

# Material Deformation at the Microstructural Scale in BCC Tantalum

**Jay D. Carroll**

**Sandia National Laboratories**

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## 1. Motivation

- Why do we need to understand grain scale deformation?

## 2. Background

- Slip in FCC and BCC metals

## 3. Relating microstructure to deformation

- Schmid factors, etc.

## 4. Predicting deformation behavior at the grain scale

- Crystal plasticity models

## 5. Conclusions

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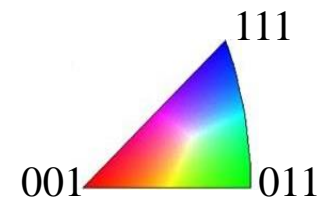
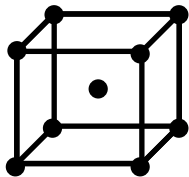
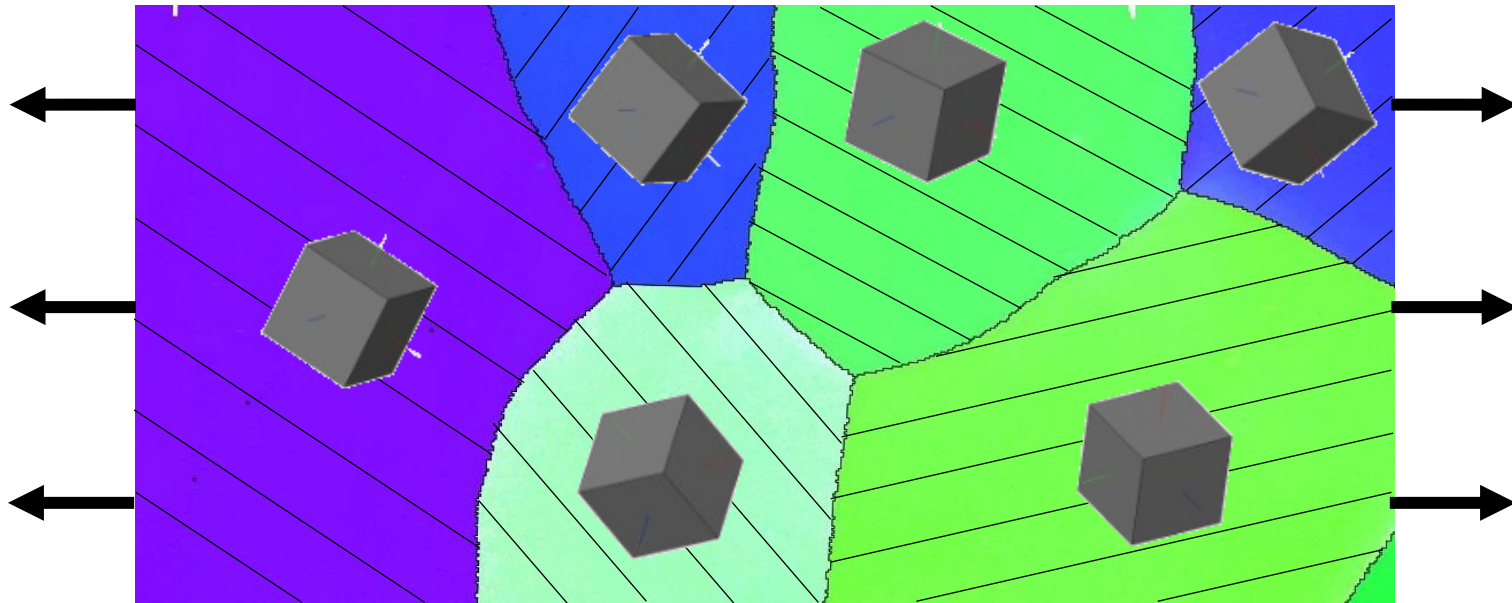
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- Crystal plasticity models

## 5. Conclusions

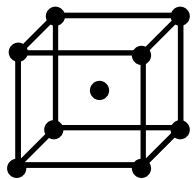
Grains are regions of the material with consistent orientation of the atomic unit cell.

4



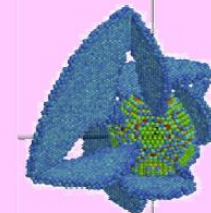
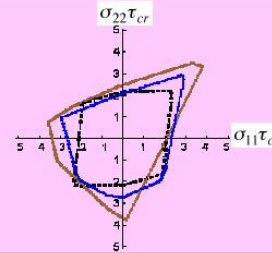
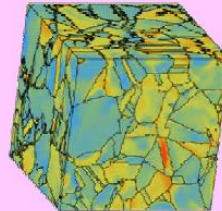
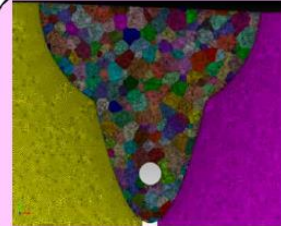
# Relate variability in structural behavior to microstructural variability

- Task 3: Predict macroscale variability from microstructural statistical models.
- **Task 2: Microscale effects on deformation behavior.**
- Task 1: Atomic/nanoscale defects and dislocation effects.

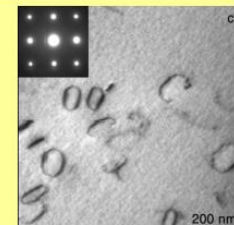
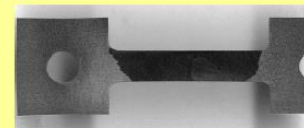
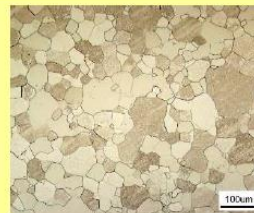
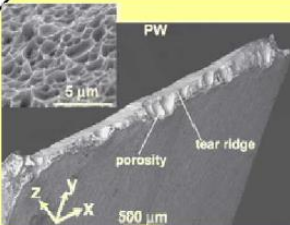


BCC  
Ta

Models



Experiments



**Material performance**  
 $10^0$  m  $10^6$  s

**Microstructural effects**  
 $10^{-3}$  m  $10^3$  s

**Single crystal behavior**  
 $10^{-6}$  m  $10^0$  s

**Atomic scale phenomena**  
 $10^{-9}$  m  $10^{-9}$  s

Atoms-up: Develop physics-based models to provide scientific insight

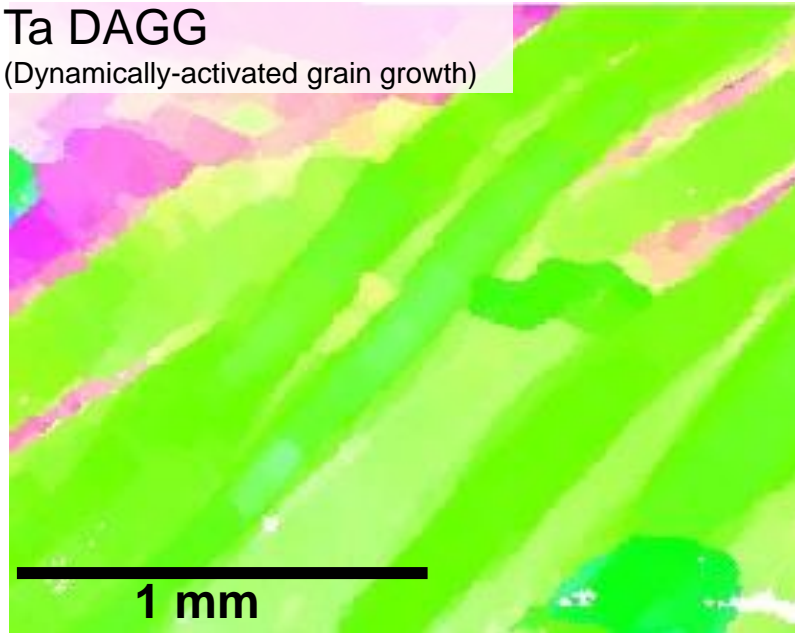
Continuum-down: Augment engineering-scale models to provide customer value

# Grain structure strongly depends on the material's history.

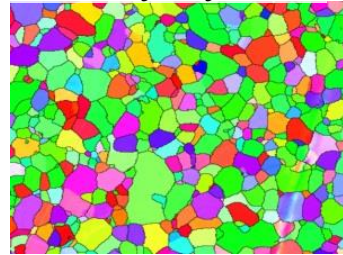
6

Ta DAGG

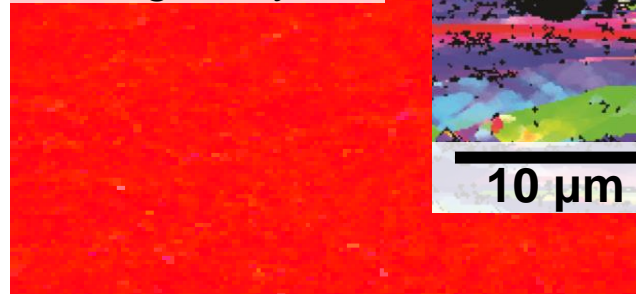
(Dynamically-activated grain growth)



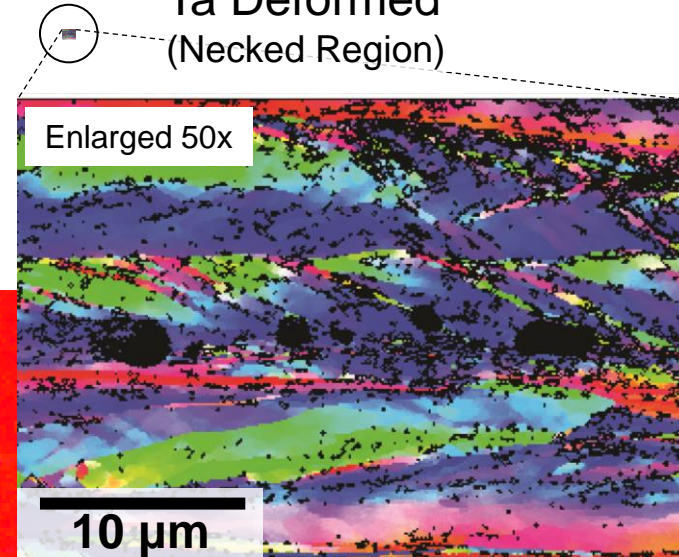
Ta Polycrystal



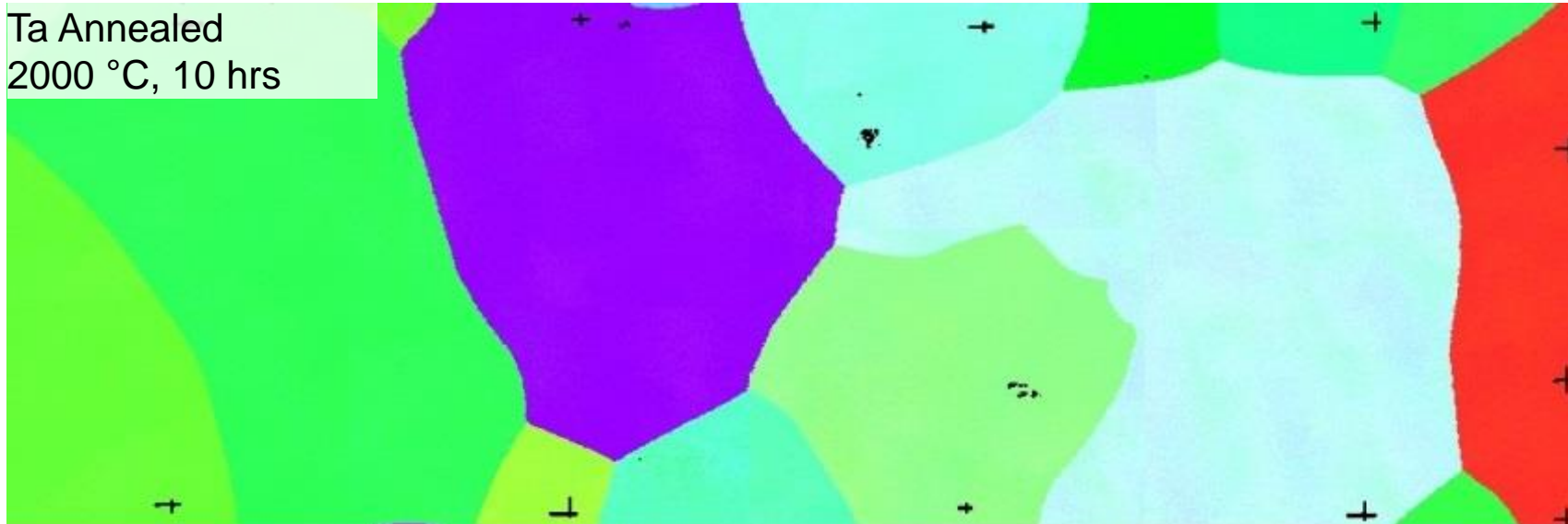
Ta Single Crystal



Ta Deformed  
(Necked Region)



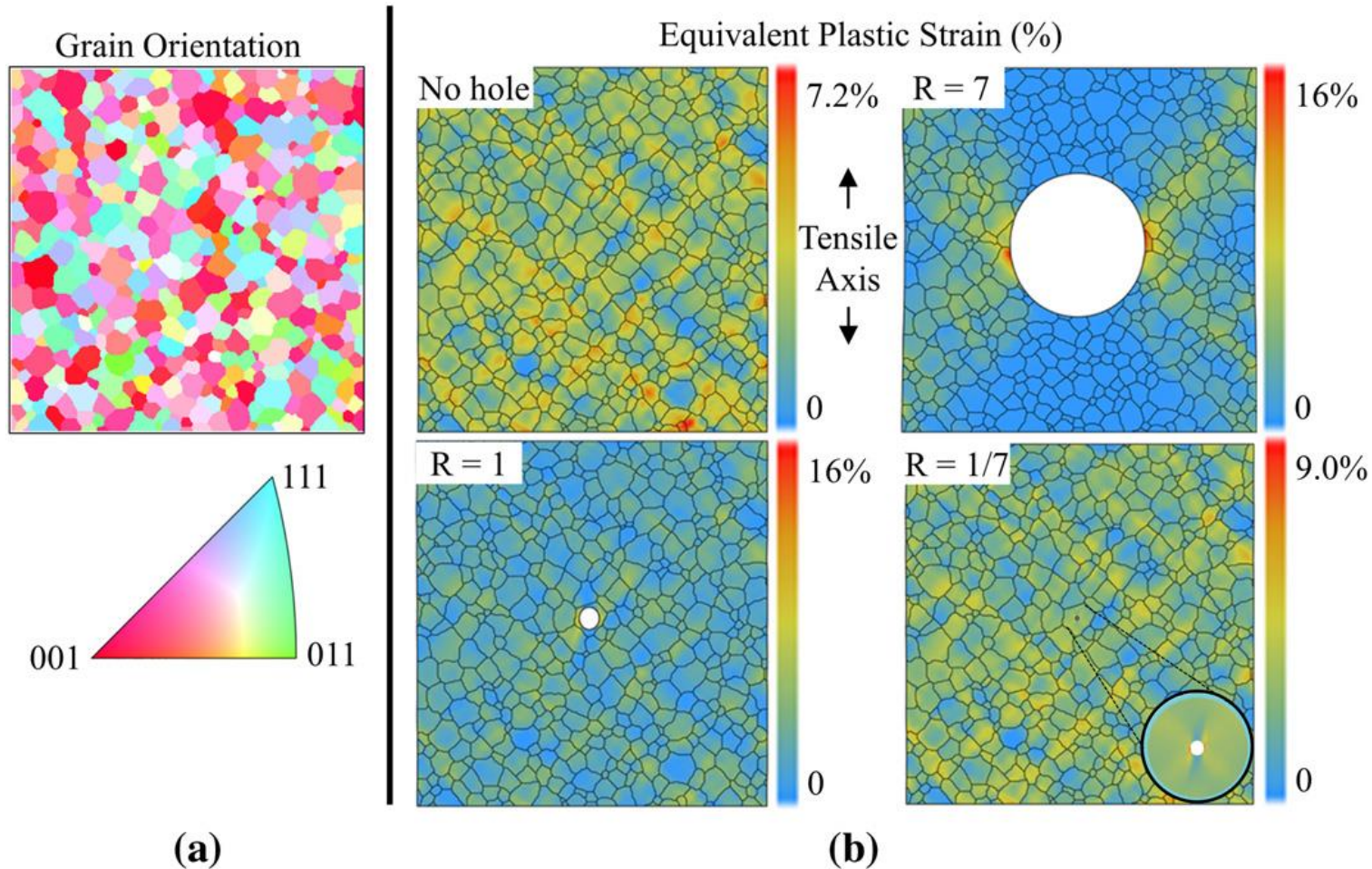
Ta Annealed  
2000  $^{\circ}\text{C}$ , 10 hrs





# For small features, microstructure can be more important than the stress concentrator.

- Model predictions



# For small features, microstructure can be more important than the stress concentrator.

## Experimental measurements

$$R = \frac{\text{Hole Size}}{\text{Grain Size}}$$

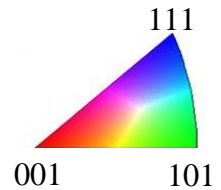
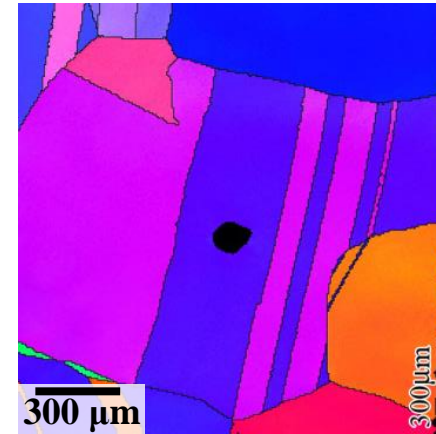
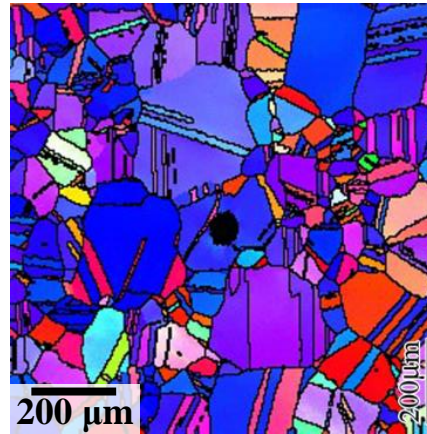
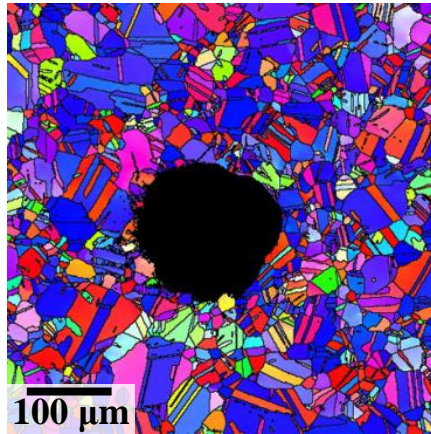
Orientation

R=7

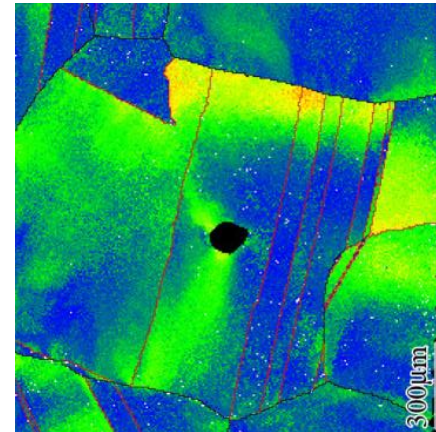
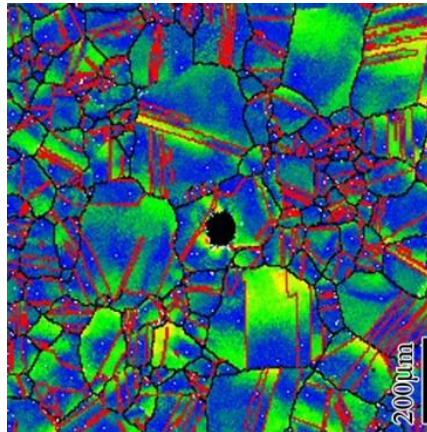
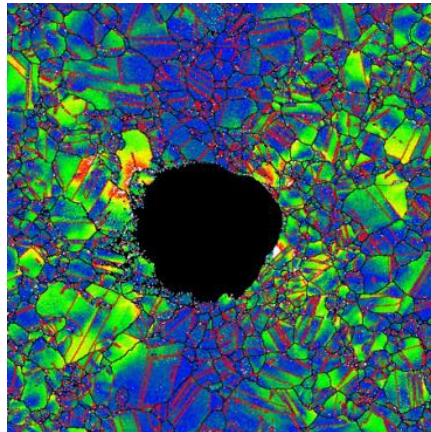
R=1

R=1/7

↑  
Tensile  
Axis  
↓



Misorientation  
(≈strain)





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## 3. Relating microstructure to deformation

- Schmid factors, etc.

