

Finite Element Predictions of Grainscale Behavior in BCC Metals

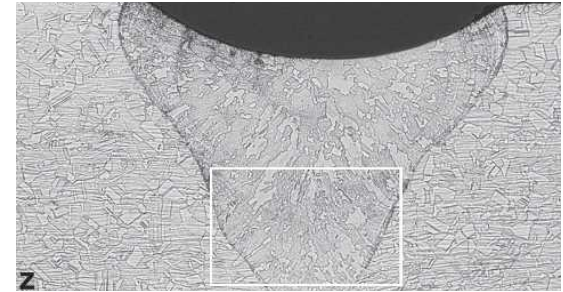
Christopher R. Weinberger

Hojun Lim, Jay Carroll, Thomas E. Buchheit,
Corbett C. Battaile and Brad L. Boyce

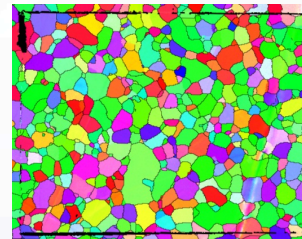
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Can We Predict Grainscale Deformation Behavior?

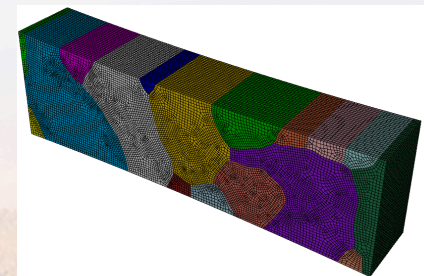
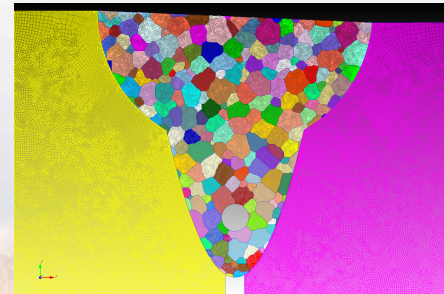
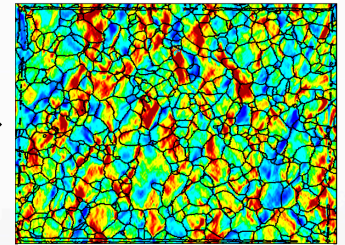
- Need the capability to model complex geometries with realistic microstructures.
- Deformation in polycrystals involves the networks of grains
 - Need expedient models grainscale deformation
 - Can we use current CPFEM models?
 - How well do they work?
 - What needs improvement?



Microstructure



Strain



Finite Element Predictions of Grainscale Behavior in BCC Metals

- Previous work in validating CPFEM models
 - Crystal Rotations (EBSD)
 - Surface Strains (DIC)
- Our Ongoing Validation of BCC Tantalum CPFEM Models
 - Our CPFEM model and experiments
 - Surface Strain Comparisons
 - Texture Evolution and Crystal Rotations
 - Failure
- Model Sensitivity
 - Mesh Sensitivity
 - Sensitivity to Initial Orientations
 - Sensitivity to Slip Planes



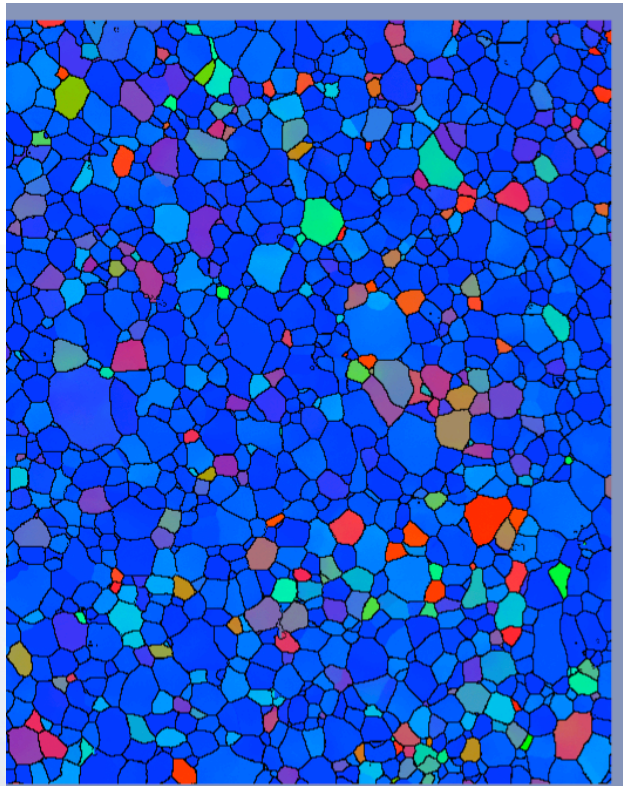


Previous Validation Attempts Using EBSD & DIC

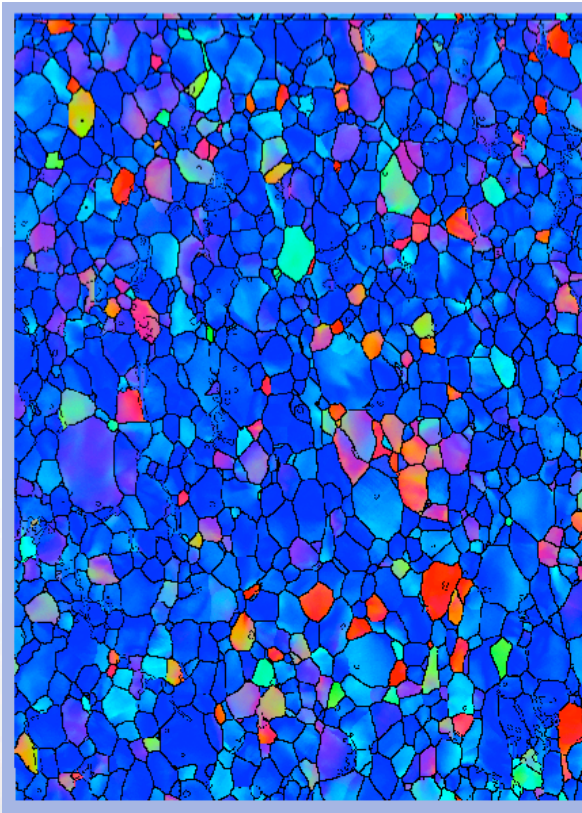
(A limited sampling...)

EBSD of Interstitial-Free Steel Polycrystals

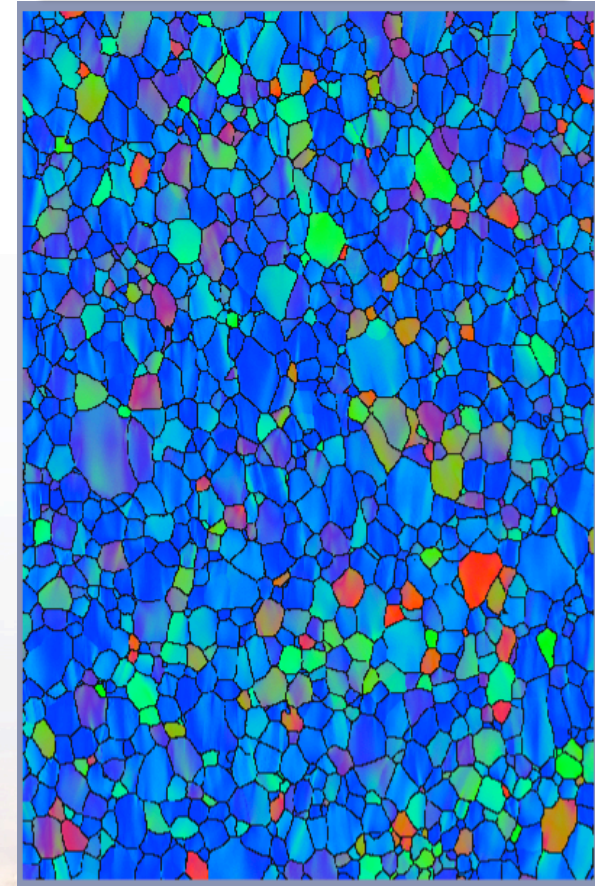
{110+112} only, surface layer



Experiment, 0% Strain



Experiment, 12% Strain



Simulation, 12% Strain



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EBSD of Indented Copper Single Crystals

The figure displays a comparison of three methods for simulating the microstructure of indented copper single crystals: viscoplastic CPFEM, experimental 3D EBSD, and dislocation-based CPFEM. The results are presented as EBSD maps for three different scans (scan 7, scan 8, and scan 9).

The columns represent the simulation/experiment types:

- viscoplastic CPFEM
- experiment 3D EBSD
- dislocation-based CPFEM

The rows represent the specific scans:

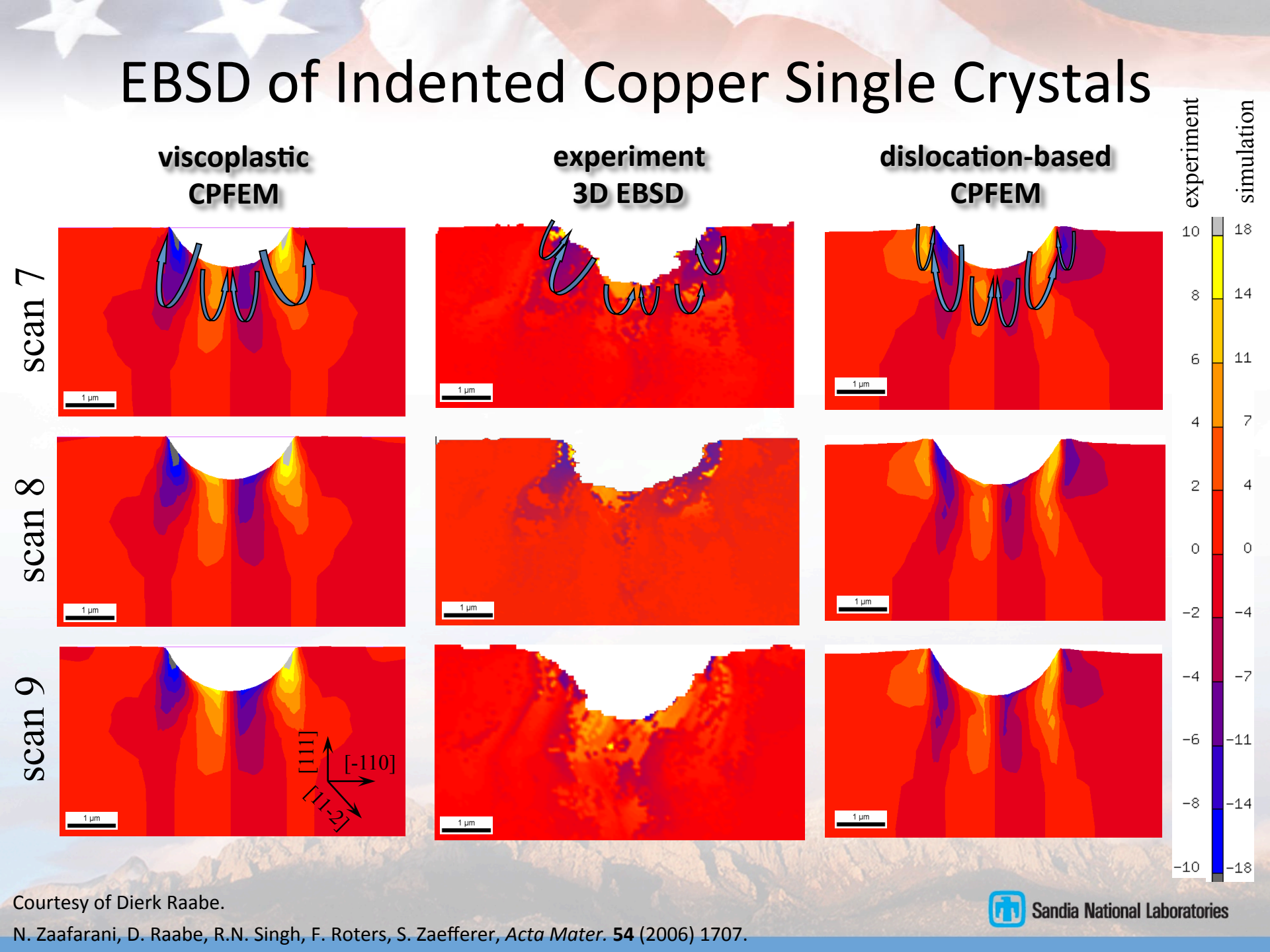
- scan 7
- scan 8
- scan 9

A color scale on the right indicates the orientation angle, ranging from -10 (blue) to 18 (yellow). Each map includes a 1 μm scale bar.

