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# Shock compression of Hydrocarbon Polymers:

A molecular dynamics approach with comparison to DFT, Continuum and Experiment

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**SAND 2011-**

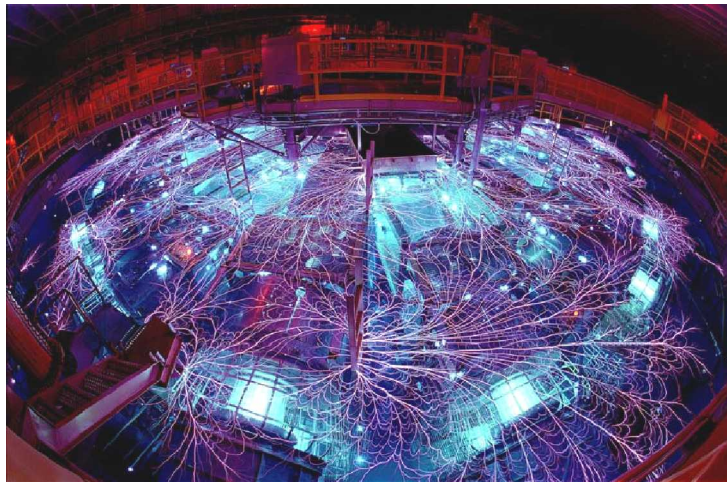
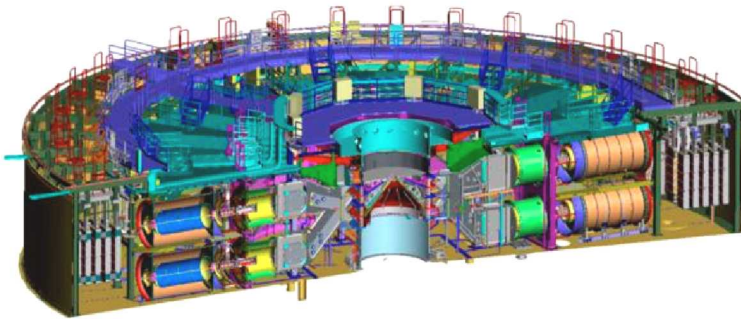


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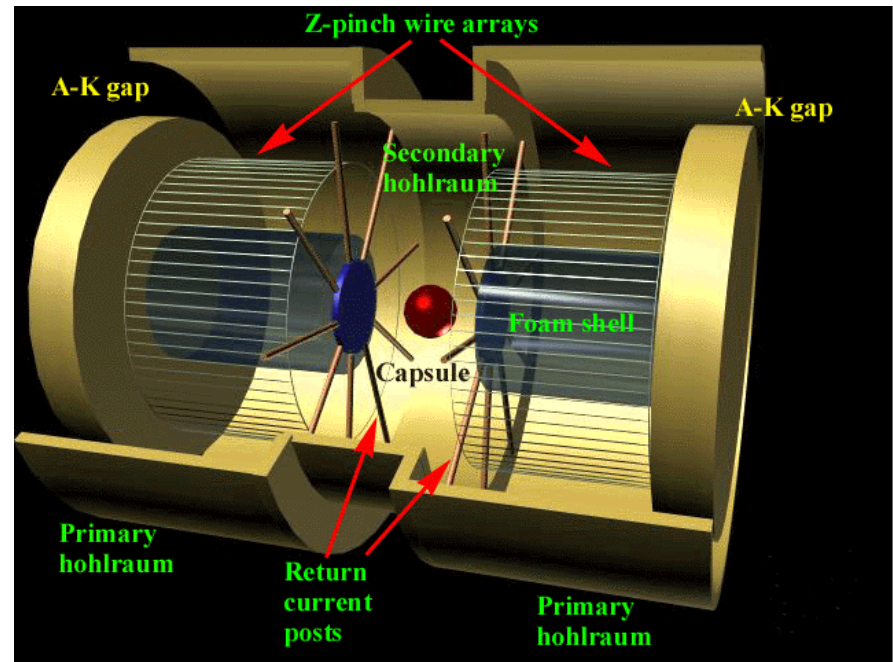
# Motivation for polymer shock simulation

## Z-machine at Sandia National Labs



Z-machine experiments allow:

- Extremely high-pressure shock studies
- Inertially confined fusion (ICF) research
- Better understand EOS for polymers and mixtures



Double-ended Z pinch with foam-shell capsule

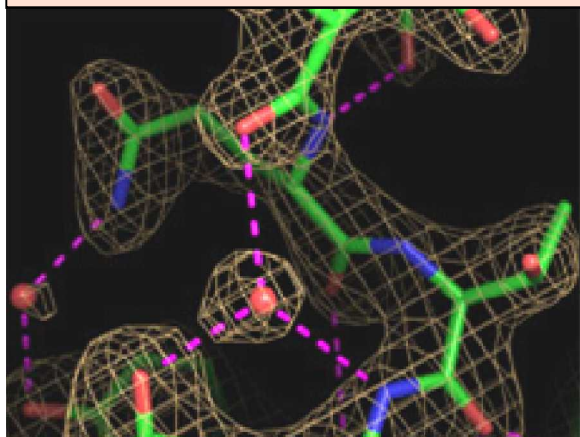


# Hierarchy of modeling approaches

Shock response in hydrocarbon foams depend on processes and structure at several length scales:

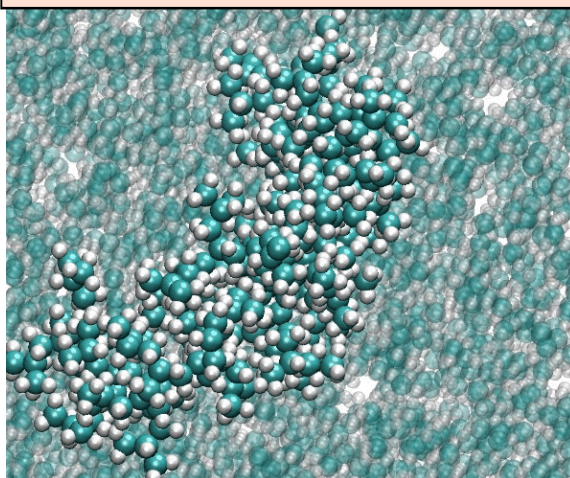
## Quantum scale

fs times & Å lengths



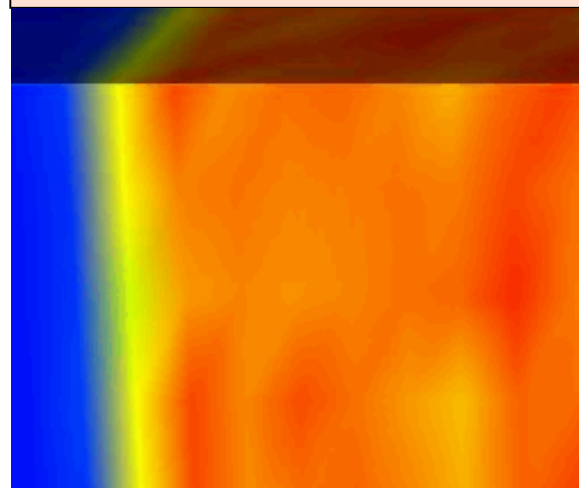
## Molecular scale

ns times & 10s nm lengths



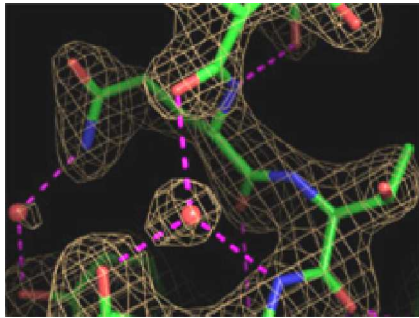
## Continuum scale

$\mu$ s times & nm-mm lengths

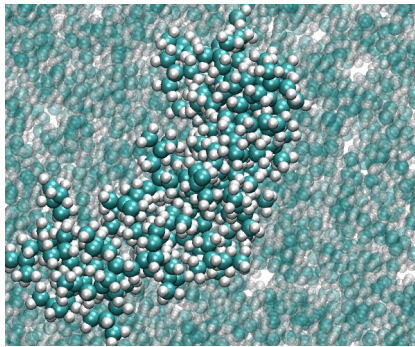


Increasing time and length scales

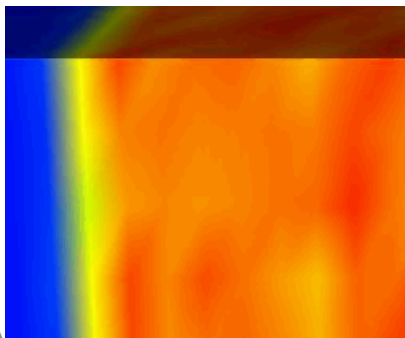
# Hierarchy of modeling approaches



- VASP 5.1 code (Georg Kresse, Vienna, Austria)
- Plane-wave basis allows controlled convergence & free electrons/ionization
- Finite-temperature DFT with Projector augmented wave core functions (PAW)
- Sandia developed AM05 functional (Armiento & Mattsson, Phys Rev B 2005)
- Cochrane, et al., *Shock Compression in Condensed Matter Proceedings*. (2011)

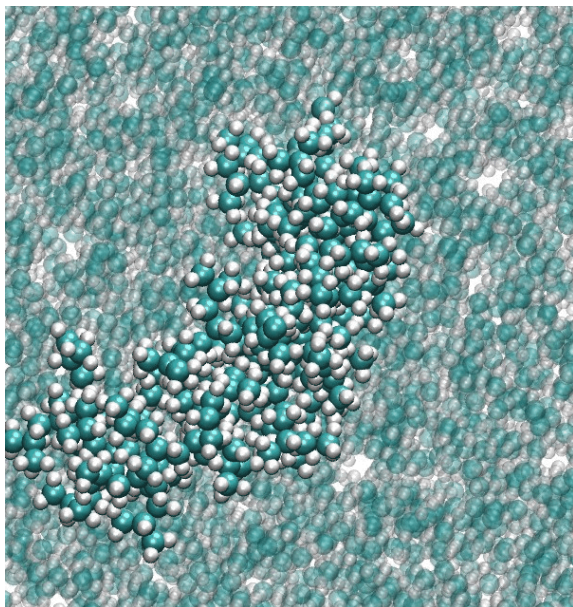


- LAMMPS code (Steve Plimpton, et al., Sandia National Labs)
- MD engine for atoms, molecules, or particles at any length/time scale
- Multiple common interatomic potentials allows for comparison
- Integrated MPI for massively parallel simulations
- <http://lammps.sandia.gov>
- Lane, et al., *Shock Compression in Condensed Matter Proceedings*. (2011)



- ALEGRA code (Sandia National Labs)
- MHD engine for 3D modeling of continuum systems
- Mesoscale response with multiple physics models
- <http://www.cs.sandia.gov/ALEGRA>
- Haill, et al., *Shock Compression in Condensed Matter Proceedings*. (2011)

# Dense polymer simulation



T.R. Mattsson, et al., *Phys Rev B*,  
81, 054103 (2010).