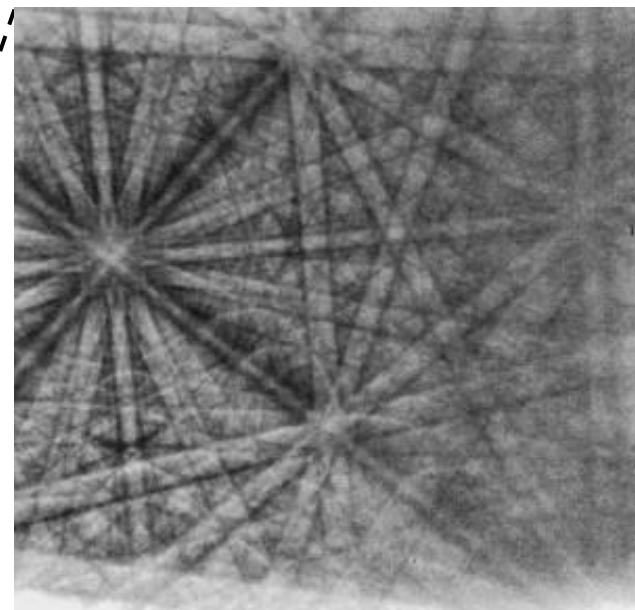
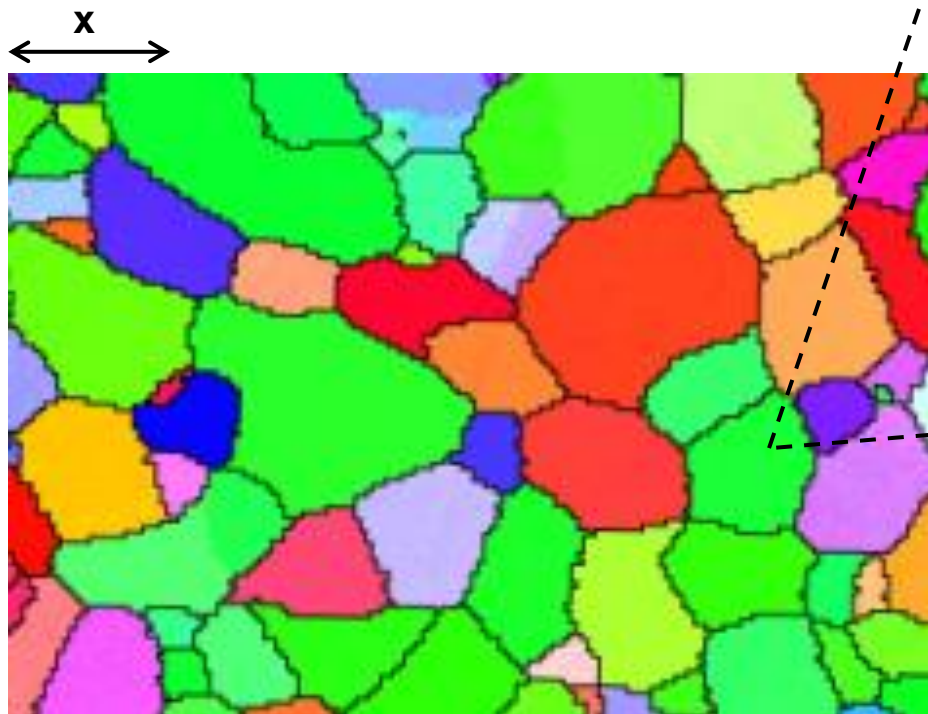
## 4. Predicting deformation behavior at the grain scale

- Crystal plasticity models

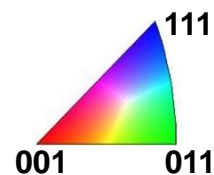
## 5. Conclusions

# Electron Backscatter Diffraction (EBSD) measures local grain orientation.

- Capture an image of Kikuchi diffraction bands.
- Compare angles of bands to a lookup table to find the crystal orientation.
- Repeat for all pixels in the map.

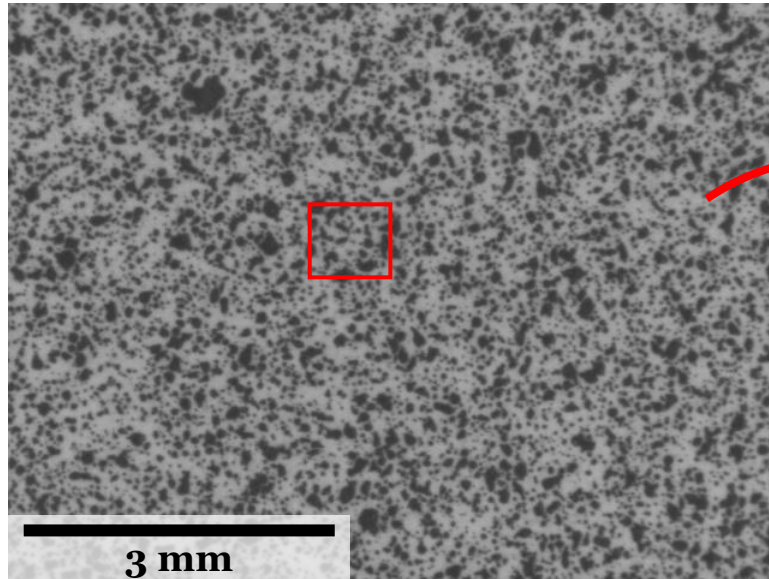


- J.A. Small, J.R. Michael, Journal of Microscopy-Oxford, v. 201 (2001).

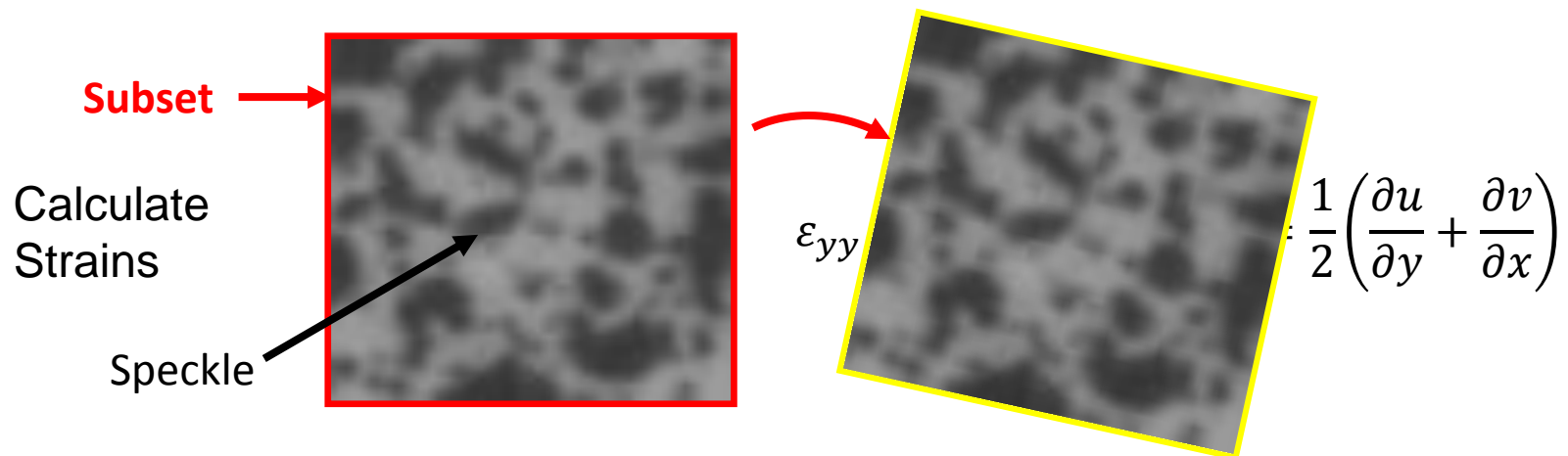
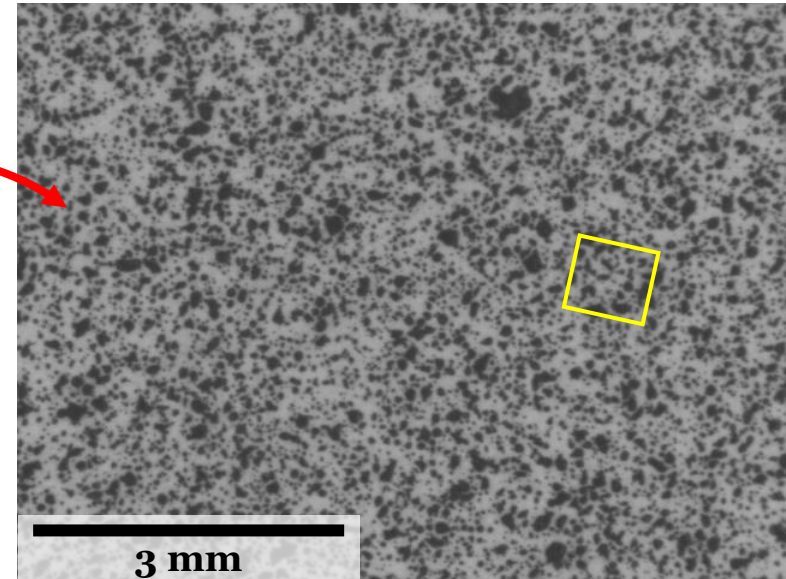


# Digital image correlation (DIC) measures full field displacements and strains by tracking speckles.

Reference

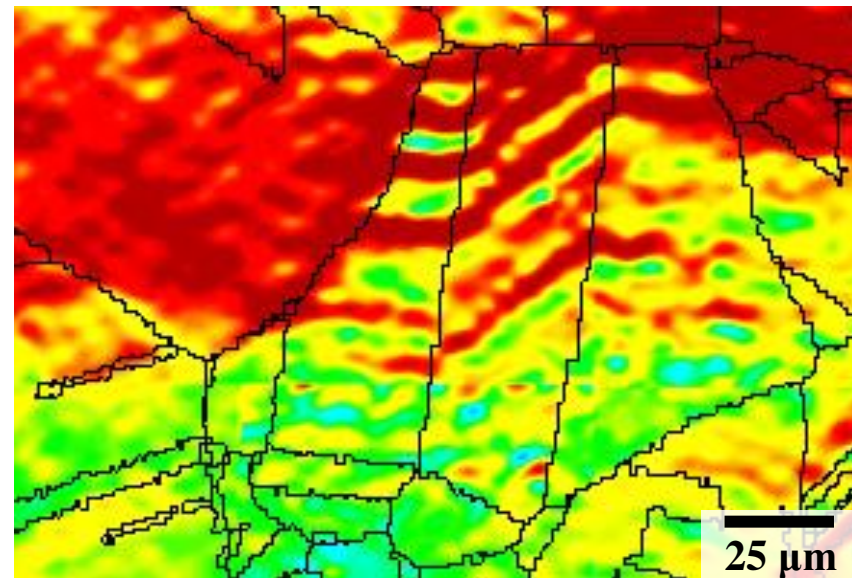
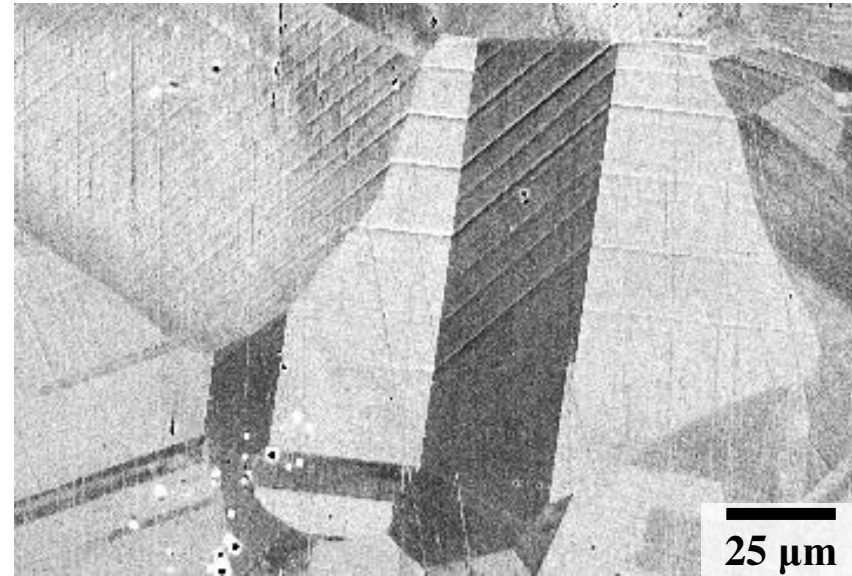
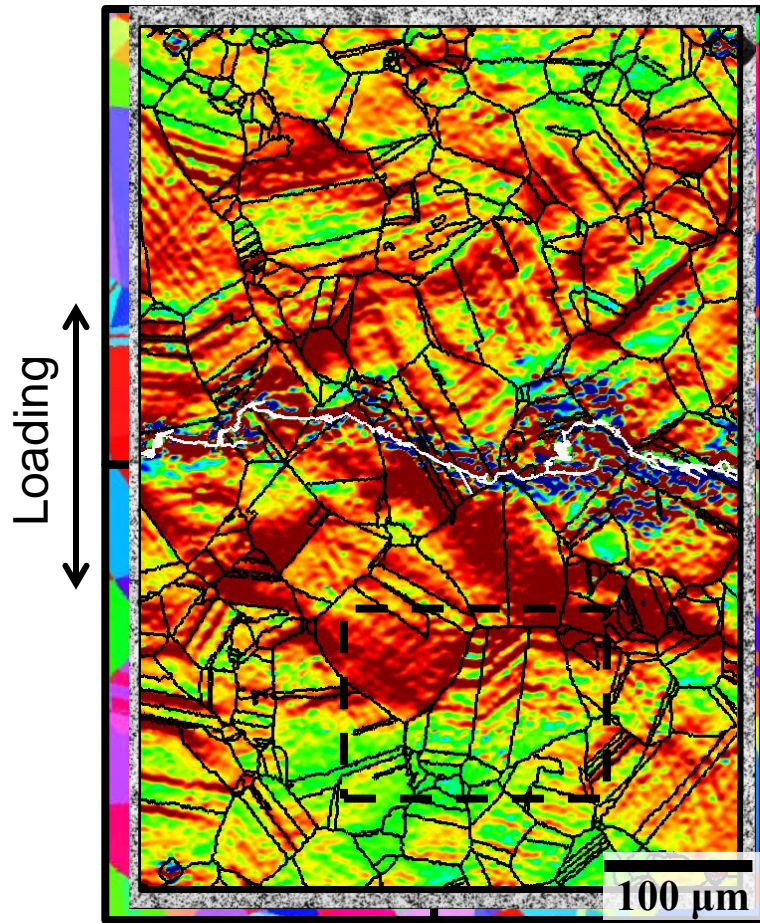


Deformed





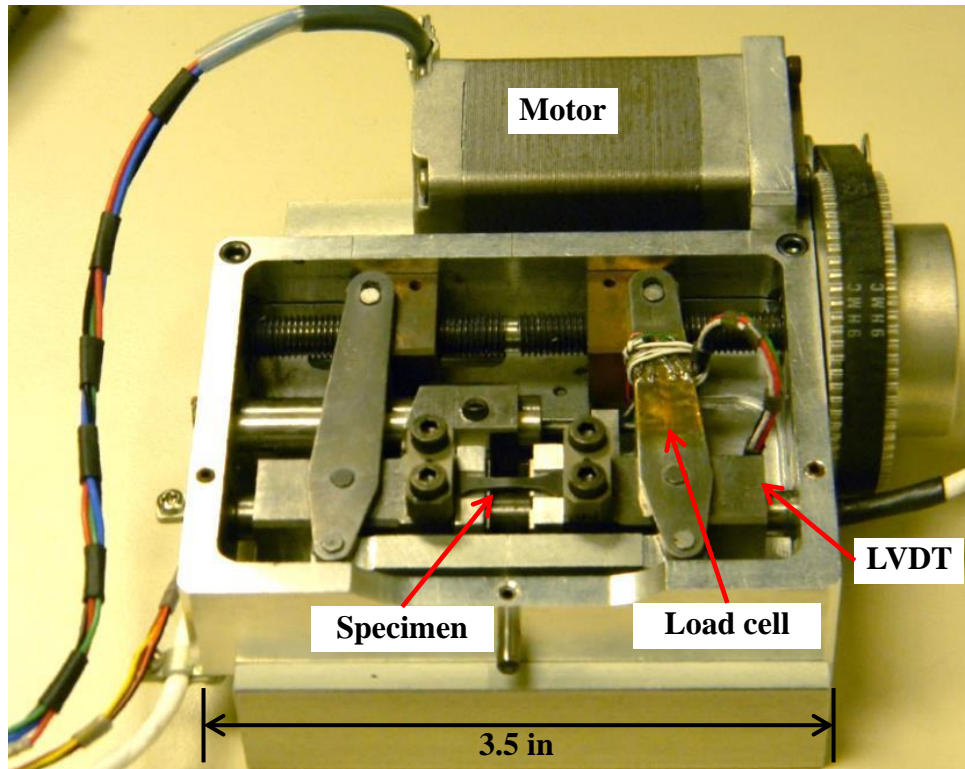
# Our high resolution experimental technique relates subgrain level strains to microstructure.



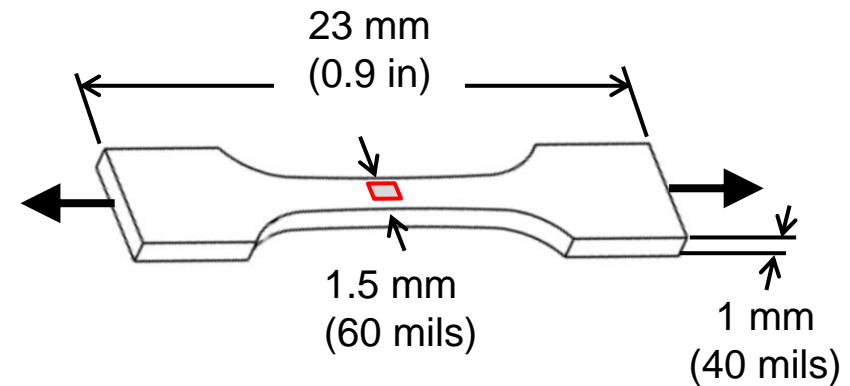
- Carroll et al., *Rev. Sci Instr.*, v. 81 (2010).
- Carroll et al., *Int J. Fracture*, v. 180 (2012).
- Carroll et al., *Int. J. Fatigue*, (in press, 2013).



