



Fracture and fatigue of commercial grade API pipeline steels in gaseous hydrogen

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Motivation

- Materials/System Selection
 - Microstructure of pipelines steels varies depending on strength level and era of manufacture
 - Need to assess performance of existing pipelines for hydrogen service as well as new pipeline steels for hydrogen service
- Engineering Requirements
 - Testing requirements in ASME Code for Pressure Piping, B31.12 Hydrogen Piping and Pipelines
 - Refers to ASME BPVC, Section VIII, Division 3, Article KD-10: Special Requirements for Vessels in High-Pressure Gaseous Hydrogen Transport and Storage Service
 - Article KD-10 requires two sets of engineering data, which are measured in gaseous hydrogen
 - Fracture mechanics properties (K_{IH})
 - Fatigue crack growth rates (da/dN)
- Science-based Engineering
 - Develop robust testing methodologies
 - Understand hydrogen-metal interactions to aid materials selection for hydrogen service



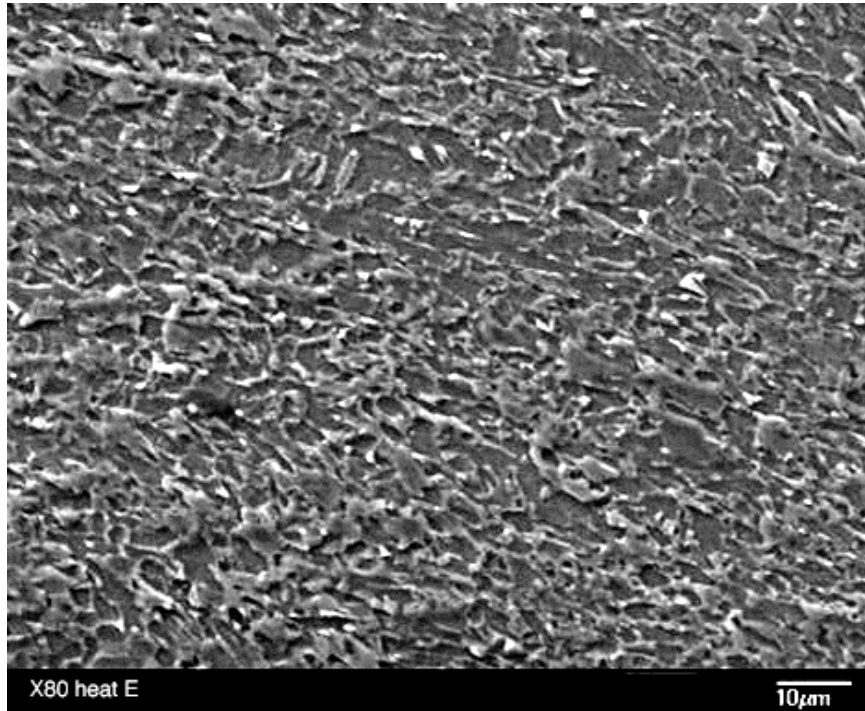
Pipeline steels being tested

Designation	Grade	Vintage (approx)	Microstructure
B	X80	2000	Ferrite / Acicular Ferrite
D	X60 HIC	2000	Polygonal Ferrite (Pearlite ~5%)
E	X70/X80	2000	Ferrite / Acicular Ferrite (microalloyed with Mo)
F	X70/X80	2000	Ferrite / Acicular Ferrite
J	X52	1990 2000	Ferrite - Pearlite (~5%)

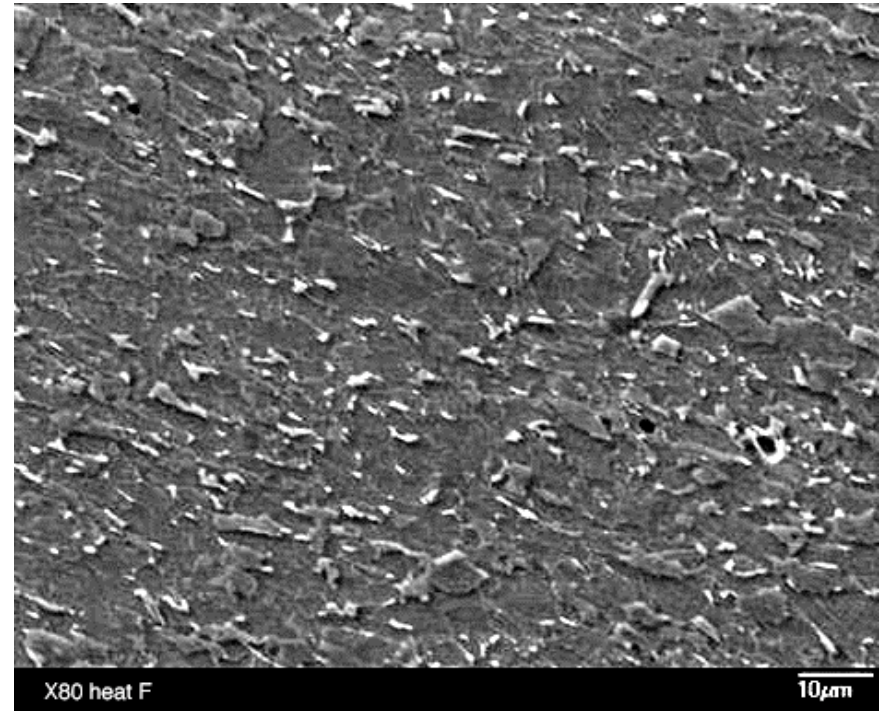
All steels tested as skelp; except alloy J, which is from pipe.



Microstructure



X80 (alloy E)
Mo additions ~0.15 wt%
Fine Acicular Ferrite
Yield strength: 593 MPa

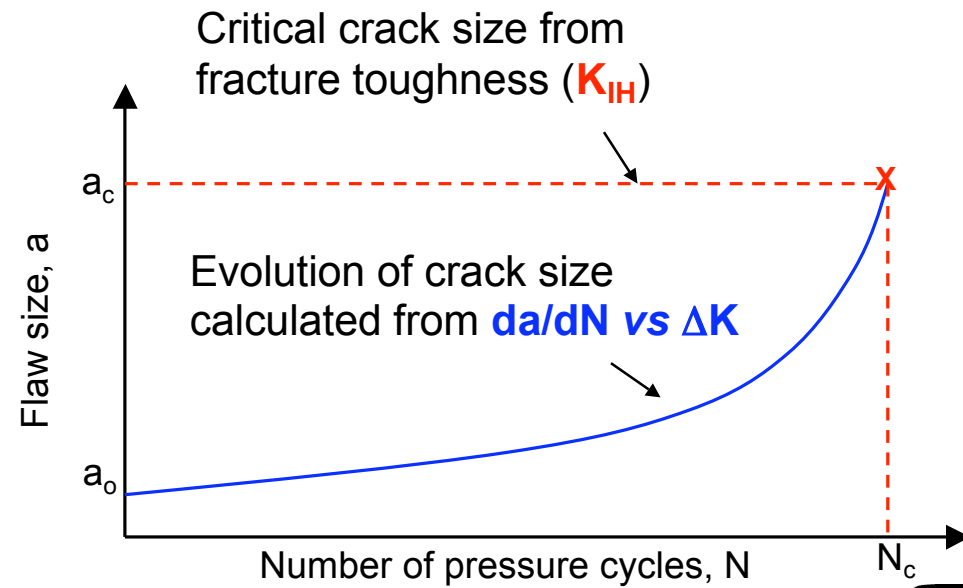
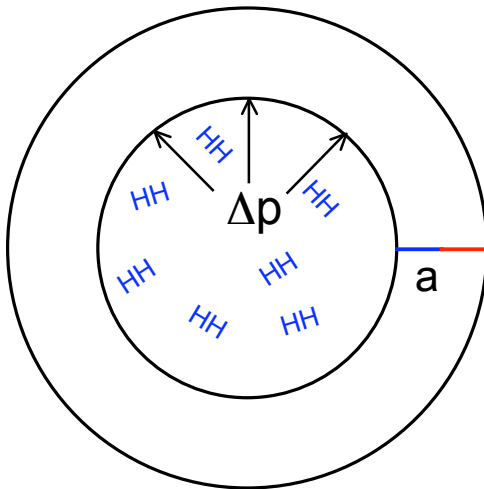