In scan 9, a crystallographic coordinate system is indicated by arrows labeled [111], [-110], and [1-2].

Courtesy of Dierk Raabe.
N. Zaafarani, D. Raabe, R.N. Singh, F. Roters, S. Zaefferer, *Acta Mater.* **54** (2006) 1707.

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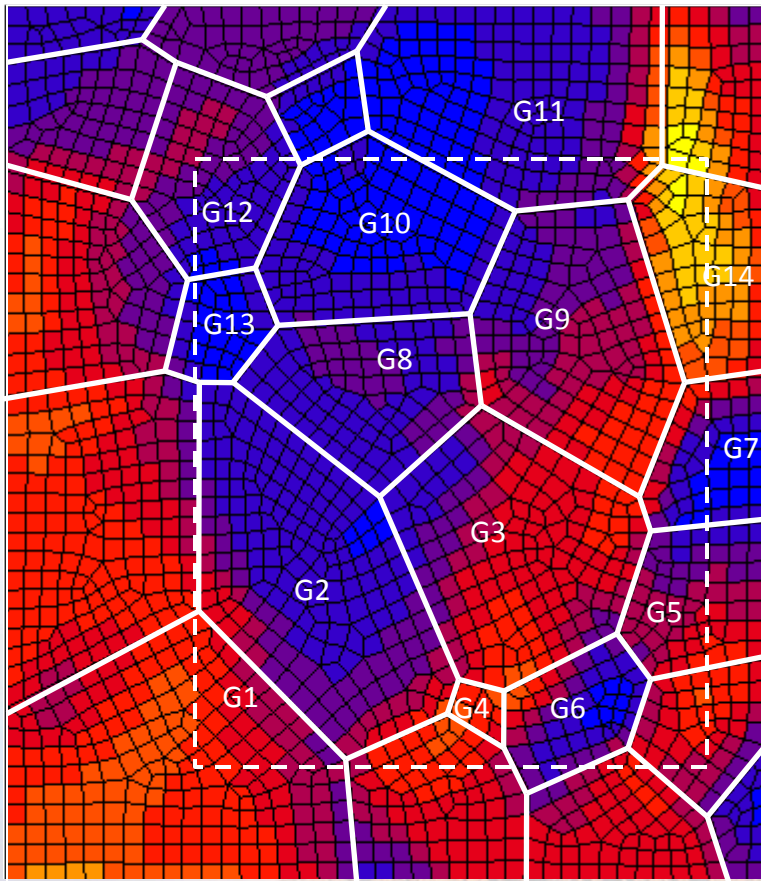
Courtesy of Dierk Raabe.

N. Zaafarani, D. Raabe, R.N. Singh, F. Roters, S. Zaefferer, *Acta Mater.* **54** (2006) 1707.

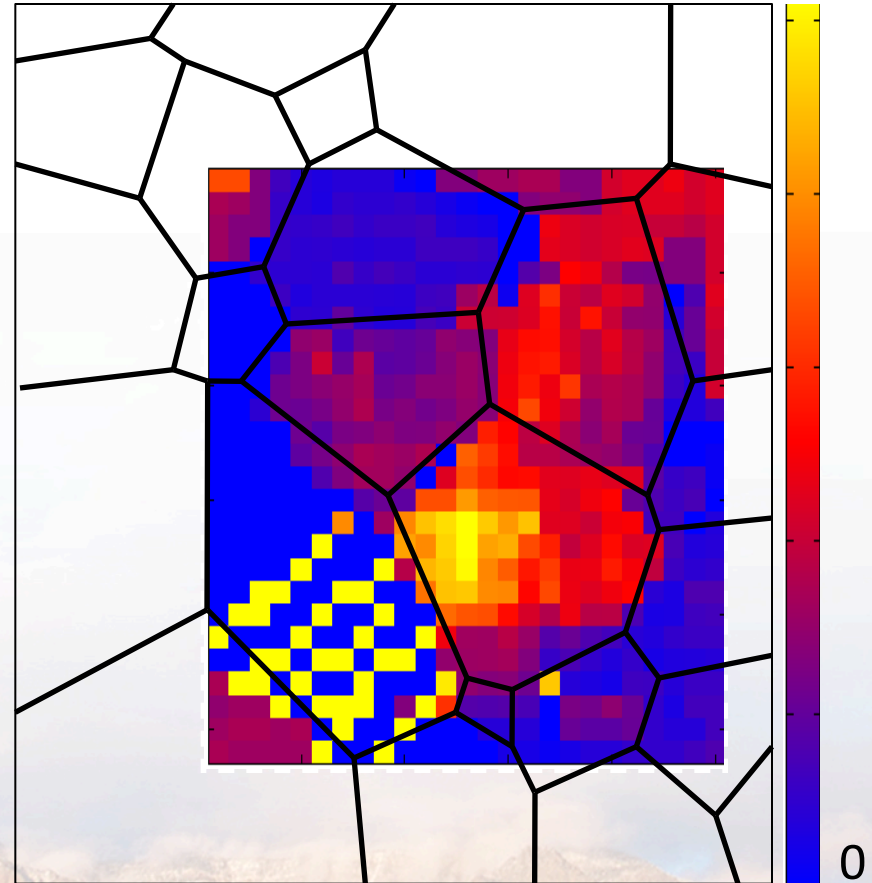
Strain Mapping in a Titanium Polycrystal

Total shear after 1.5% applied strain:

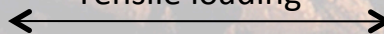
Simulation



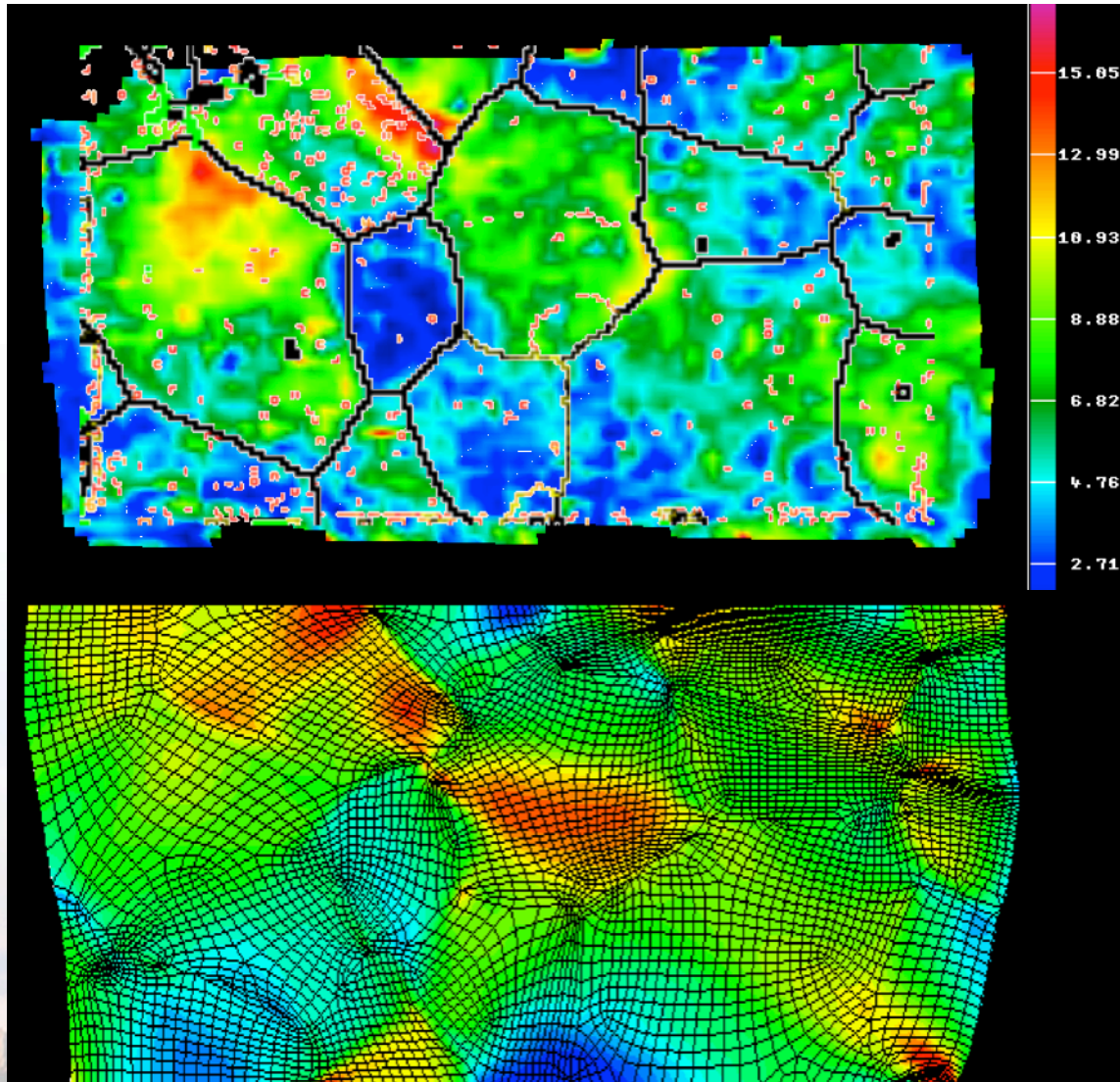
Experiment



Tensile loading



Strains in an Al Oligocrystal



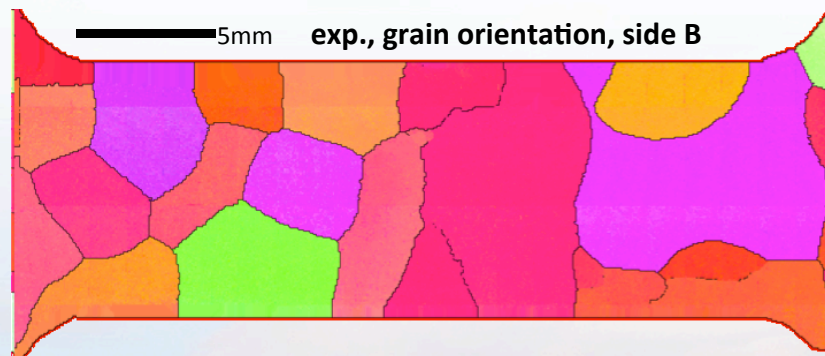
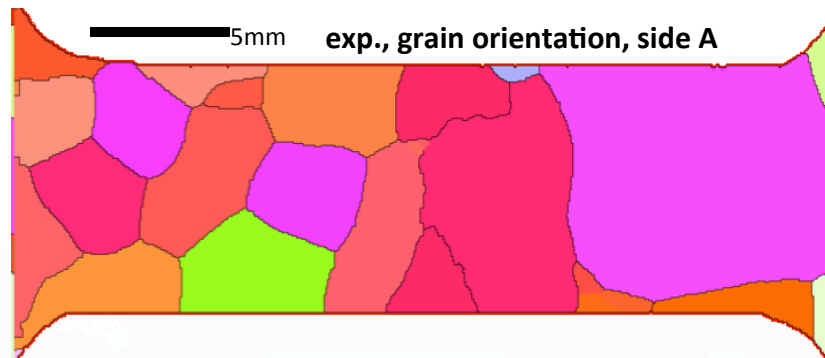
Experiment
(DIC, EBSD)
Mises strain