# An in situ load frame developed at Sandia allows loading inside the SEM.



- Can make DIC and EBSD measurements at load.

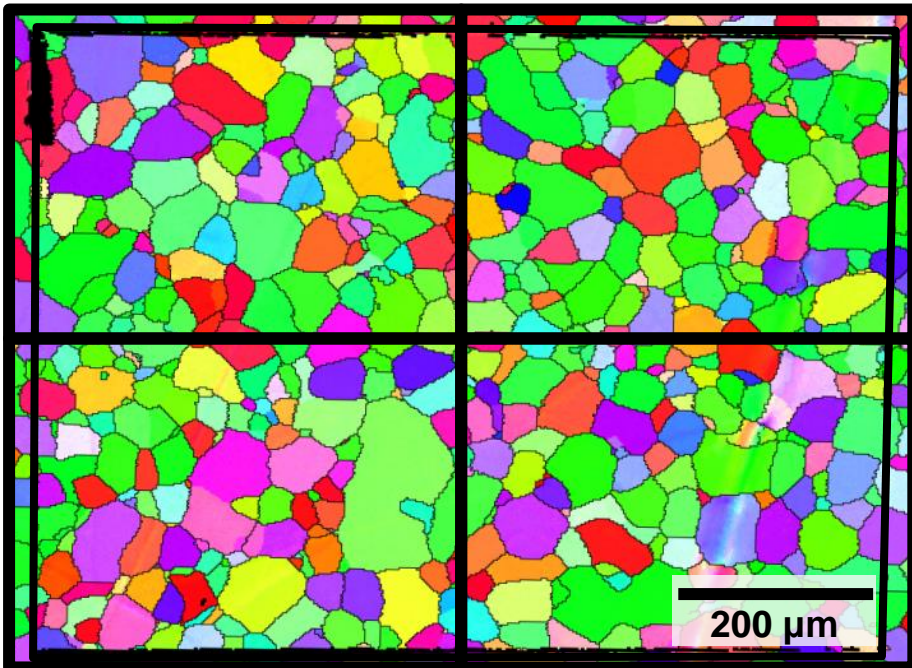


Tapered gage section is narrower at center.

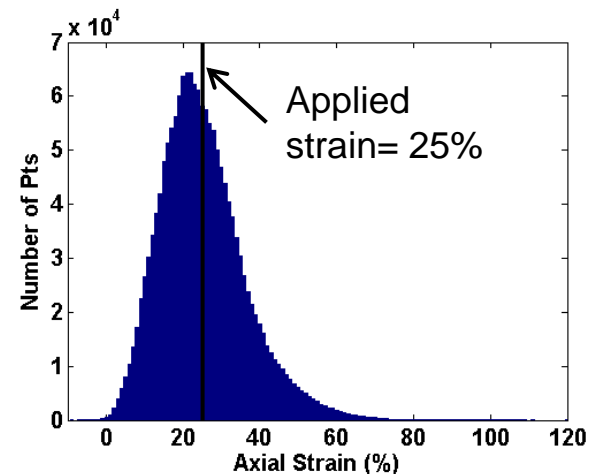
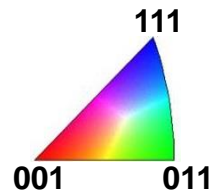
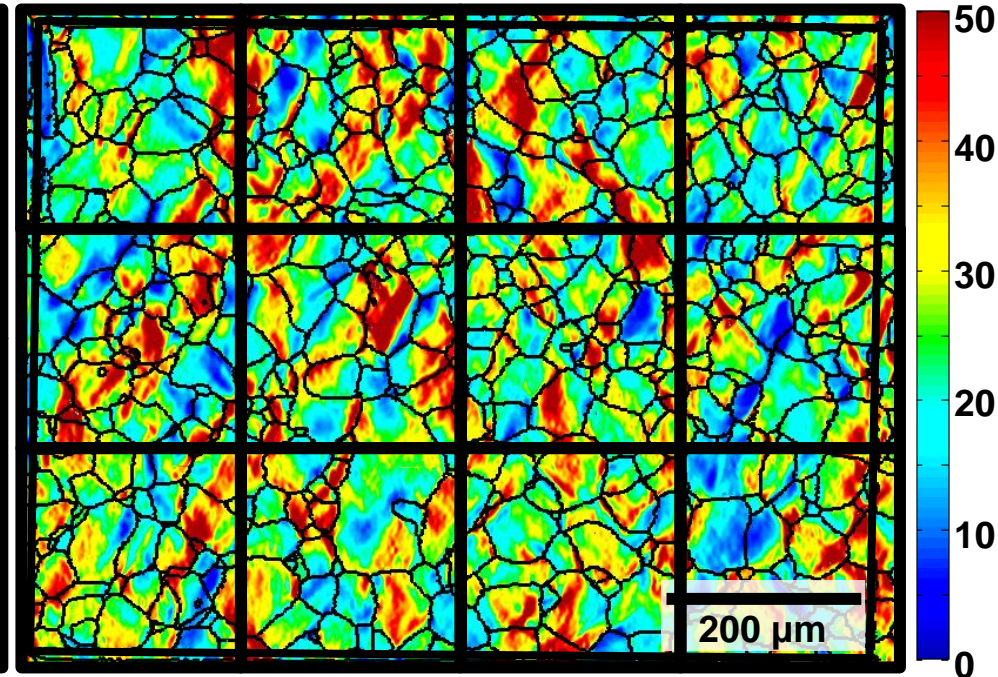
# Compare grain structure to local strain measurements in polycrystal Ta (BCC metal).

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Microstructure (EBSD)



Effective Strain (%) (DIC)



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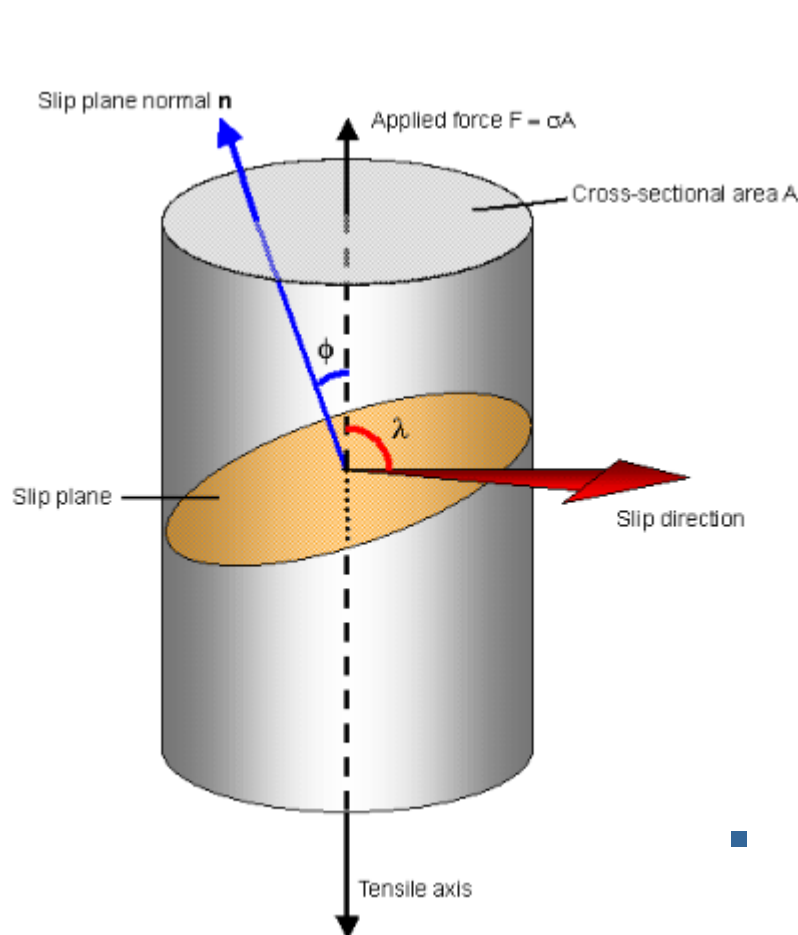
- Schmid factors, etc.

## 4. Predicting deformation behavior at the grain scale

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# The Schmid Factor is measures how prone each plane is to crystallographic slip.



[http://www.doitpoms.ac.uk/tlplib/slip/slip\\_geometry.php](http://www.doitpoms.ac.uk/tlplib/slip/slip_geometry.php)

Shear stress on system

Applied stress

$$\tau = \frac{F}{A} \underbrace{\cos\phi \cos\lambda}_{\text{Schmid factor}}$$

$$0 < \text{Schmid} < 0.5$$

**Hard. Slip will not happen on this system.**

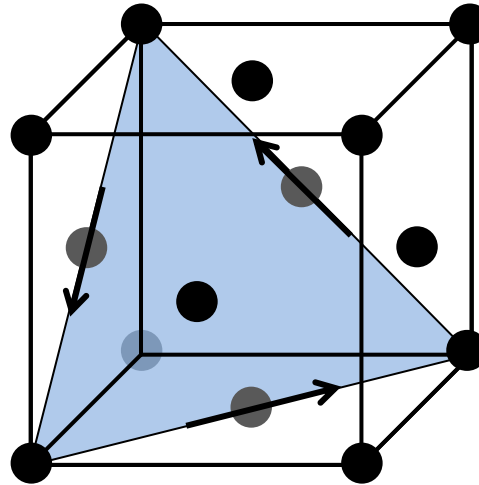
**Soft. Slip will probably happen on this system.**

- The Schmid factor of a grain is the max Schmid factor of all slip systems considered.



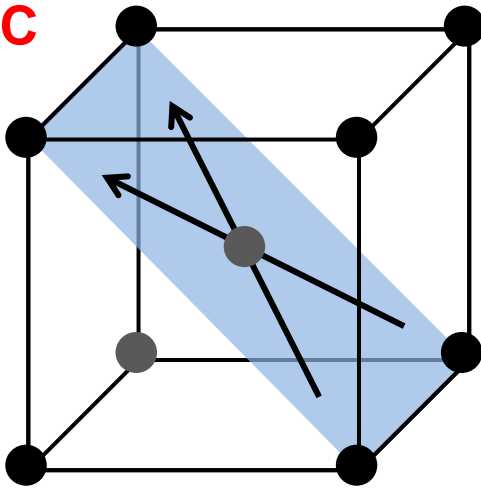
# Images showing slip planes and directions in FCC and BCC unit cells.

**FCC**

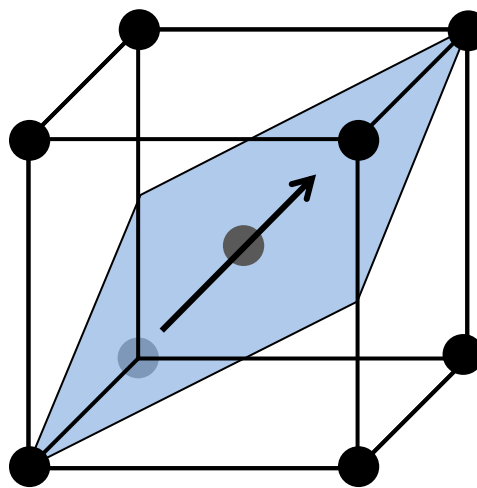


12 Slip systems:  
Four {111} Planes each with  
three  $\langle 110 \rangle$  Slip directions

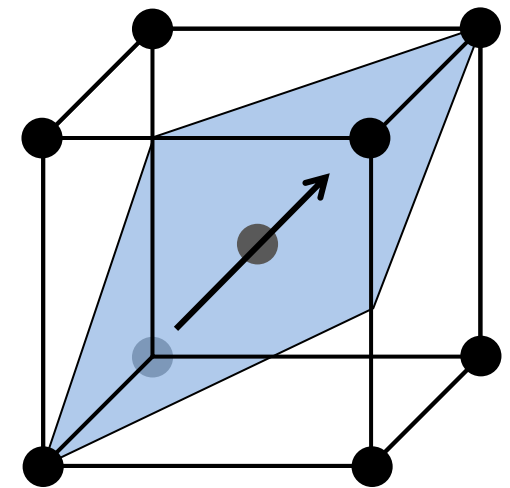
**BCC**



12 {110} slip systems  
6 Planes each with  
two  $\langle 111 \rangle$  Directions

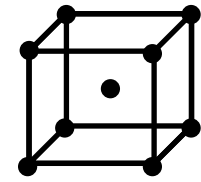


12 {112} slip systems  
12 Planes each with  
one  $\langle 111 \rangle$  Direction



24 {123} slip systems  
24 Planes each with  
one  $\langle 111 \rangle$  Direction

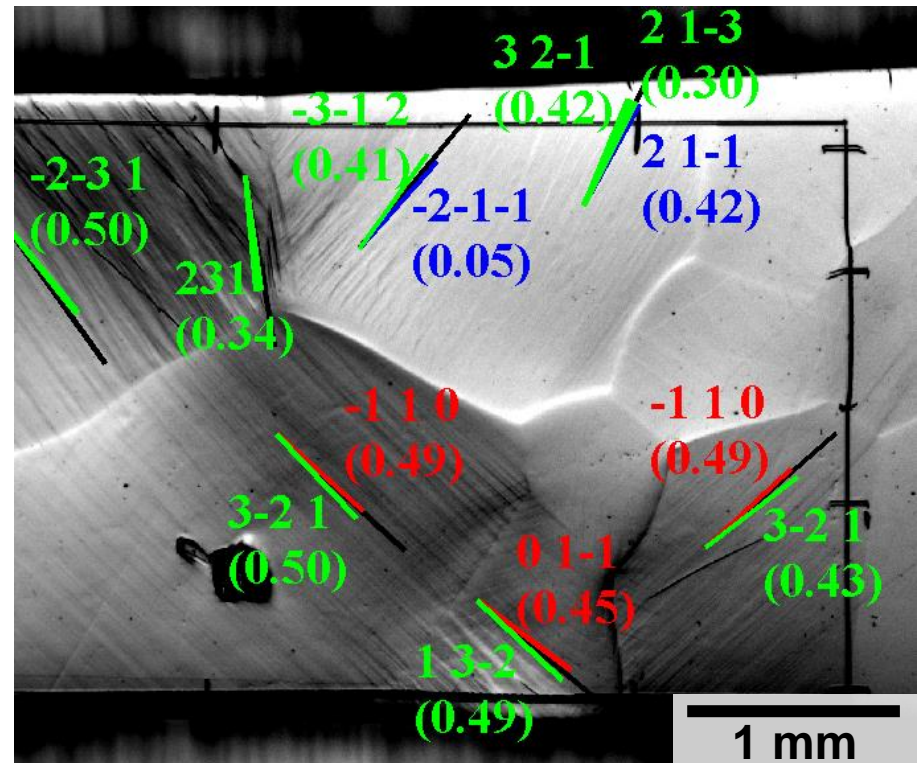
# Special challenge with BCC materials: Identifying active slip systems.



- Slip is in the  $\langle 111 \rangle$  direction, but on which plane?
- At the atomic level,  $\{110\}$  is the most likely slip plane, but no consensus.
- Microscopically, slip can be on  $\{110\}$ ,  $\{112\}$  or the maximum resolved shear stress plane containing a  $\langle 111 \rangle$  direction.

Material	Slip plane	...	References
Tungsten	$\{110\}$		118
Tungsten	$\{112\}$		118
$\alpha$ -iron	$\{110\}$		118
$\alpha$ -iron	$\{112\}$		118
Chromium	$\{112\}$		118
Vanadium	$\{112\}$		118
Tantalum	$\{112\}$		133
Tantalum	$\{112\}$		134
Tungsten	$\{110\}$		130
Molybdenum	$\{110\}$		130
Molybdenum	$\{110\}$		130
Molybdenum	$\{110\}$		130
Niobium	$\{110\}$		130
Niobium	$\{112\}$		130
Niobium	$\{110\}$		130
$\alpha$ -iron	$\{110\}$		130
$\alpha$ -iron	$\{112\}$		130
$\alpha$ -iron	$\{112\}$		130
Vanadium	$\{110\}/\{112\}$		130
Vanadium	$\{110\}/\{112\}$		130
Vanadium	$\{112\}$		130

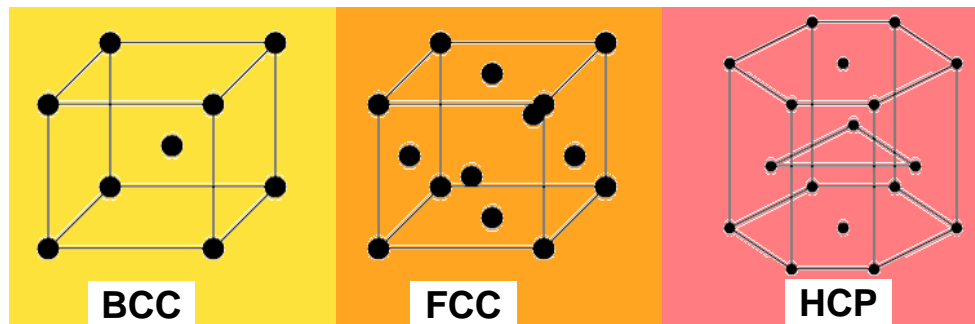
Weinberger C.R., et al.,  
International Materials Reviews (2013)



Carroll J.D., Clark B.G., Buchheit T.E., Michael J.R., Boyce B.L., *Materials Science and Engineering A*, v. 581 (2013).