X80 (alloy F)
Acicular Ferrite w/
30% polygonal ferrite
Yield strength: 552 MPa



Article KD-10 provides design method for Hydrogen containing pressure vessels

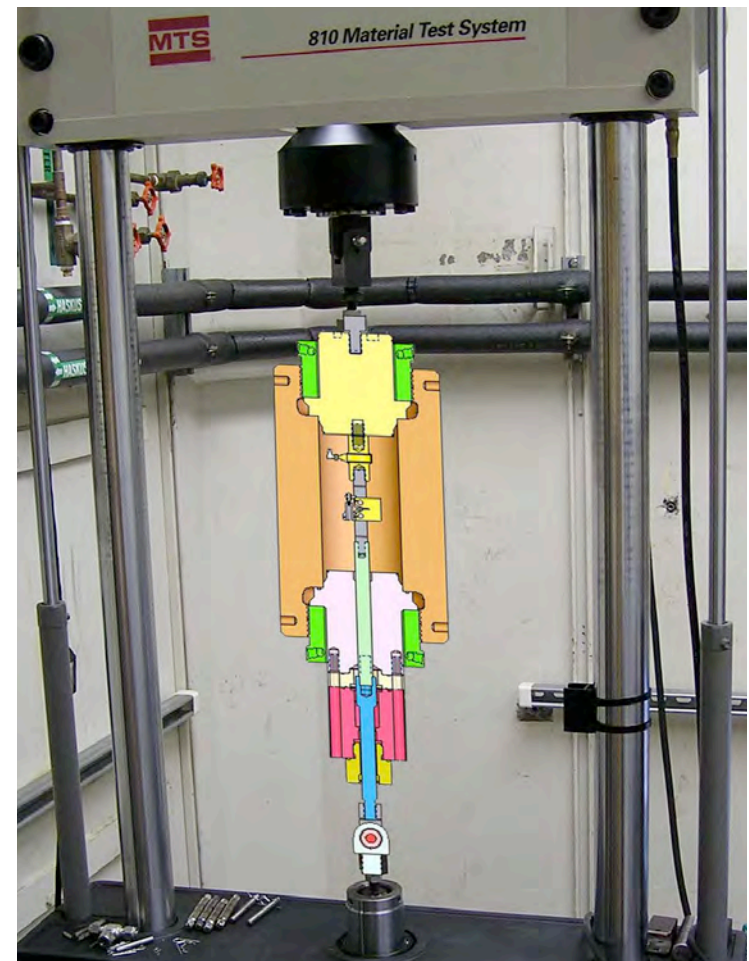
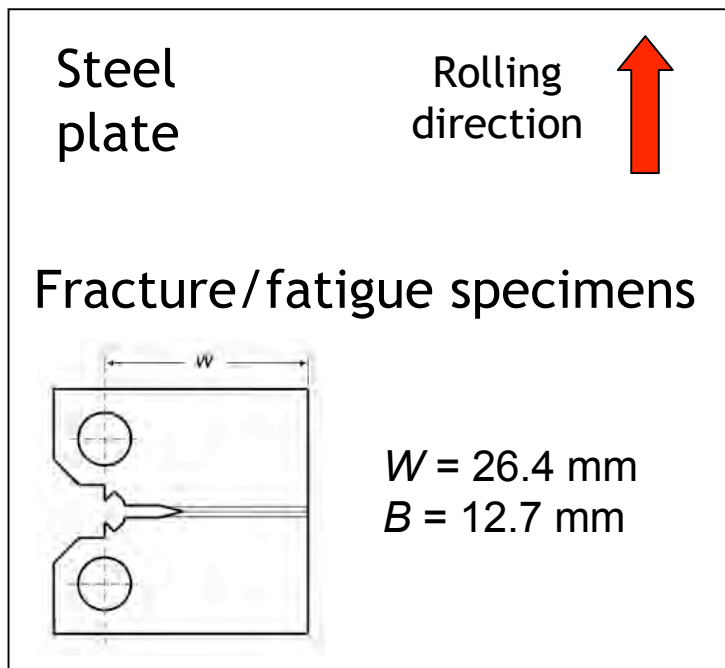
- Initial flaw assumed (a_0)
- Crack propagates in fatigue at rate determined by measured crack growth rate (da/dN)
- At critical crack length (a_c), fracture resistance (K_{IH}) is exceeded and crack propagation is unstable





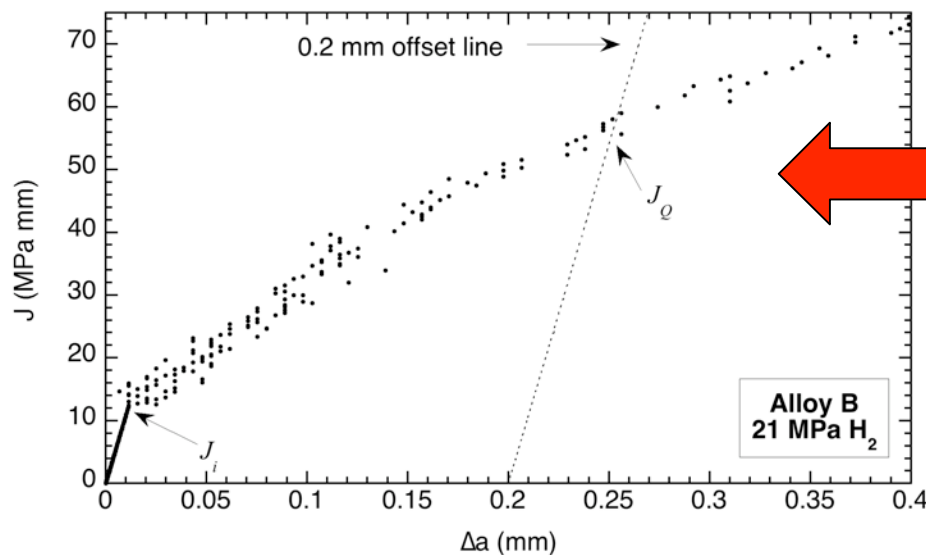
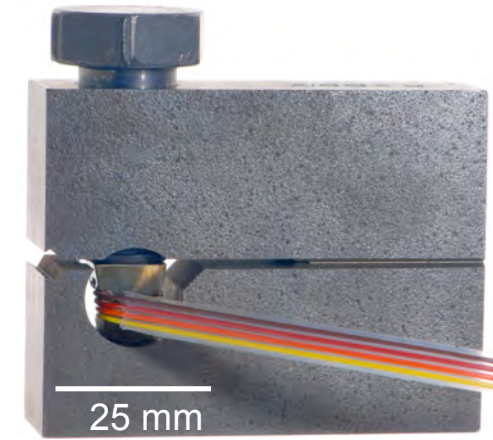
Fracture mechanics tests were conducted in gaseous hydrogen at pressure of 21 MPa