**Polyethylene** - simplest possible linear carbon backbone structure forms semi-crystalline solids

**PMP (TPX)** - branched hydrocarbon with bulky side chains which is good for producing amorphous foams

## Molecular Dynamics

**OPLS**, Jorgensen et al., JACS, 118, 11225 (1996)

**Borodin-Smith exp-6**, JPCB, 110, 6279 (2006)

**AIREBO**, Stuart et al., JCP, 112, 6472 (2000)

**ReaxFF**, vanDuin et al., JPCA, 112, 1040 (2008)

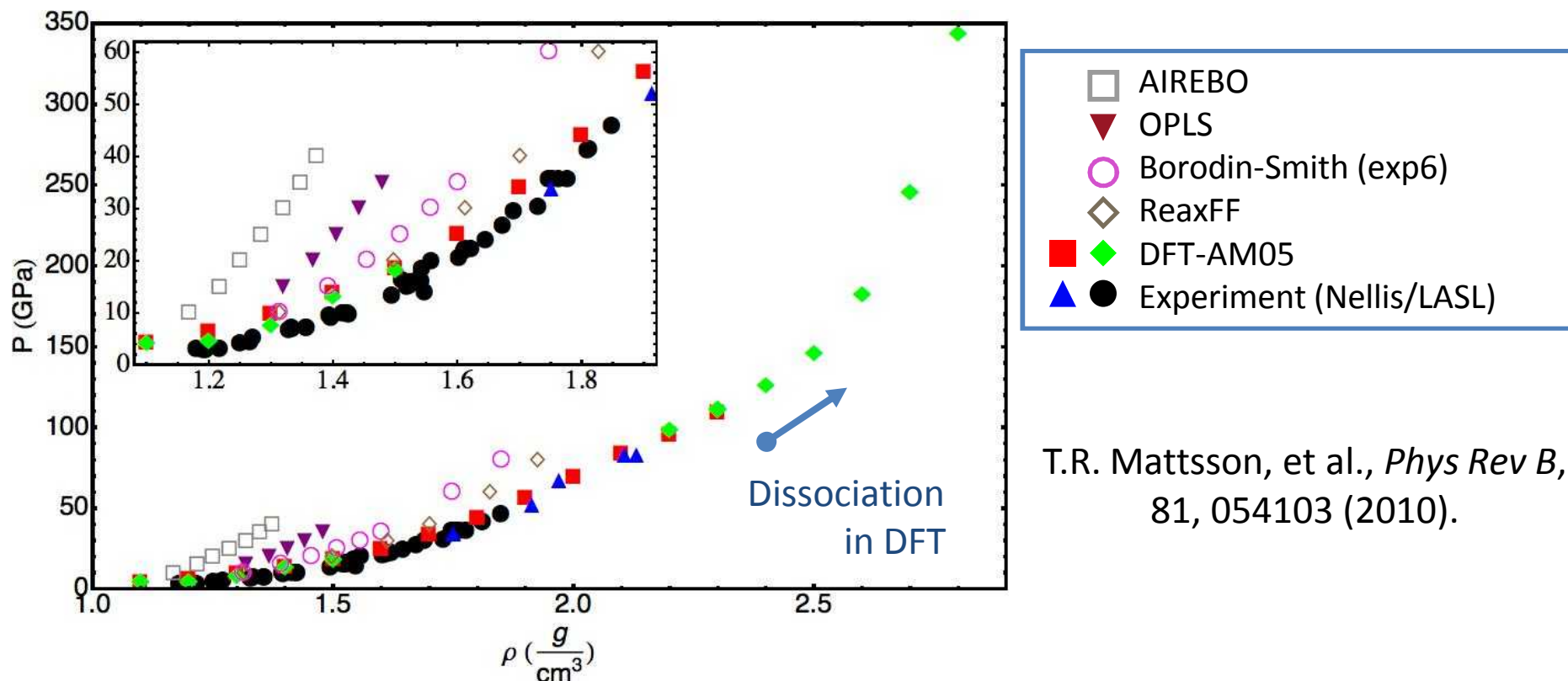
- 22,000+ atom PE sample and 45,000+ atom PMP
- Uniaxial Hugoniotstat method used to compress