# Crystal structure of metals in the periodic table.

H																		He			
453.69 Li bcc	1560 Be hcp											B	C	N	O	F	Ne				
370.87 Na bcc	923 Mg hcp	HCP		BCC		FCC										933.47 Al fcc	Si	P	S	Cl	Ar
336.53 K bcc	1115 Ca fcc	1814 Sc hcp	1941 Ti hcp	2183 V bcc	2180 Cr bcc	1519 Mn fcc	1811 Fe bcc	1768 Co hcp	1728 Ni fcc	1357.8 Cu fcc	692.68 Zn hcp	301.91 Ga fcc	Ge	As	Se	Br	Kr				
312.46 Rb bcc	1050 Sr fcc	1799 Y hcp	2128 Zr hcp	2750 Nb bcc	2896 Mo bcc	2430 Tc hcp	2607 Ru hcp	2237 Rh fcc	1828 Pd fcc	1235 Ag fcc	594 Cd fcc	430 In fcc	505 Sn fcc	904 Sb fcc	Te	I	Xe				
301.59 Cs bcc	1000 Ba bcc	*	2506 Hf hcp	3290 Ta bcc	3422 W bcc	3186 Re hcp	3306 Os hcp	2446 Ir fcc	1768 Pt fcc	1337.33 Au fcc	234.32 Hg fcc	577 Tl hcp	600.61 Pb fcc	544.7 Bi fcc	527 Po fcc	At	Rn				
Fr	973 Ra bcc	**	Rf	Db	Sg	Bh	Hs	Mt	Ds	Rg	Cn	Uut	Fl	Uup	Lv	Uus	Uuo				
*	1193 La dhcp	1068 Ce fcc	1208 Pr dhcp	1297 Nd dhcp	1315 Pm dhcp	1345 Sm fcc	1099 Eu bcc	1585 Gd hcp	1629 Tb hcp	1680 Dy hcp	1734 Ho hcp	1802 Er hcp	1818 Tm hcp	1097 Yb fcc	1925 Lu hcp						
**	1323 Ac fcc	2115 Th fcc	1841 Pa fcc	1405.3 U fcc	917 Np fcc	912.5 Pu fcc	1449 Am dhcp	1613 Cm dhcp	1323 Bk dhcp	1173 Cf dhcp	1133 Es fcc	Fm	Md	No	Lr						



The top number in the cell is the melting point (in K)

dhcp: double hexagonal close packed



unusual structure

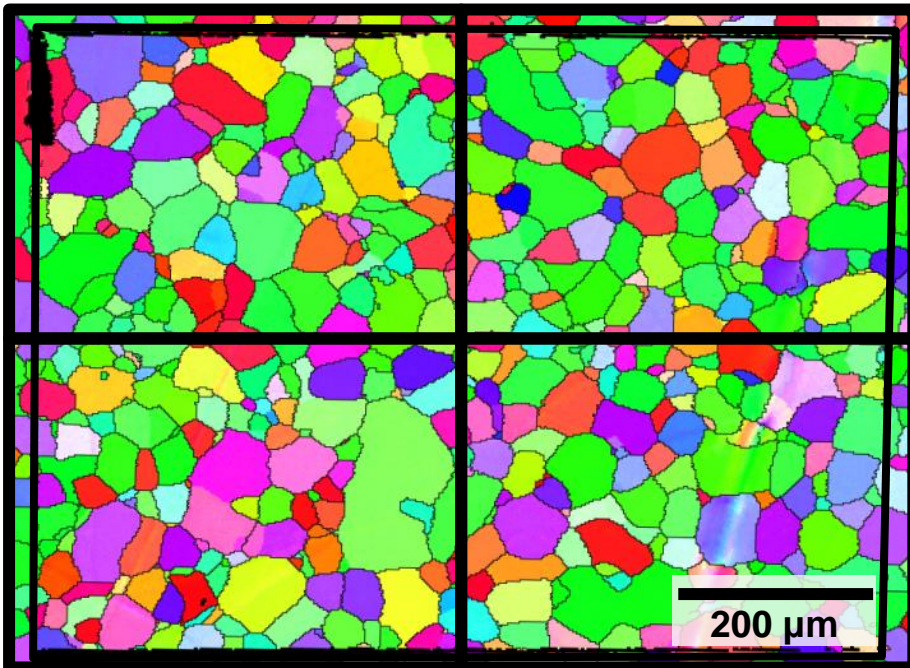
nonmetal

unknown or uncertain

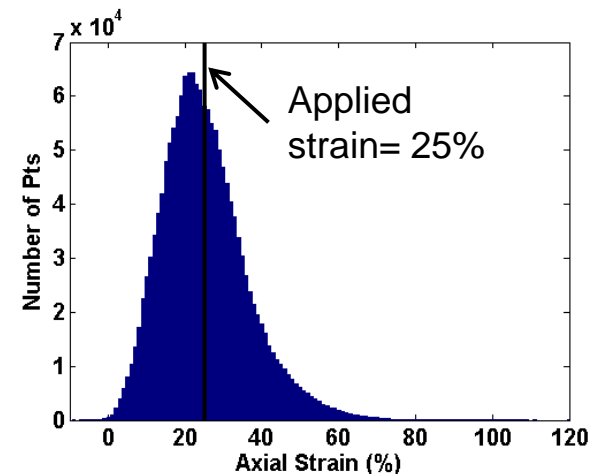
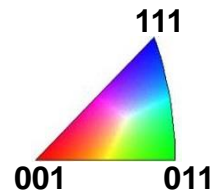
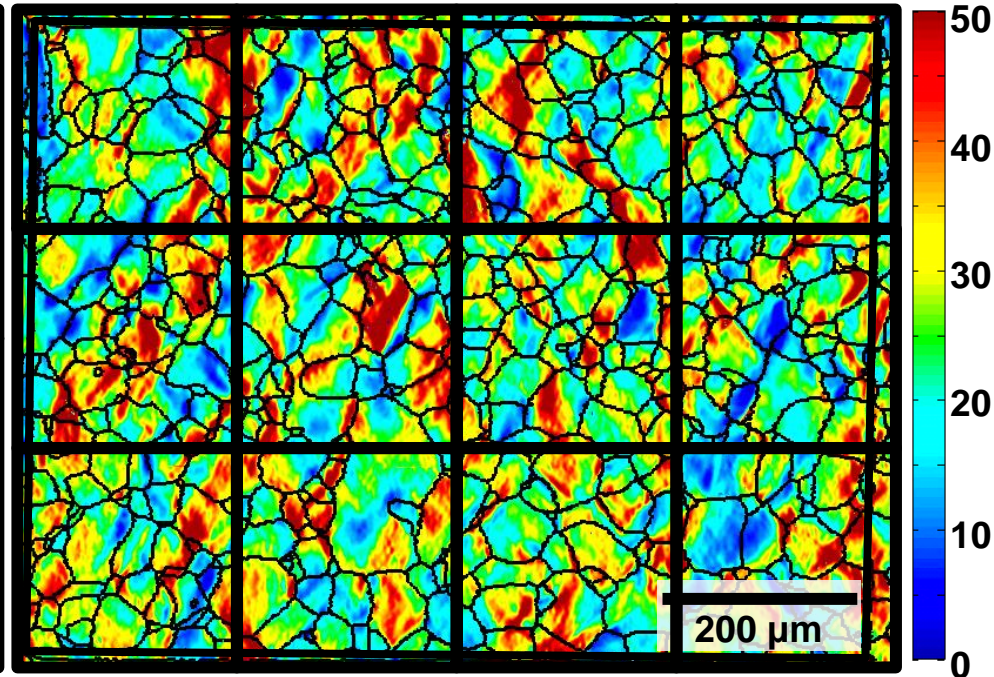
# Compare grain structure to local strain measurements in polycrystal Ta (BCC metal).

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Microstructure (EBSD)



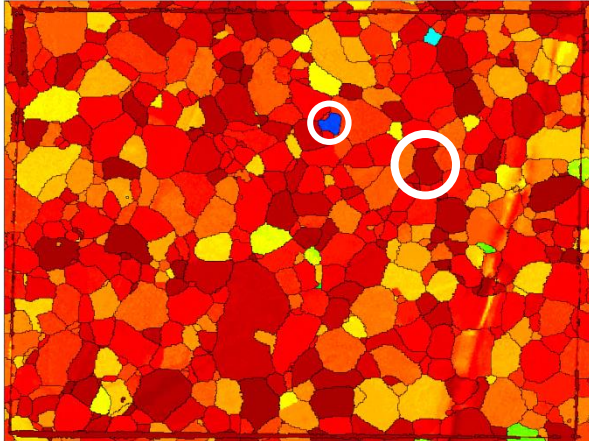
Effective Strain (%) (DIC)



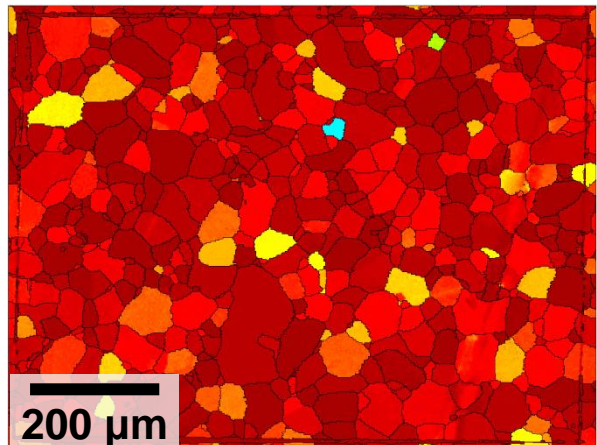


# The effects of six microstructural parameters on local strain were considered.

Schmid  $\{110\} \langle 111 \rangle$

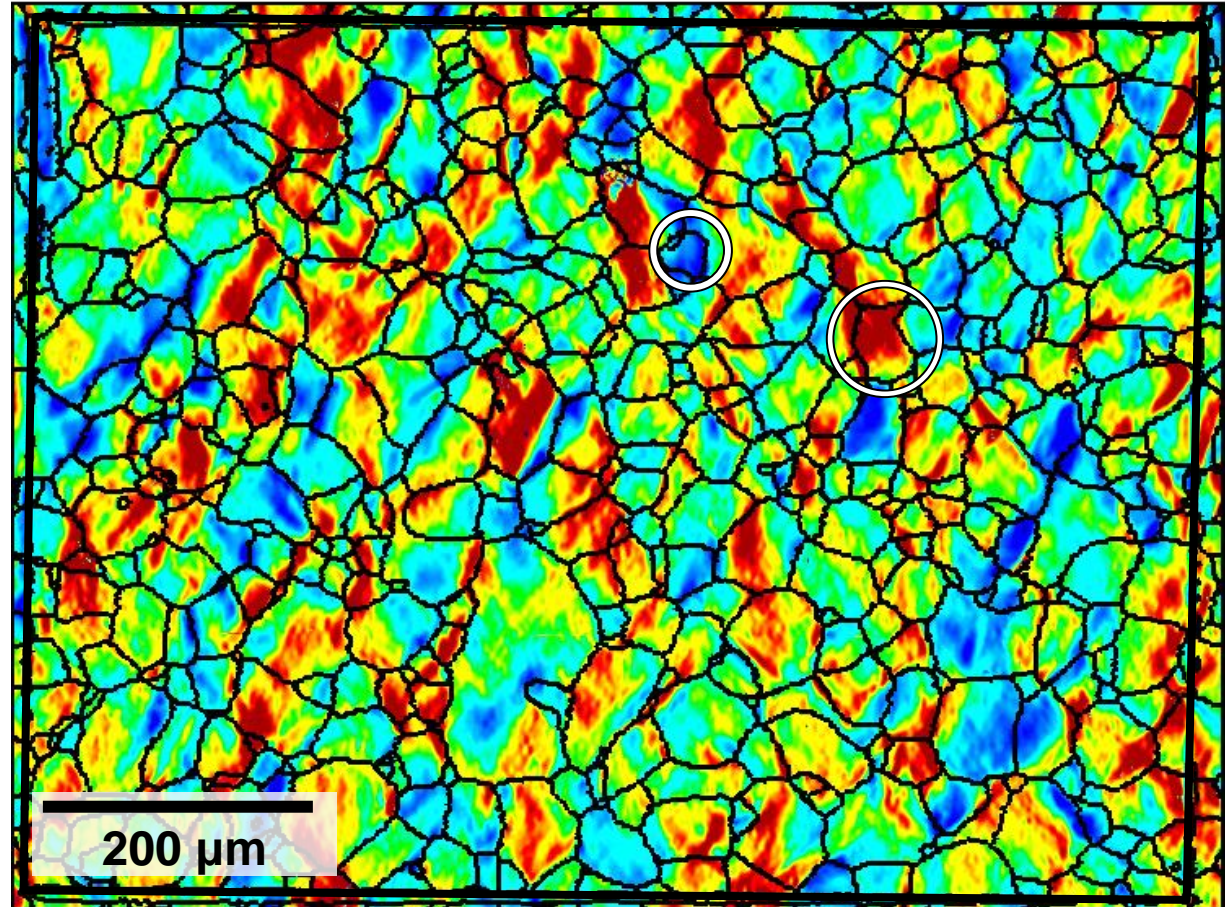


Schmid Factor of 3 Planes



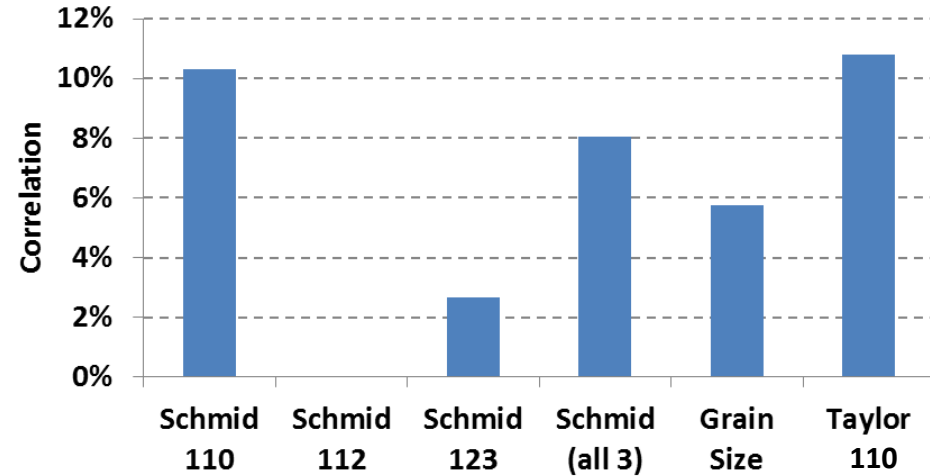
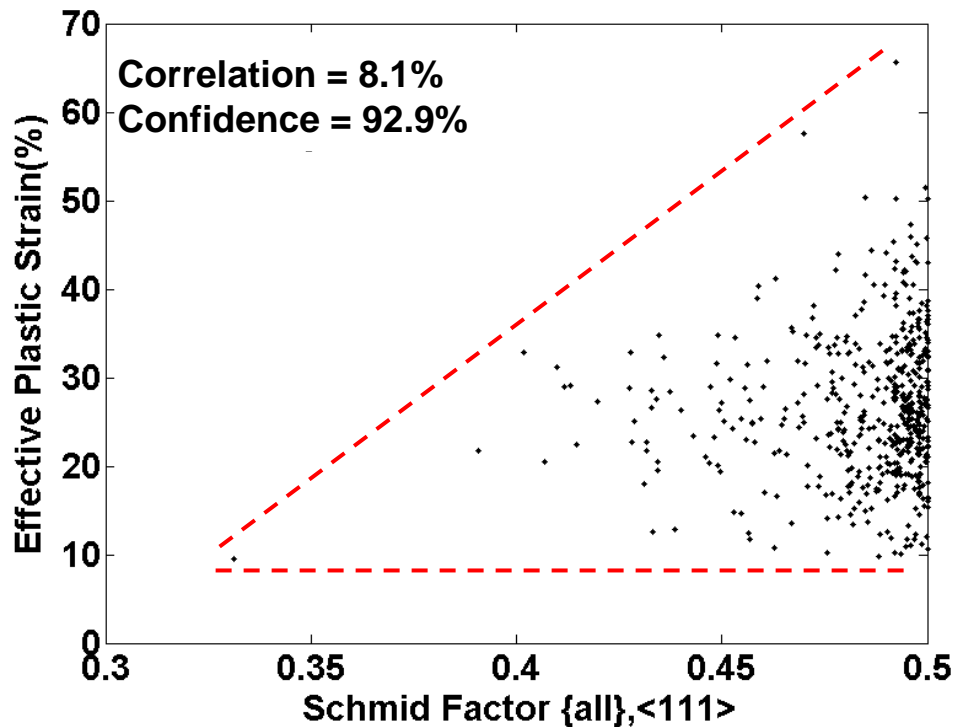
0.25 0.3 0.35 0.4 0.45 0.5  
Hard Soft

Effective Plastic Strain (at 25% Applied Strain)



0 10 20 30 40 50  
 $\epsilon_{eff} (\%)$

# There is some correlation between microstructure and average strain within each grain.

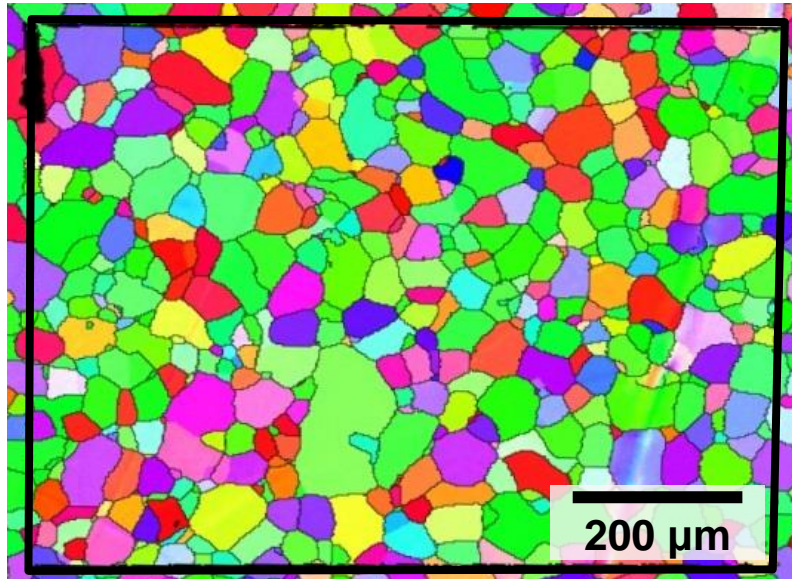


Parameter	Correlation	Confidence
Schmid 110	10%	98%
Schmid 112	-0.4%	6%
Schmid 123	2.7%	45.1%
Schmid (all 3)	8.1%	92.9%
Grain Size	5.8%	80%
Taylor 110	11%	99%

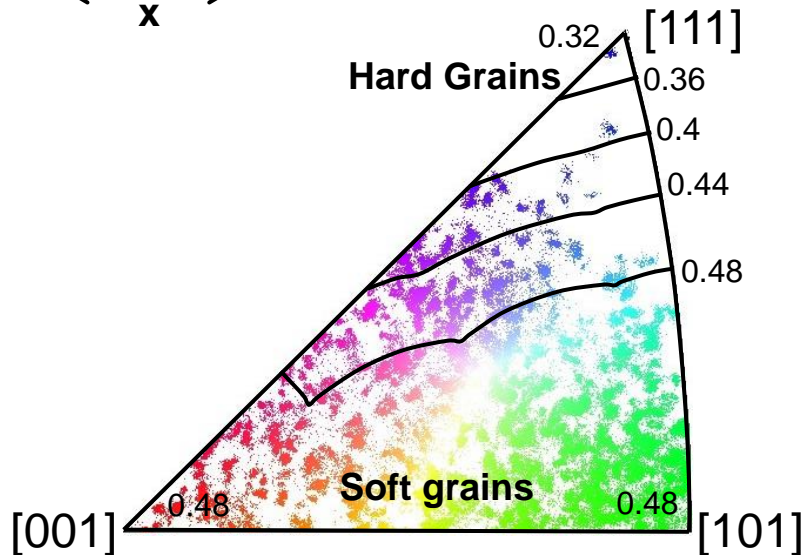
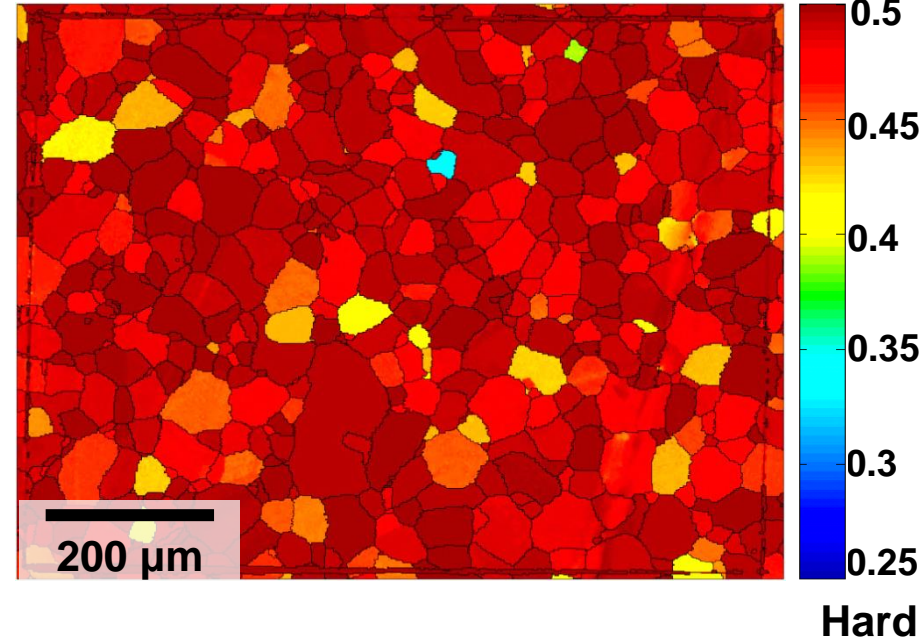


In BCC metals, only the  $\langle 111 \rangle$  grains are hard (low Schmid factors).