Simulation
(Viscoplastic
CP-FEM)
Mises strain

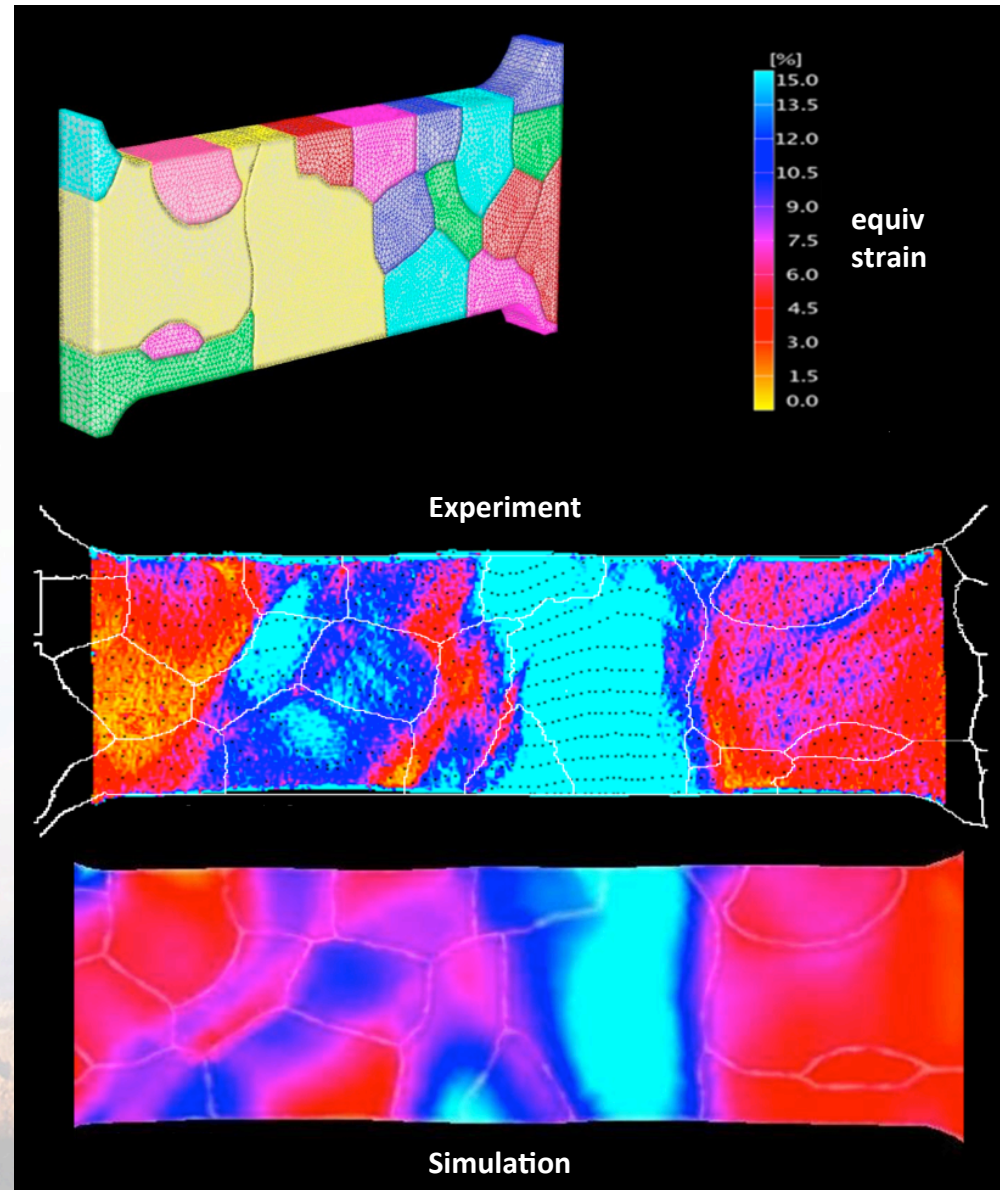
Courtesy of Dierk Raabe.

D. Ma, F. Roters, D. Raabe, *Acta Mater.* **54** (2006) 2169 and 2181.

Strains in another Al Oligocrystal



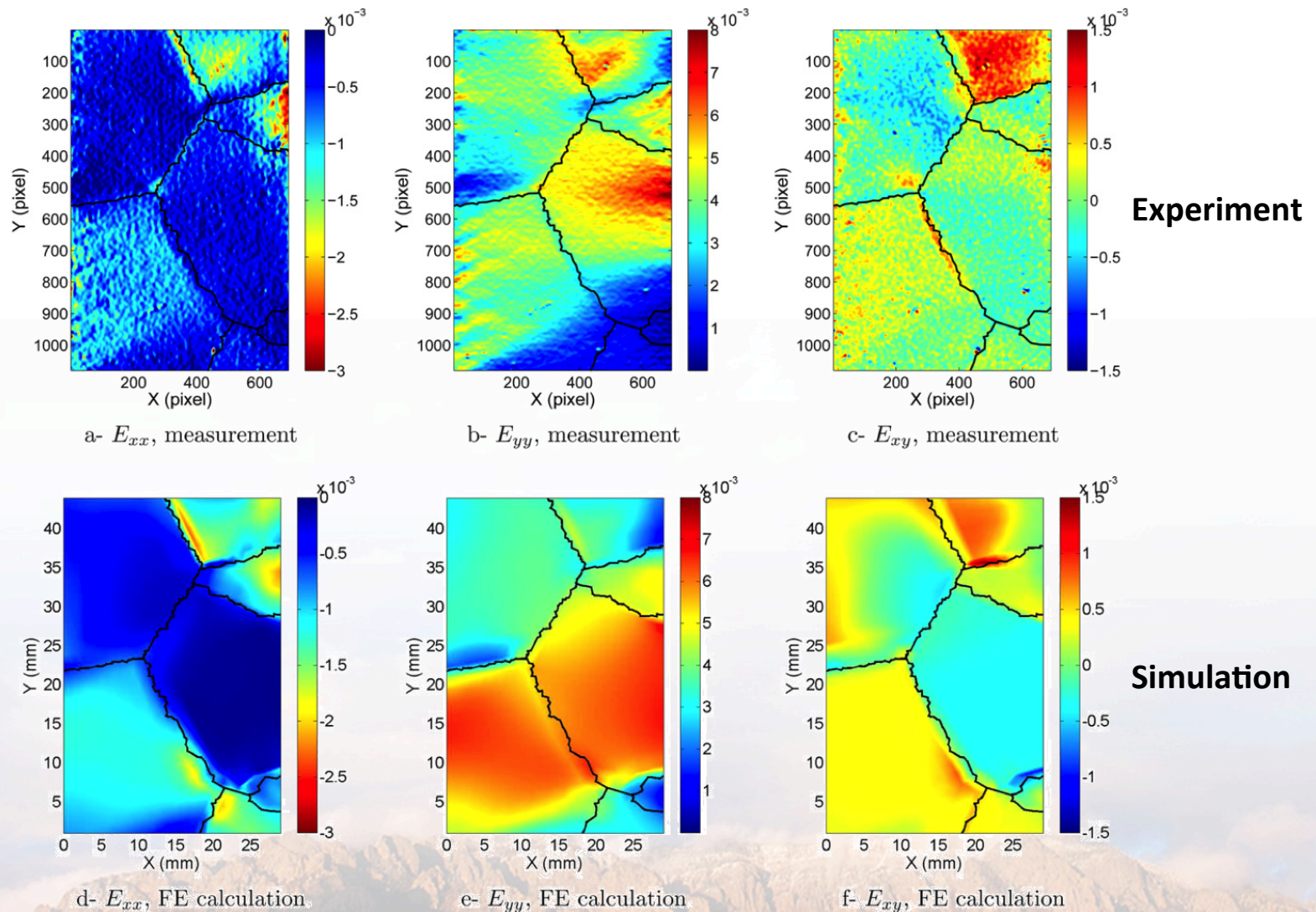
21mm X 8mm X 1mm



Courtesy of Dierk Raabe.

Z. Zhao, M. Ramesh, D. Raabe, A.M. Cuitino, R. Radovitzky, *Int. J. Plast.* **24** (2008) 2278.

Strains in yet another Al Oligocrystal

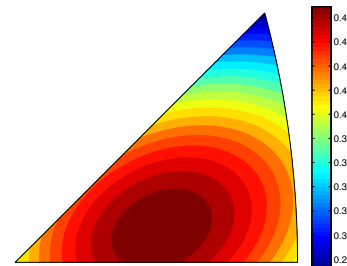
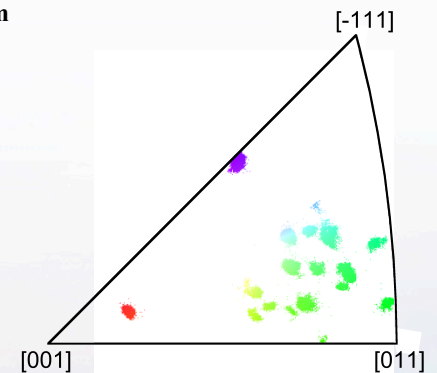
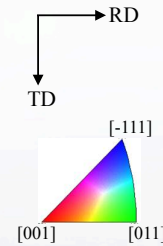
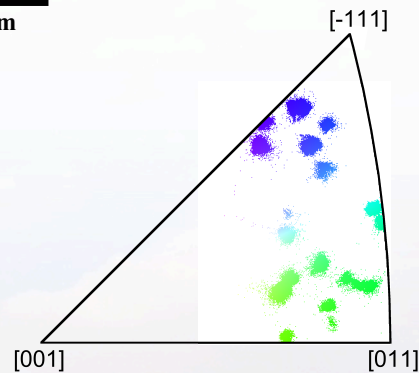
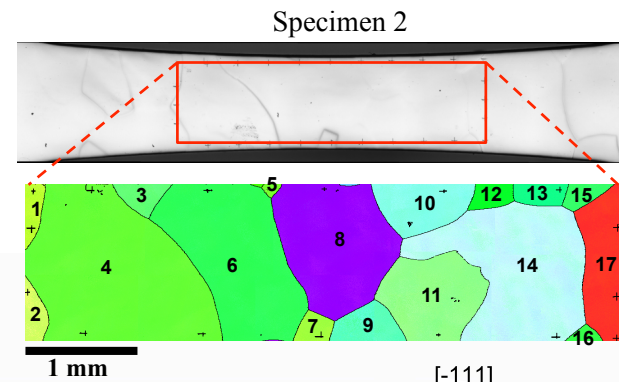
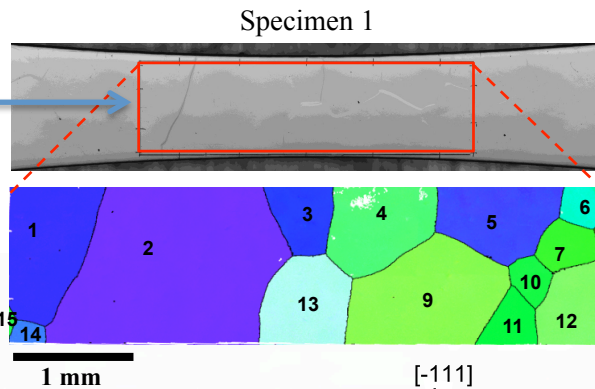




Validation of BCC Tantalum CPFEM Models

Oligocrystals

Area
Of
Interest



BCC CPFEM Formulation

Slip rate: $\dot{\gamma}^\alpha = \dot{\gamma}_0^\alpha \left(\frac{\tau^\alpha}{g^\alpha} \right)^{1/m}$ (Hutchinson, 1976)

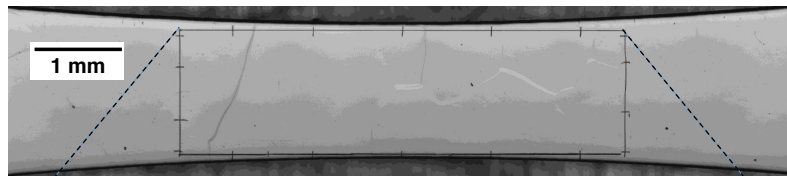
Slip resistance: $g^\alpha = \max(\tau_{\text{cr}}^\alpha - \tau_{\text{ns}}^\alpha, 0) + \tau_{\text{obs}}^\alpha$ (Weinberger, 2012)