- Fatigue crack growth: da/dN
 - ASTM E647, constant load amplitude
- Fracture resistance: K_{IH}
 - ASTM E1820, elastic-plastic analysis using J-R curve determination



Crack extension could not be initiated in constant displacement tests

- Article KD-10 refers to test procedure in ASTM E1681
- Crack extension could not be achieved in bolt-loaded (constant displacement) specimens of steels with yield strength less than 550 MPa following ASTM E1681 in gaseous hydrogen at pressure of 103 MPa



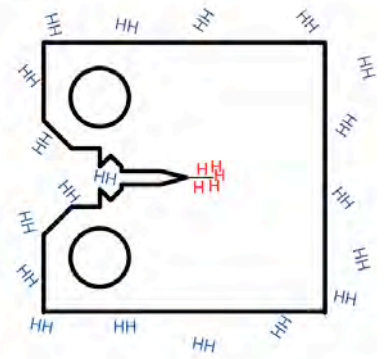
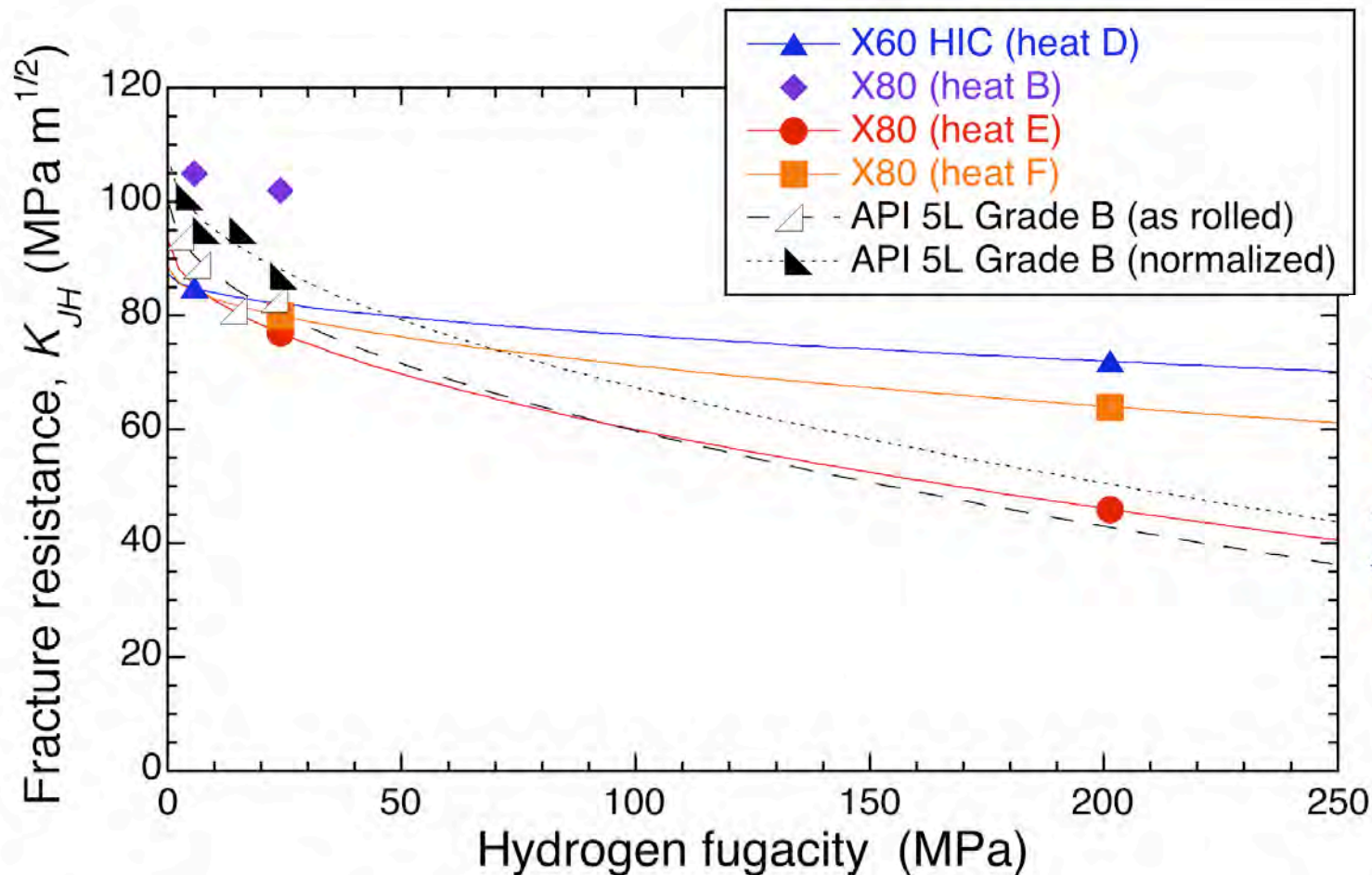
JR-curve construction was used to evaluate fracture toughness (ASTM E1820)