Grain Orientation



Schmid Factor  
 $\{110\}$ ,  $\{112\}$ ,  $\{123\}$ ,  $\langle 111 \rangle$

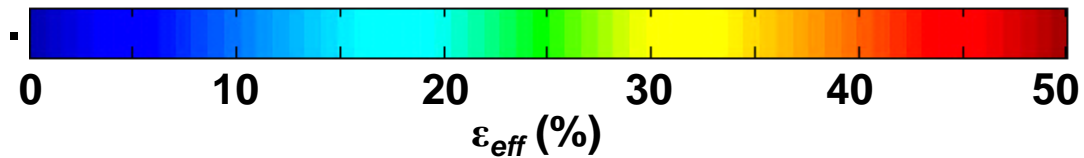
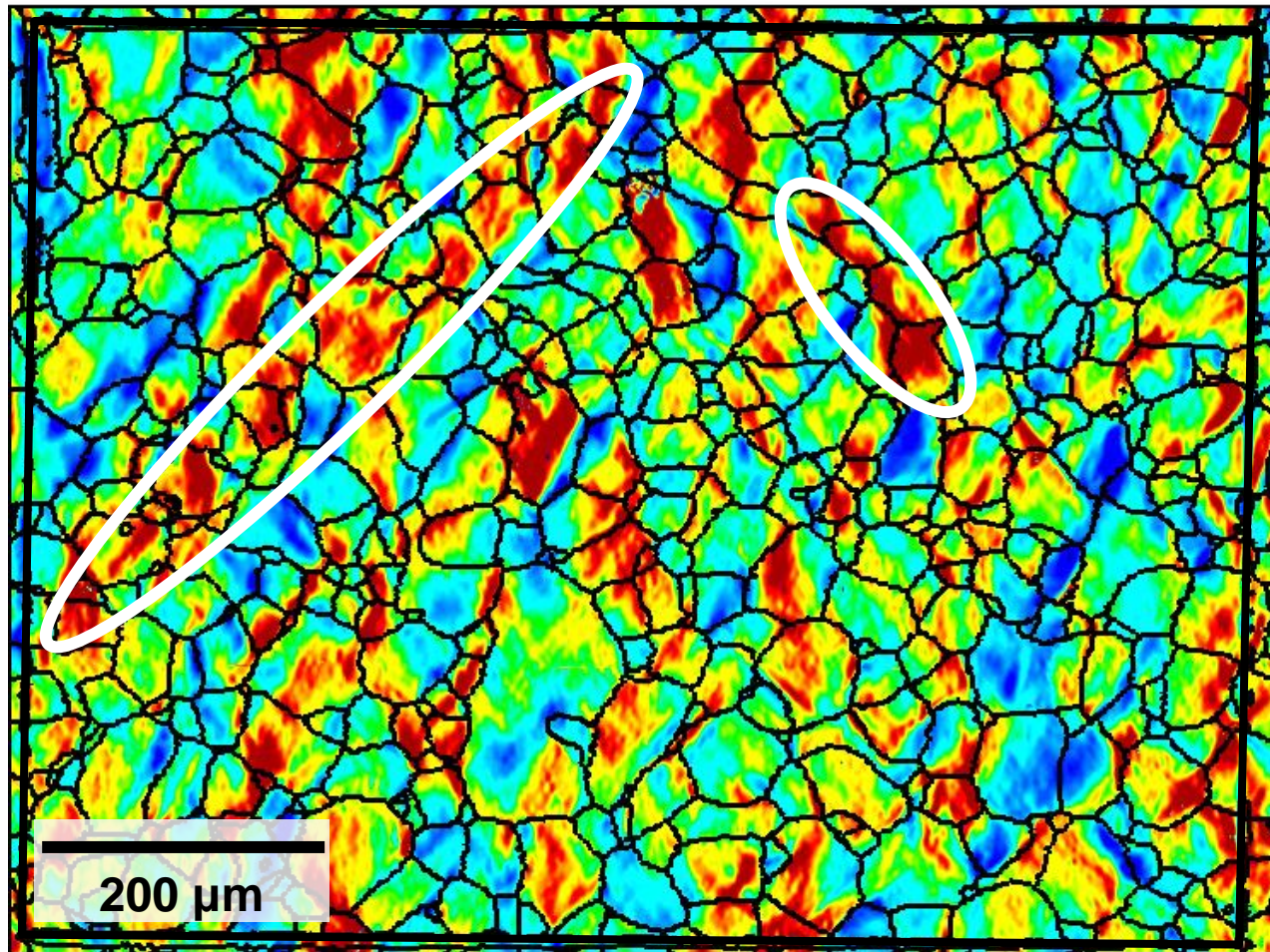


Schmid contours  
 $\{110\}$ ,  $\{112\}$ ,  $\{123\}$   
 $\langle 111 \rangle$

# Neighborhood effects are apparent in strain fields.

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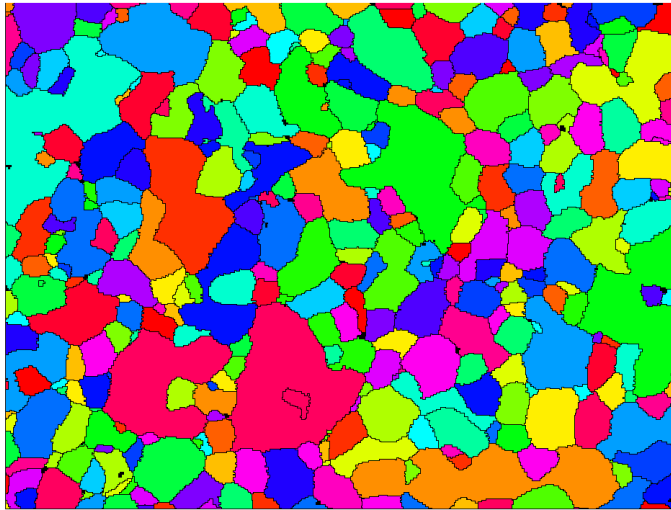
Local Effective Strain at 25% Applied Strain





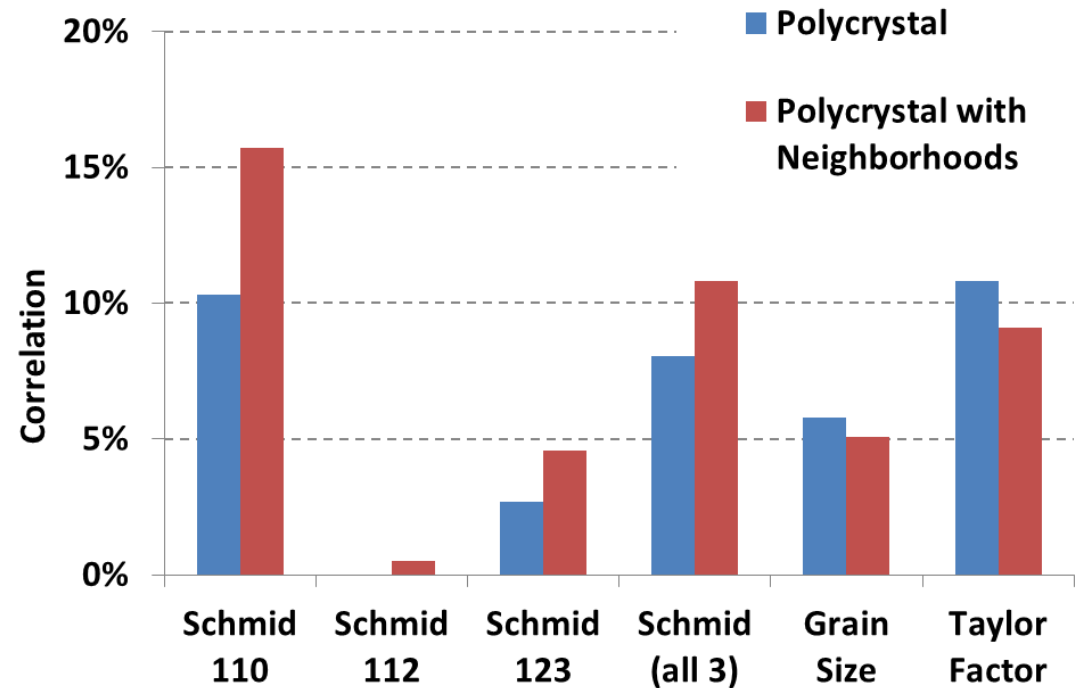
# Grain neighborhoods can be identified by grouping grains with similar orientations.

Grain Neighborhoods  
(random colors)



200 μm

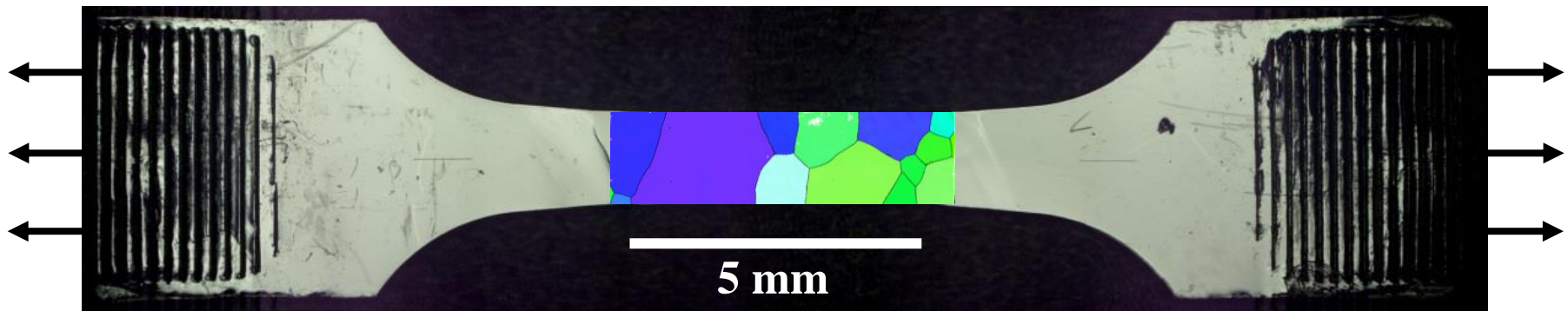
Misorientation angle of  $20^\circ$  defines GBs instead of  $5^\circ$ .



Parameter	Polycrystal		Neighborhoods	
	Correlation	Confidence	Correlation	Confidence
Schmid 110	10%	98%	16%	100%
Schmid 112	-0.4%	6%	0.5%	8%
Schmid 123	2.7%	45.1%	4.6%	61%
Schmid (all 3)	8.1%	92.9%	10.8%	96%
Grain Size	5.8%	80%	5.1%	65%
Taylor Factor	11%	99%	9.1%	91%

# Oligocrystals

Specimens where deformation is controlled by a few grains (3–20).

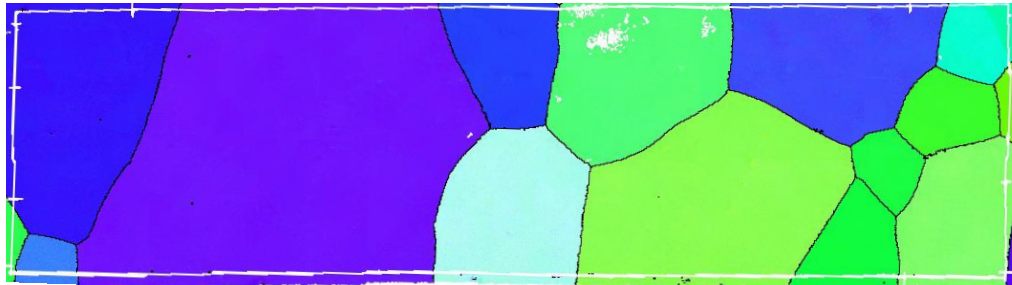


- Ta oligocrystals were made by annealing.
- This has much fewer neighborhood effects.

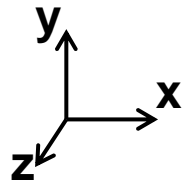
# This oligocrystal also has more hard grains.

27

Grain orientation



x



1 mm



[001]

[101]

Schmid contours  
{110},{112},{123}  
<111>

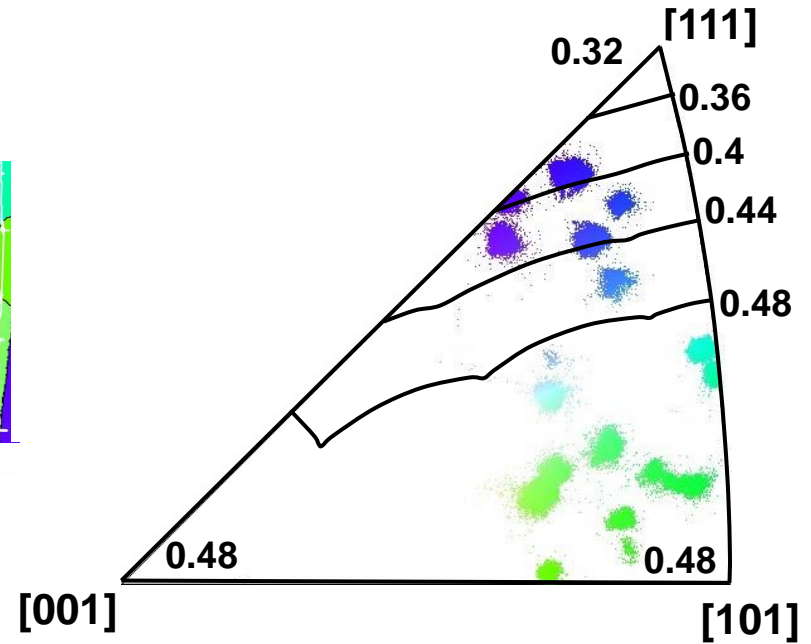
Schmid {110}<111>



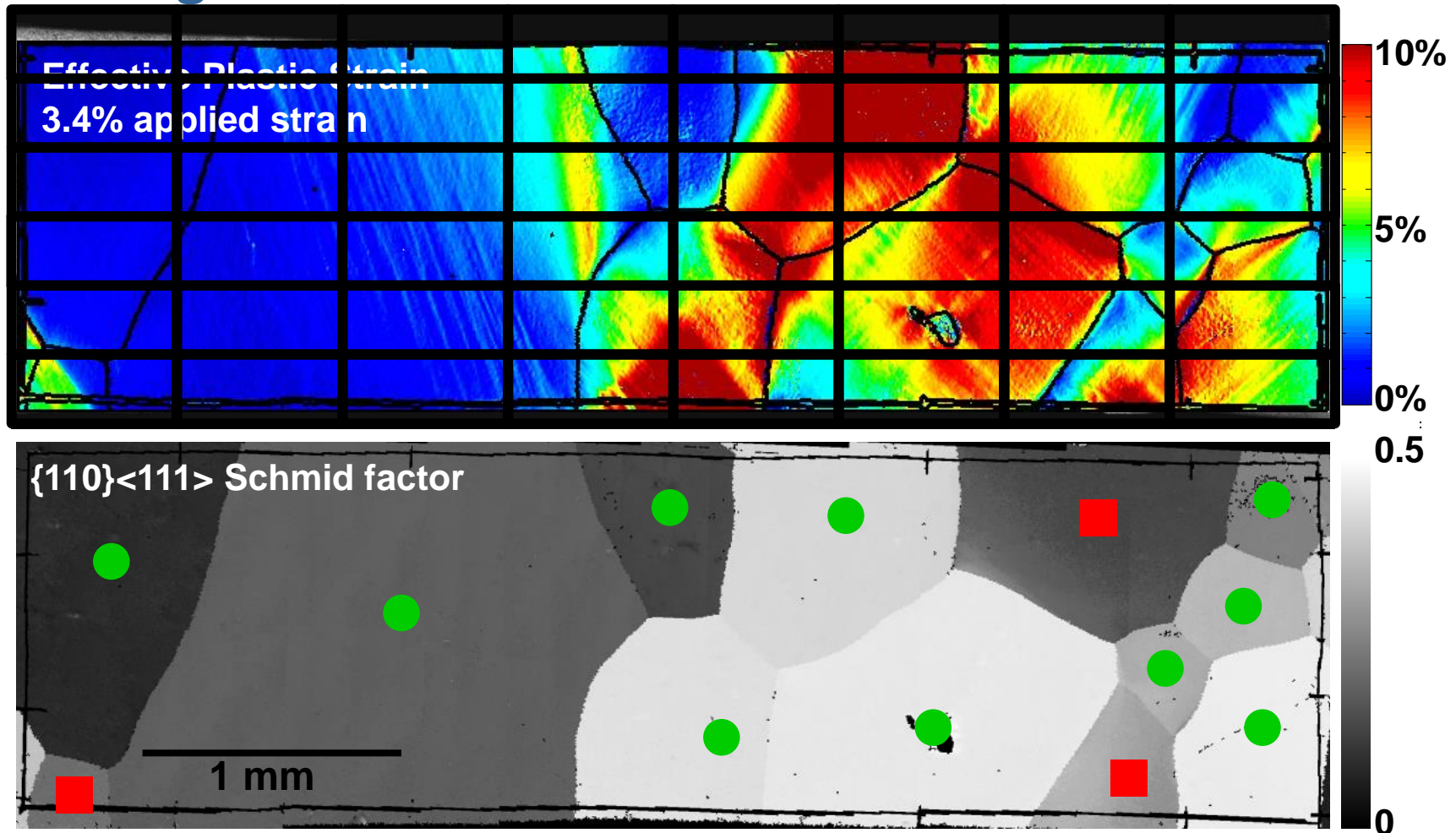
0.5 (soft)

0.35 (hard)