└──────────┬───────────> Obstacle stress
 └──────────> Lattice friction

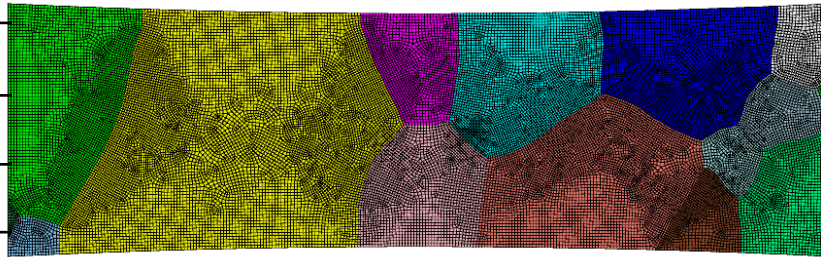
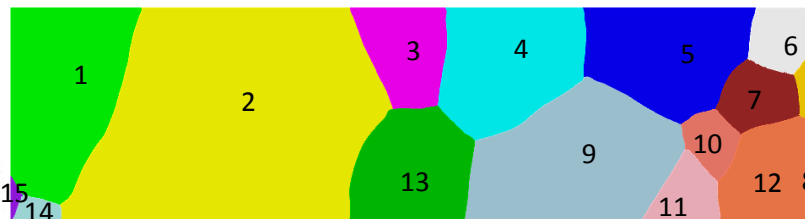
Obstacle stress: $\tau_{\text{obs}}^\alpha = \alpha \mu b \sqrt{\sum_{\beta=1}^{NS} \rho^\beta}$ (Taylor, 1934)

$$\rho^\alpha = \left(\kappa_1 \sqrt{\sum_{\beta=1}^{NS} \rho^\beta} - \kappa_2 \rho^\alpha \right) \cdot |\dot{\gamma}^\alpha| \quad (\text{Kocks, 1976})$$

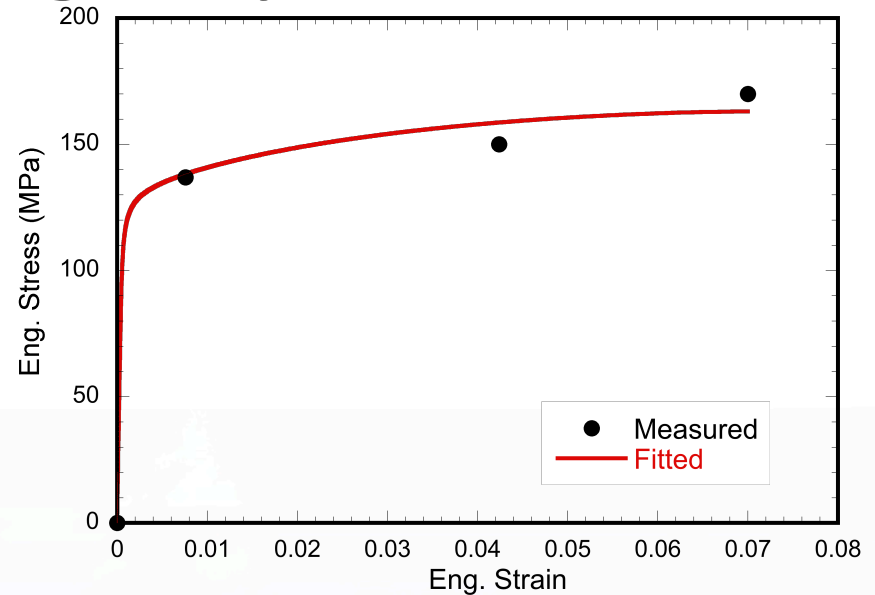
Tantalum Oligocrystal 1



Specimen



Finite Element Mesh



$$\bar{\tau} = 28 \text{ MPa}$$

Elastic constants (GPa)

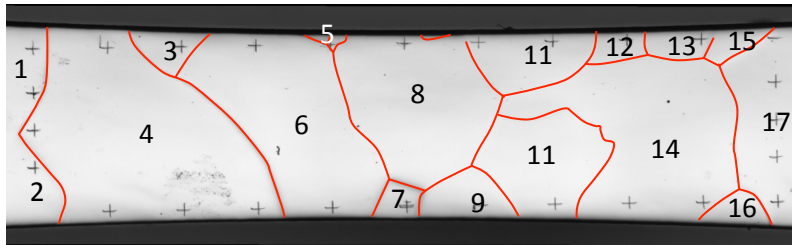
C_{11}	C_{12}	C_{44}
267	161	82.5

Hardening coefficients

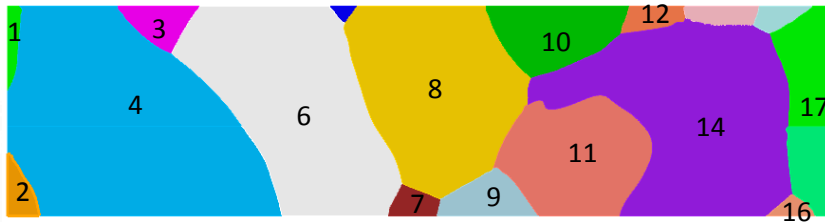
$k_1 \text{ (m}^{-1}\text{)}$	k_2
1.4×10^6	14



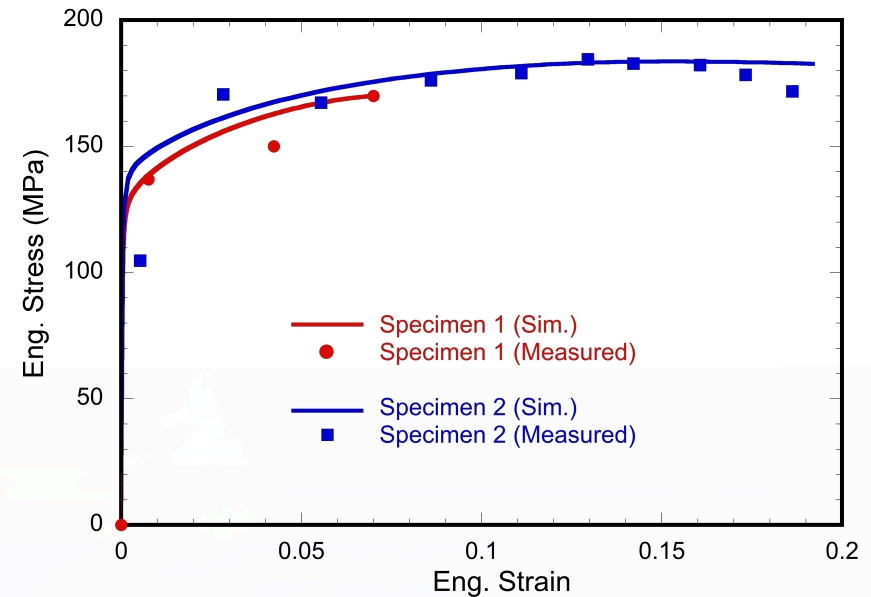
Tantalum Oligocrystal 2



Specimen



Finite Element Mesh



$$\bar{\tau} = 37 \text{ MPa}$$

Elastic constants (GPa)

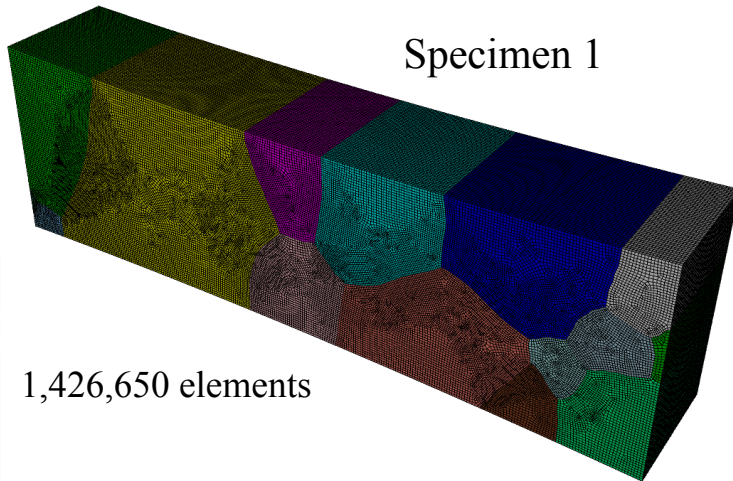
Hardening coefficients

C_{11}	C_{12}	C_{44}	$k_1 \text{ (m}^{-1}\text{)}$	k_2
267	161	82.5	1.4×10^6	14

(Same as for Oligocrystal #1)