- monotonic loading
- potential difference method

$$K_J = \sqrt{\frac{JE}{(1-\nu^2)}}$$

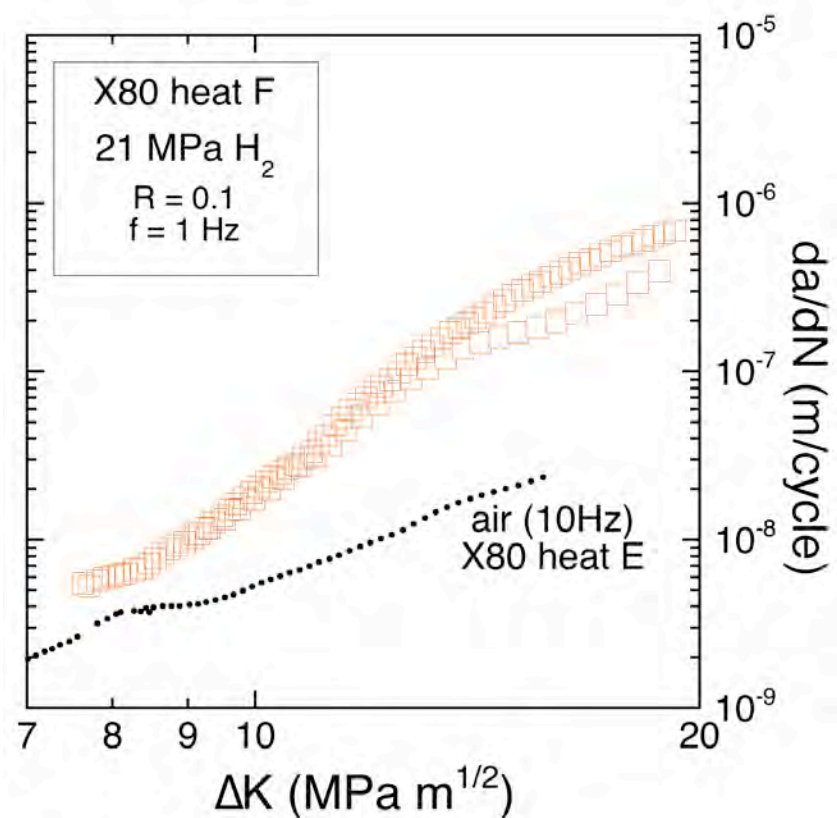
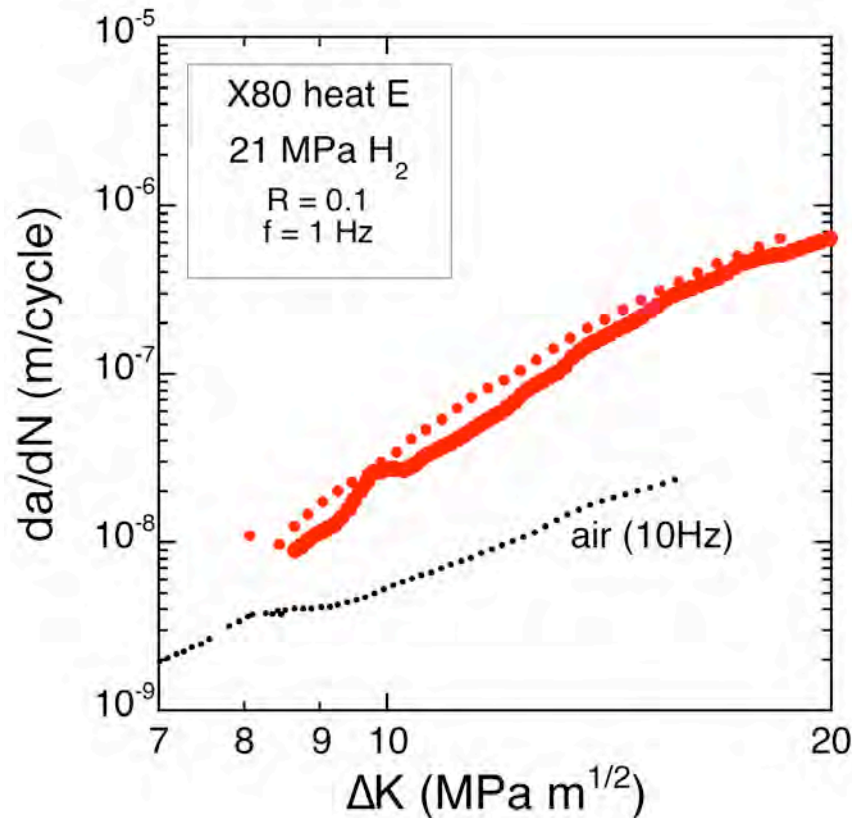
Fracture resistance in gaseous hydrogen depends on pressure

- Curves represent empirical fit assuming square root dependence on fugacity ($K \propto f^{1/2}$)
- API 5L Grade B: data from literature





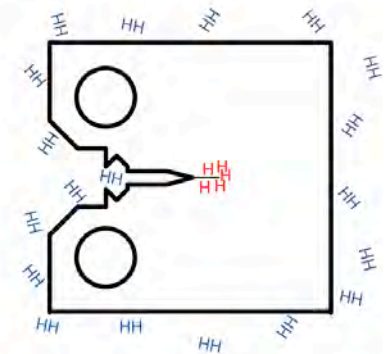
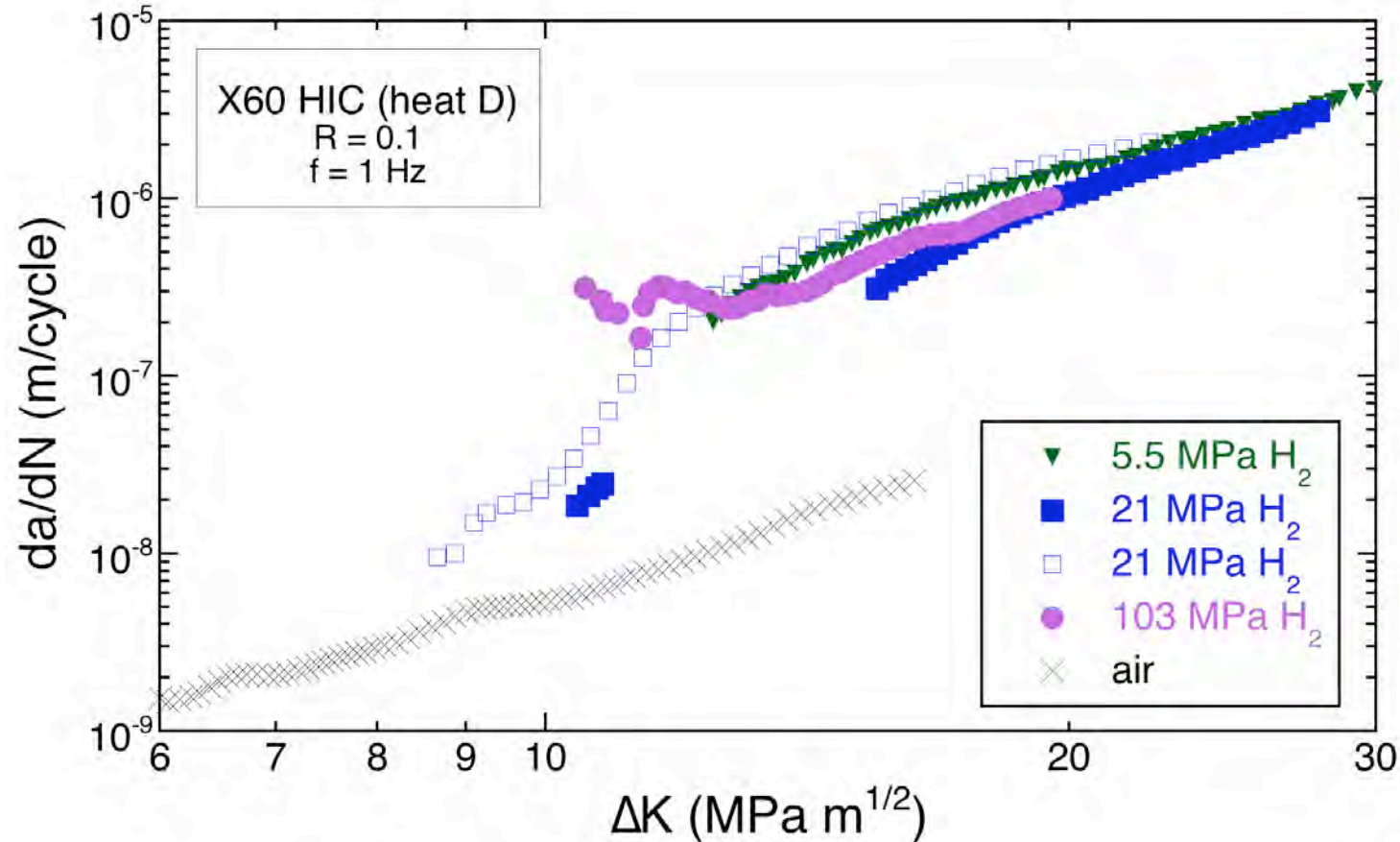
Fatigue crack growth is strongly affected by gaseous hydrogen



At “high” ΔK (>15 MPa $m^{1/2}$): $(da/dN)_H \geq 10 \times (da/dN)_{air}$
At “low” ΔK (<7 MPa $m^{1/2}$): $(da/dN)_H \Rightarrow (da/dN)_{air}$