1 mm



# Strain accumulation agrees with Schmid factor for most grains.

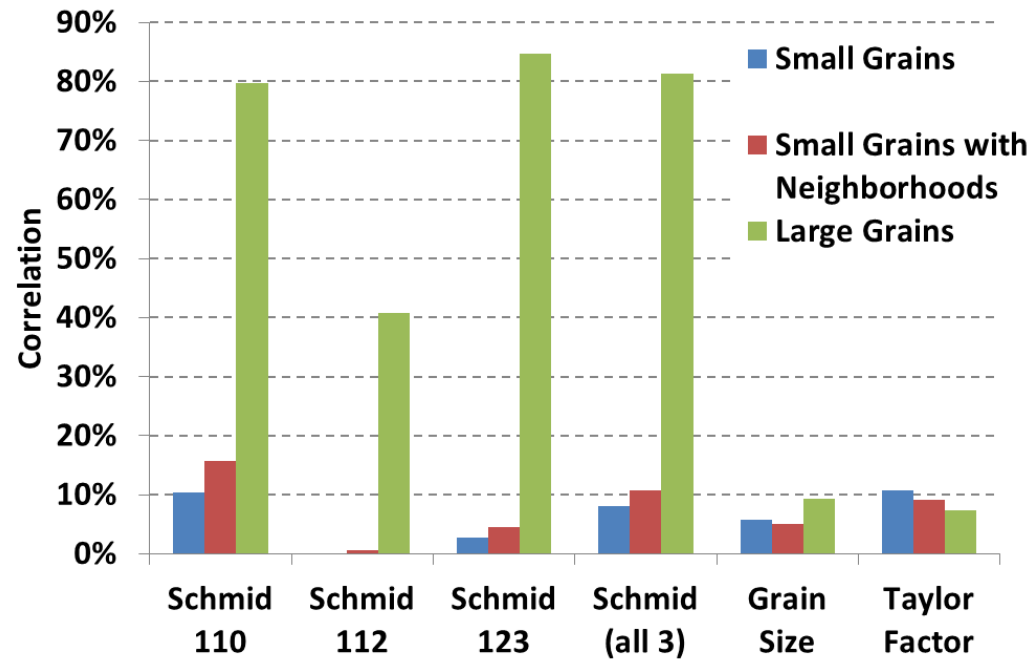
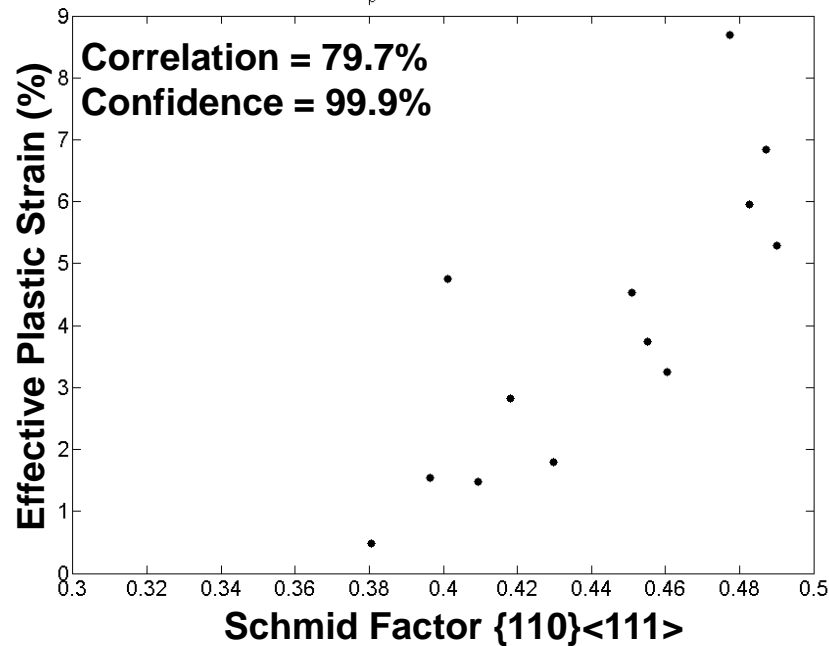


- 10 out of 13 grains have strain and Schmid agree (77%).
- Neighbor effects may explain the other cases.



# Oligocrystal shows strong correlations between Schmid factor and strain.

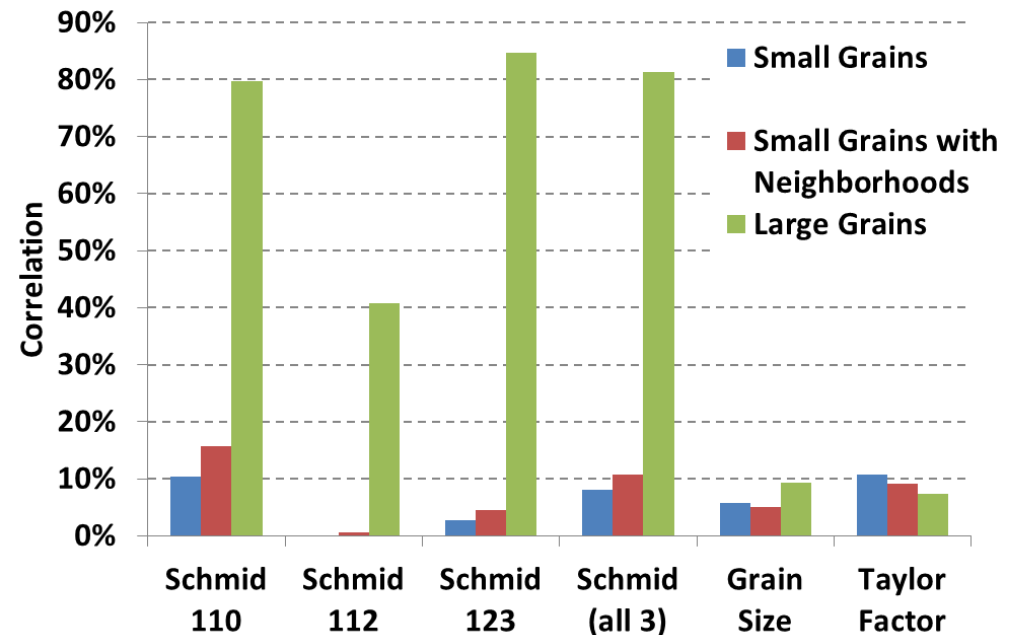
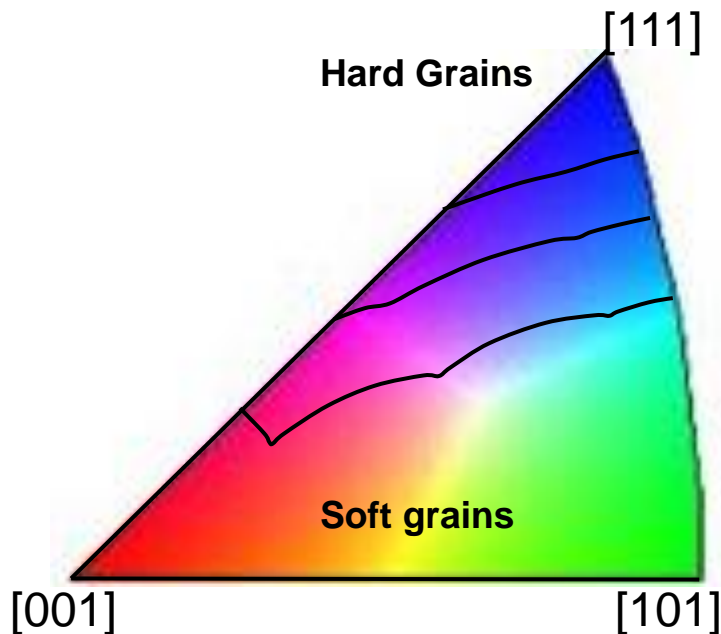
D Avg( $\text{Seff}_p$ ) vs. Schmid By Grain 110



	Polycrystal		Neighborhoods		Oligocrystal	
Parameter	Correlation	Confidence	Correlation	Confidence	Correlation	Confidence
Schmid 110	10%	98%	16%	100%	80%	98%
Schmid 112	-0.4%	6%	0.5%	8%	41%	83%
Schmid 123	2.7%	45%	4.6%	61%	85%	100%
Schmid (all 3)	8.1%	93%	10.8%	96%	81%	100%
Grain Size	5.8%	80%	5.1%	65%	9.3%	24%
Taylor Factor	11%	99%	9.1%	91%	7.3%	19%

# Summary on relating microstructure to deformation.

- In BCC metals, the only hard grains are those with the  $\langle 111 \rangle$  direction aligned near the tensile axis.
- $\{110\}$  are the most likely slip planes in Ta.
- Strain accumulation is related to Schmid factor.
- Grain neighbors are important!



## 1. Motivation

- Why do we need to understand grain scale deformation?

## 2. Background

- Slip in FCC and BCC metals

## 3. Relating microstructure to deformation

- Schmid factors, etc.

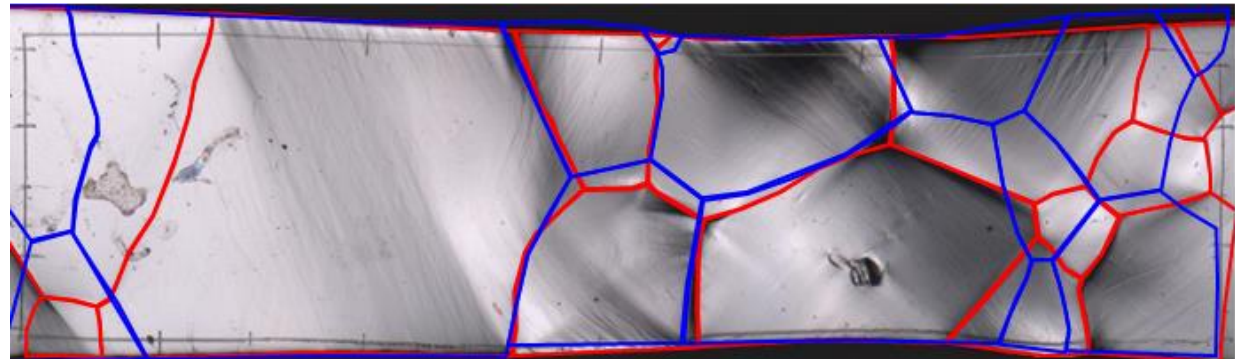
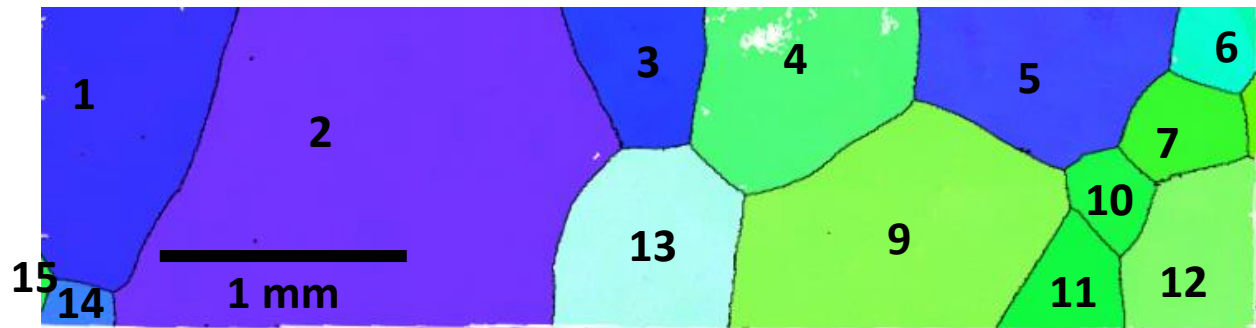
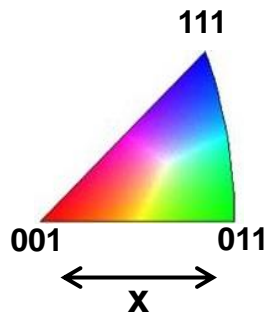
## 4. Predicting deformation behavior at the grain scale

- Crystal plasticity models

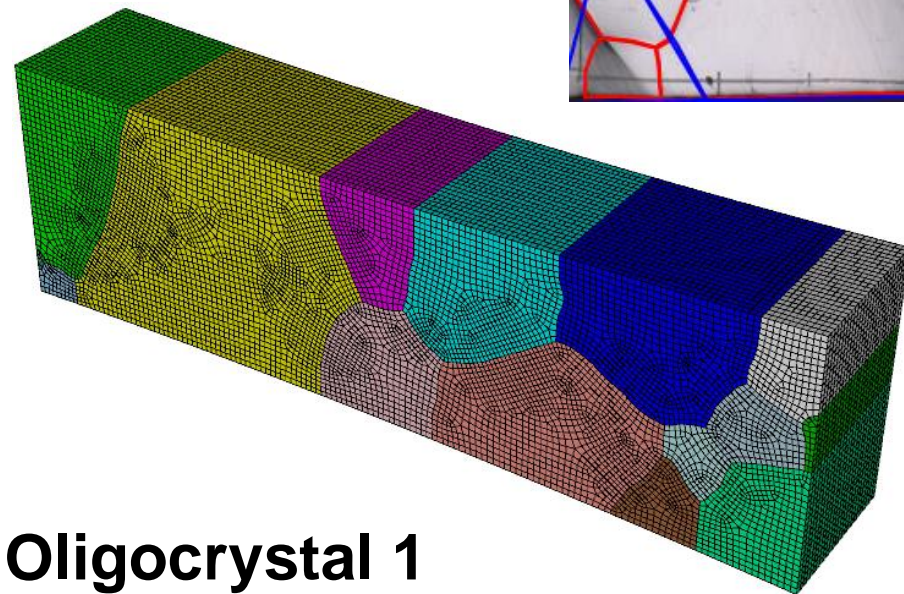
## 5. Conclusions



Oligocrystals with pseudo-2D grains provide more accurate comparisons between models and experiments.



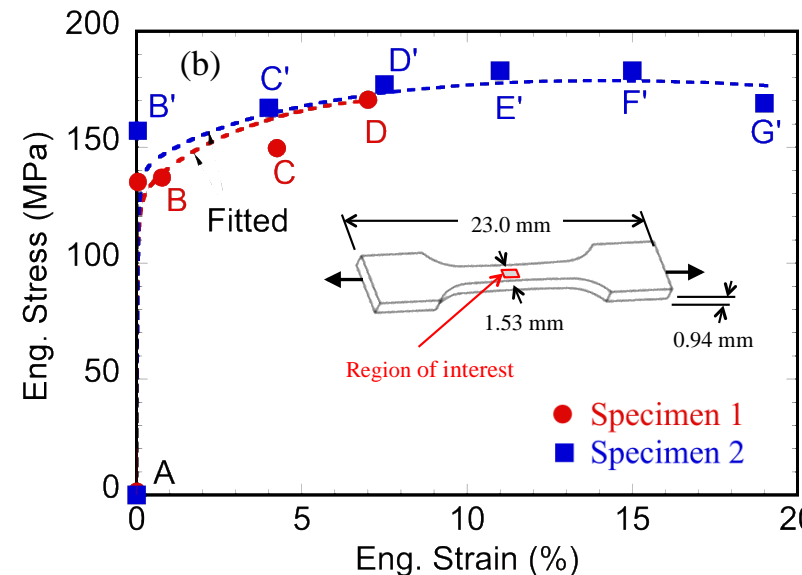
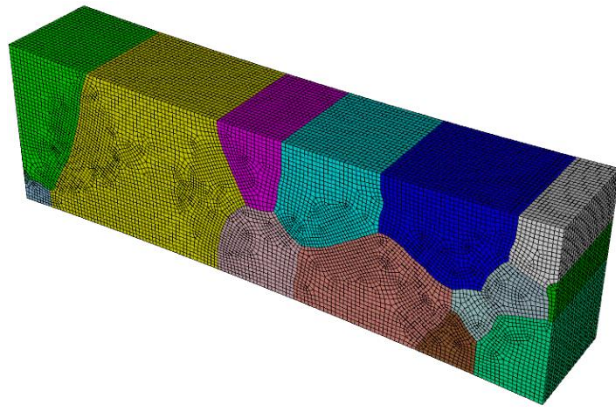
- Front grain boundary
- Back grain boundary



Oligocrystal 1

# Crystal plasticity finite element model.

- FEM code (JAS-3D) developed at Sandia.
- Dislocation density based hardening.
- Temperature and rate dependent, based on kink pair theory.
- Hexahedral elements (8 nodes).
- 50 elements through specimen thickness.
  - ~1.5 million total elements
  - ~30,000 surface elements

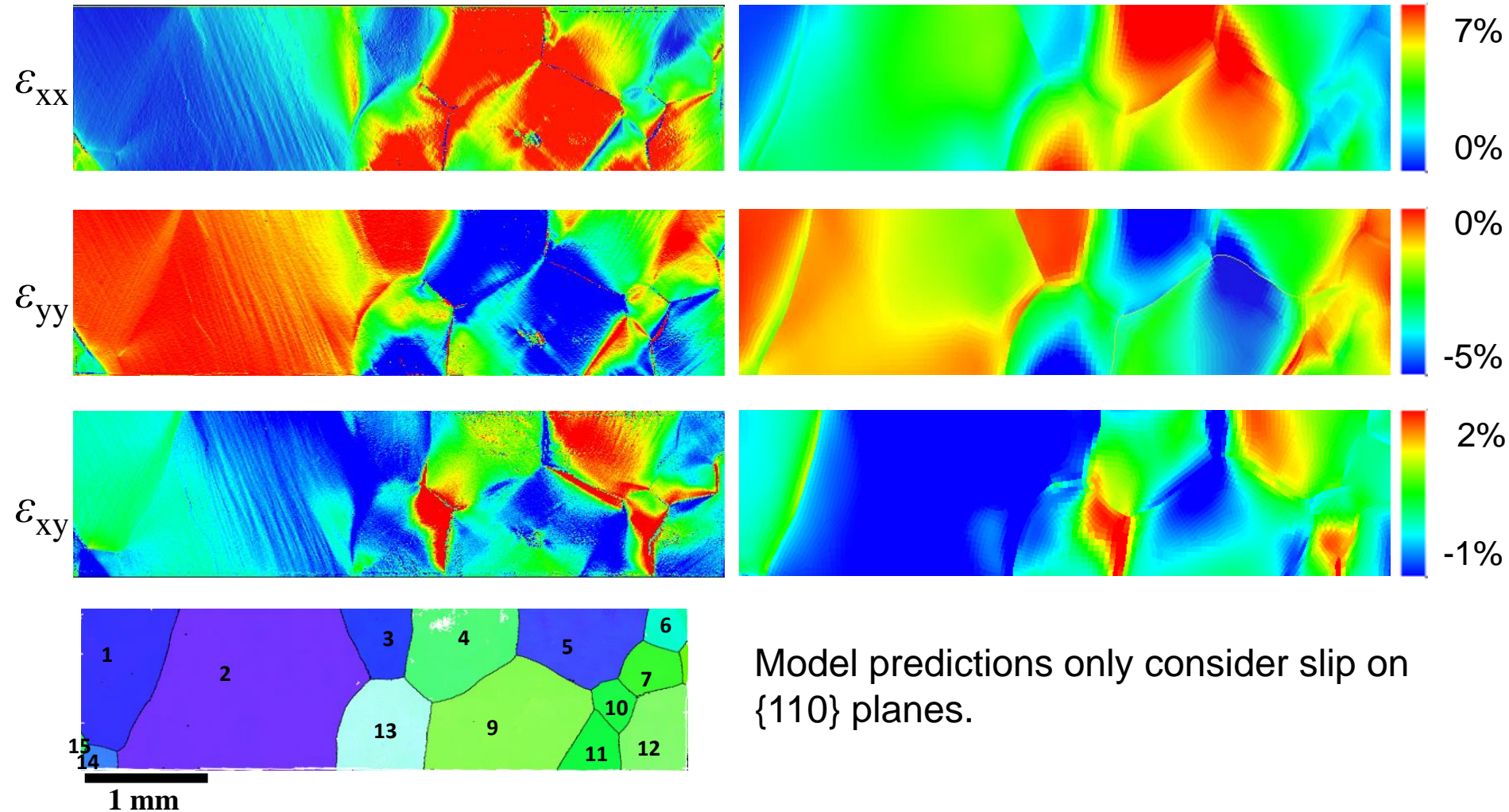


Parameters	Values	Parameters	Values	Parameters	Values
$\tau_0^{LT}$	406 MPa	$C_{11}$	267 GPa	$k_a$	$1.4 \times 10^6 (m^{-1})$
$\tau_0^{EI}$	320 MPa	$C_{12}$	161 GPa	$k_b$	14
$2H_k$	0.85 eV	$C_{44}$	82.5 GPa	$\bar{\tau}_1$	27 MPa
$\dot{\gamma}_0$	$2.99 \times 10^6 s^{-1}$	$b$	2.87 Å	$\bar{\tau}_2$	37 MPa

# Model predictions of strain agree well with experimental measurements in most places.

Experimental Strains (DIC)

Model Strains (CP-FEM)



Model predictions only consider slip on  $\{110\}$  planes.