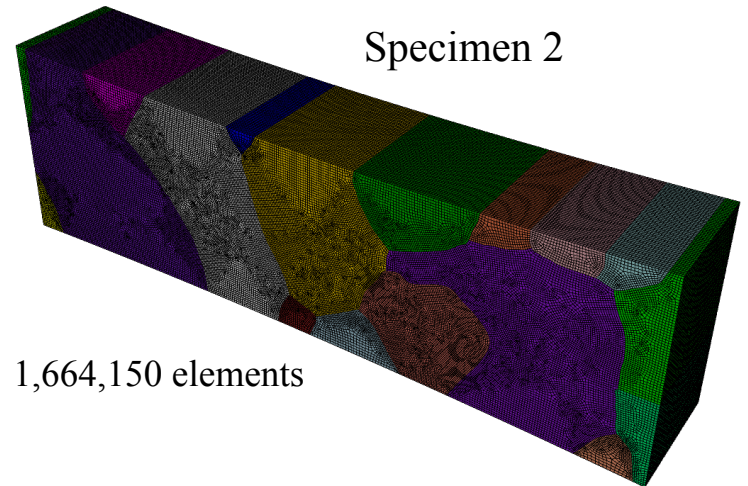
Columnarity

Specimen 1

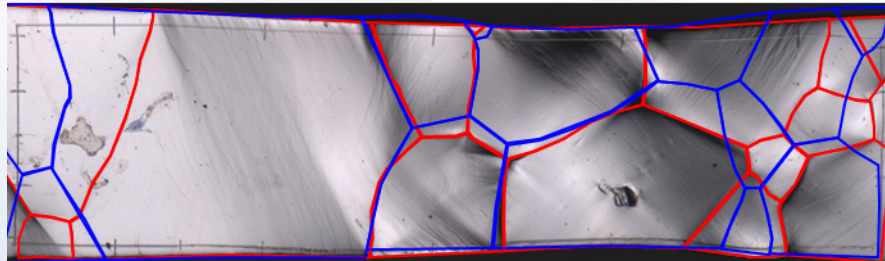


1,426,650 elements

Specimen 2



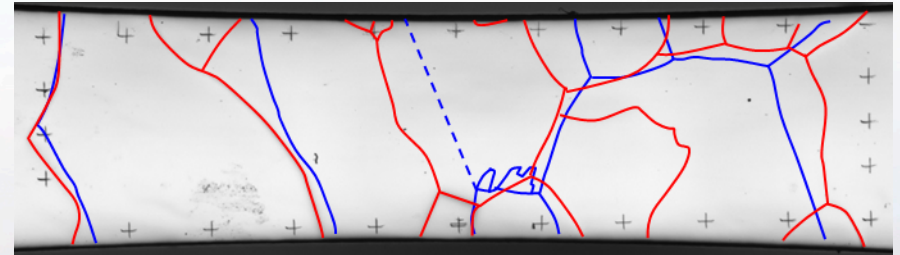
1,664,150 elements



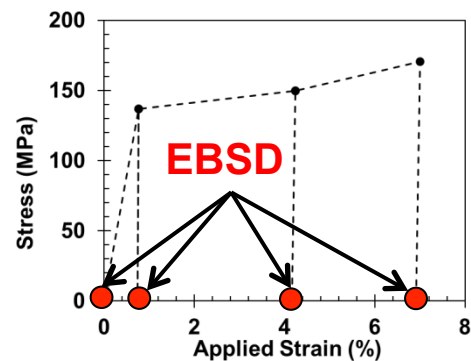
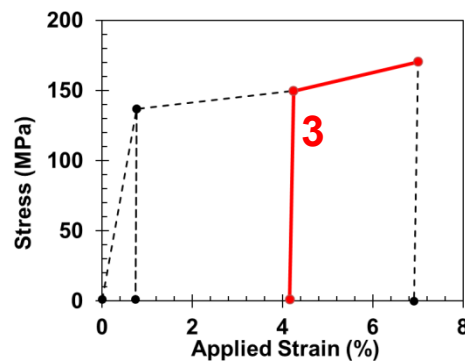
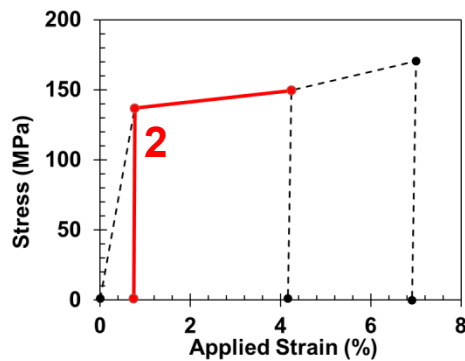
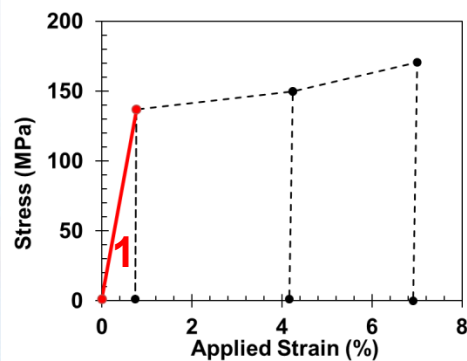
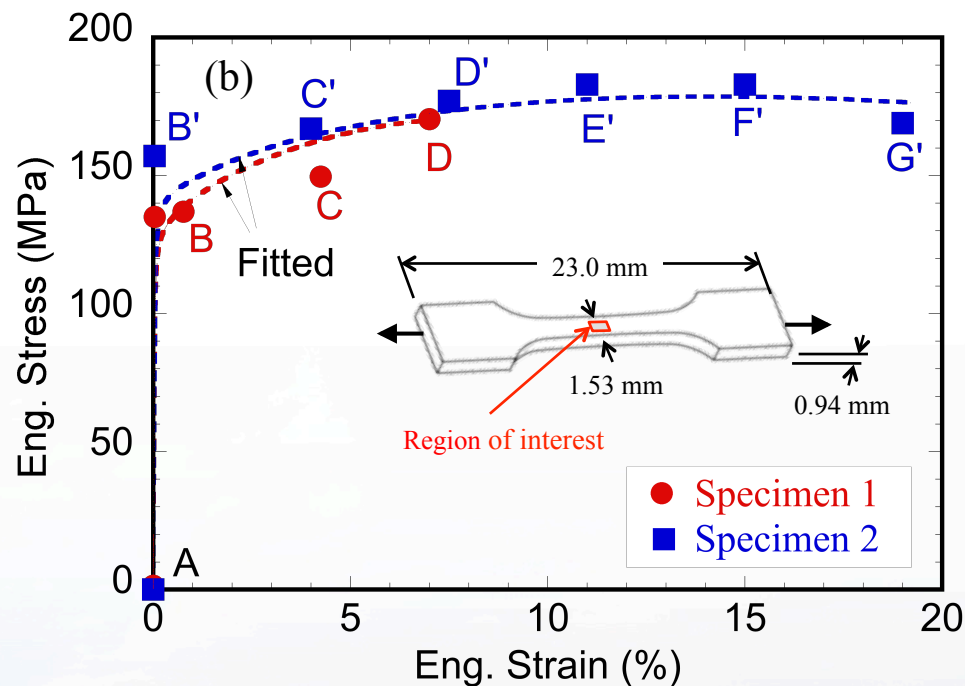
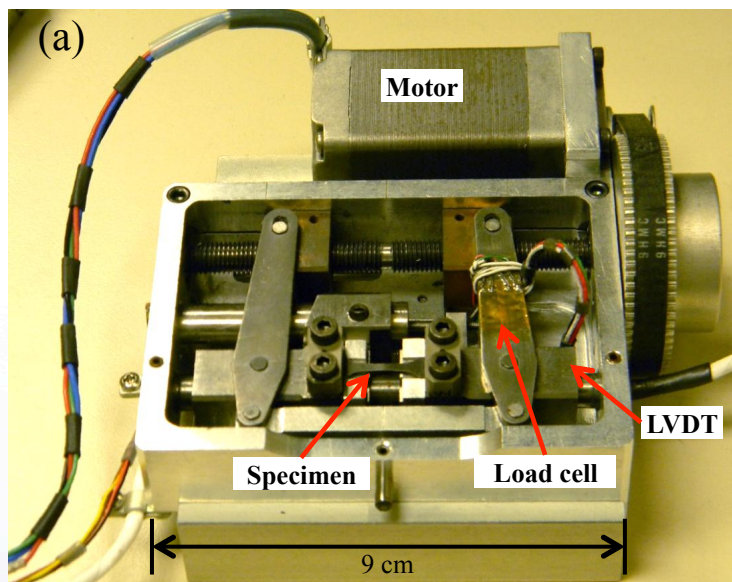
Grain boundary (Front)

1 mm

Grain boundary (Back)

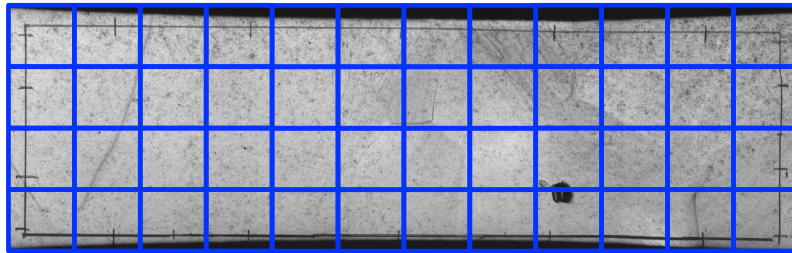


Loading of Specimens and EBSD



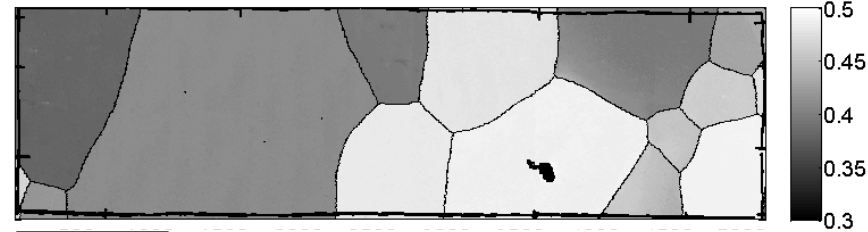
Digital Image Correlation

Undeformed
1.5 mm



5 mm

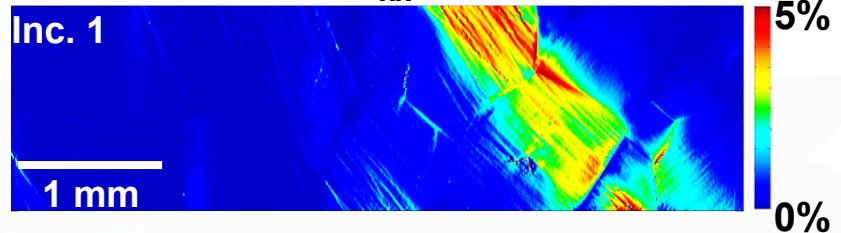
41 MPix (11,500 x 3,600)



