

Phase and texture evolution in solution deposited PZT thin films

Krishna Nittala¹, Geoff L. Brennecka², Jon F. Ihlefeld², Bruce A. Tuttle²,
Douglas S. Robinson³, and Jacob L. Jones¹

¹*Department of Materials Science and Engineering, University of Florida, Gainesville, FL, USA*

²*Sandia National Laboratories, Albuquerque, NM, USA*

³*Advanced Photon Source, Argonne National Laboratory, Argonne, IL, USA*



Supported by the National Institute for NanoEngineering (NINE) and the Laboratory Directed Research and Development (LDRD) program at Sandia National Laboratories.

Sandia is a multiprogram laboratory operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Company, for the United States Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000.

Use of the Advanced Photon Source, an Office of Science User Facility operated for the U.S. Department of Energy (DOE) Office of Science by Argonne National Laboratory, was supported by the U.S. DOE under Contract No. DE-AC02-06CH11357.

Outline

1. Introduction

- Motivation
- Methodology

2. Phase evolution (UF)

- Effect of Nb doping
- Pb content in solution

3. Phase and Texture evolution (APS)

- Effect of heating rate
- Cation segregation in thin film

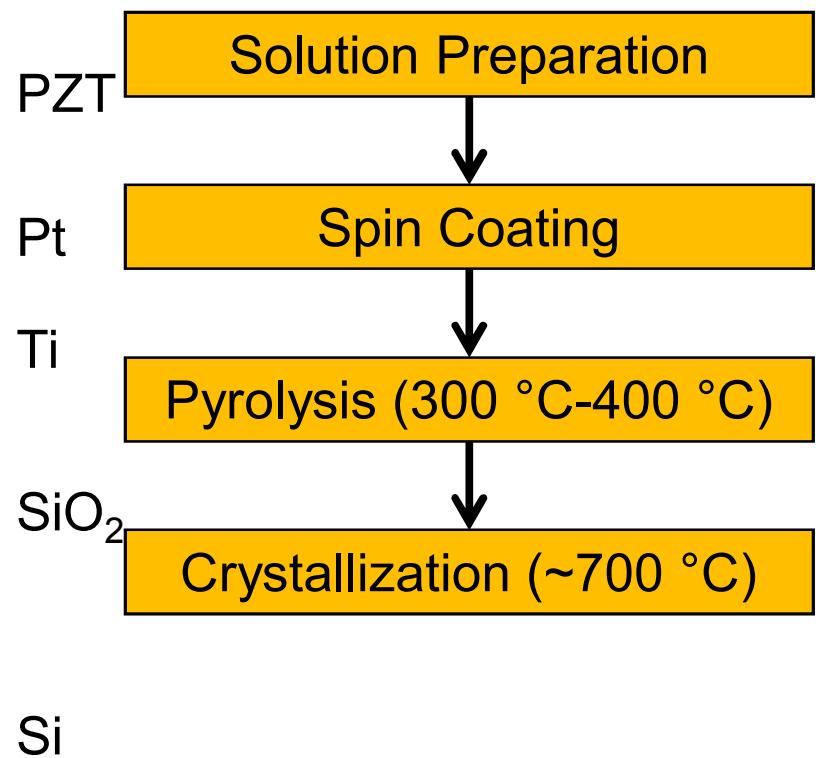
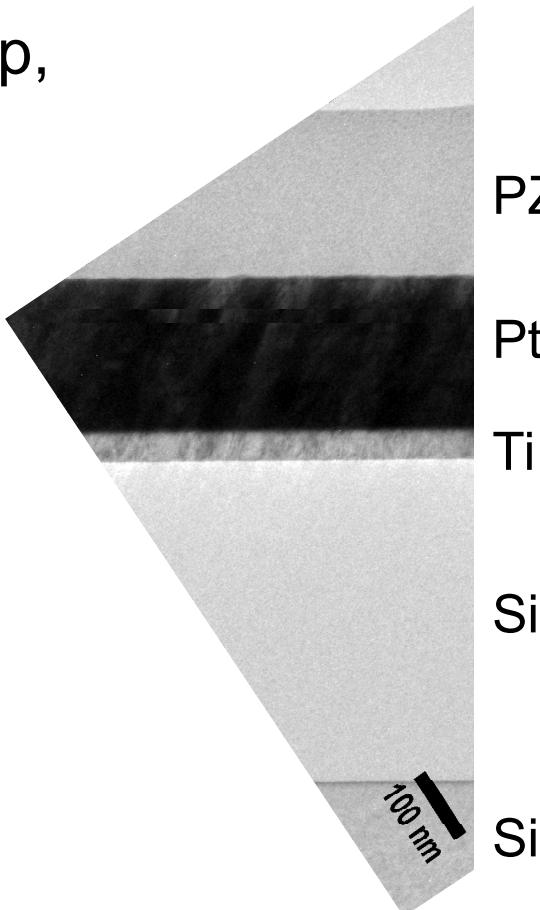
4. Ongoing and Future work

5. Conclusions

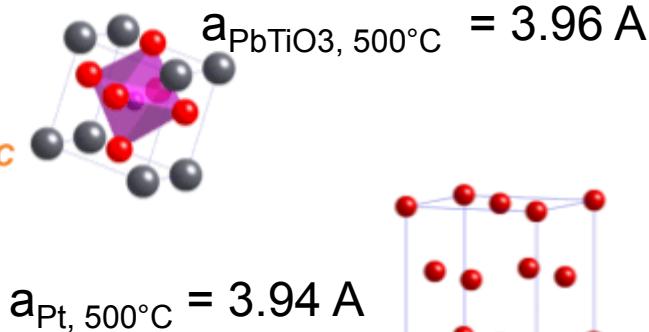
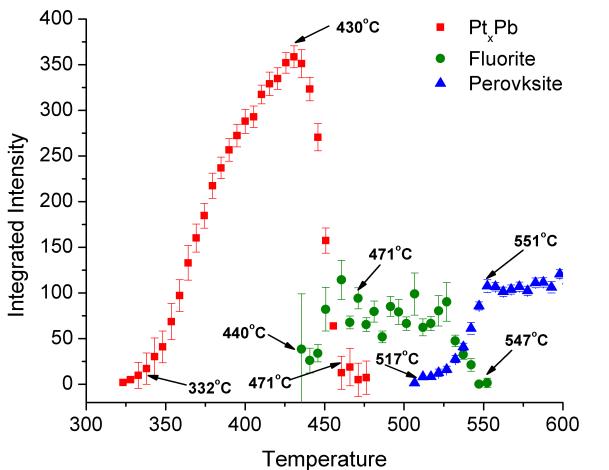
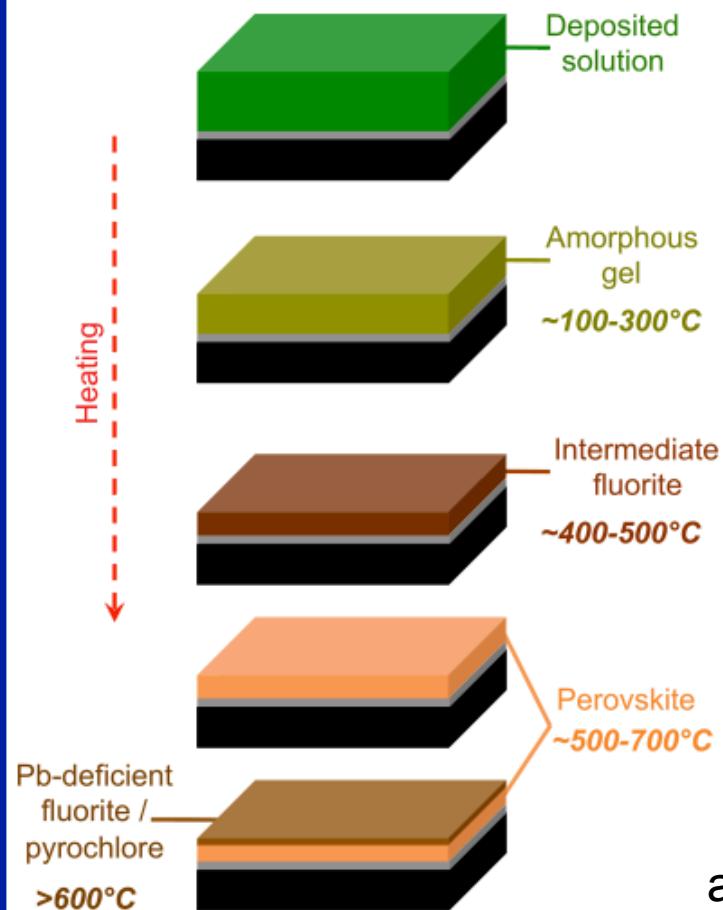
Solution deposition of PZT films

- Solution deposition is cheap, simple, and versatile (deposit on foil, platinized Si, etc.)

- PZT films are commonly deposited on platinized silicon substrates



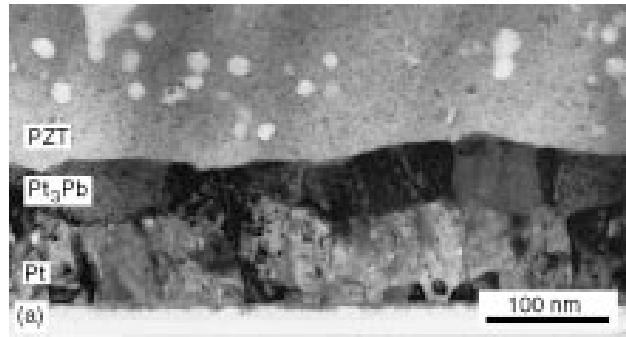
Phase evolution during processing



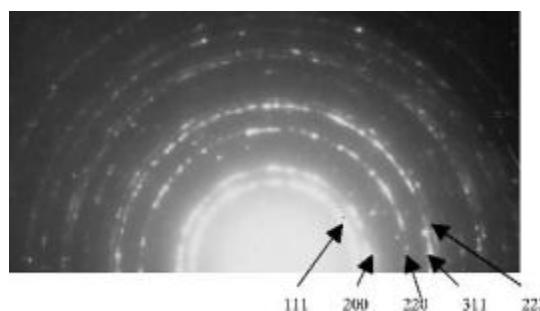
- Ensure complete conversion of fluorite type phase to perovskite phase
- Avoid formation of Pb deficient fluorite type phase
- Minimize reaction between film and the electrode

Phase evolution: influence on texture

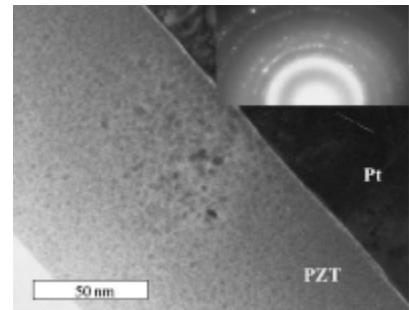
- Heating rate: (111) texture is caused due to nucleation of perovskite on Pt_xPb .³
- Fluorite crystallinity: degree of crystallinity controls the final film texture.⁴
- Pt_3Ti ⁶ and TiO_2 ⁷ seeds at the film-Pt interface.