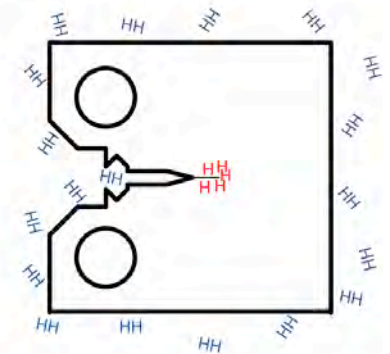
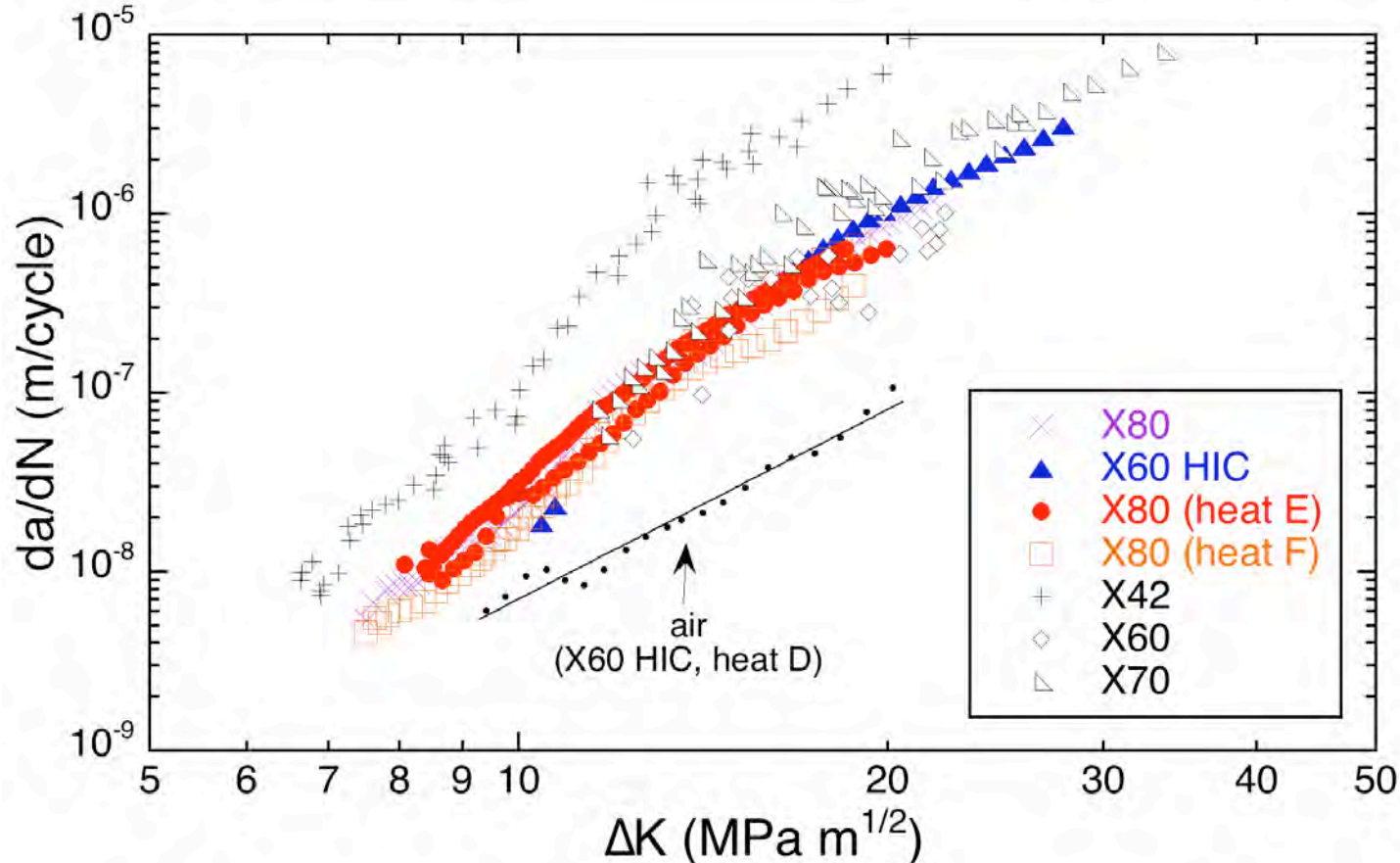
Fatigue crack growth in gaseous hydrogen does not depend on pressure

- Additional testing is necessary to clarify trends for fatigue crack growth at “low” ΔK in high pressure gaseous hydrogen



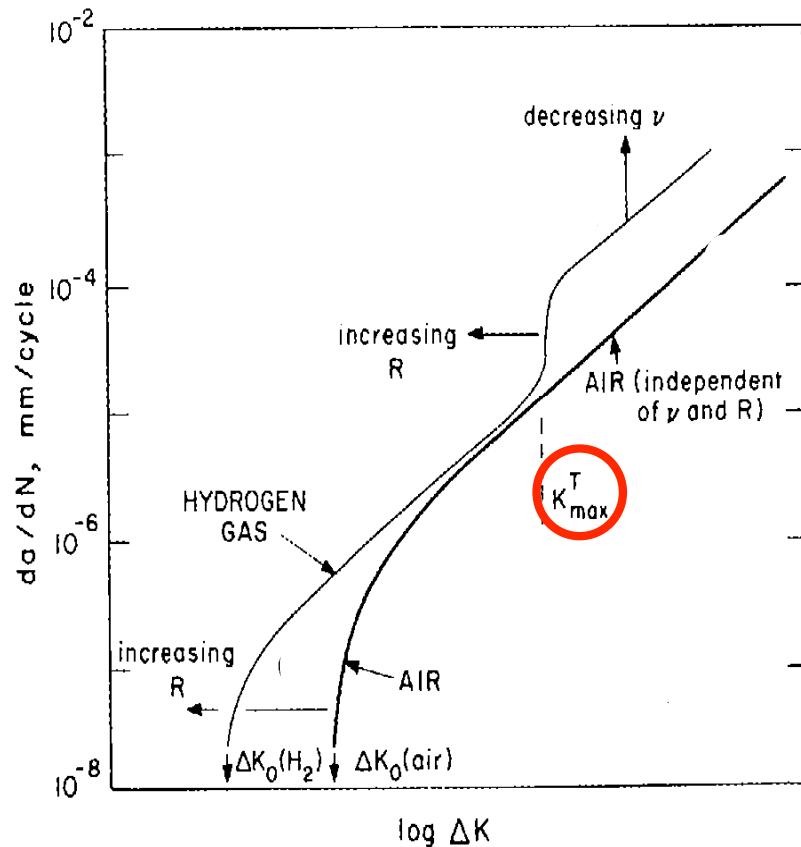
Trends for fatigue crack growth are consistent for a range of pipeline steels

Fatigue crack growth rates in gaseous hydrogen (~20 MPa pressure) appear to be consistent for a broad range of pipeline steels