## Density Functional Theory

**AM05** Functional

- Several hundred atom samples of PE and PMP

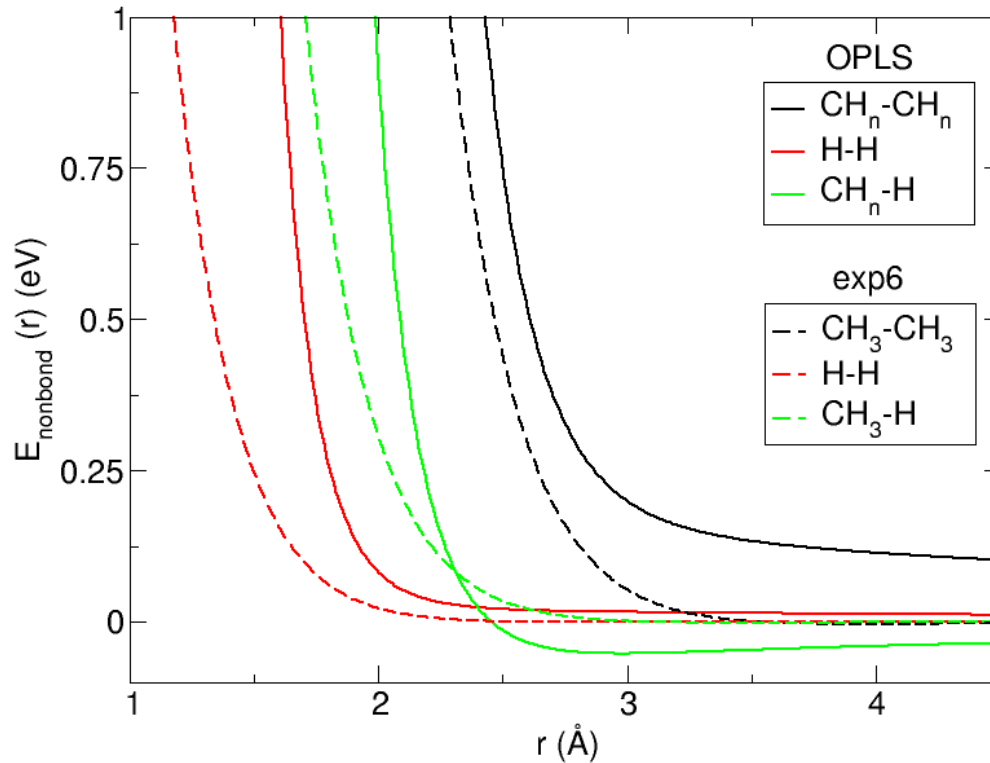
# Polyethylene shock Hugoniot



T.R. Mattsson, et al., *Phys Rev B*,  
81, 054103 (2010).

- AIREBO and OPLS both give significantly too stiff shock response at all pressures
- Borodin-Smith and ReaxFF better choices for weak shocks in polyethylene
- Only the DFT-AM05 simulation of high fidelity also for strong shocks
- Significant deviations already well before the regime where dissociation occurs
- Reactive properties of force-fields are not important for weak shocks