Pt_xPb phase forms at the interface of the Pt electrode and the thin film.⁵



Pyrolysis: 350°C, 10s
(111) texture⁴



Pyrolysis: 450°C, 2 min
(100) texture

³S. Y. Chen and I. W. Chen, J. Am. Ceram. Soc. 81 (1998) 97.

⁶T. Tani, PhD Thesis (UIUC, Urbana - Champaign, 1993).

⁴G. J. Norga et al, J. Mater. Res. 18 (2003) 1232.

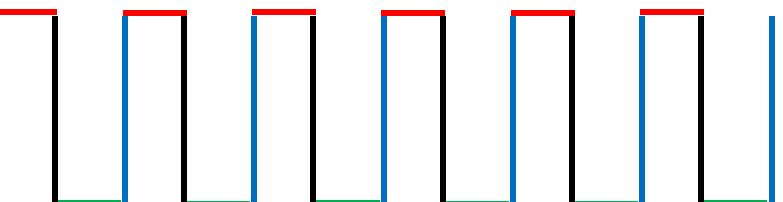
⁷P. Muralt, J. Appl. Phys. 100 (2006) 051605.

⁵Z. Huang et al, J. Appl. Phys. 85 (1999) 7355.

Typical approaches for characterization of texture and phase

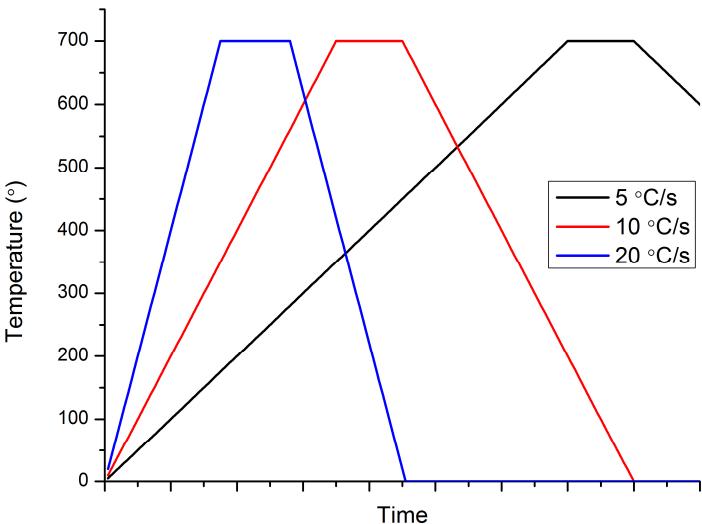
Typical experimental methodologies

- Isothermal heating
- Quenching
- XRD characterization
- Rapid heating



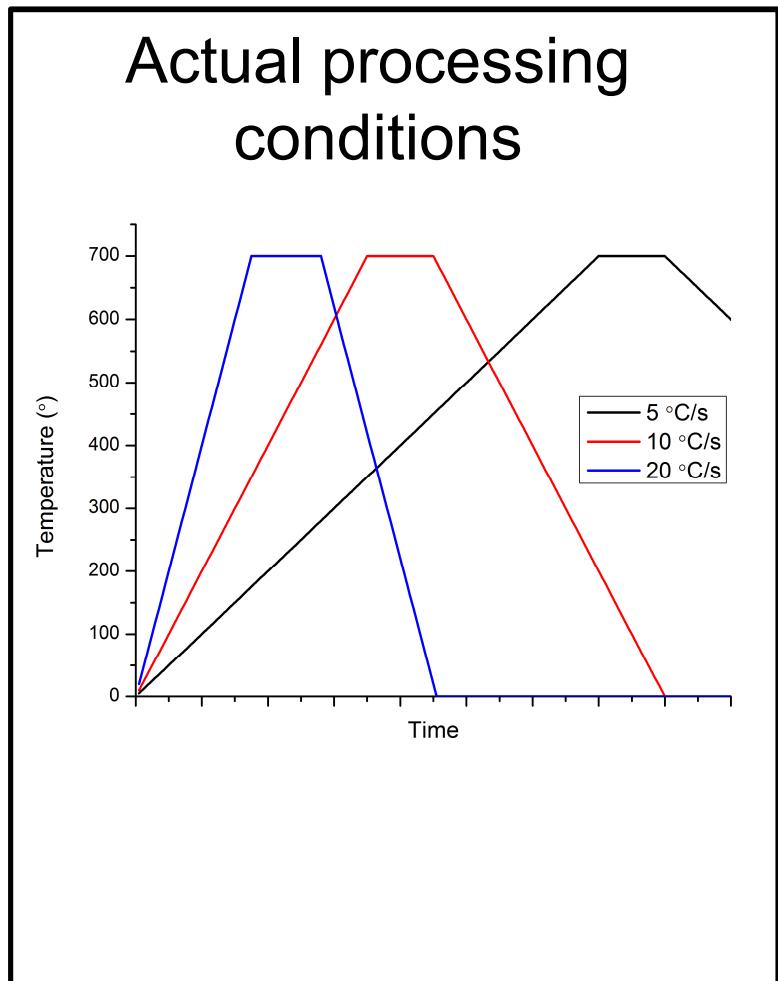
Repeated until the thin film completely crystallizes

Actual processing conditions



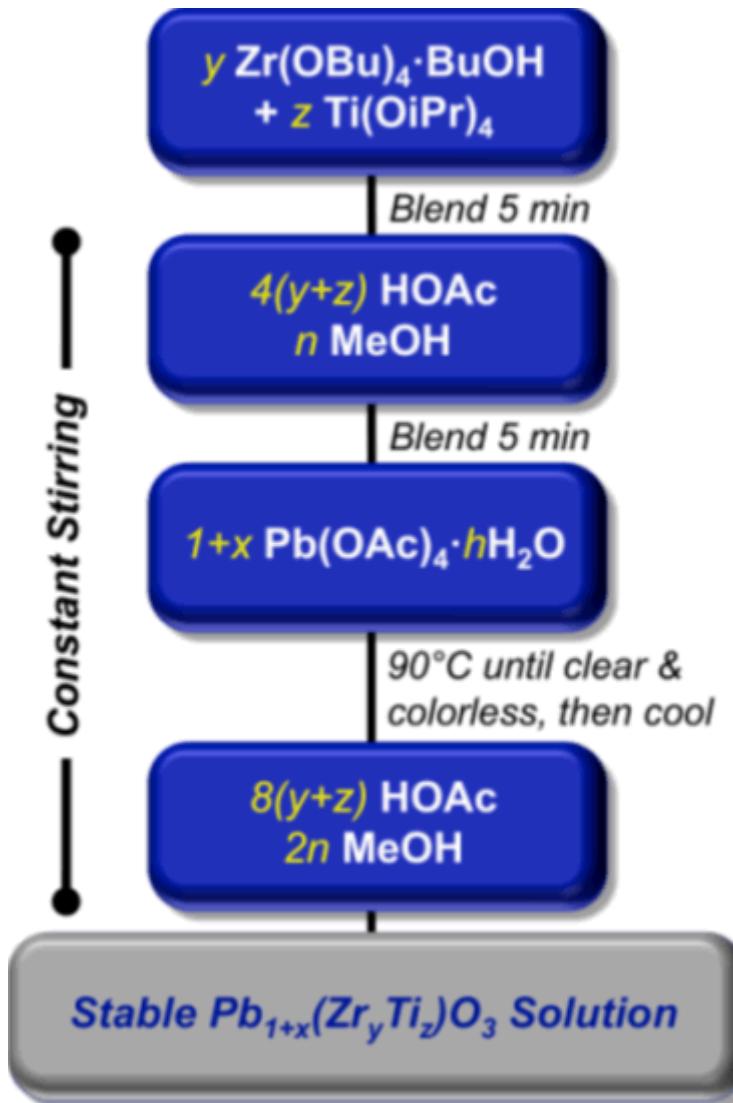
Objective of present investigation

- *in situ* characterization of phase and texture evolution during crystallization to understand the factors affecting final film texture in PZT thin films
- Two different types of *in situ* experiments were performed:
 - Laboratory X-ray (UF)
 - Synchrotron X-ray (APS)



Solution preparation and Film deposition

- Inverted Mixing Order (IMO) process was used for preparation of solutions
- Films deposited on platinized silicon substrates (Pt/Ti/SiO₂/Si 170 nm/40nm/300nm/Si)
- Films were spin coated for 30s at 3000 rpm
- Films were pyrolyzed at 300 °C after each deposition step



R. A. Assink and R. W. Schwartz, Chem. Mater. 5 (1993) 511.

G. L. Brennecka *et al.* J. Am. Ceram. Soc. 93 (2010) 3935.

Outline

1. Introduction

- Motivation
- Methodology

2. Phase evolution (UF)

- Effect of Nb doping
- Pb content in solution

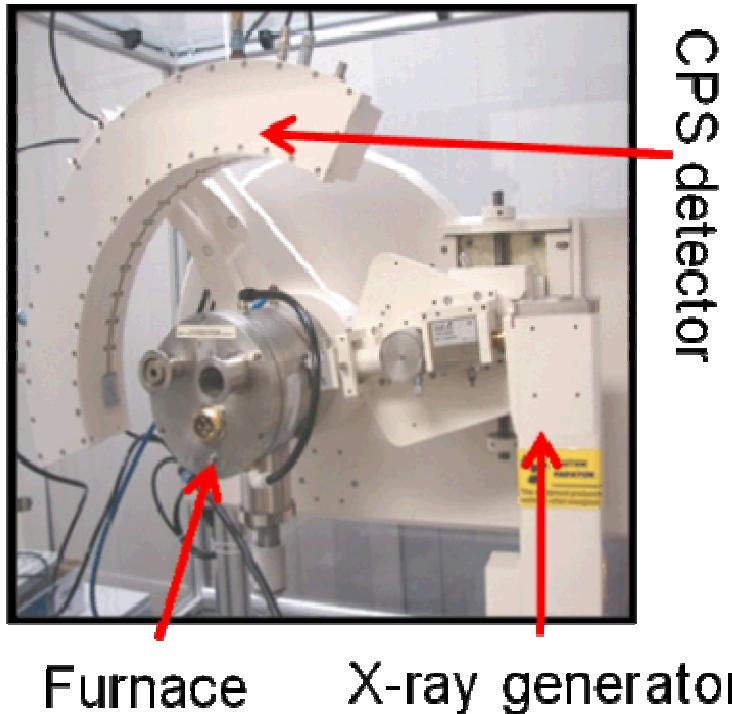
3. Phase and Texture evolution (APS)