Fatigue crack growth results are consistent with previous studies

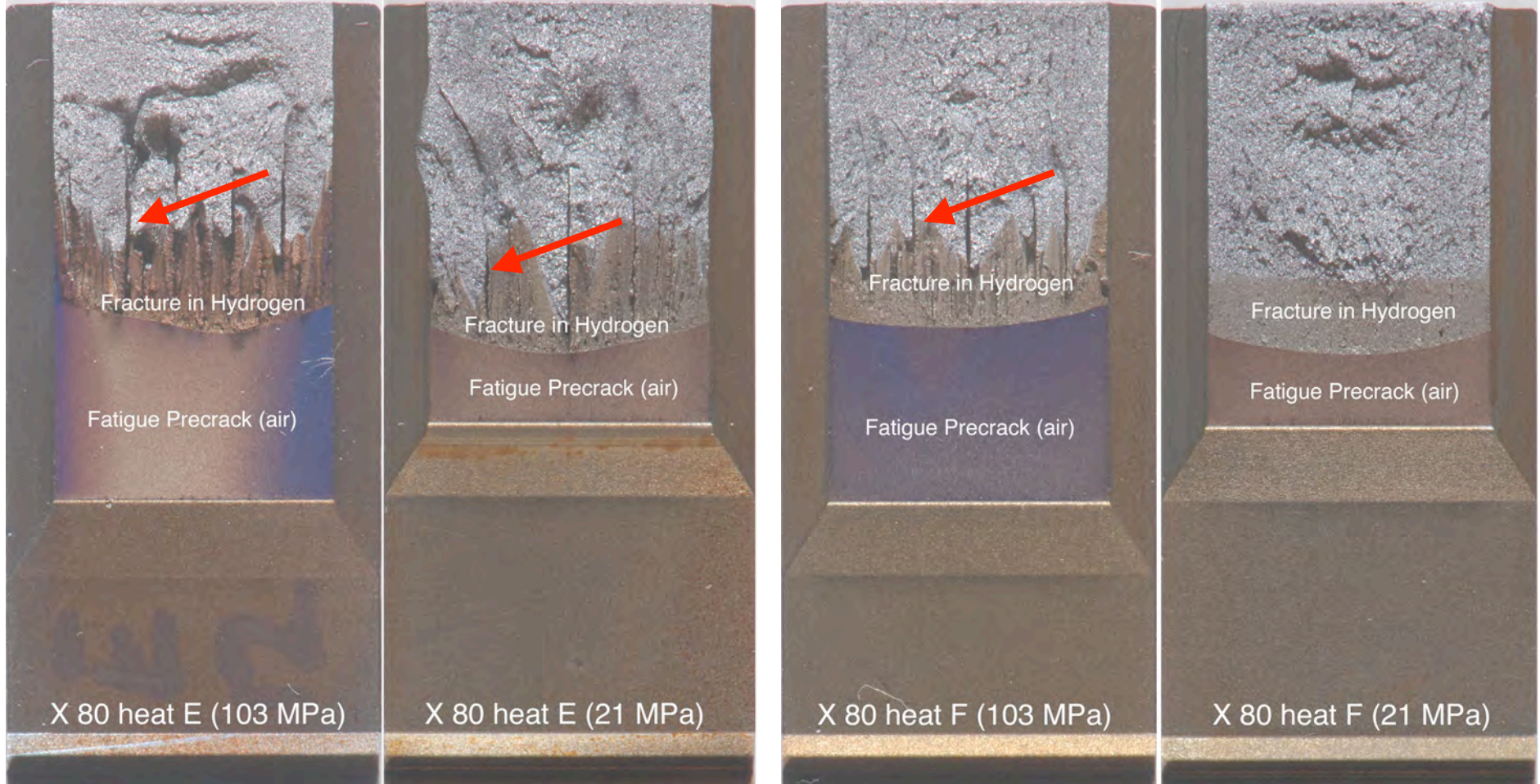


After: Suresh and Ritchie *Metal Science* 16 (1982) p.529

- Fatigue crack growth in gaseous hydrogen cannot be idealized as simple power law
- Hydrogen-assisted crack growth is “triggered” at some critical stress intensity factor called K_{max}^T
- $K_{max}^T < 15 \text{ MPa m}^{1/2}$ for low-strength carbon steels in gaseous hydrogen at pressure greater than 5 MPa
- More work is necessary to characterize transition region



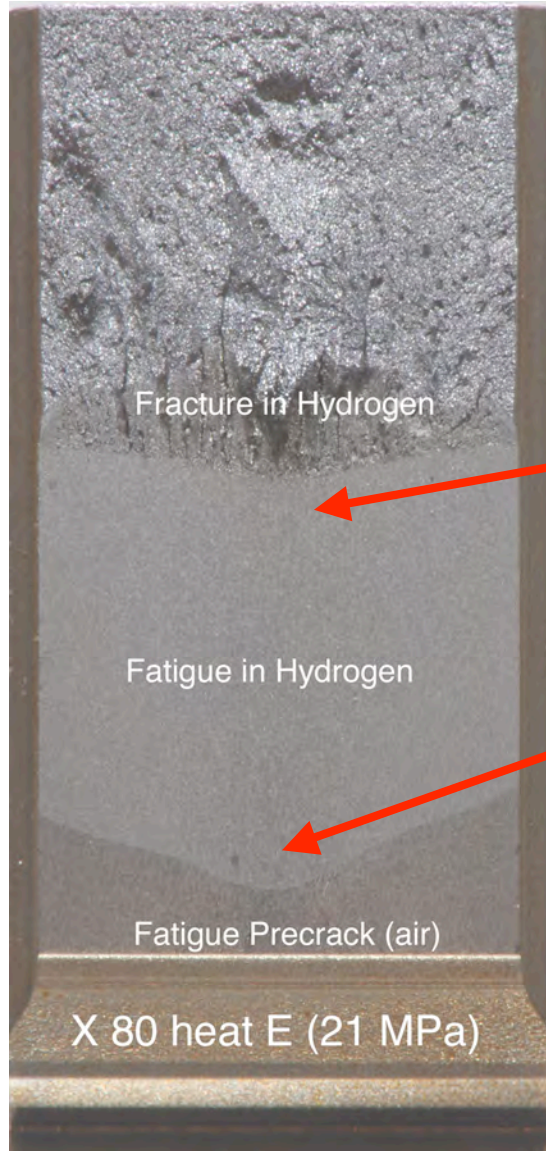
Delamination is promoted in high-pressure hydrogen, at high stress



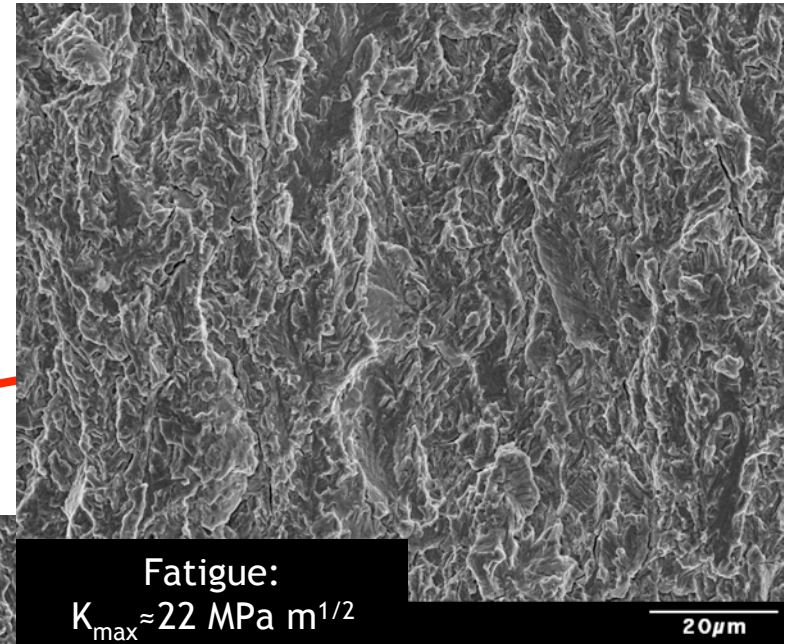
Delamination is not observed in fatigue (low stress)