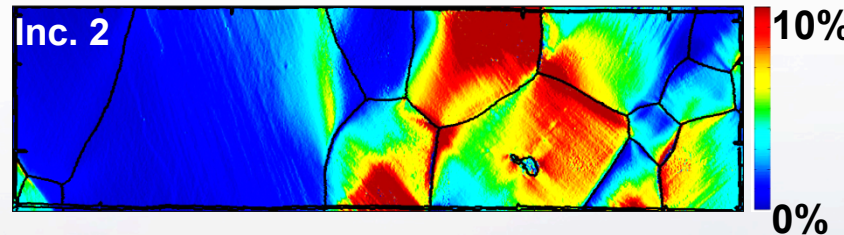
1 mm

ϵ_{xx}

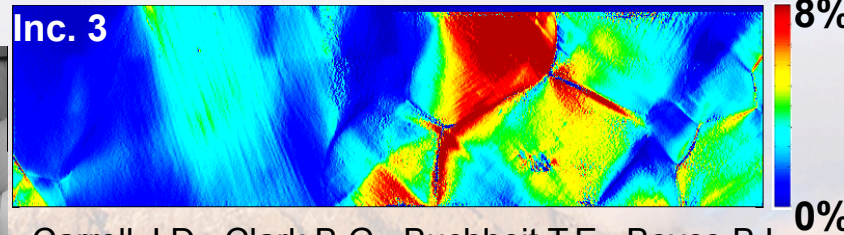
Inc. 1



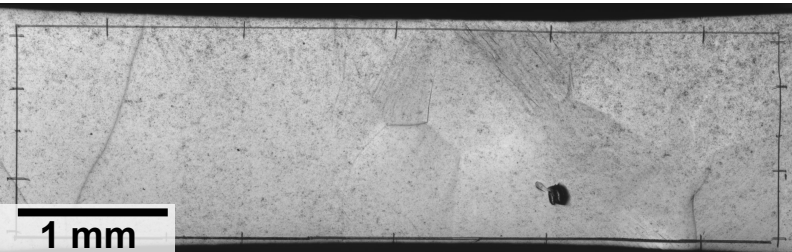
Inc. 2



Inc. 3

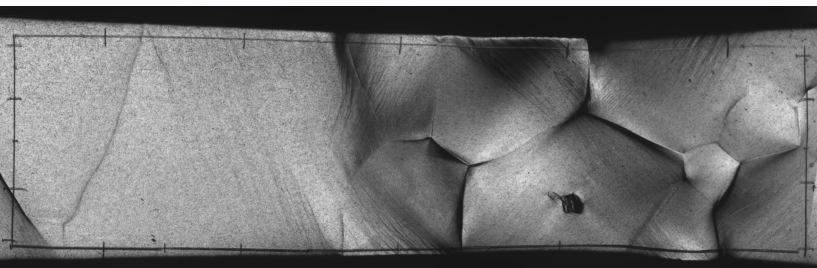


Strain 1:
0.75% Inc.
0.75% Total

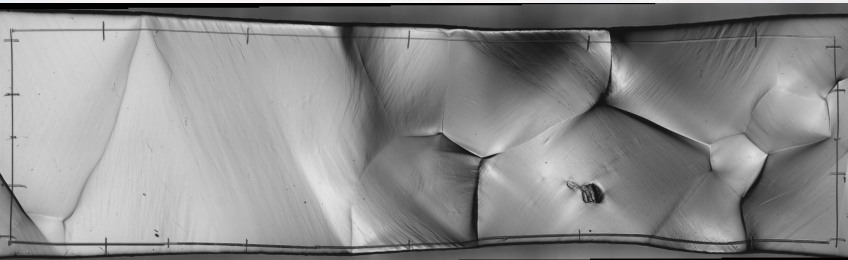


1 mm

Strain 2:
3.4% Inc.
4.2% Total



Strain 3:
2.6% Inc.
6.8% Total

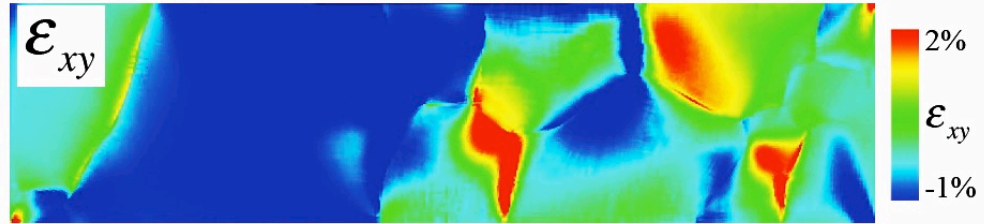
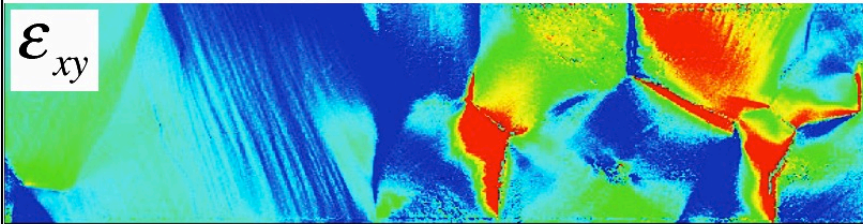
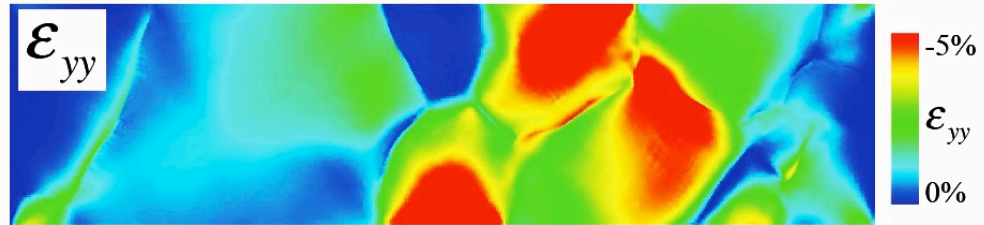
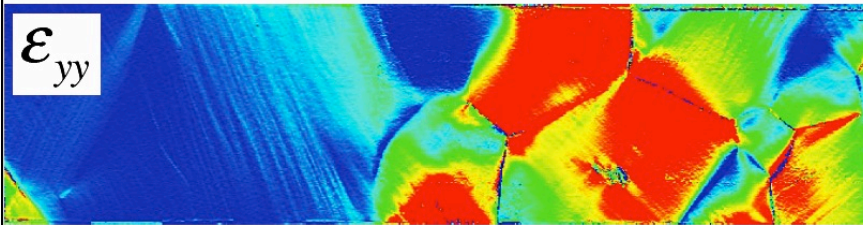
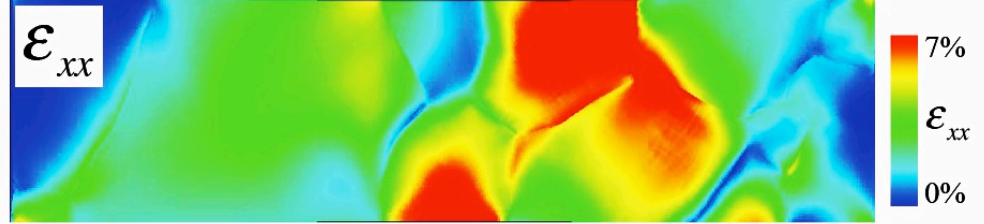
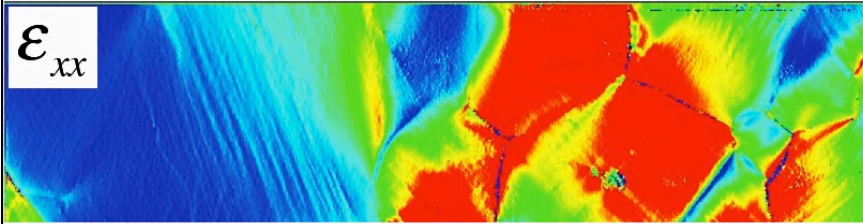


Carroll J.D., Clark B.G., Buchheit T.E., Boyce B.L.,
Weinberger C.R., Mater. Sci. Eng. A (2013)

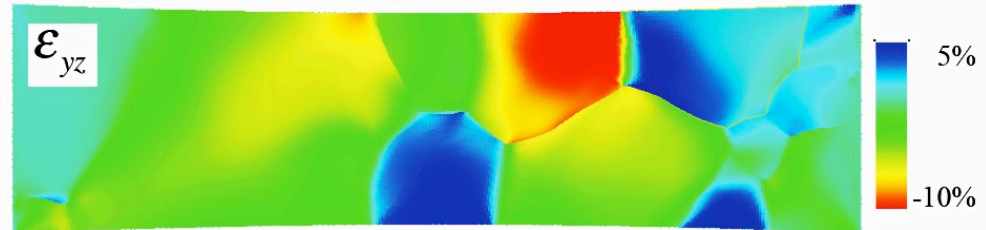
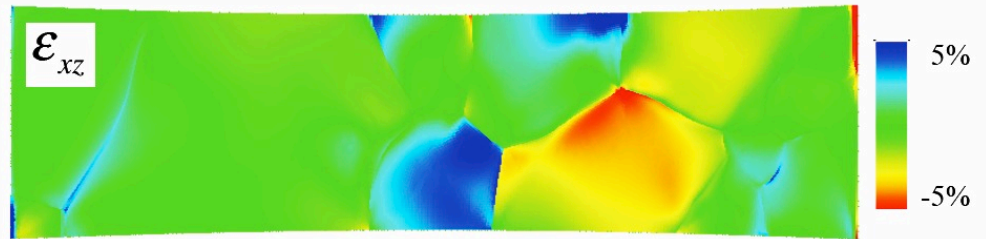
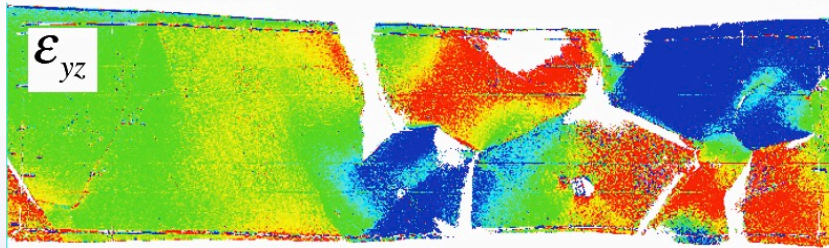
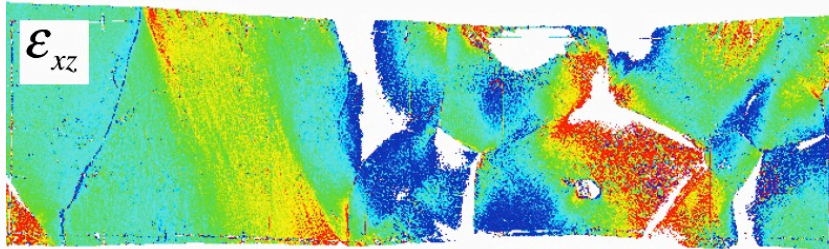


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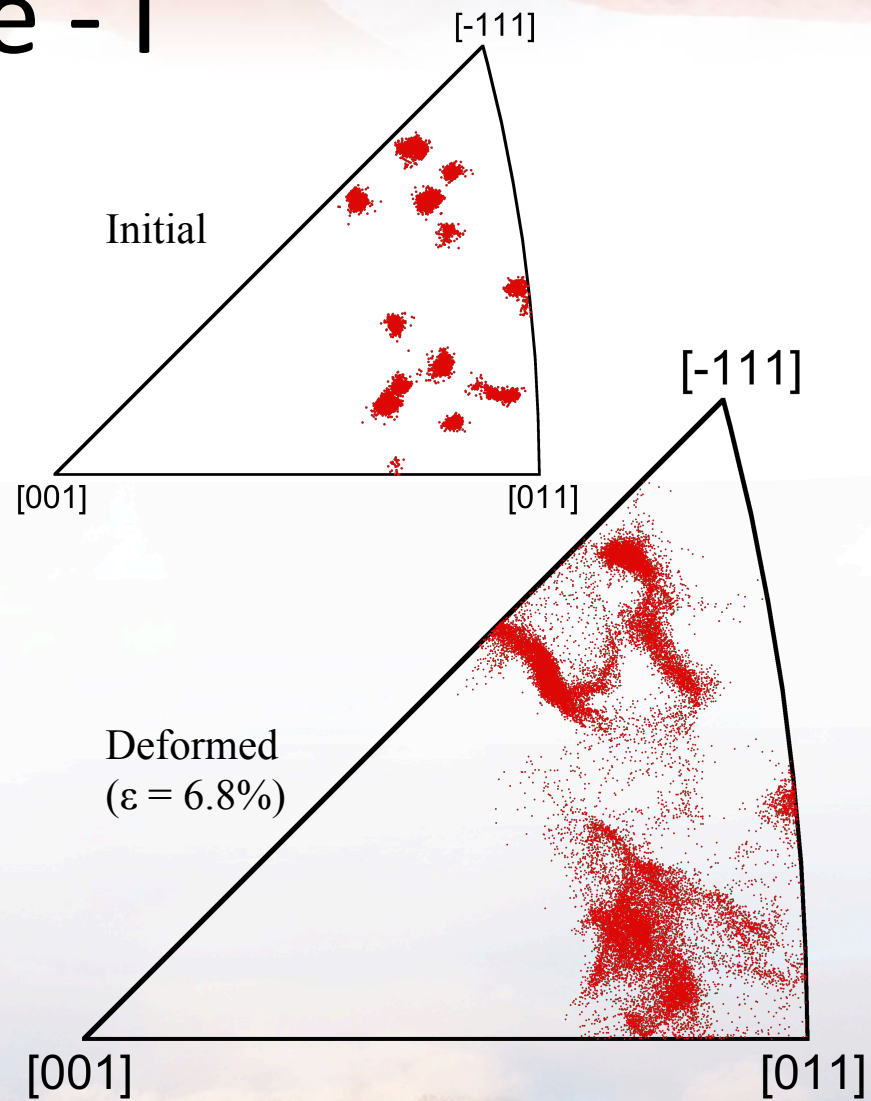
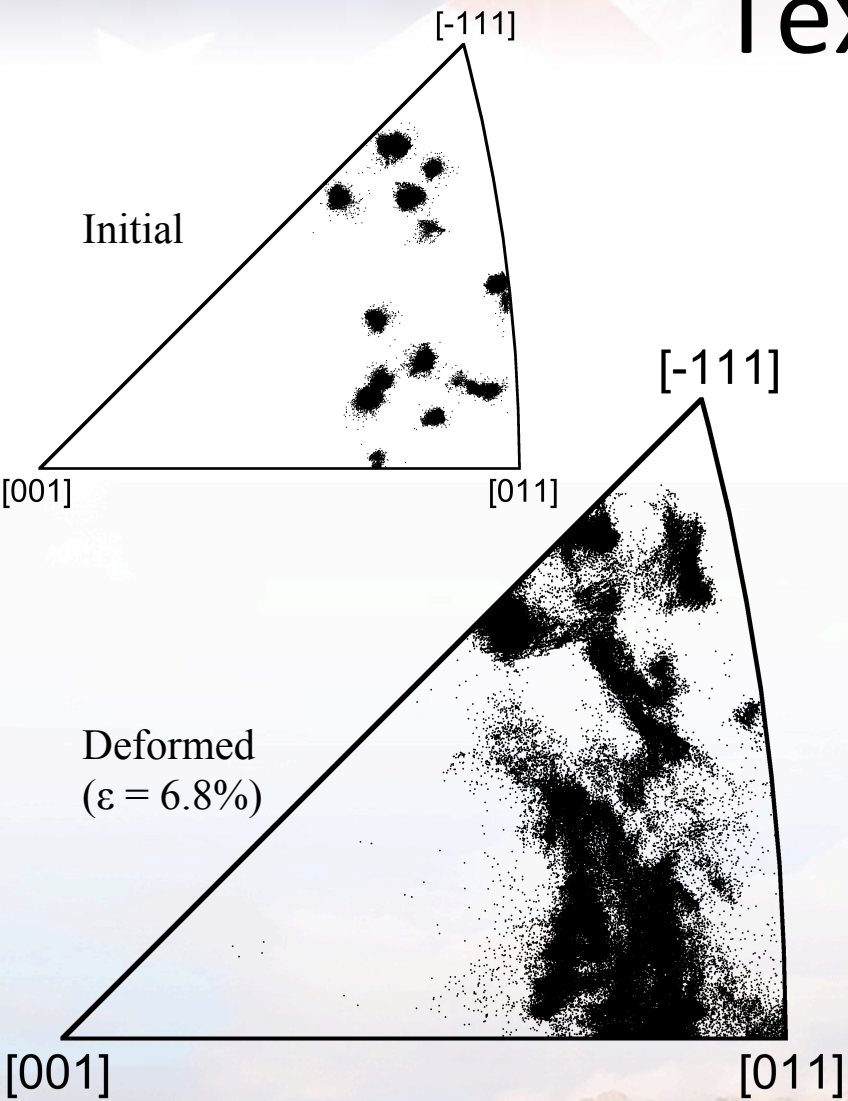
Predicted In-Plane Strains