- Effect of heating rate
- Cation segregation in thin film

4. Ongoing and Future work

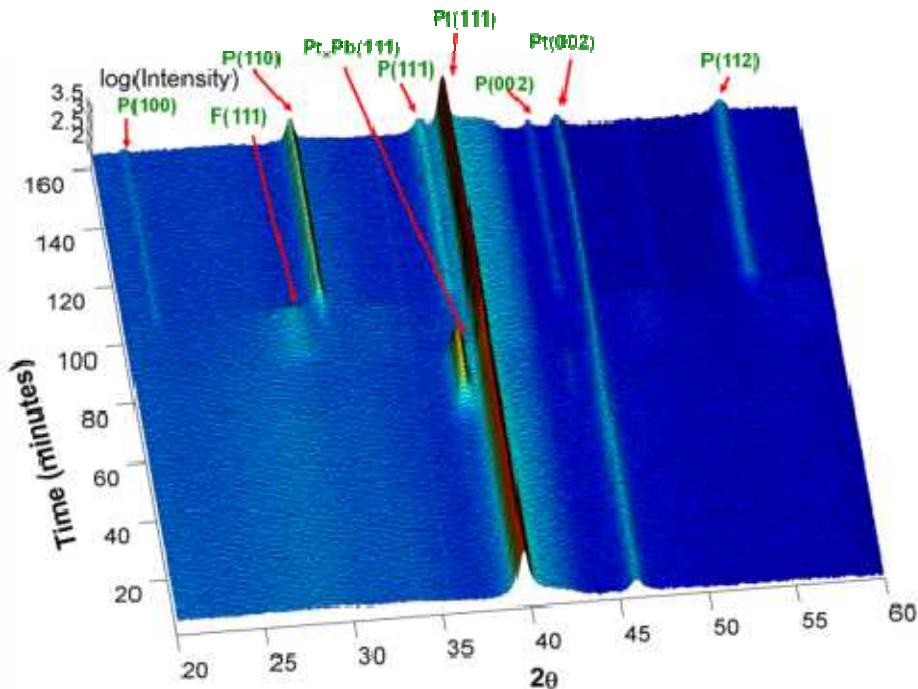
5. Conclusions

Laboratory XRD



Furnace X-ray generator

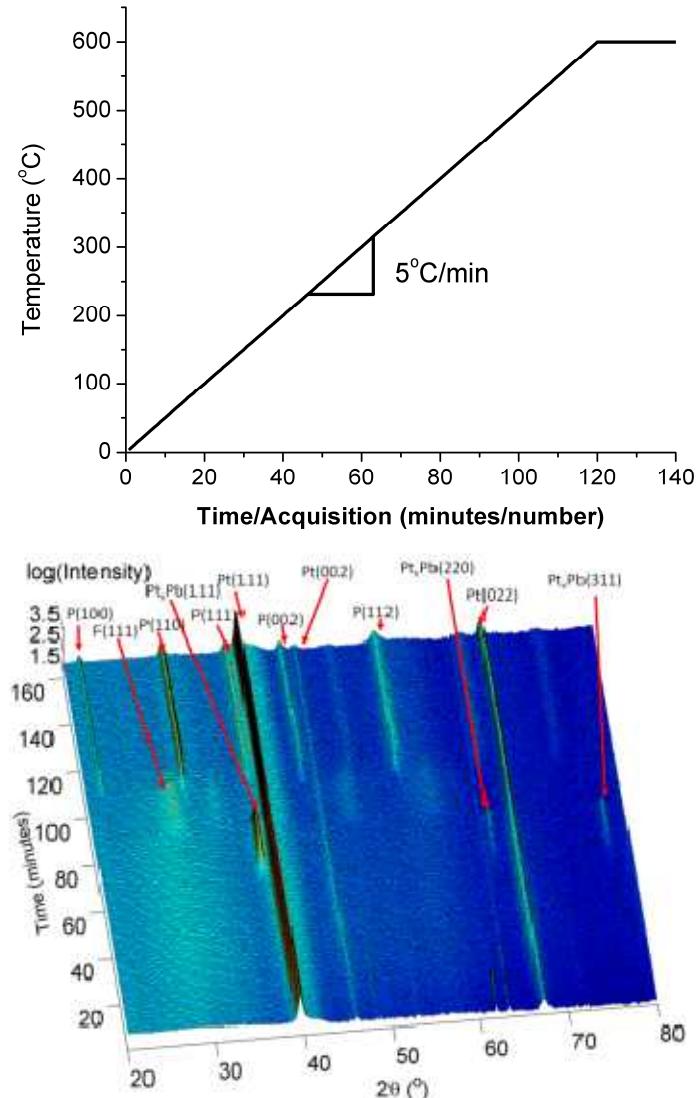
- Inel diffractometer with furnace attachment at UF
- 1-D detector allows for rapid acquisition of diffraction data



- Diffraction data measured is represented as a contour plot
- Plot shows the evolution of phases during crystallization

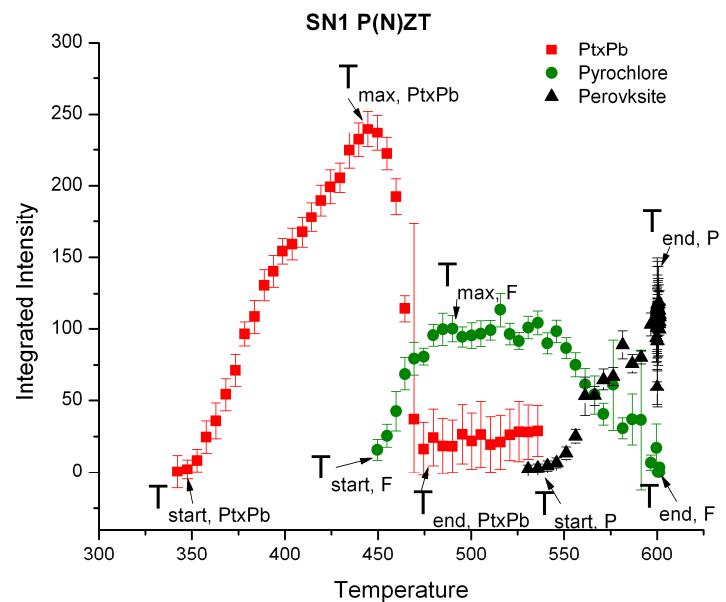
Effect of Nb content

- PZT is routinely doped with Nb and La
- Effect of dopants on phase evolution in solution deposited films is not well understood
- Films were heated 5 °C/min while continuous diffraction patterns were taken



Nb doping: influence on phase evolution

- Temperature at which the Pt_xPb peaks is observed is increased
- Fluorite phase forms ($T_{\text{start, F}}$) at higher temperature in Nb-doped PNZT compared to undoped PZT thin films
- PNZT: Perovskite formation is complete at 624 °C



	$T_{\text{start,PtxPb}}$	$T_{\text{max,PtxPb}}$	$T_{\text{end,PtxPb}}$	$T_{\text{start, F}}$	$T_{\text{max,F}}$	$T_{\text{end,F}}$	$T_{\text{start,P}}$	$T_{\text{max,P}}$
PNZT 4/52/48	350.8	446.5	474.7	446.3	481.3	596.2	539.5	624.0
PZT 52/48	339.4	442.0	475.1	433.5	472.1	585.5	543.3	590.4
Difference	11	4.5	~0	13	9	11	-4	34

Summary: Nb doping

- Nb doping observed to effect stability of the Pt_xPb metastable phase
- Effect on Fluorite and perovskite formation is consistent with report by Klissurska et al.

Outline

1. Introduction

- Motivation
- Methodology

2. Phase evolution (UF)

- Effect of Nb doping
- Pb content in solution

3. Phase and Texture evolution (APS)

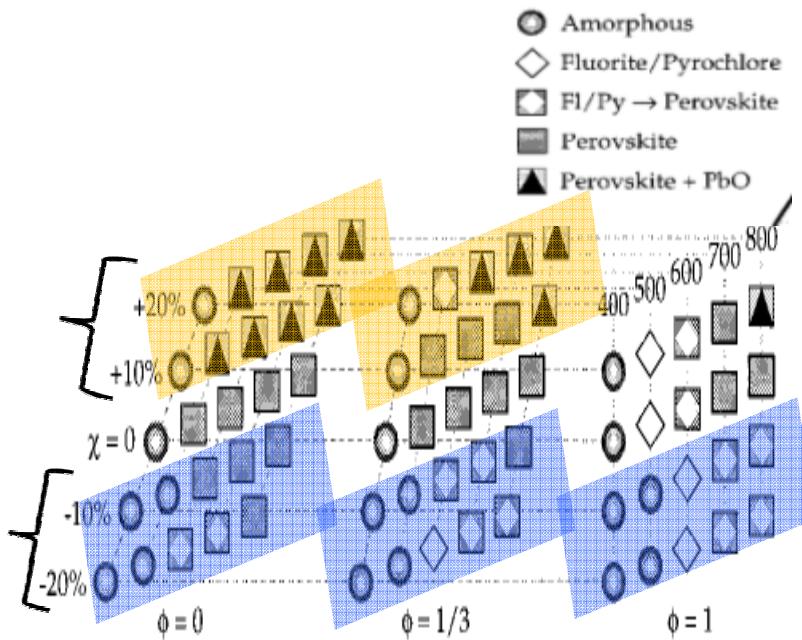
- Effect of heating rate
- Cation segregation in thin film