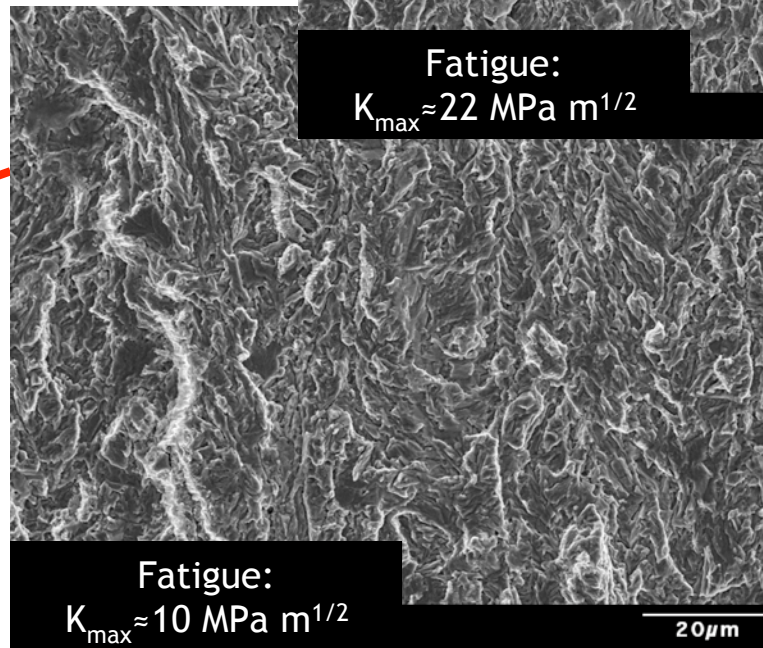
Fatigue fracture surface in hydrogen is not a strong function of $K_{\max} > K_{\max}^T$