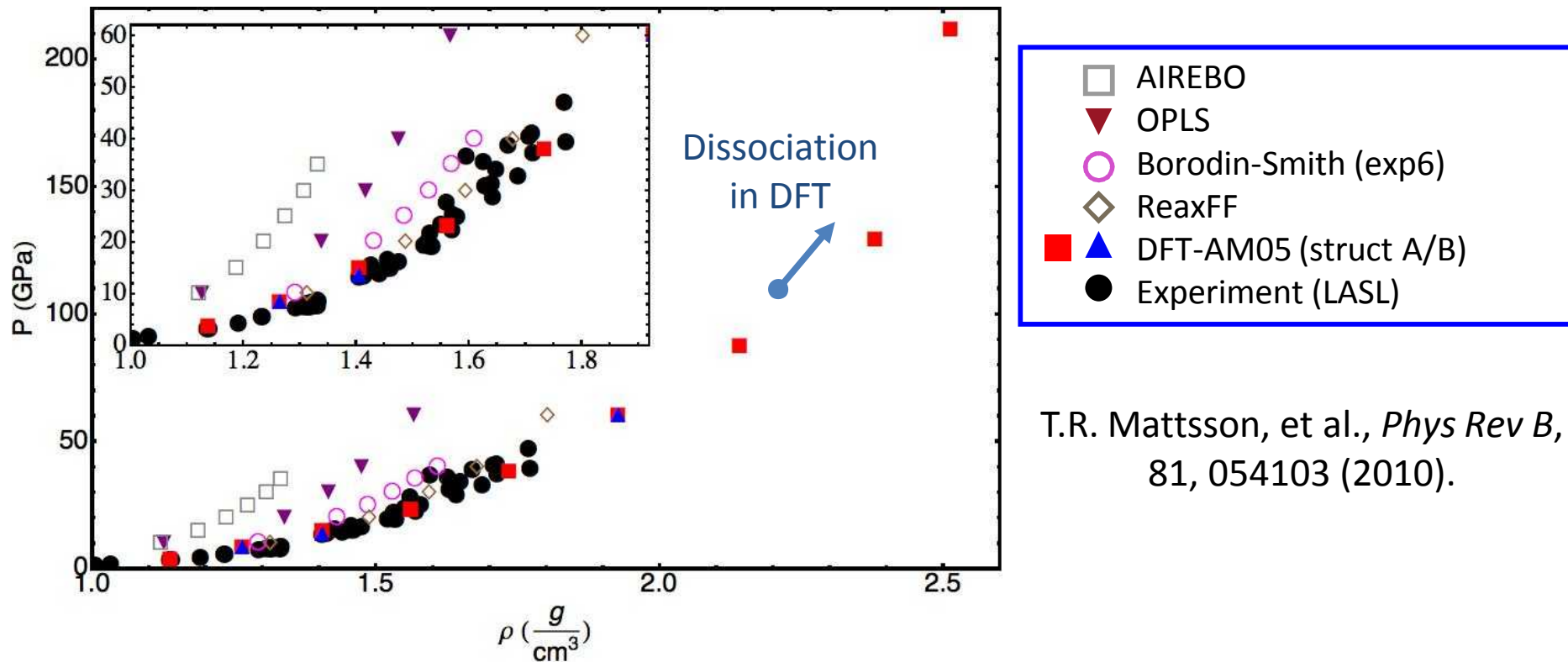
# Core repulsion model is critical



- AIREBO and OPLS share L-J 12-6 functional form for core interaction
- Borodin-Smith exp-6 has a weaker exponential form, more physical shock behavior
- Potentials parameterized near ambient conditions where core interactions are small, unlike in high-density shock conditions
- Behavior under shocks are difficult to predict from equilibrium properties



# Poly(4-methyl 1-pentene) shock Hugoniot

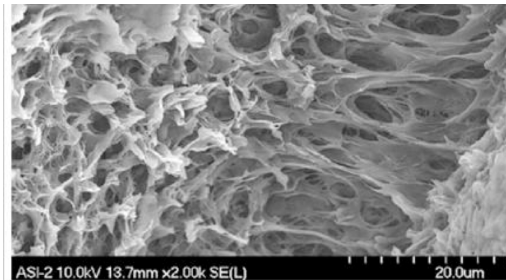


T.R. Mattsson, et al., *Phys Rev B*,  
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# Foam introduction



## Densities

- Experimental foam –  $0.309 \pm 0.025$  g/cc
- Simulated foam – 0.300 g/cc

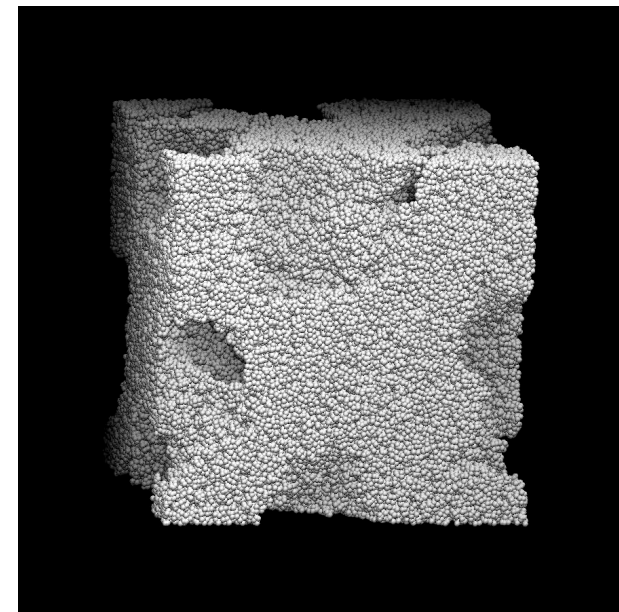
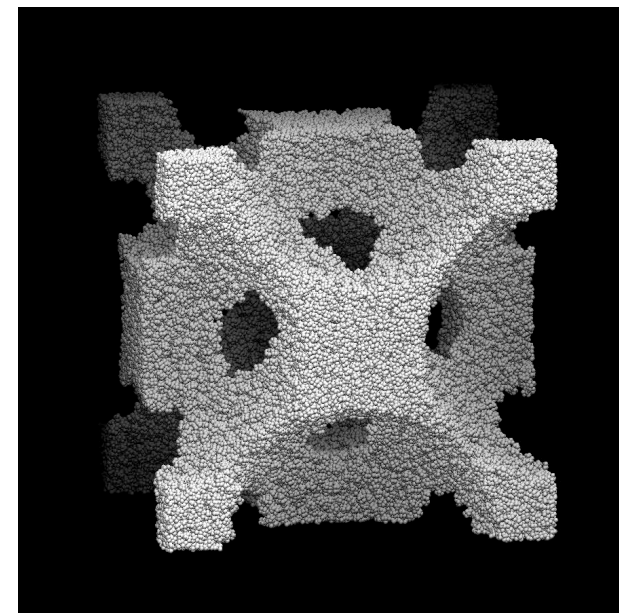
## Experimental fabrication:

- 0.833 g/cc PMP dissolved in cyclohexane & evaporated

## Simulation fabrication:

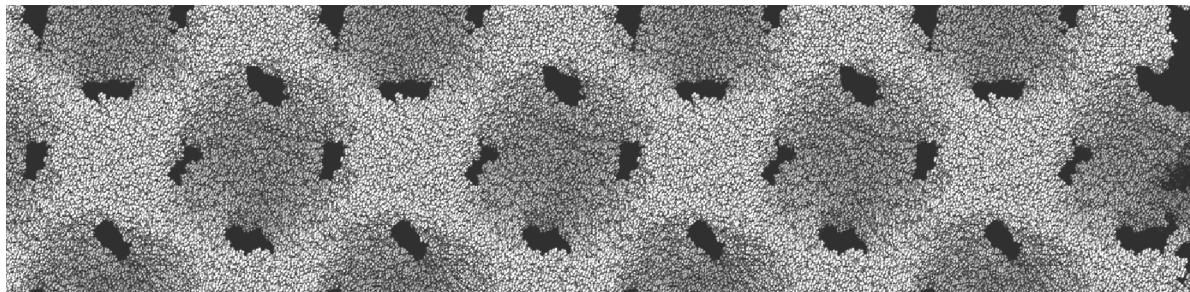
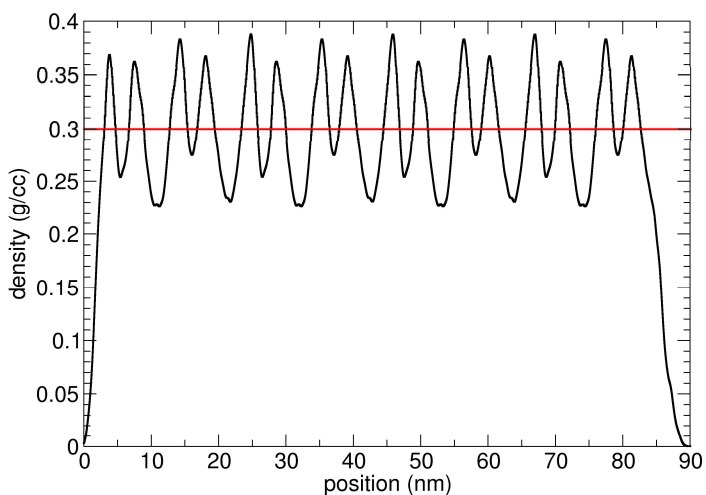
- 0.833 g/cc PMP perforated with growing spherical indenters placed on an fcc lattice, creating a unit cell of foam with unit cell of  $20 \times 20 \times 20$  nm.

ReaxFF potential was used exclusively.

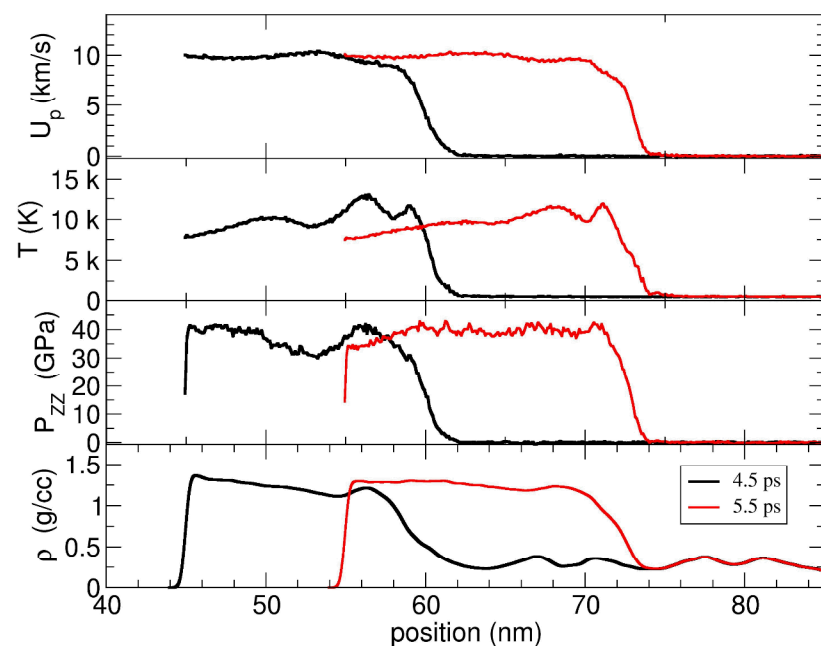


S. Root, et al., (to be submitted 2011).

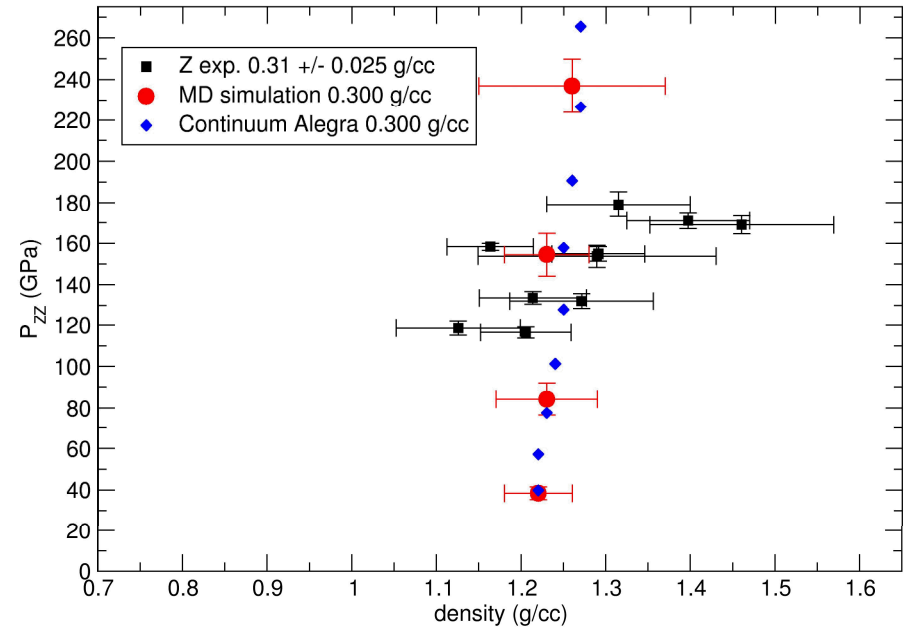
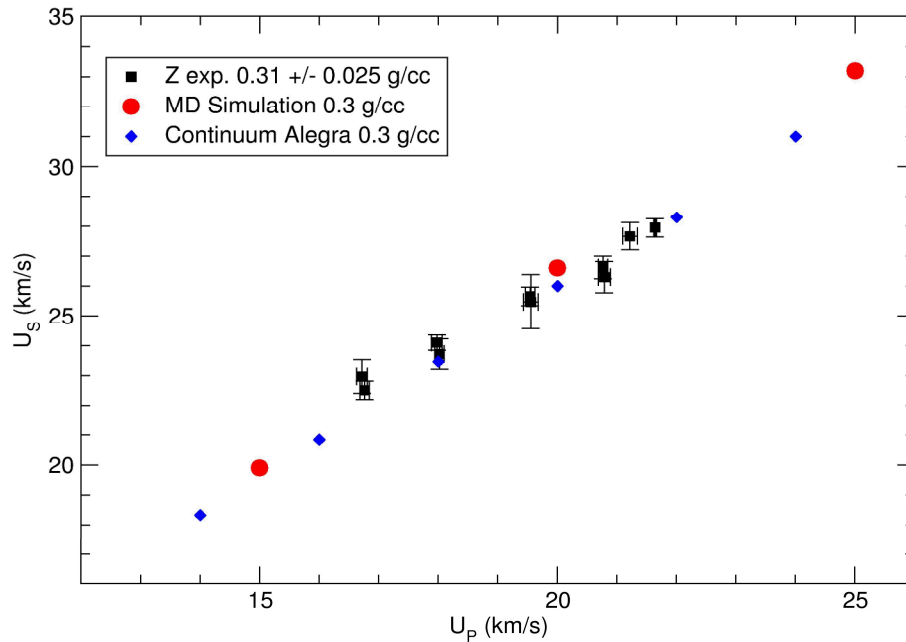
# Foam shock methodology