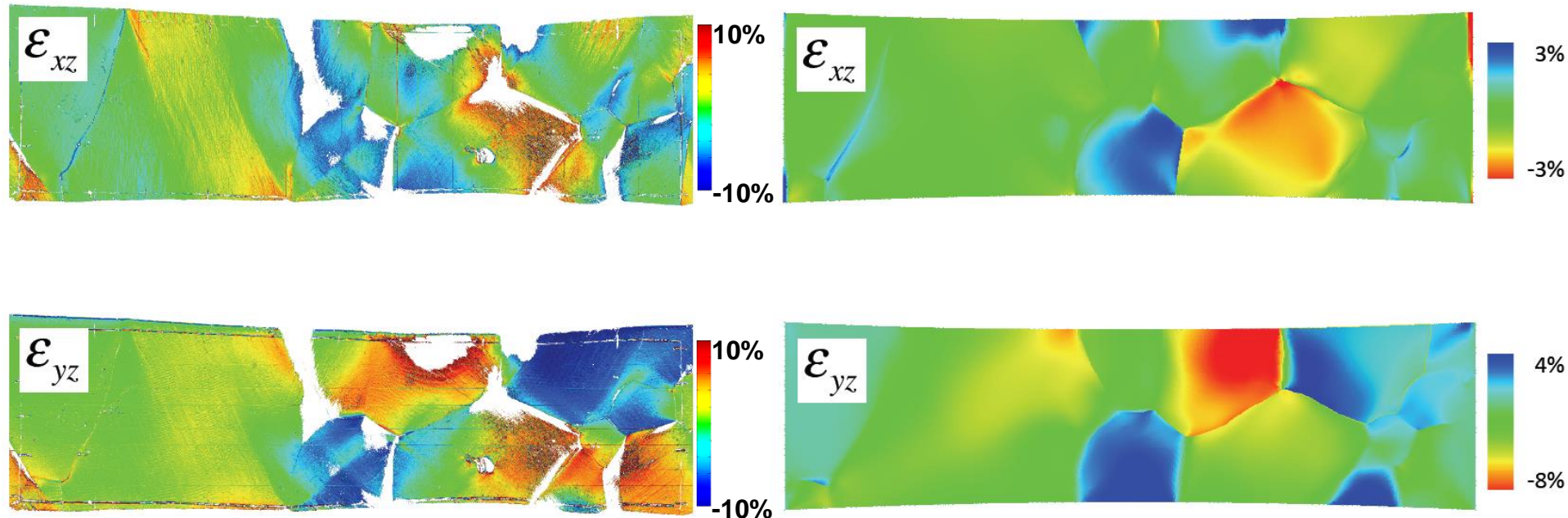


# Compare out-of-plane strain (profilometry).

Oligocrystal 1 (7% deformation)

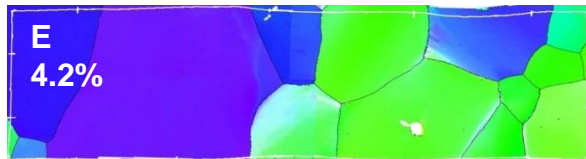
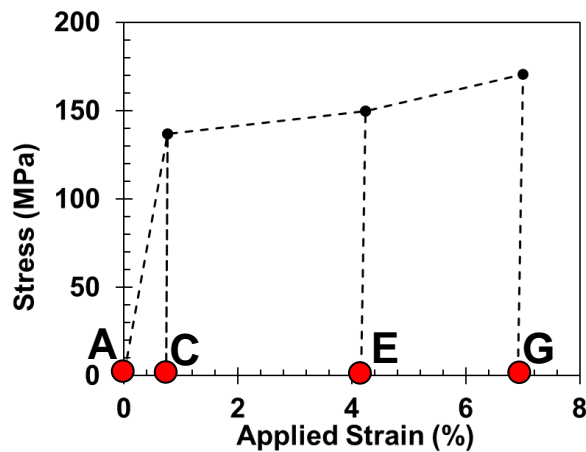
Profilometry Measurements

CP-FEM Predictions



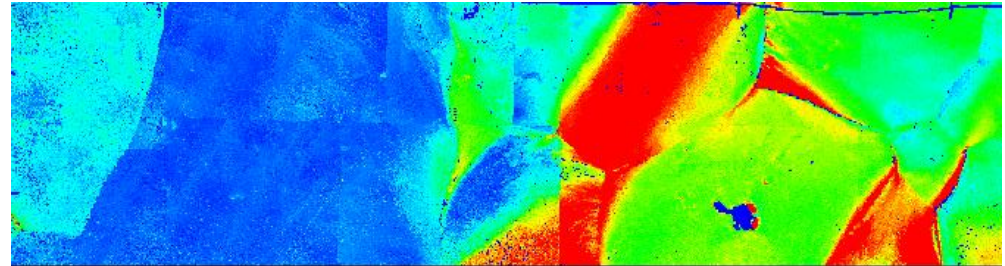
- Good qualitative agreement within grains.
- Grain boundaries have large strains not captured by model.

# Compare orientation changes between model and experiment.

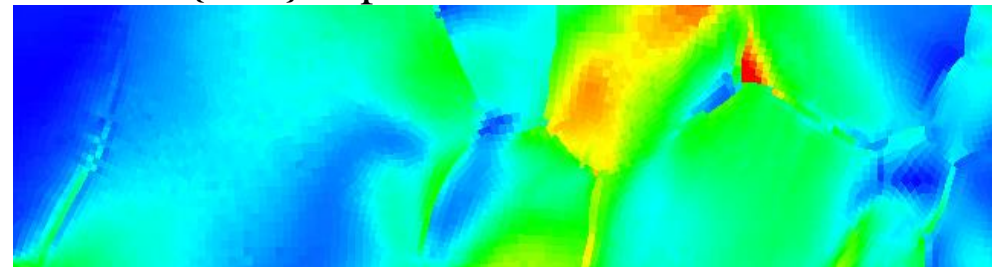


## Orientation Change A → E

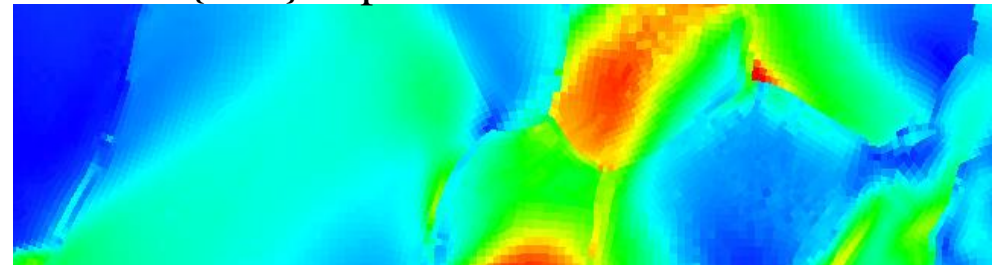
EBSD



CP-FEM {110} slip

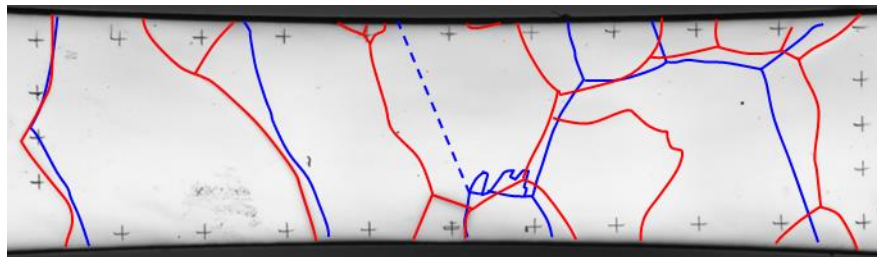
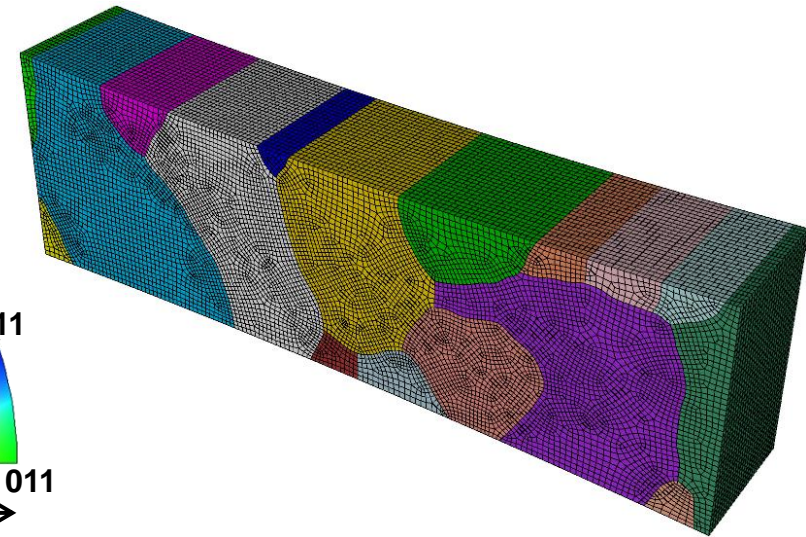
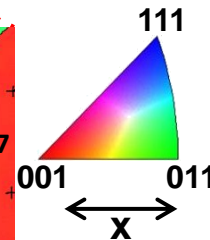
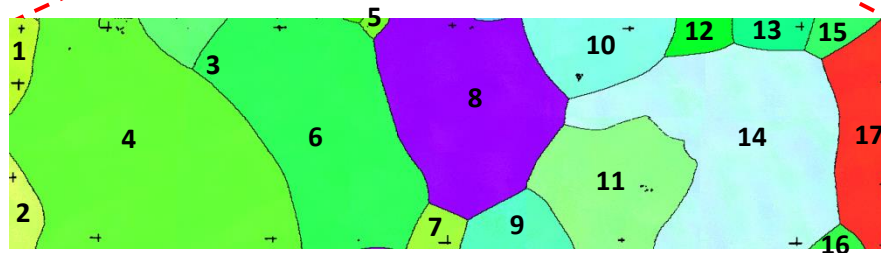
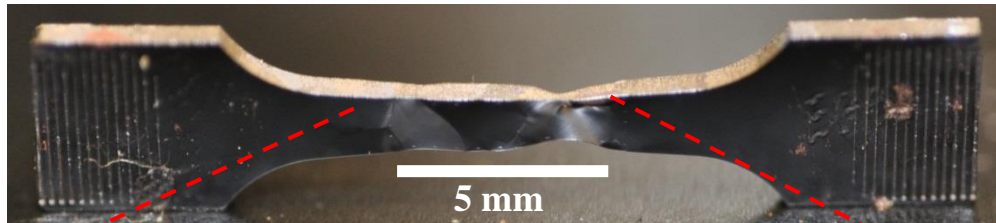


CP-FEM {112} slip

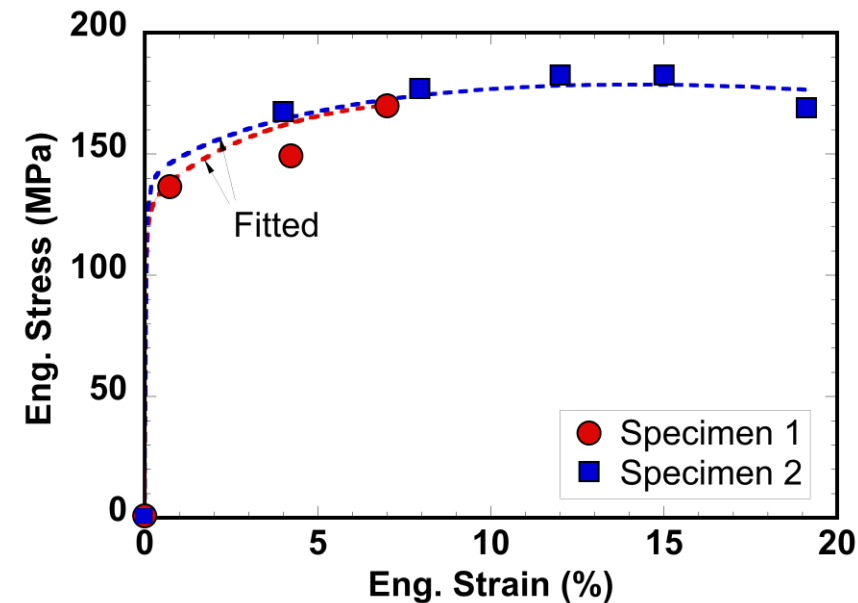


# Oligocrystal 2. Loading to 19.2% to see failure initiation.

37



— Grain boundary (Front) — Grain boundary (Back)

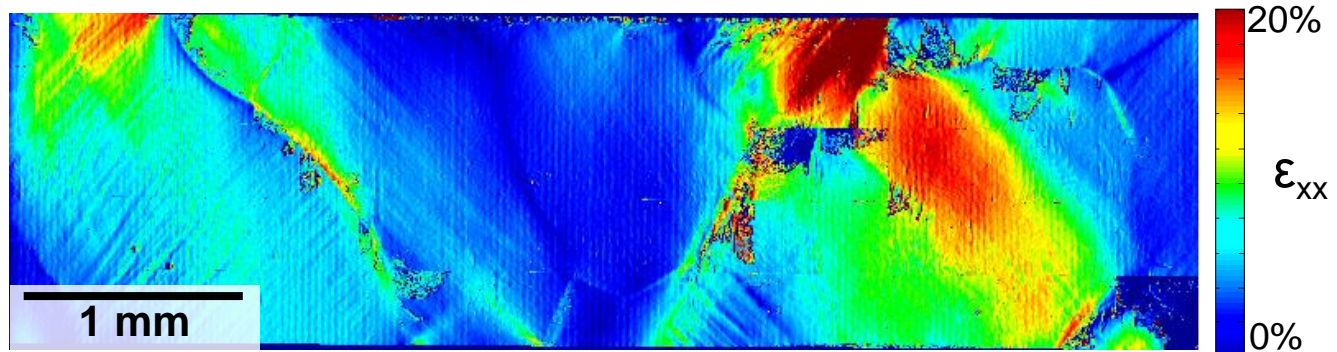


EBSD before and after only.