More secondary cracking and flat features at higher K_{\max}



Fatigue:
 $K_{\max} \approx 22 \text{ MPa m}^{1/2}$

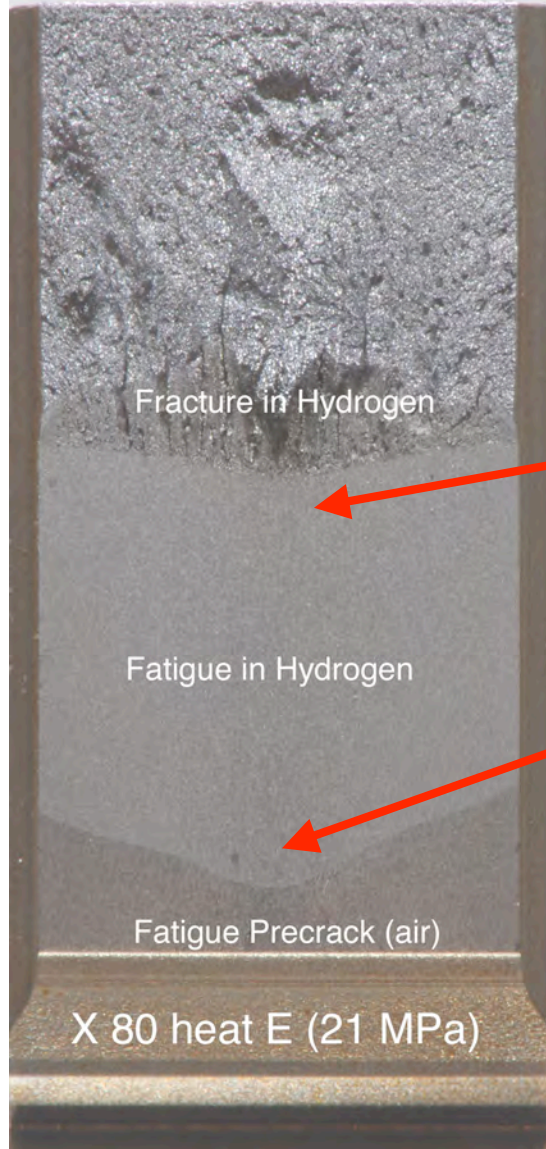


Fatigue:
 $K_{\max} \approx 10 \text{ MPa m}^{1/2}$

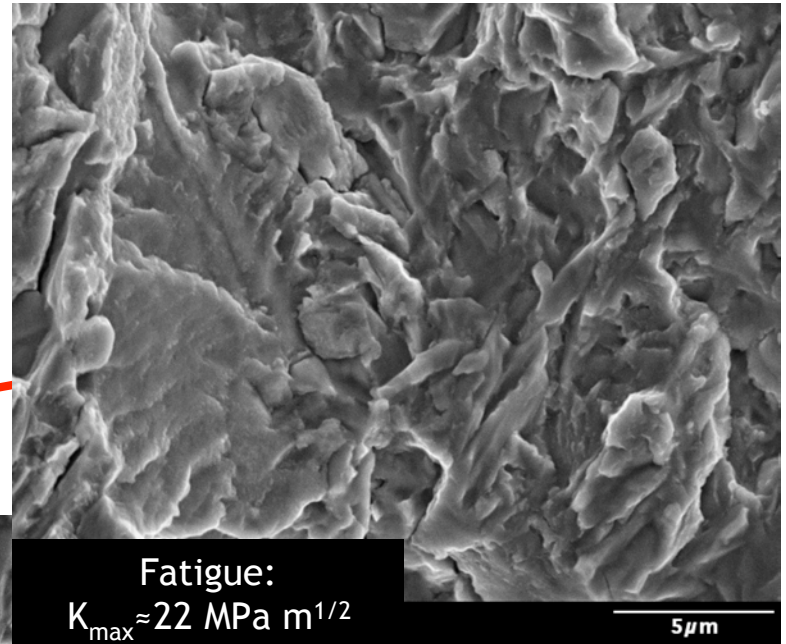
Alloy E
21 MPa H₂
R=0.1



Fatigue fracture surface in hydrogen varies as a function of K_{max}

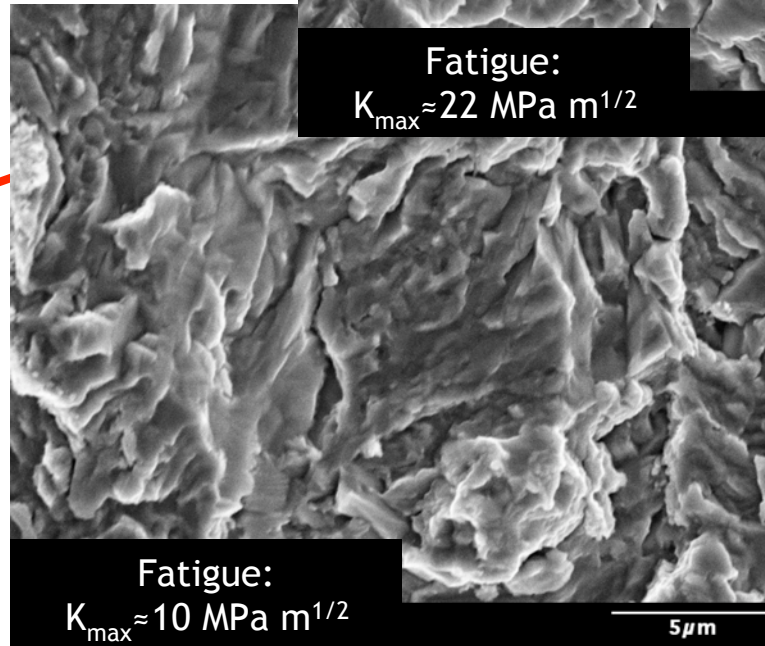


More secondary cracking and flat features at higher K_{max}



Fatigue:
 $K_{max} \approx 22 \text{ MPa m}^{1/2}$

5 μm

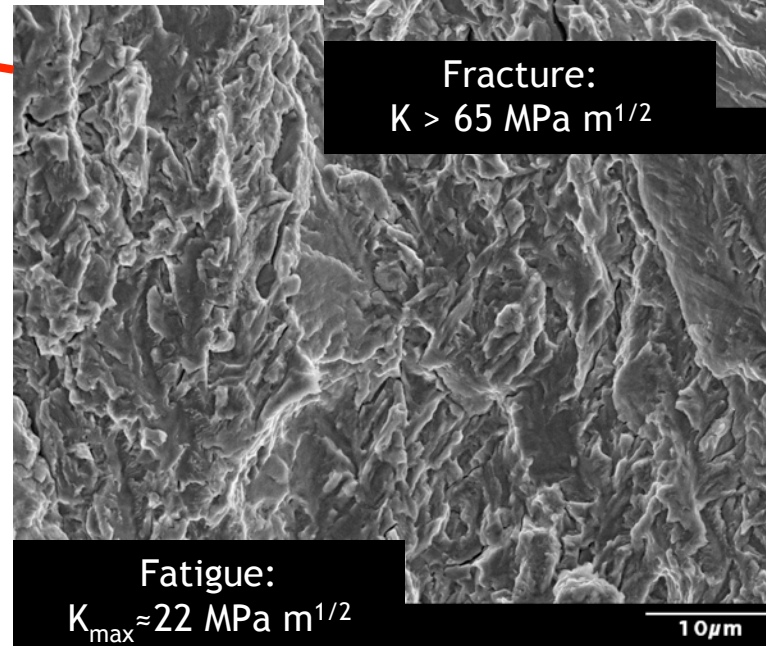
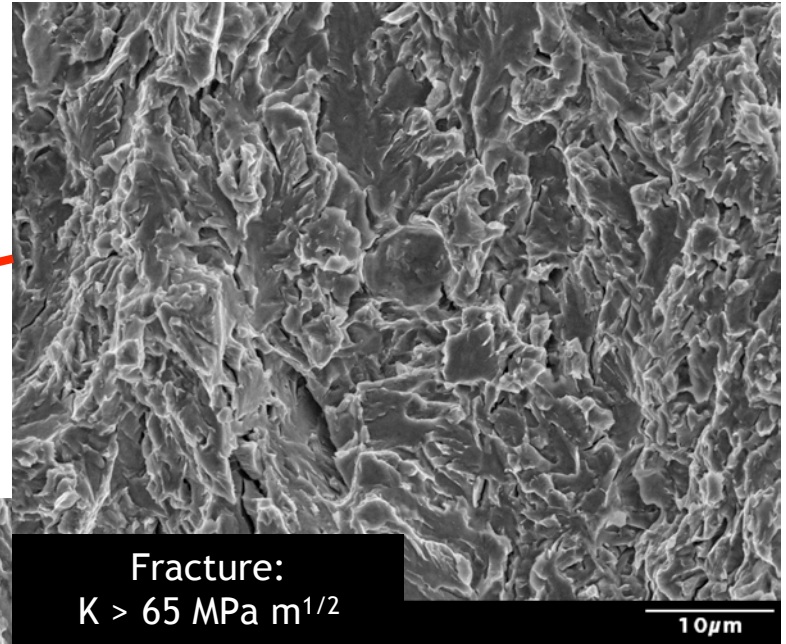
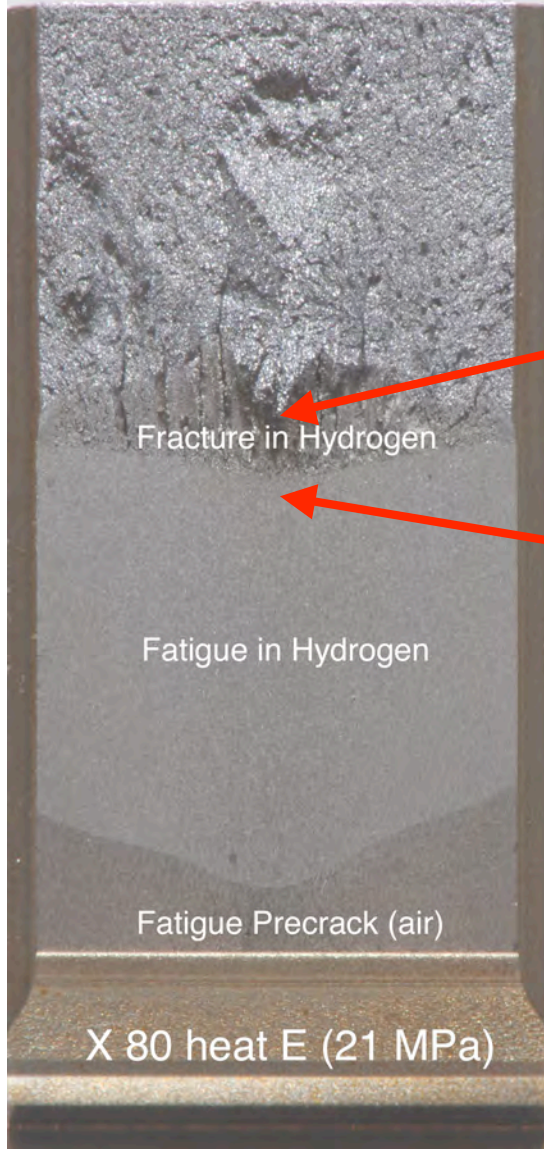


Fatigue:
 $K_{max} \approx 10 \text{ MPa m}^{1/2}$

5 μm

Alloy E
21 MPa H₂
R=0.1

Fracture surface in fatigue at high K_{max} is similar to fracture under rising load



Alloy E
21 MPa H₂
R=0.1



Summary

Determination of fracture resistance and fatigue crack growth rates of pipeline steels in gaseous hydrogen are motivated by ASME B31.12 code

- Fracture toughness according to BPVC VIII.3 KD-10 could not be determined
- Elastic-plastic fracture analysis of a range of pipeline steels
 - fracture resistance: $K_{JH} = 80\text{-}100 \text{ MPa m}^{1/2}$
 - Hydrogen gas pressure ~5 to 25 MPa
 - Low fracture resistance at higher pressure
- Fatigue crack growth is accelerated by a factor of 10 at high ΔK , but converges with air data at low ΔK
- Fatigue crack growth rates appear less sensitive to alloy and hydrogen pressure than fracture resistance
 - At “high” ΔK , crack growth rates are not dependent on pressure in the range of 5 to 100 MPa