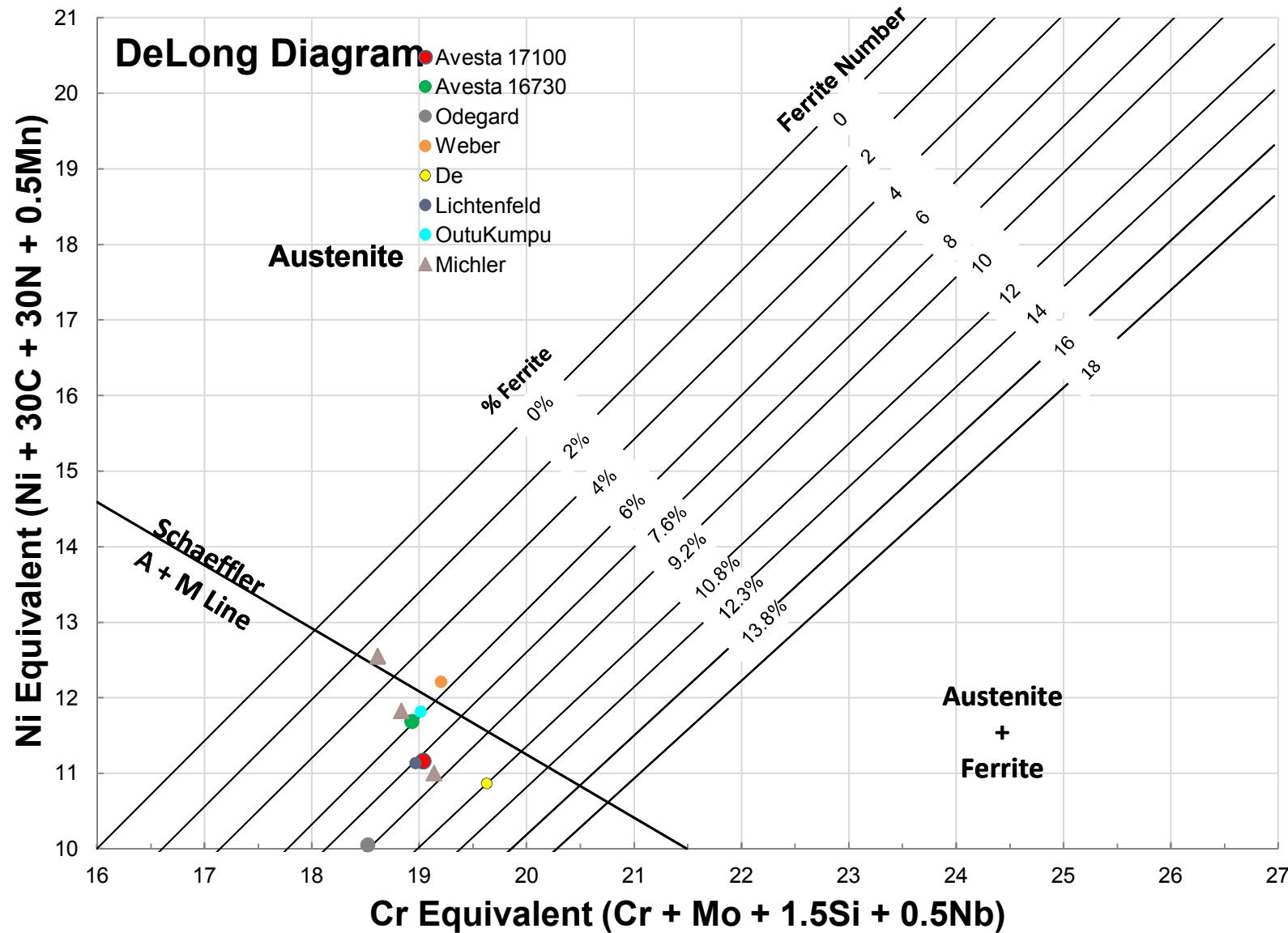
Weld Metal Constitution Diagram



Weld Metal Constitution Diagram

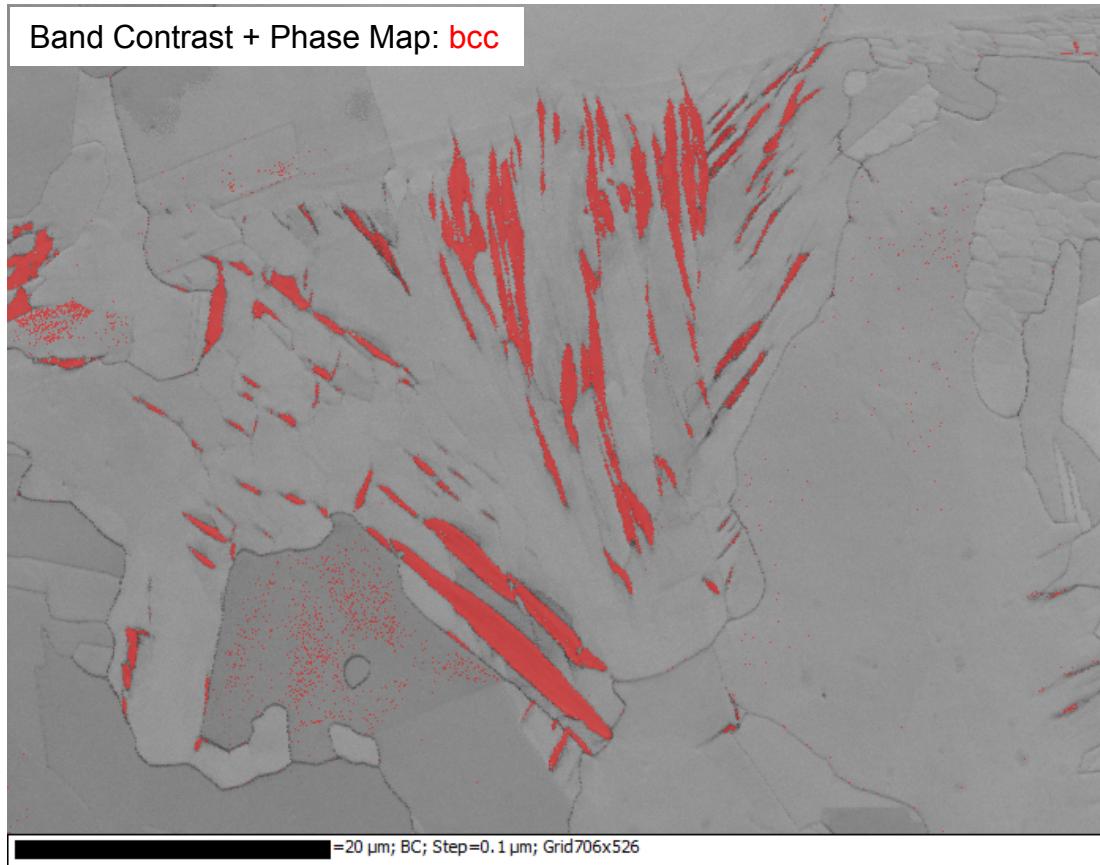


Weld Metal Constitution Diagram



LN2-Quenched Laser Weld – No Tensile Deformation

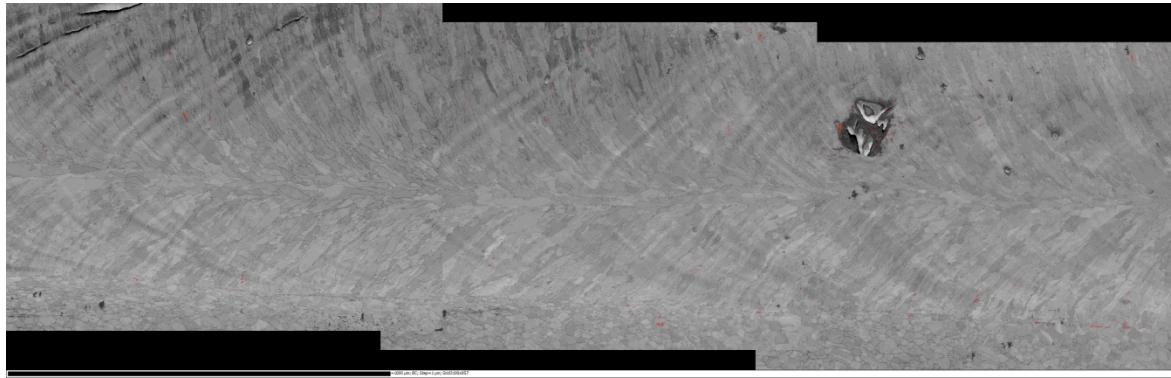
Electropolished laser butt weld after quenching to LN2



LN2 increased the amount of bcc indexed phase near fusion boundary slightly. No obvious increase in bcc indexed phase in base material.

Laser Weld: Strained (tensile loading) + LN2 Quench

Straining of weld in tension results in no measurable increase in bcc indexed phase



Band Contrast + Phase Map: **bcc**

Straining of weld in tension followed by LN2 results in no measurable increase in bcc indexed phase