4. Ongoing and Future work

5. Conclusions

Effect of Pb content on phase evolution

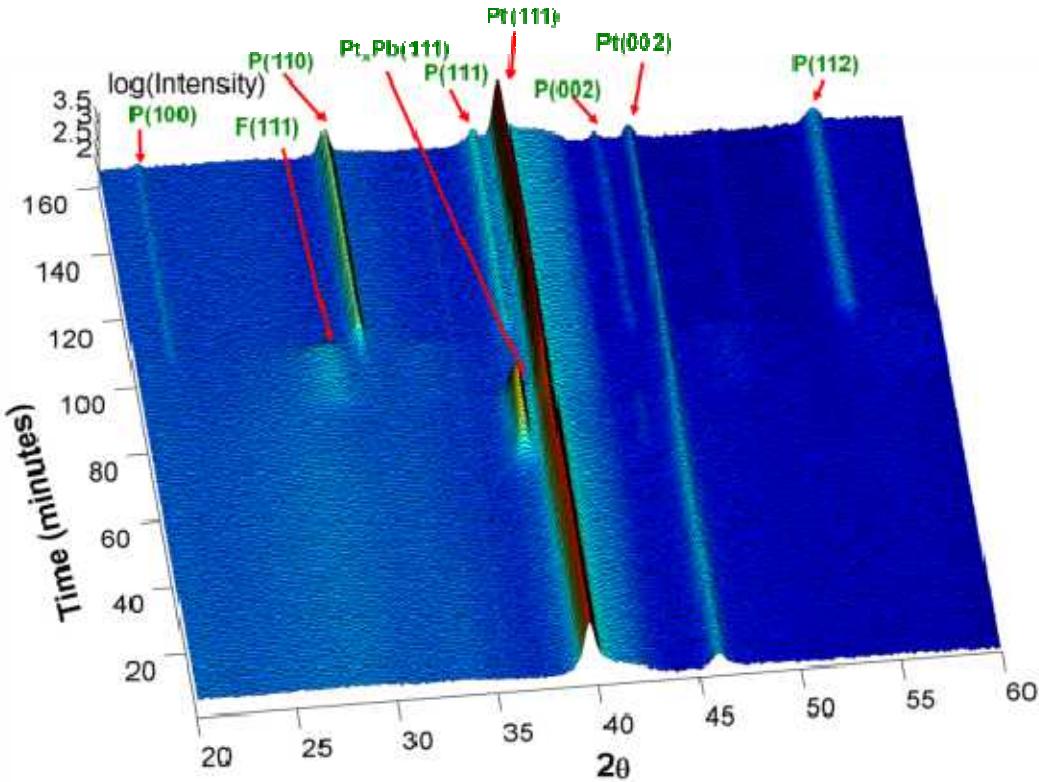
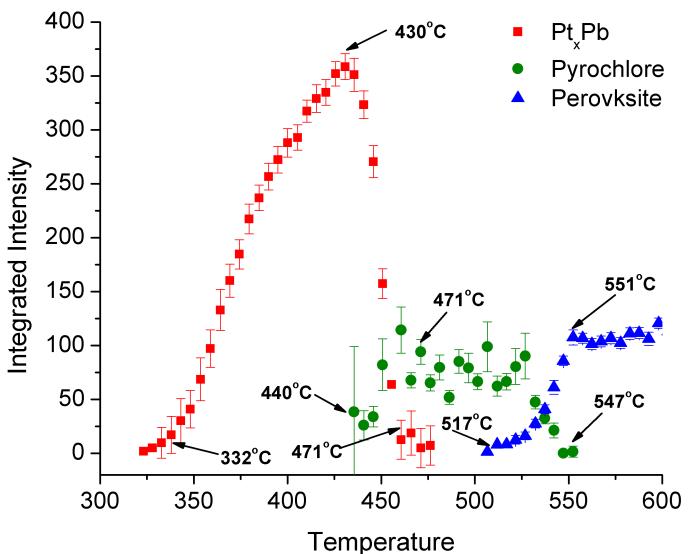
- Pb content was observed to influence phase evolution for solution derived powders
- Presence of substrate introduces additional factors of influence
- Phase evolution of PLZT (6/52/48) 20% Pb films was investigated



Experimental conditions
Heating rate: 5 C/min
Acquisition time: 60 s

Phase evolution: Pb excess

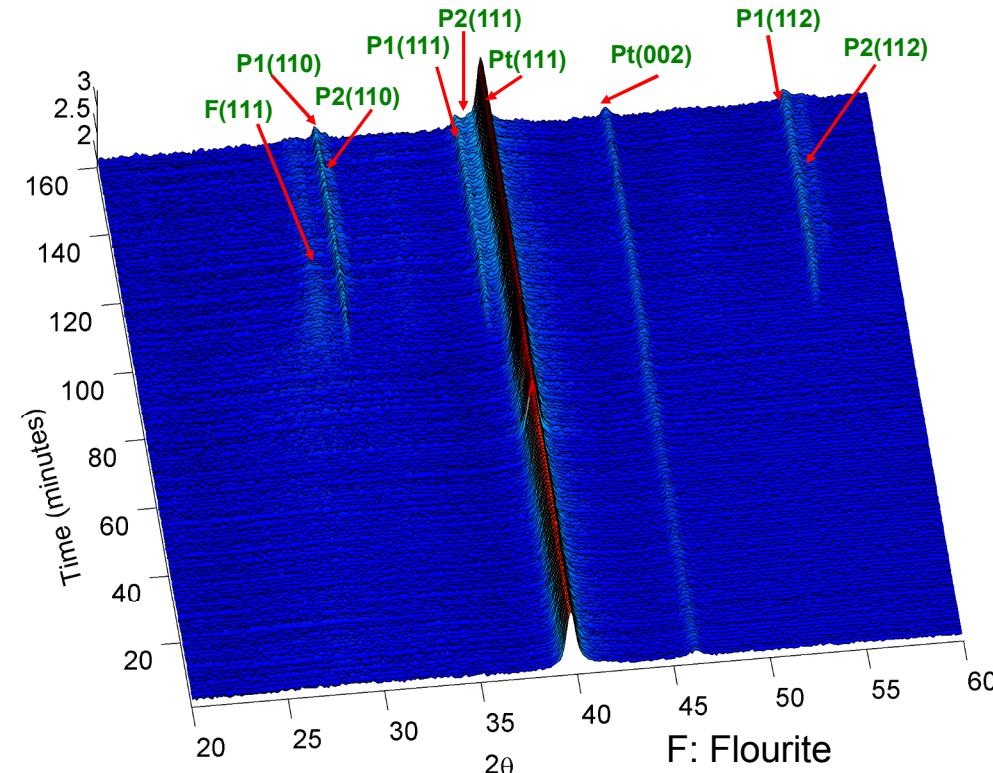
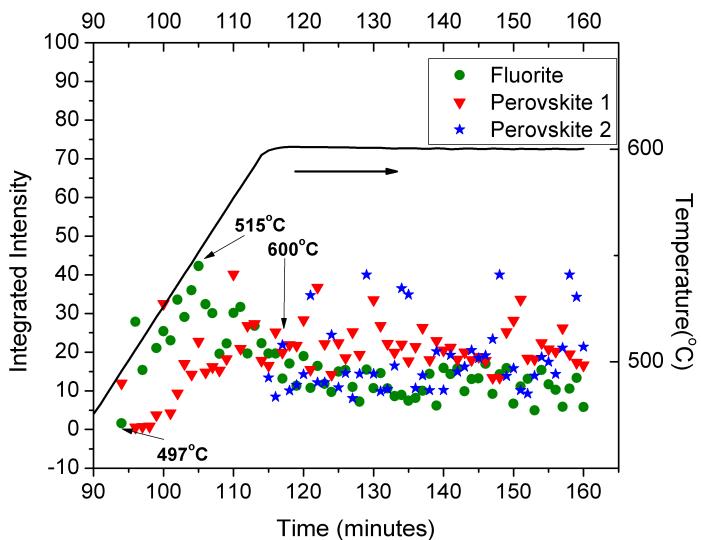
- Pt_xPb is the first phase to form
- Fluorite phase (F) is observed as Pt_xPb starts to disappear



- Fluorite finally disappears and Perovskite (P) appears
- Result agrees with previous studies

Phase evolution: Pb deficient

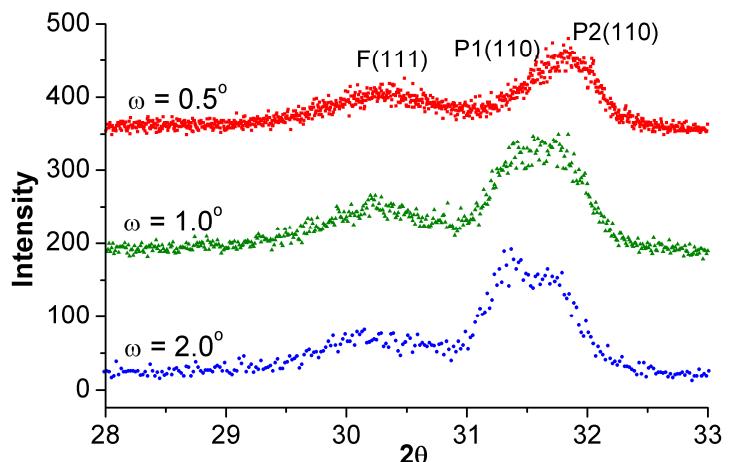
- Fluorite(F) and Perovskite(P1) phase observed to form together at 515°C
 - A secondary perovskite phase(P2) is observed to form on holding at 600°C