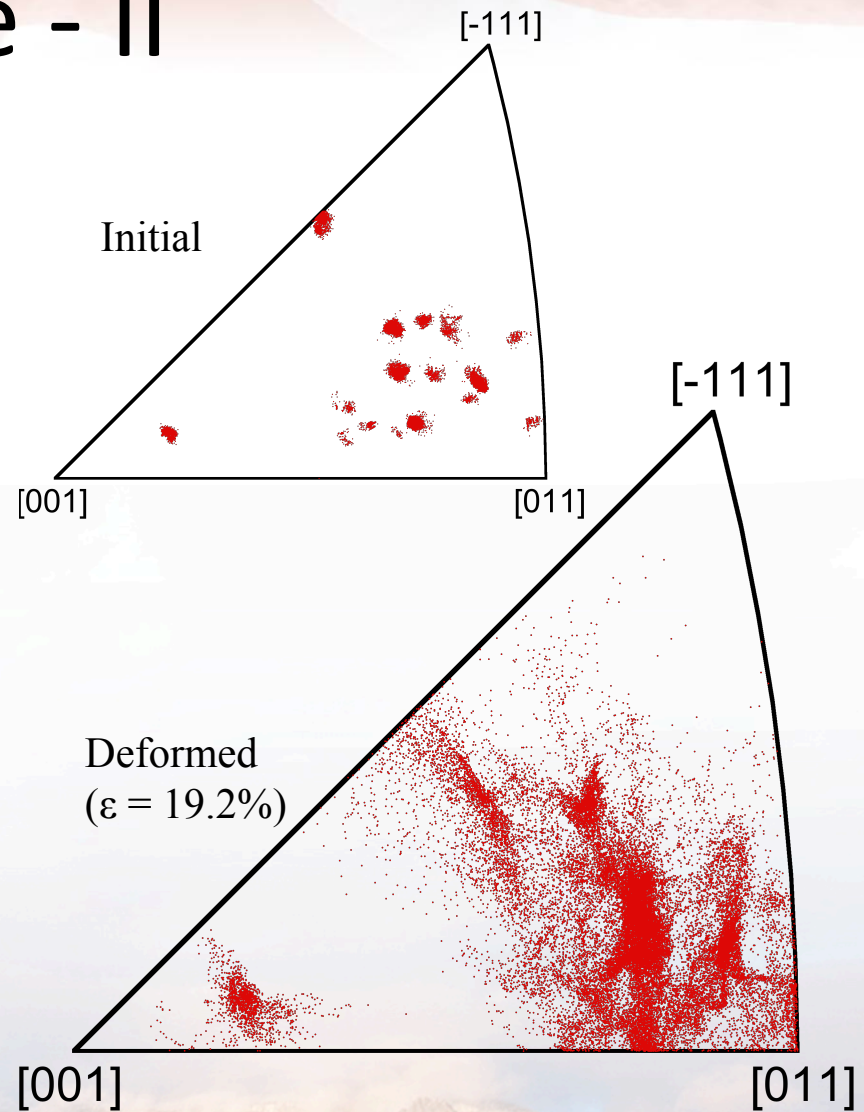
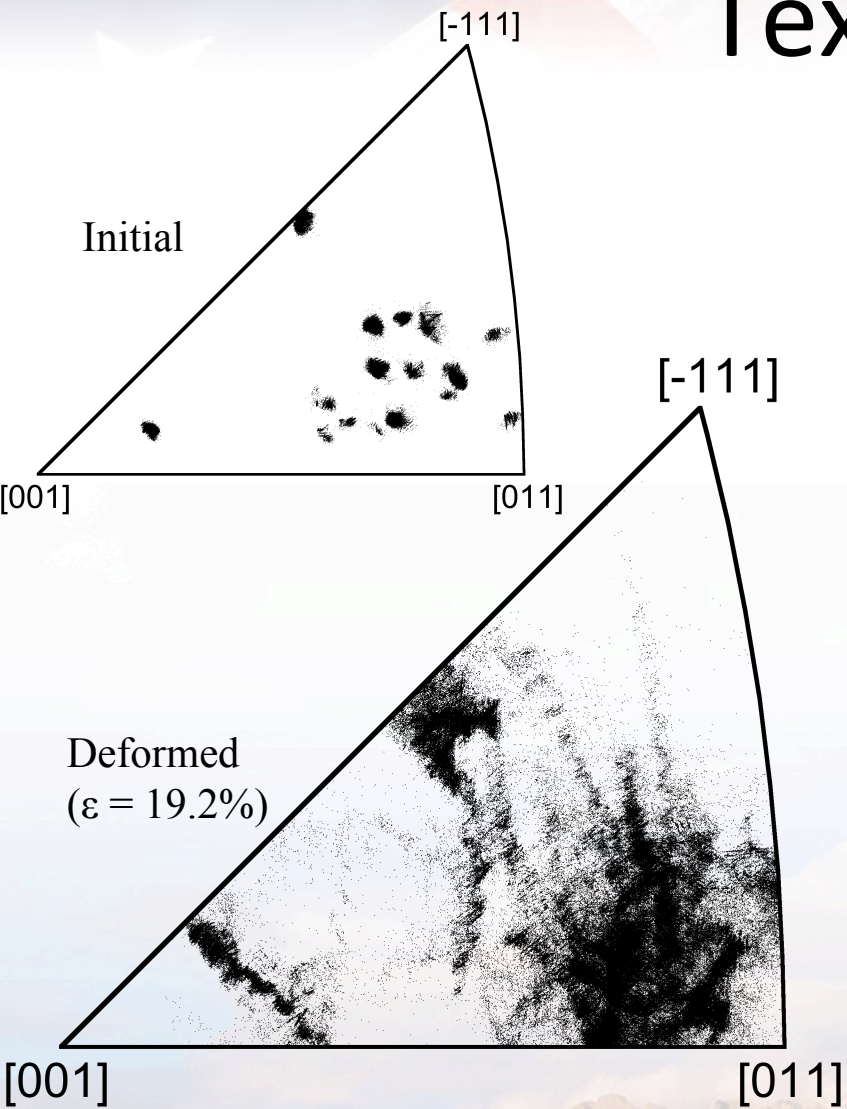
Out-Of-Plane Strains



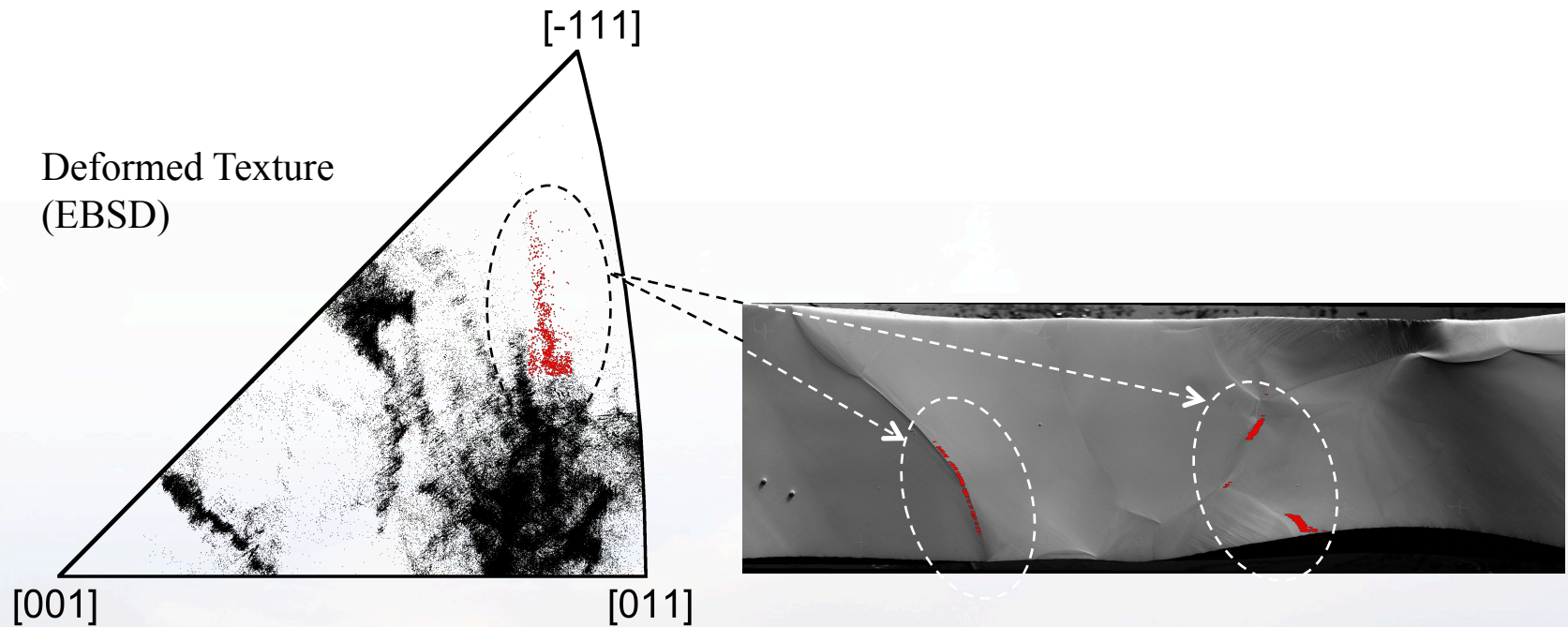
Texture - I



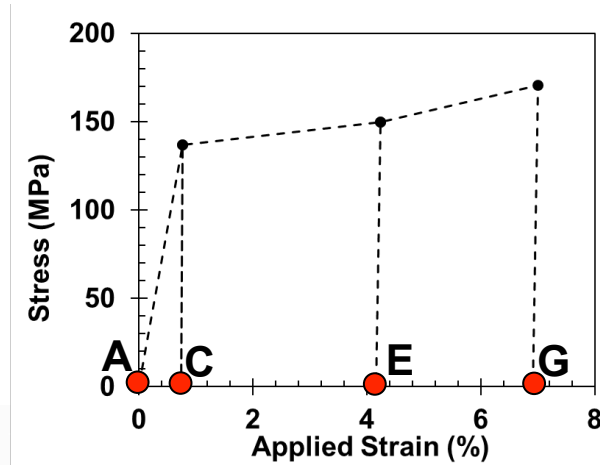
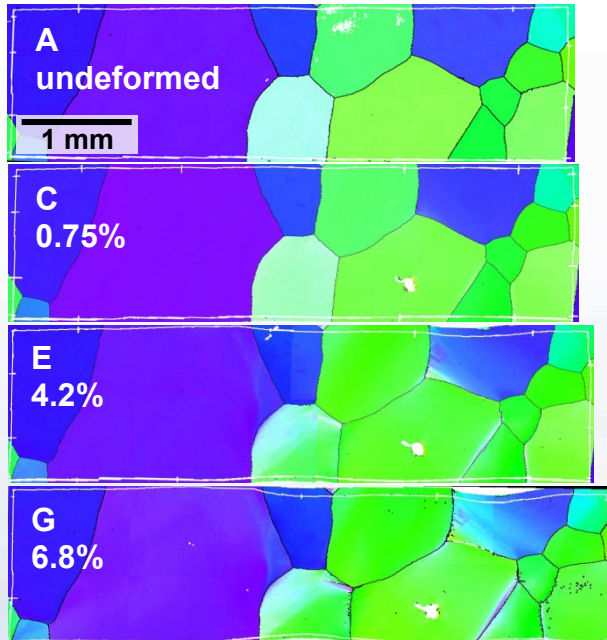
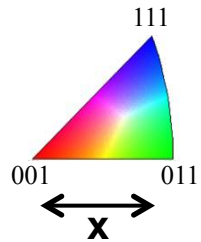
Texture - II