- Shock Hugoniotstat method could not be used, so long simples were constructed from foam unit cells.
- Shock is driven by warm momentum mirror using standard NEMD techniques.
  - Piston velocities ranged from 10 to 30 km/s.
  - Very short simulation timesteps (0.01 to 0.05 fs) were required
- Profiles were calculated from per-atom variables

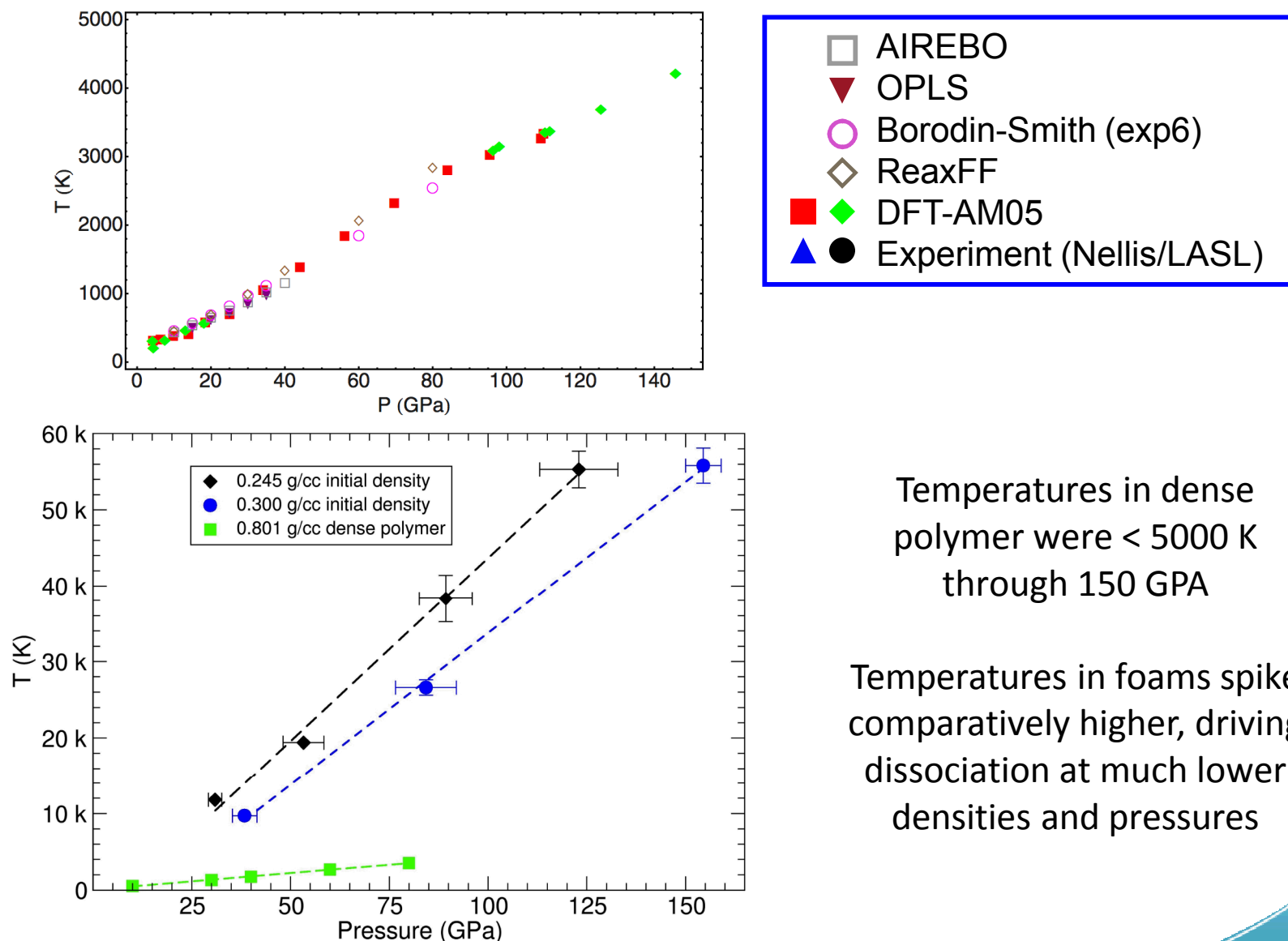


# Foam results – Hugoniot data



- Good quantitative agreement between Z experimental data, and both MD and continuum simulations for  $\sim 0.310$  g/cc foam shock loading
- Spread in experimental data comes from variation in sample density and inhomogeneity.

# Temperature dense polymer vs foam



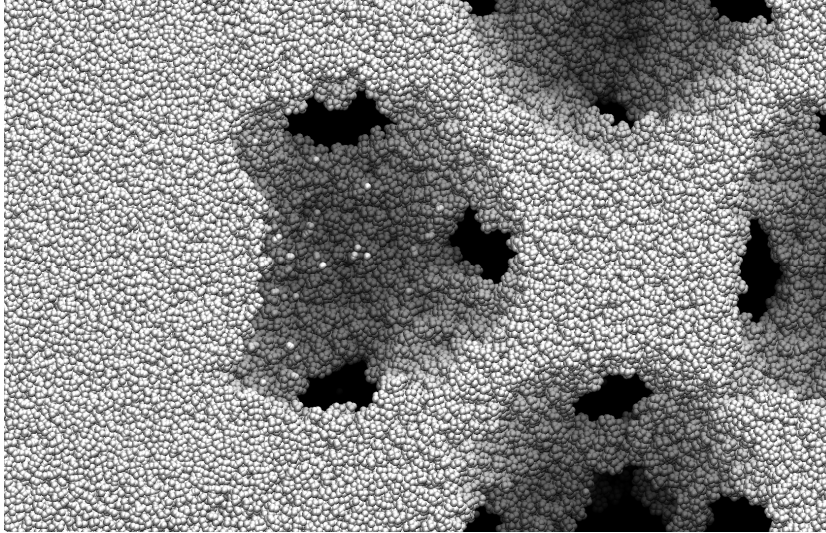
Temperatures in dense polymer were < 5000 K through 150 GPa

Temperatures in foams spike comparatively higher, driving dissociation at much lower densities and pressures

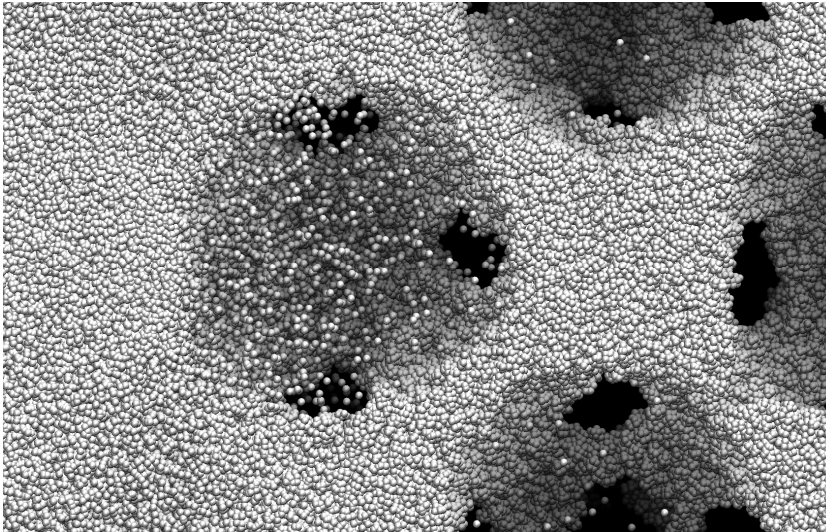


# Foam results – dissociation and vapor

10 km/s piston – 0.300 g/cc PMP foam



25 km/s piston – 0.300 g/cc PMP foam



- Dissociation and vaporization at much lower flyer velocities.
- Accelerated free surface expansion gives rise to void hot spots
- Vaporization and surface blow out broadens the shock profile
- In percolating voids vapor material can behave qualitatively differently than in closed voids.

# Summary and conclusions

- For dense polymers, **shock response depends more on accurate potentials than on structure**. Thus, DFT is excellent in dense polymers
- **MD potentials exist, such as ReaxFF, which capture the quantitative shock response** of both dense polymers and foams up to 50 GPa in dense polymers and higher in foams
- Good agreement found with experiment and continuum simulation for shock compressed polymer foam. Much is left to explore in determining the **role of inhomogeneity, detailed structure and length scales on shock response**

## Future work and possible collaboration opportunities

- Nanoparticle-polymer composites
- Void scale, structure effects and void percolation

## Acknowledgments

Gary Grest, Aidan Thompson, Thomas Haill, Seth Root, Kyle Cochrane, Mike Desjarlais, Thomas Mattsson