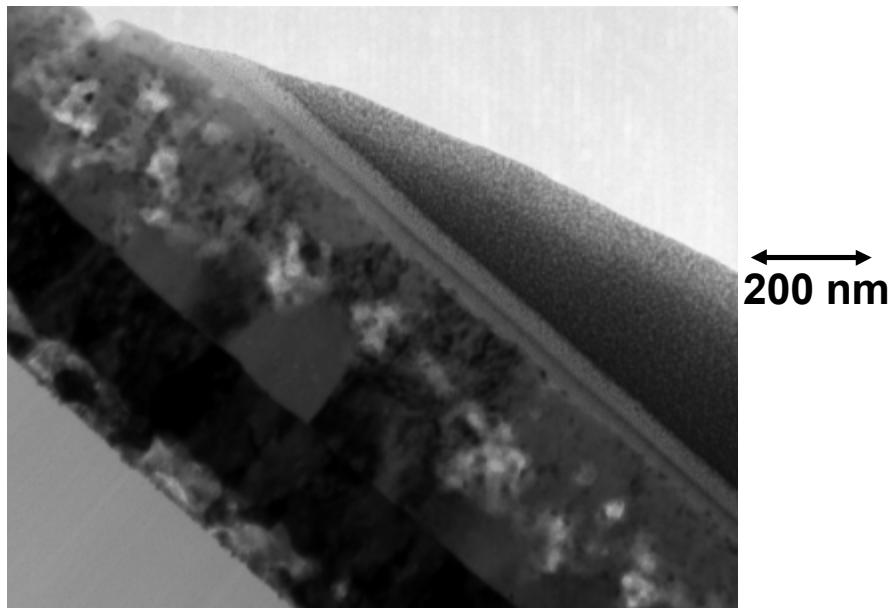
F: Fluorite
 P1: Perovskite 1
 P2: Perovskite 2

Composition	Lattice parameter (Å)
Pb-excess; Perovskite	4.08
Perovskite 1	4.08
Perovskite 2	4.02

Formation of secondary perovskite phase



GIXRD of Pb-deficient film



- Secondary Perovskite could possibly form from Fluorite phase

Summary: Effect of Pb content

- No Pt_xPb was observed in Pb-deficient thin films
- Reaction between the thin film and electrode is reduced
- Temperatures of formation of the fluorite phase is increased for Pb-deficient films
- Perovskite formation is observed at the same temperature for both Pb-excess and Pb-deficient films

Outline

1. Introduction

- Motivation
- Methodology

2. Phase evolution (UF)

- Effect of Nb doping
- Pb content in solution

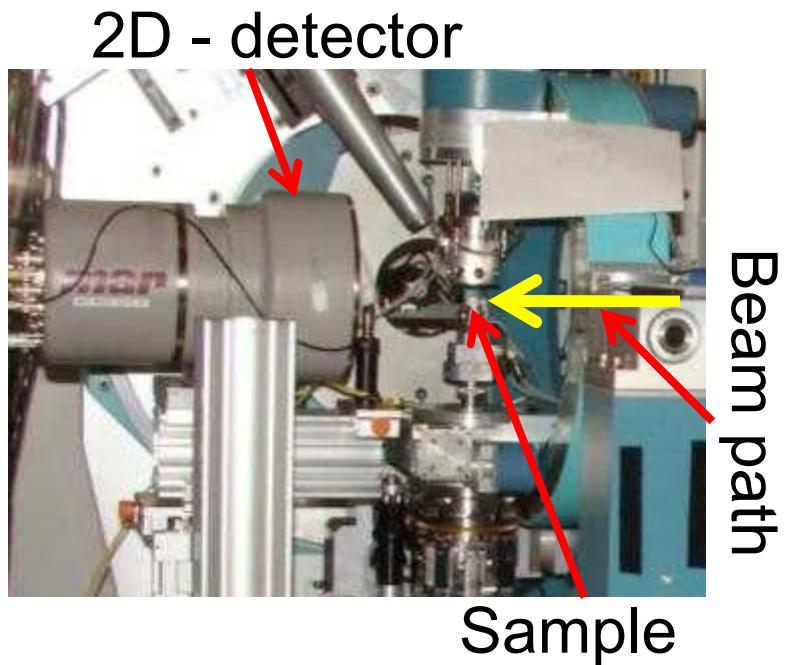
3. Phase and Texture evolution (APS)

- Effect of heating rate
- Cation segregation in thin film

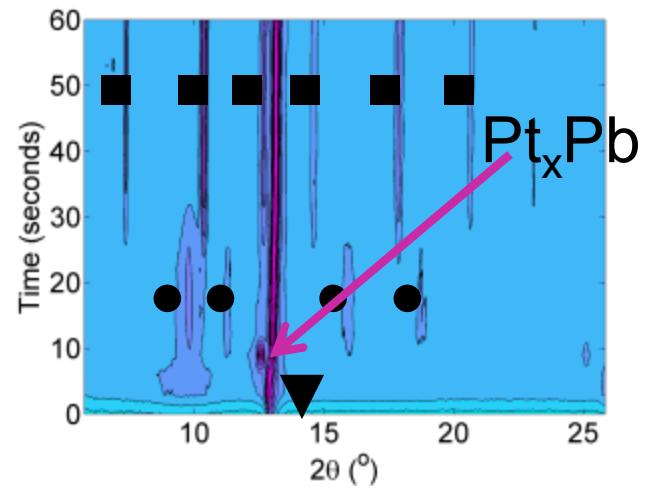
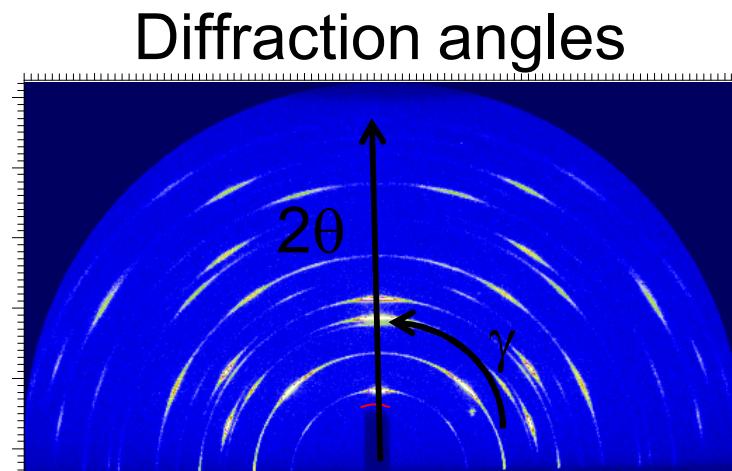
4. Ongoing and Future work

5. Conclusions

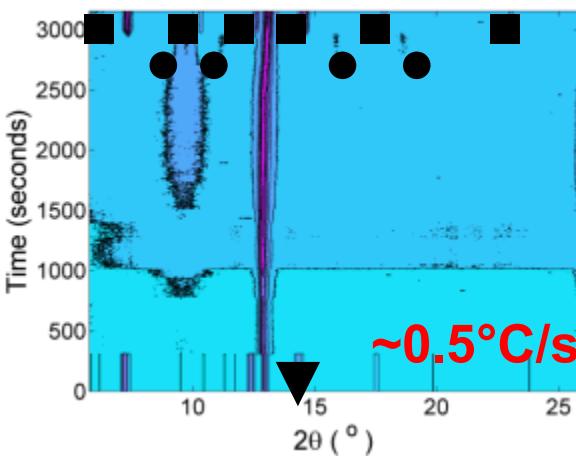
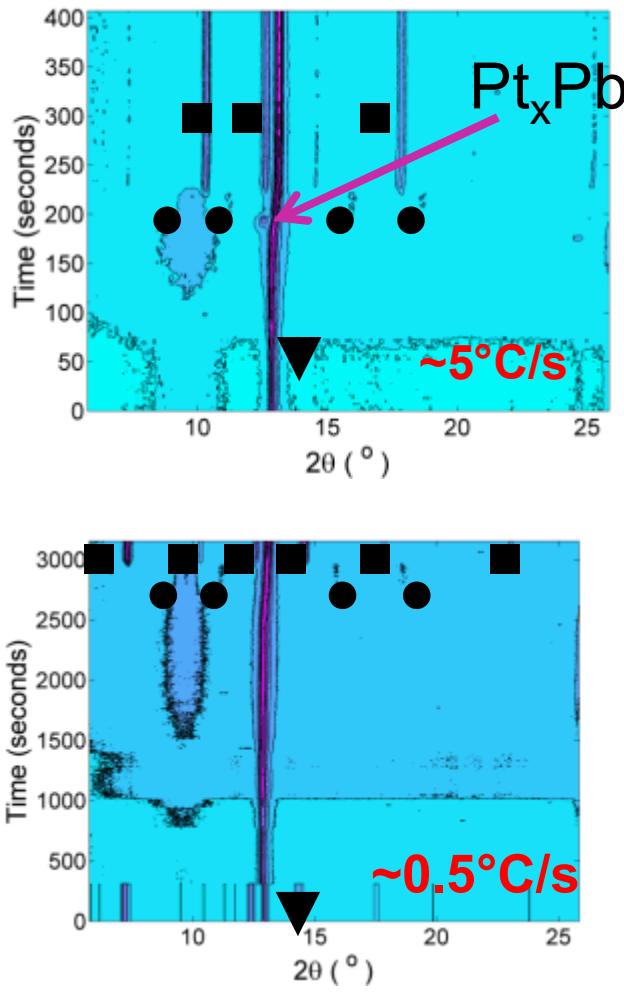
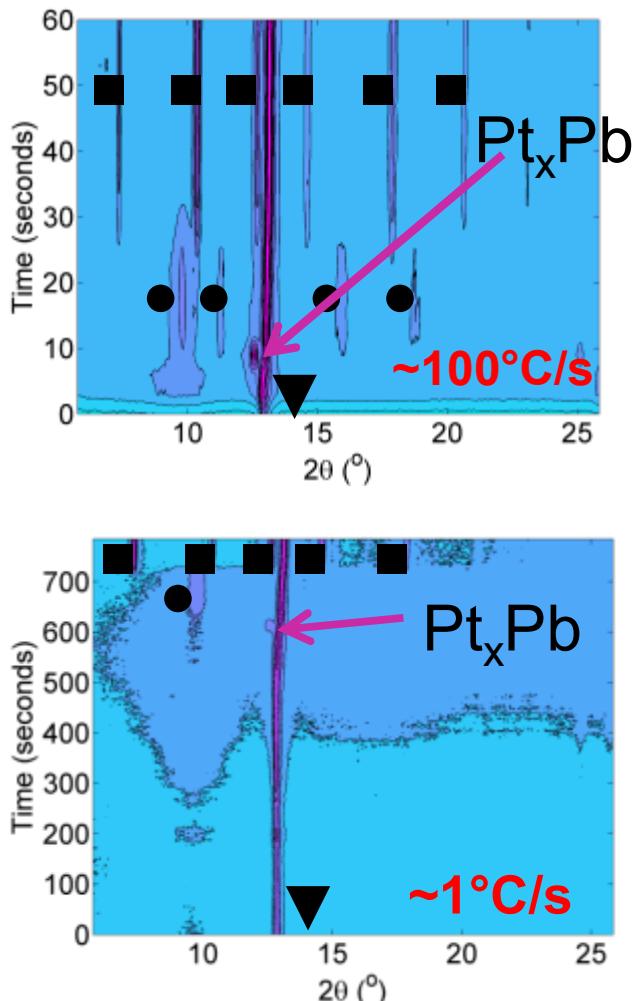
Setup at APS