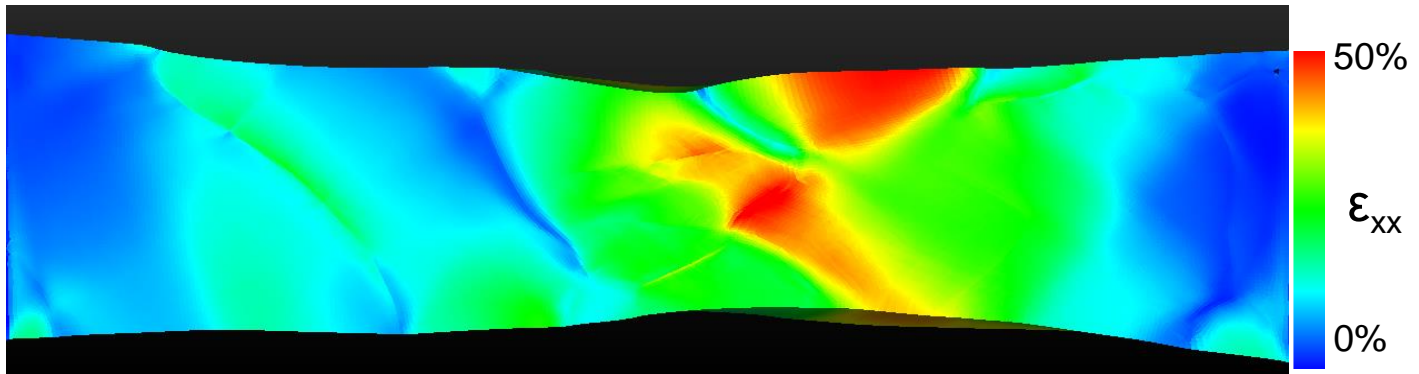


# Each model matches experiment better in some locations.

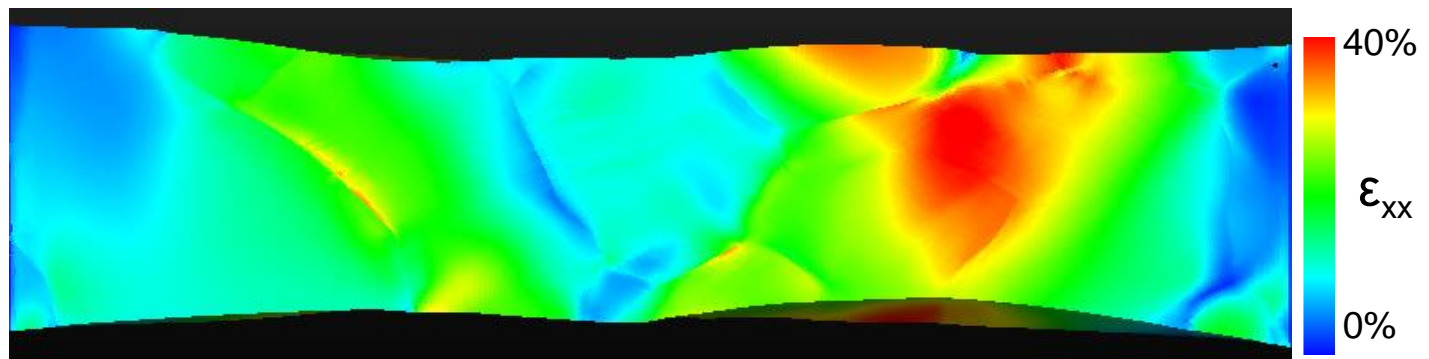
Experiment  
DIC  
~10% strain



Model  
{110} slip  
19% strain



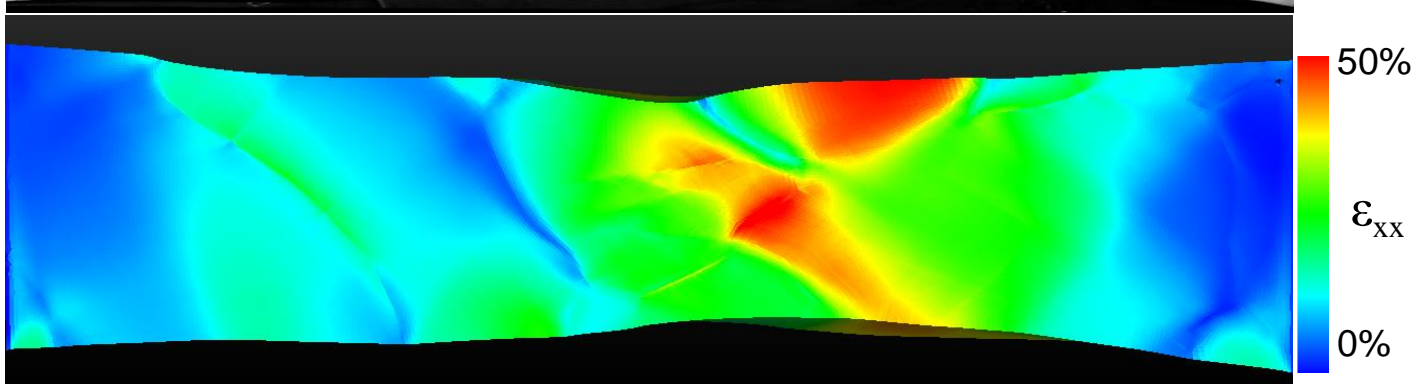
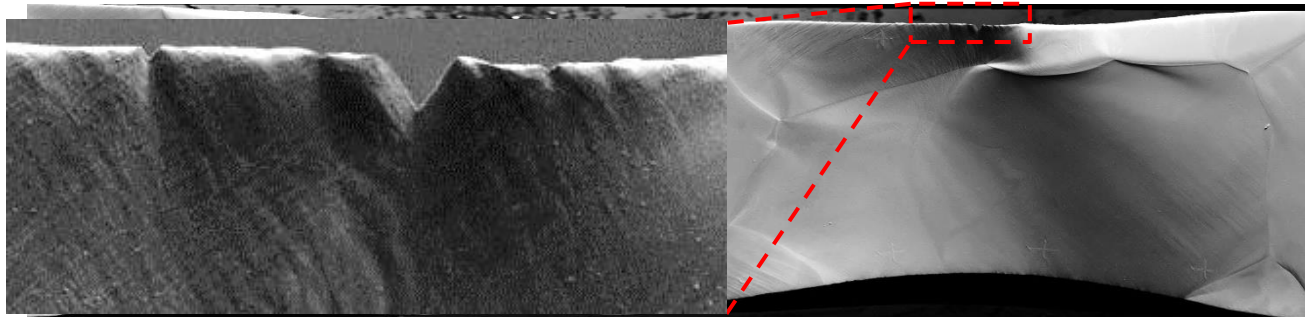
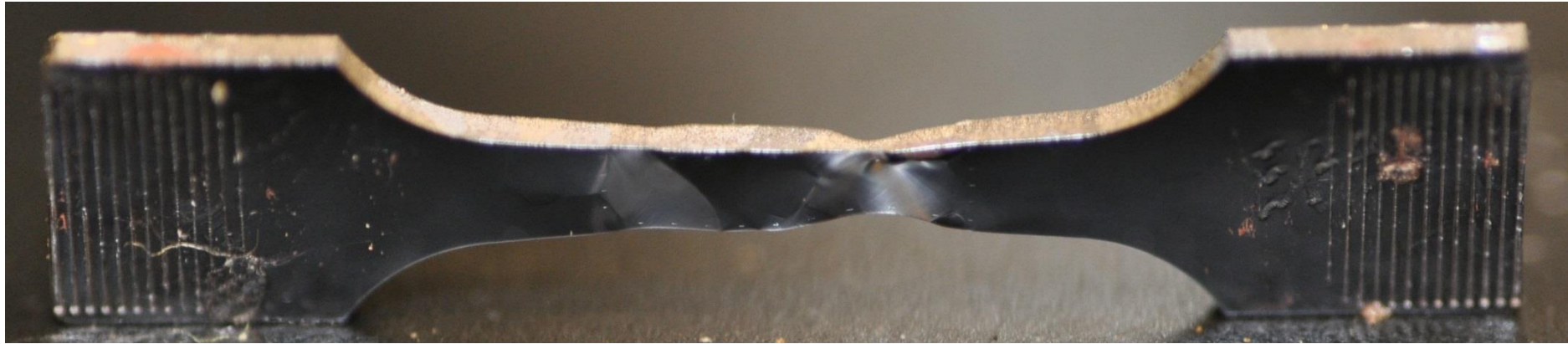
Model  
{112} slip  
19% strain





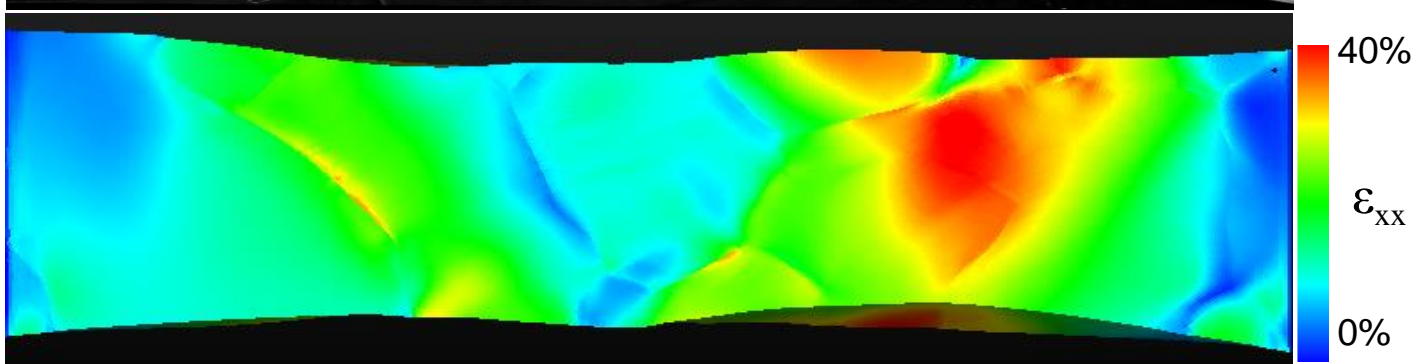
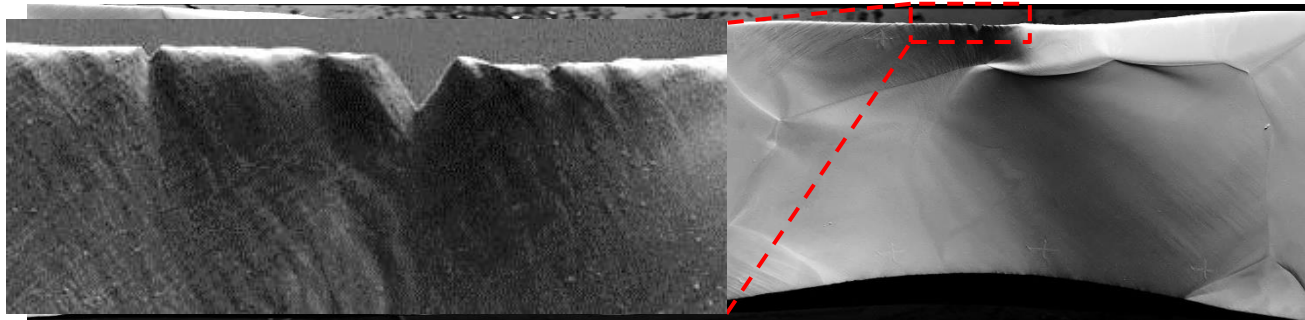
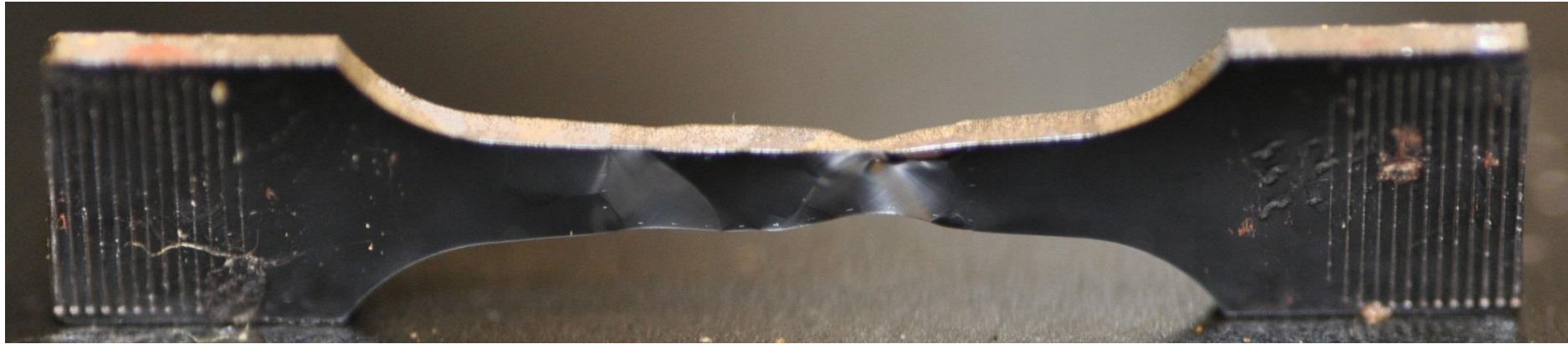
# Model predicts high strain at location of observed crack initiation.

39



# Model predicts high strain at location of observed crack initiation.

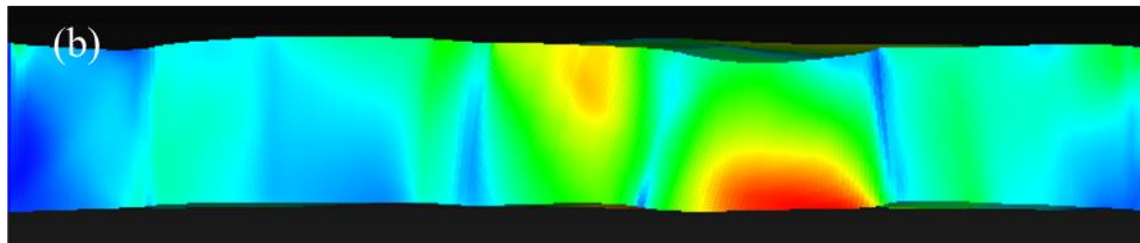
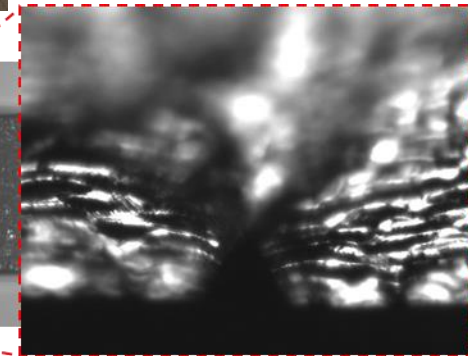
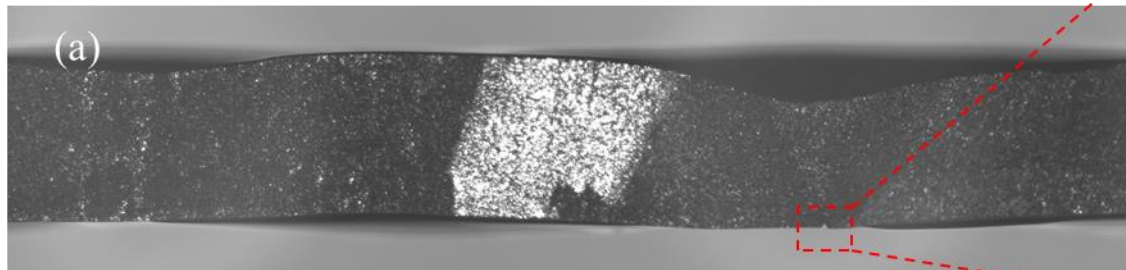
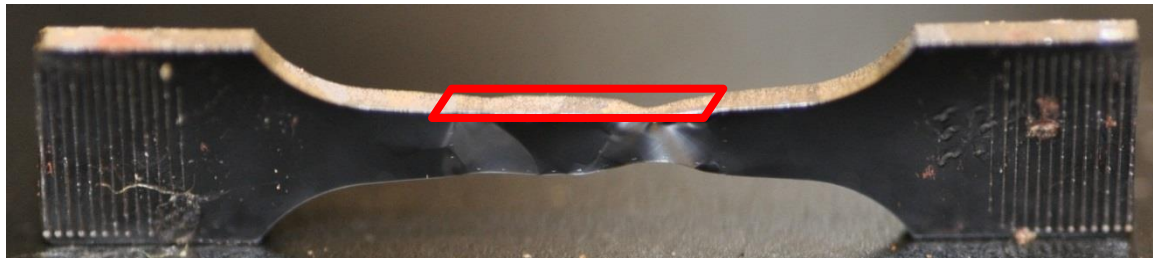
40



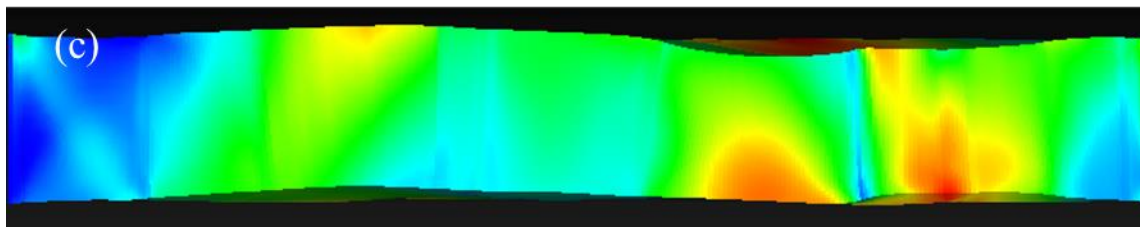
{112} slip

# The pseudo-2D model predicts failure in 3D (on front surface).

41



{110} slip



{112} slip

## 1. Motivation

- Why do we need to understand grain scale deformation?

## 2. Background

- Slip in FCC and BCC metals

## 3. Relating microstructure to deformation

- Schmid factors, etc.

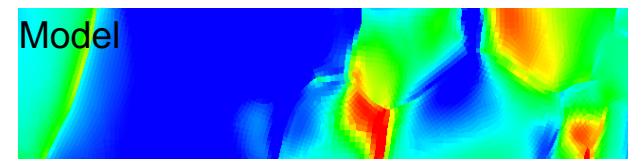
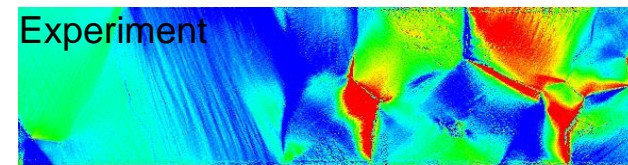
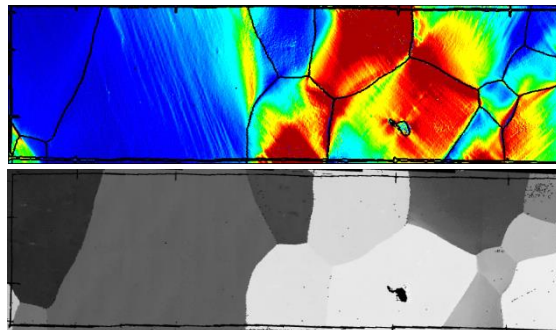
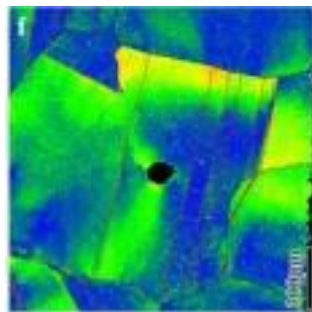
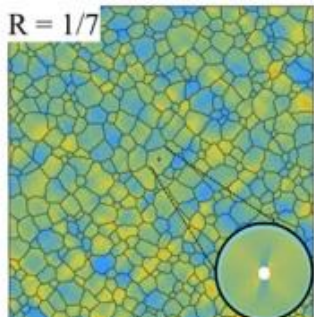
## 4. Predicting deformation behavior at the grain scale

- Crystal plasticity models. **Conclusions**
- **Time to wake up.**



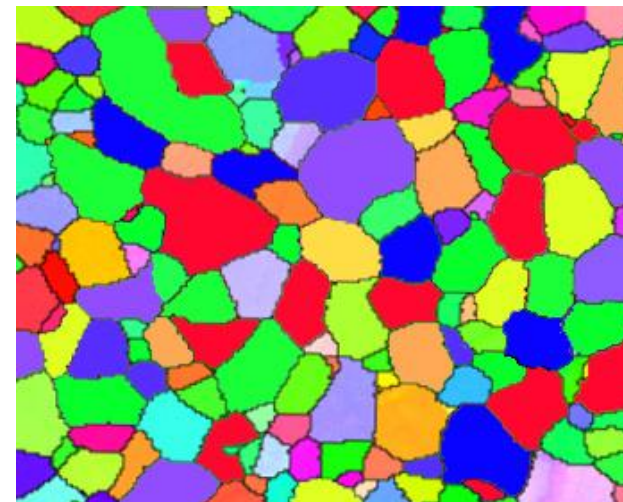
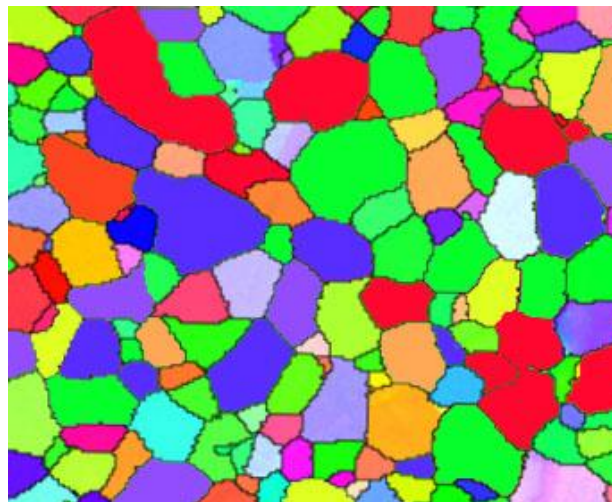
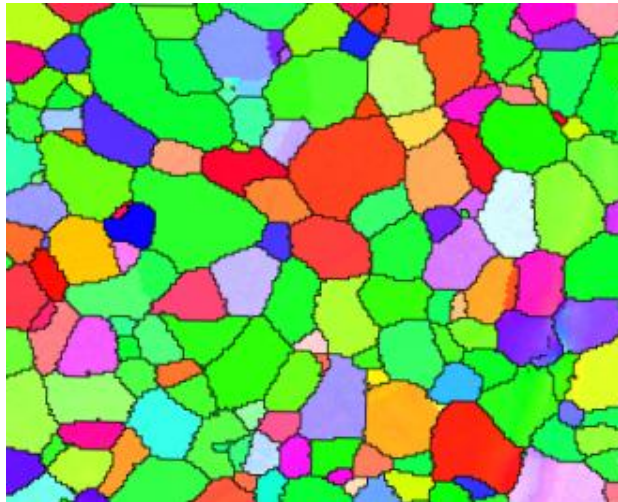
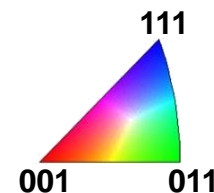
# Conclusions

1. Need to consider microstructural effects when specimen dimensions are comparable to grains size.
2. Local microstructure can be more important than small geometrical stress concentrators.
3. Schmid factor is predictive of deformation.
4. Grain neighbor effects are important!
5. Our crystal plasticity model agrees with experiments!



# Next steps in modeling

1. Refine and verify model further.
  - Apply to polycrystals
  - Other materials
  - 3D
2. Run the model on similar, different microstructures.
3. Calculate variability in material properties from statistical microstructural variability.



## Sandia National Laboratories

- **Brad Boyce**
- **Hojun Lim**
- **Corbett Battaile**
- **Tom Buchheit**
- **Chris Weinberger**
- **Blythe Clark**
- **Alice Kilgo**
- **Bonnie Mckenzie**
- **Joe Michael**
- **Chuck Walker**
- **Nick Argibay**

## University of Illinois at Urbana-Champaign

- **John Lambros**
- **Wael Abuzaid**
- **Mallory Casperson**
- **Huseyin Sehitoglu**