- Synchrotron X-ray source (APS)
- Heating rates: ~ 100 $^{\circ}\text{C/s}$ to 1 $^{\circ}\text{C/s}$
- 2-D detector captures texture and phase information
- 1s acquisition time, continuous acquisition



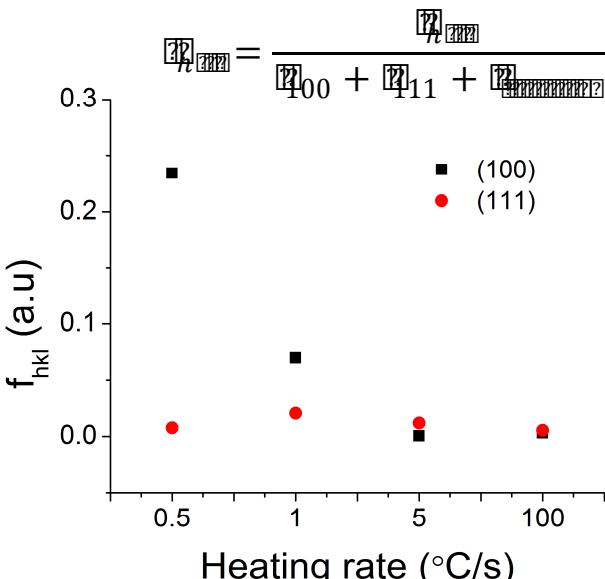
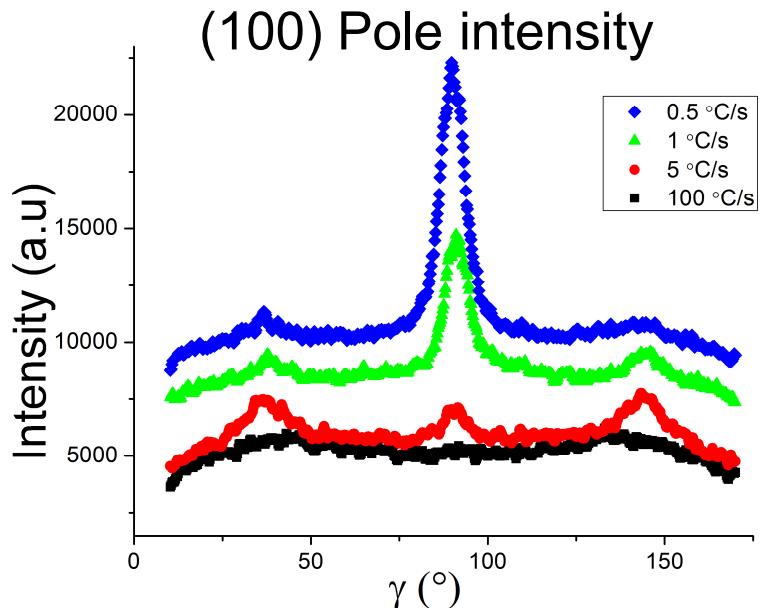
Heating rate influences phase evolution



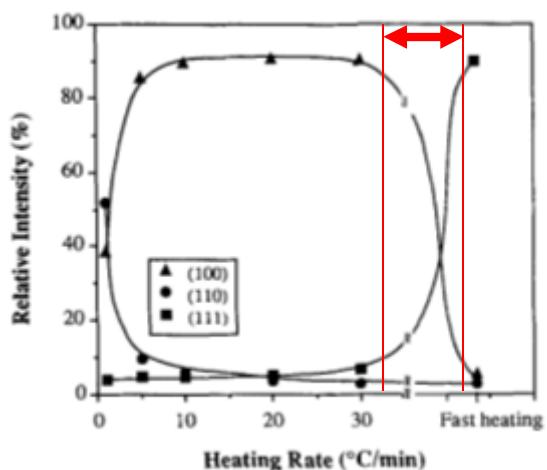
▼ Pt ■ Perovskite

- Sequence of phase evolution: (1) Pt_xPb , (2)Fluorite, (3)Perovskite
- No overlap in Pt_xPb and perovskite phases
- Amount of Pt_xPb formed decreases with decreasing heating rate
- Crystallinity of fluorite phase changes with heating rate
- Fluorite

Variation of texture with heating rate



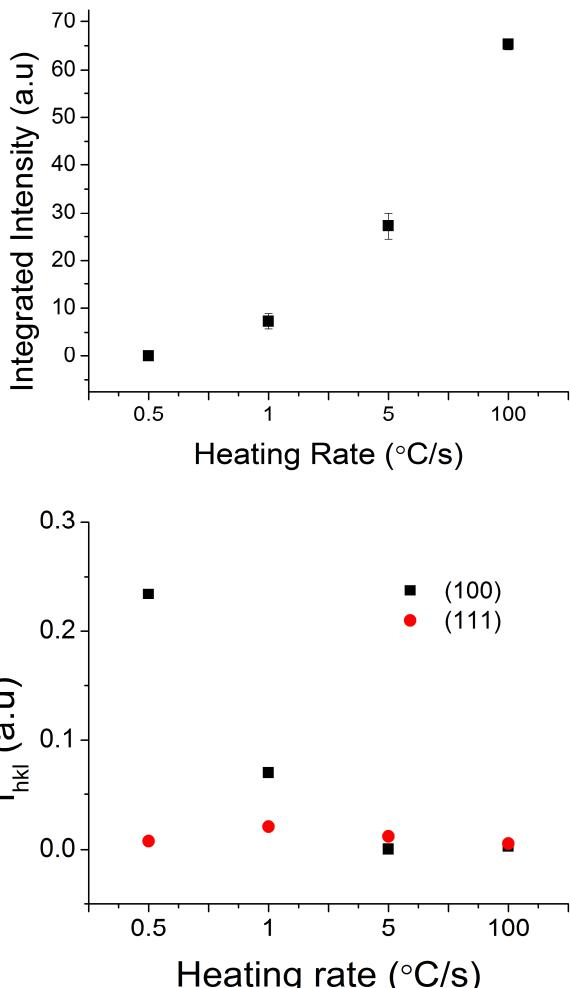
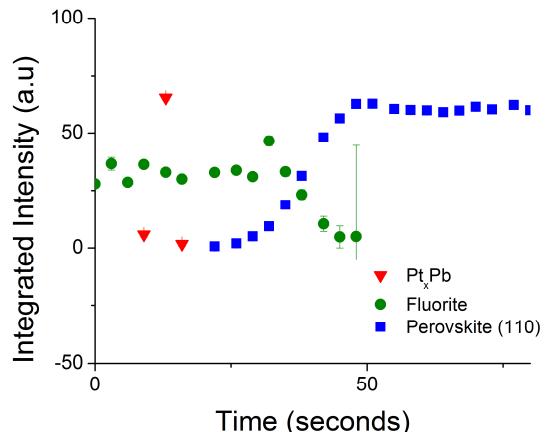
- (100) texture decreases with increasing heating rate
- In fast heating rates, homogenous nucleation may dominate over heterogeneous nucleation



Pt_xPb might not seed (111) texture

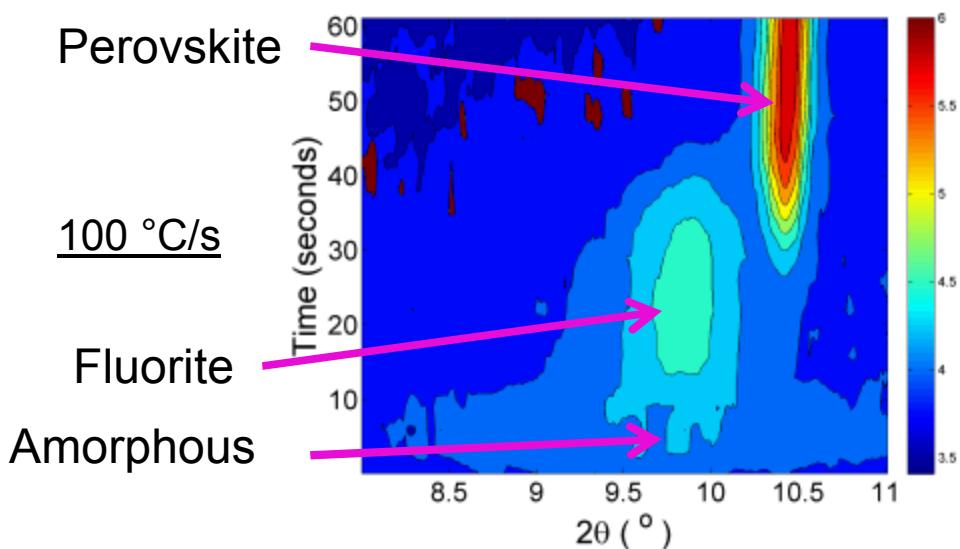
- Maximum intensity of Pt_xPb increases with heating rate
- Observed stability of Pt_xPb is consistent with *ex situ* observations*
- No overlap is observed between the Pt_xPb and perovskite
- Weak (111) or random texture obtained for samples with intense Pt_xPb formation

Phase evolution for 100°C/s



Fluorite: evolution and texture

- During crystallization, the fluorite phase is observed to always precede the perovskite phase
- The broad peak characteristic of the amorphous phase continuously transforms into the (111) - fluorite peak
- Trend is observed to be consistent for all the heating rates investigated
- No preferred orientation was observed in the fluorite phase



- Fluorite phase may not seed the (111) orientation in these films

Summary: Phase and texture evolution

- The observed phase evolution sequence is: 1. Pt_xPb , 2. fluorite, and 3. perovskite
- No evidence for seeding of texture by Pt_xPb or fluorite phase is observed

Outline

1. Introduction

- Motivation
- Methodology

2. Phase evolution (UF)

- Effect of Nb doping
- Pb content in solution

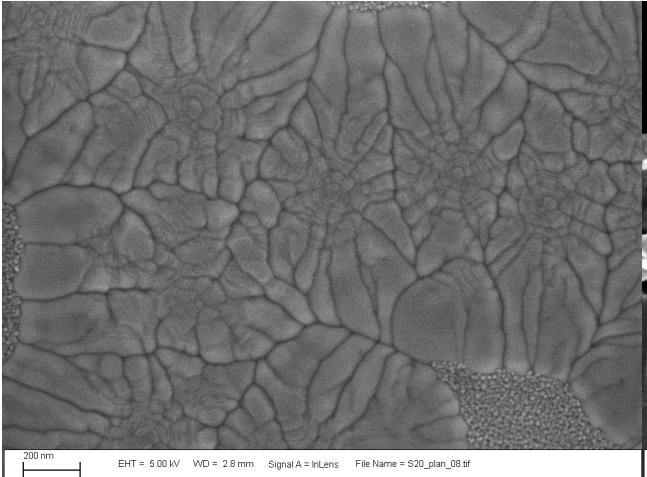
3. Phase and Texture evolution (APS)

- Effect of heating rate
- Cation segregation in thin film

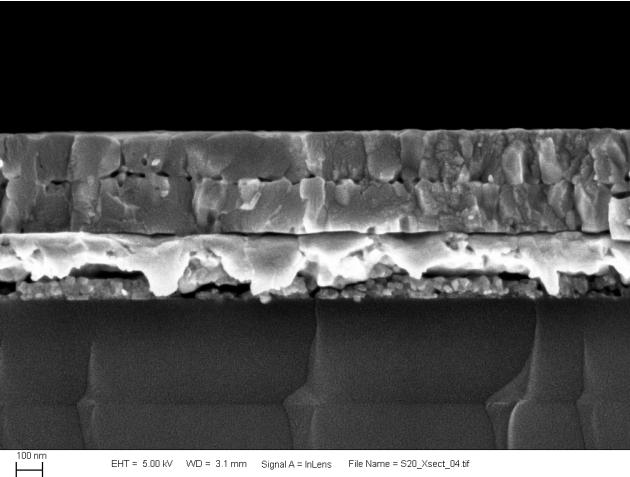
4. Ongoing and Future work

5. Conclusions

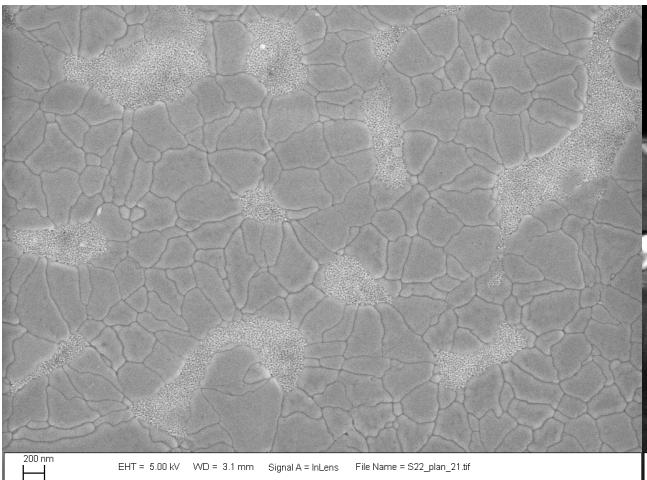
Microstructural characterization



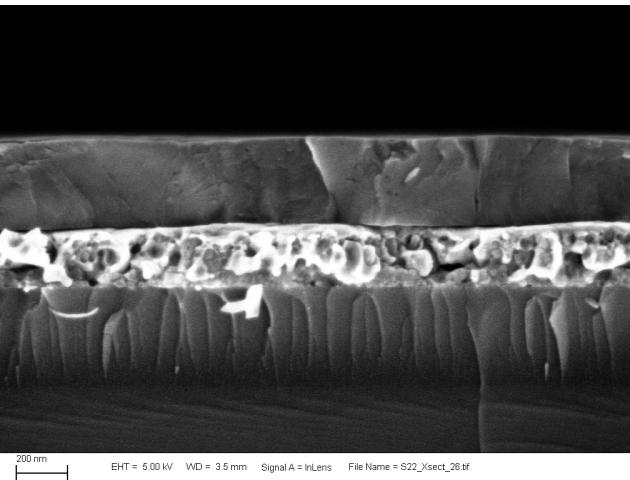
$\sim 100^\circ\text{C/s}$



$\sim 100^\circ\text{C/s}$



$\sim 0.5^\circ\text{C/s}$

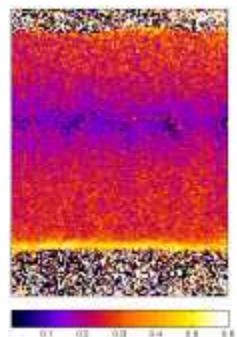


$\sim 0.5^\circ\text{C/s}$

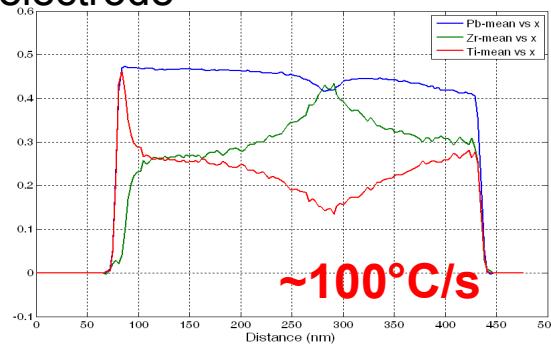
- Rosette type grain structure in 100 °C/s
- Film crystallized in two layers
- Porosity observed in the middle
- Homogenous nucleation
- Columnar type grains observed for 5 °C/s, 1°C/s and 0.5 °C/s
- Nucleation at the film – electrode interface
- Grain size observed to increase with decreasing heating rate

Chemical mapping along thickness

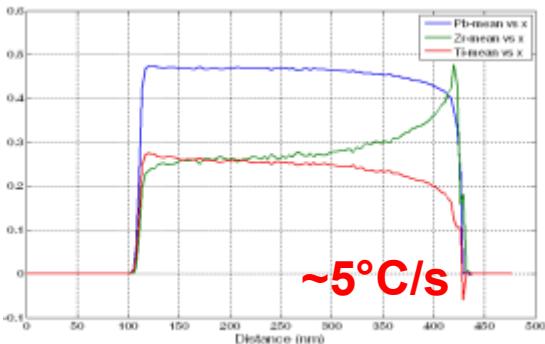
Ti cation maps



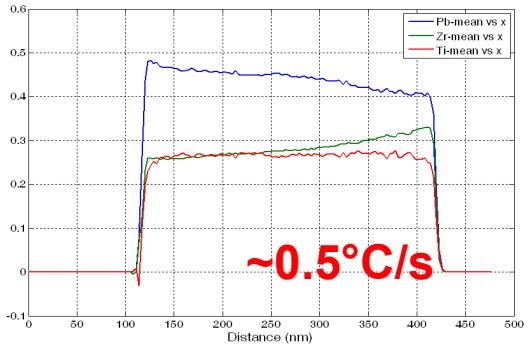
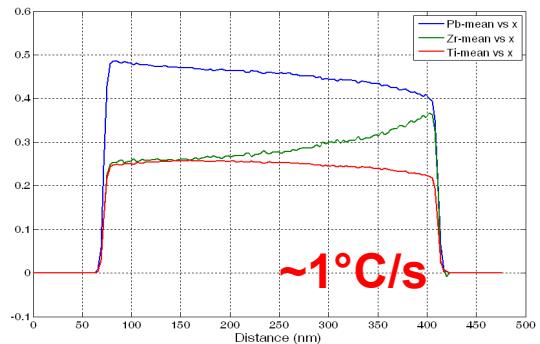
Pt electrode



Pt electrode

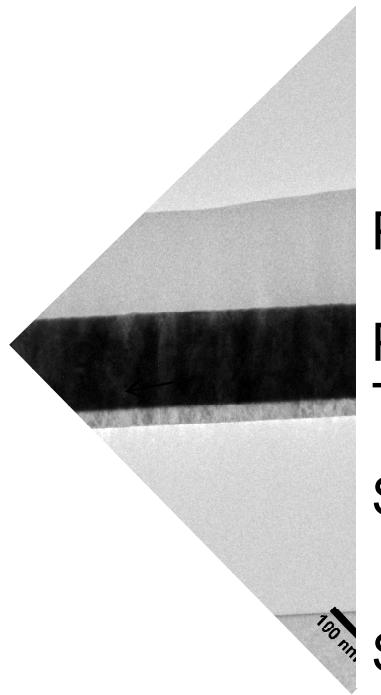


- Preferential Ti segregation near interface
- Zr/Ti segregation through thickness

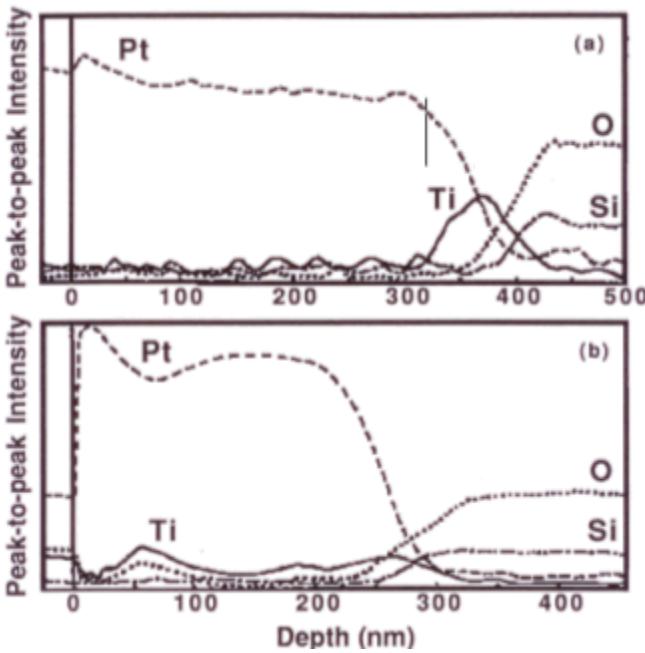


- No preferential Ti near interface
- Some Zr/Ti segregation through thickness, but less than faster ramp rates.

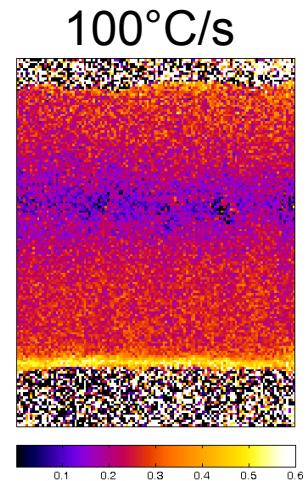
Ti segregation at interface: influence of kinetics



PZT
O₂
Pt
Ti
SiO₂
Si

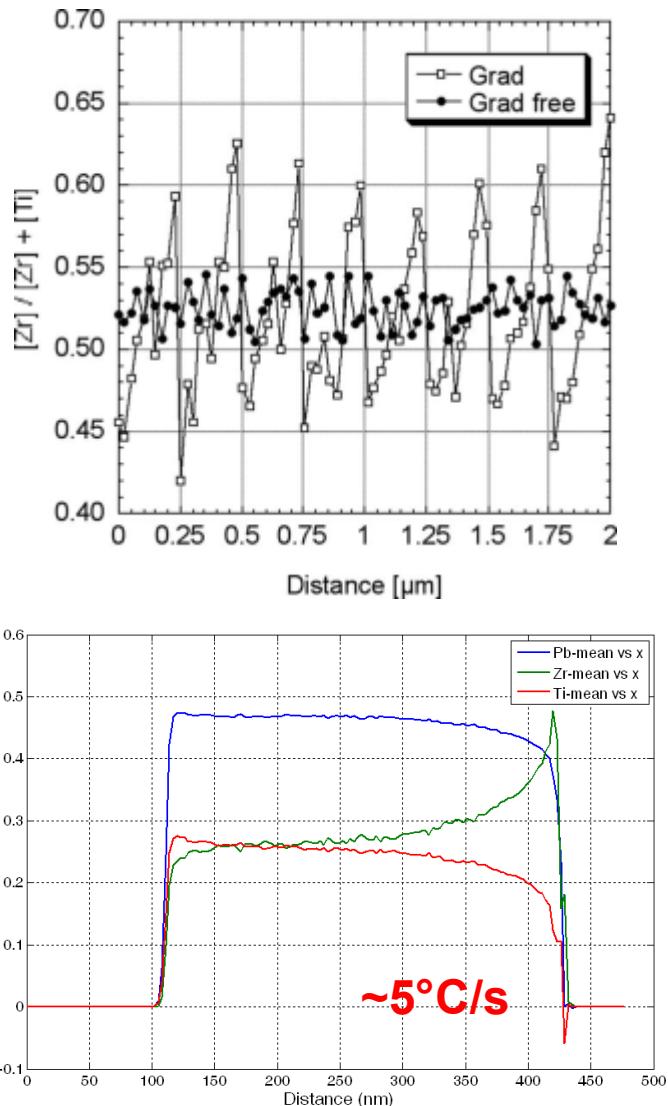


- Pt allows for diffusion of both Oxygen and Titanium
- Diffusion of Titanium species decreases with oxidation
- Considerable Oxygen deficiency in the film allows for diffusion of Titanium to the top of the Platinum electrode



Zr/Ti variation across thickness of film

- Calame and Muralt reported Zr/Ti segregation across the thickness of the film due to preferential nucleation of Ti-rich composition
- Zr/Ti segregation similar to that observed for in this study
- Intense Ti segregation observed in this study is not reported
- Could be limited by resolution of the probing technique



Conclusions: Cation segregation

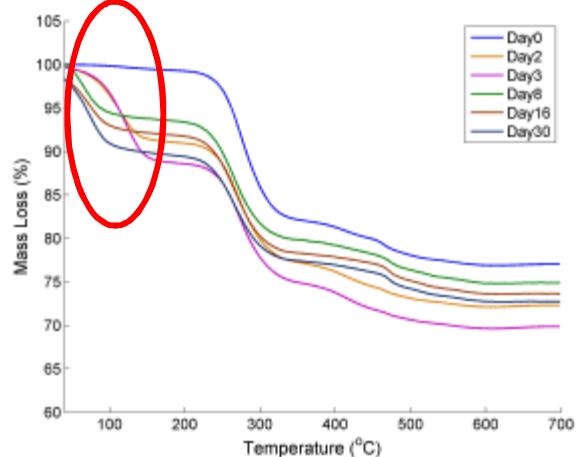
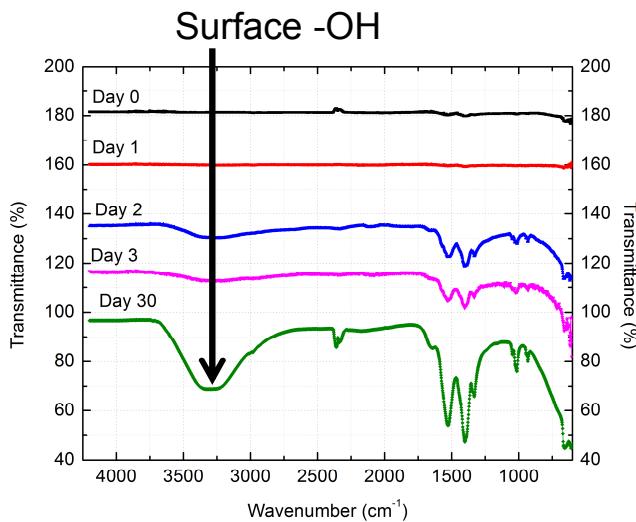
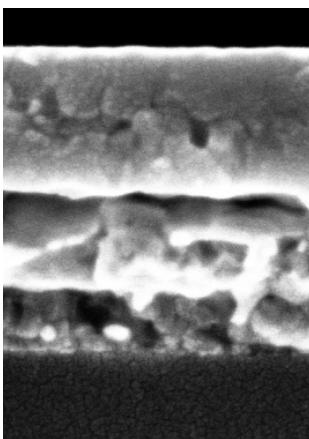
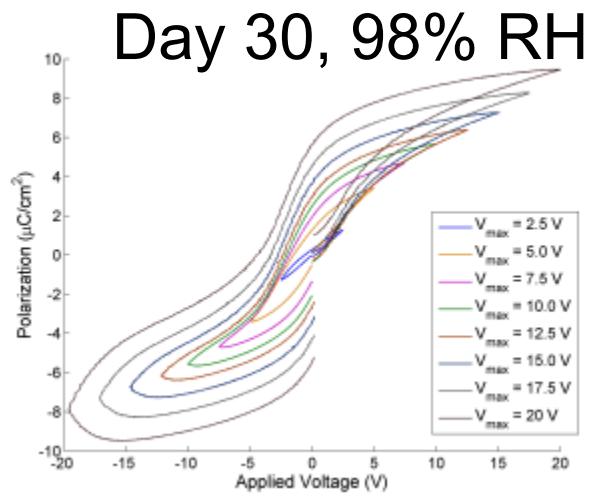
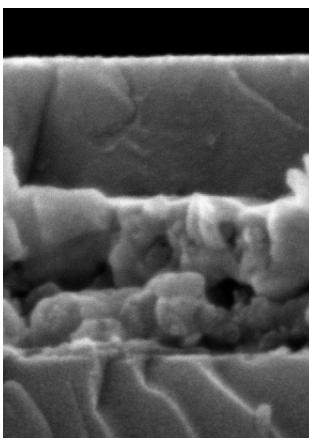
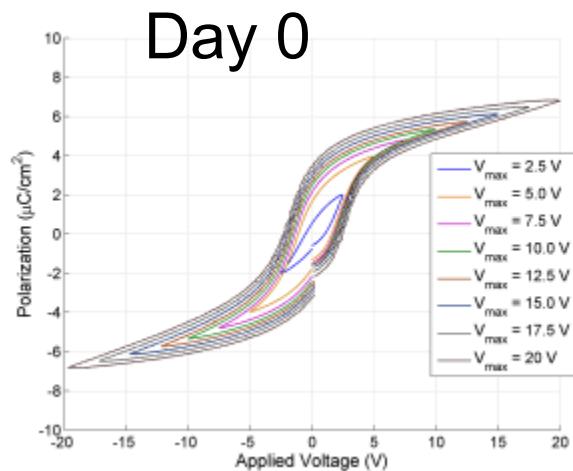
- Intense Ti segregation is observed at the thin film – electrode interface for fast heating rates
- Zr/Ti segregation is observed to decrease with decrease in heating rate
- Ti segregation could be due to preferential nucleation of Ti-rich composition or diffusion of Ti from the adhesion layer

Ongoing and Future work

The effect of the following factors on phase and texture evolution is being investigated

- Adhesion layer
- Solution chemistry (IMO vs Sol-gel)
- Aging in the gel state
- Heating geometry

Aging of pre-crystallized thin films



Degradation of electrical properties is observed for films aged in high RH conditions

Conclusions

- Nb doping influences the stability of PtxPb , fluorite and perovskite phases
- Pb deficiency in the starting solution leads to decreased reaction between film and electrode
- No evidence for seeding of orientation of the PZT phase by the intermediate phases was observed
- Ti segregation was observed to be more intense for faster heating rates

Phase and texture evolution in solution deposited PZT thin films

Krishna Nittala¹, Geoff L. Brennecka², Jon F. Ihlefeld², Bruce A. Tuttle²,
Douglas S. Robinson³, and Jacob L. Jones¹

¹*Department of Materials Science and Engineering, University of Florida, Gainesville, FL, USA*

²*Sandia National Laboratories, Albuquerque, NM, USA*

³*Advanced Photon Source, Argonne National Laboratory, Argonne, IL, USA*



Supported by the National Institute for NanoEngineering (NINE) and the Laboratory Directed Research and Development (LDRD) program at Sandia National Laboratories.

Sandia is a multiprogram laboratory operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Company, for the United States Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000.

Use of the Advanced Photon Source, an Office of Science User Facility operated for the U.S. Department of Energy (DOE) Office of Science by Argonne National Laboratory, was supported by the U.S. DOE under Contract No. DE-AC02-06CH11357.

Sungwook Mhin and Katherine Dunnigan