Texture Issues Occur Near GBs

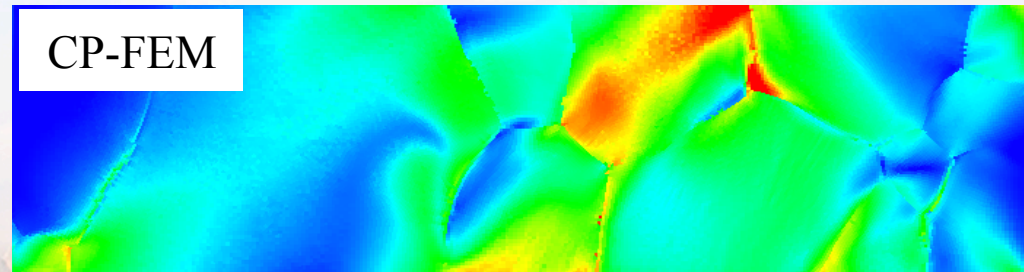
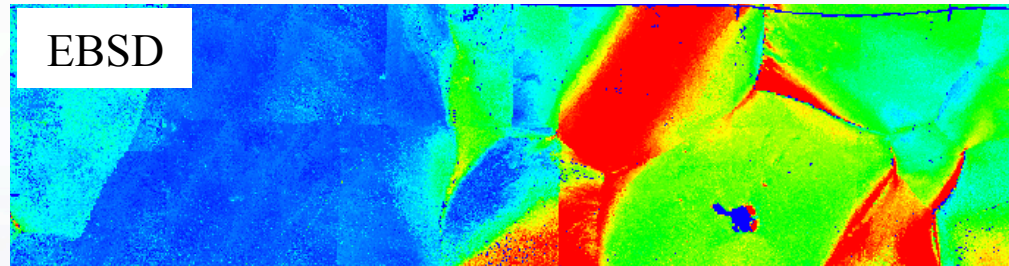


Misorientation

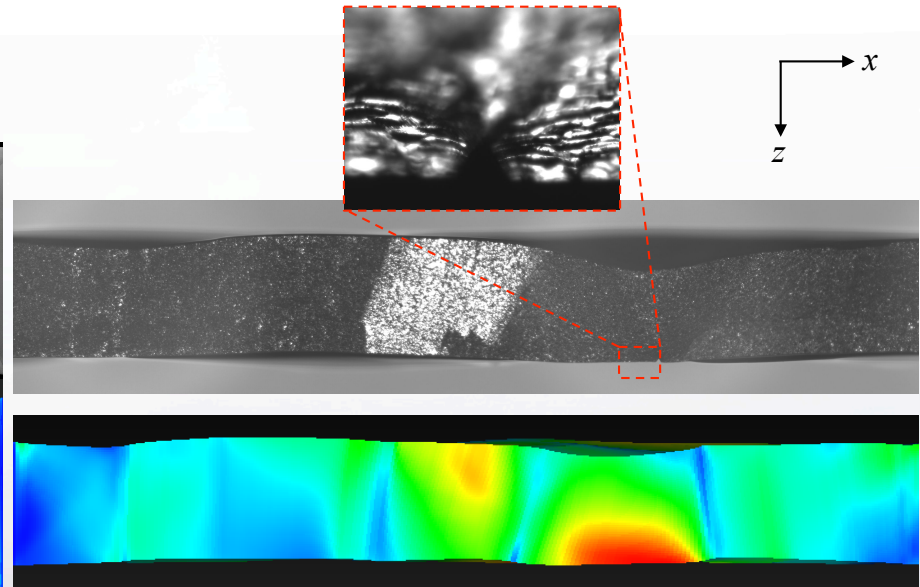
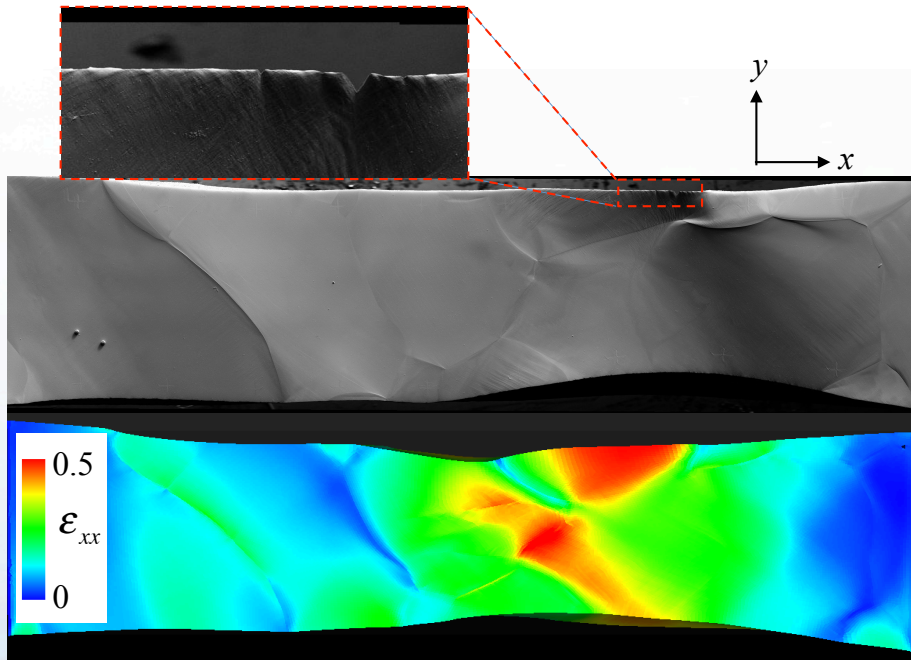
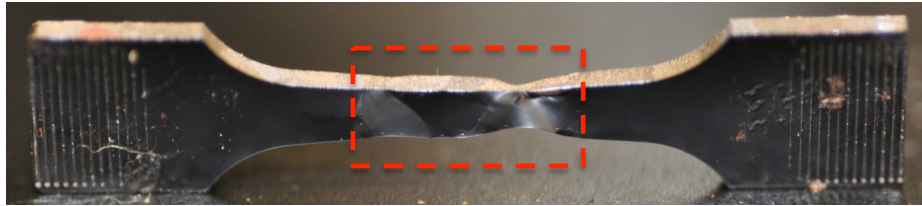


$$\Delta g_{AB} = g_B g_A^{-1}$$

$$\cos \theta = \frac{1}{2} (\Delta g_{11} + \Delta g_{22} + \Delta g_{33} - 1)$$



Failure Predictions

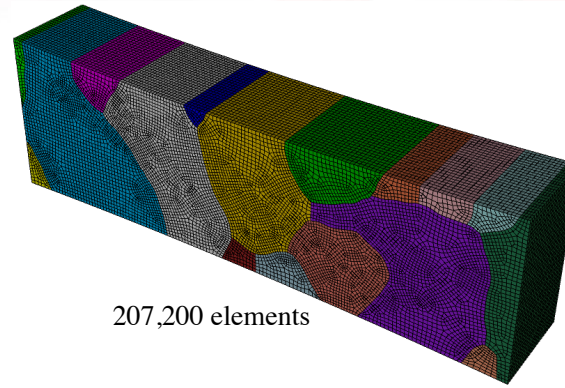
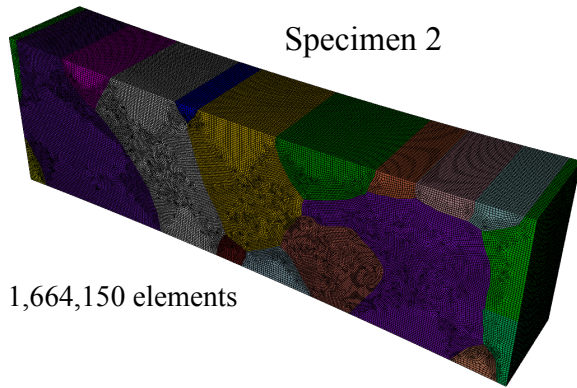


The background of the slide features a faded American flag in the upper left corner and a mountain range under a cloudy sky in the lower half.

Model Sensitivity

- Sensitivity of Mesh
- Sensitivity of Orientation Choices
- Sensitivity of Slip plane Choice

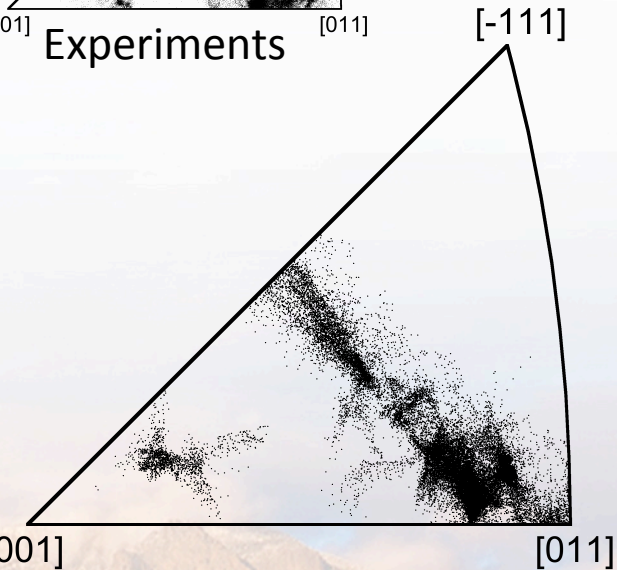
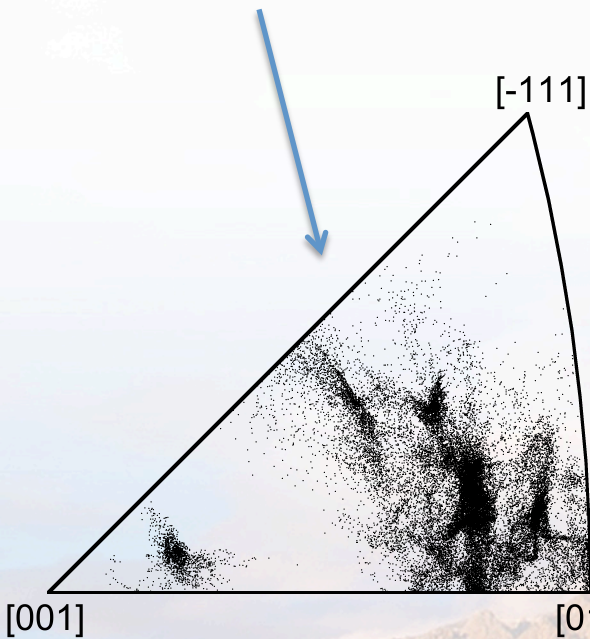
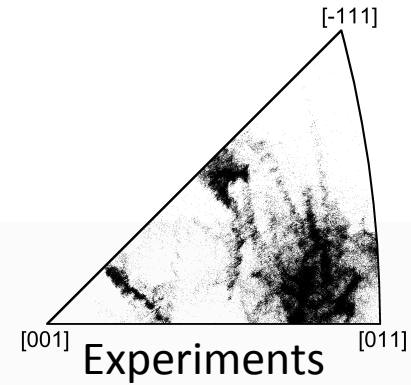
Mesh



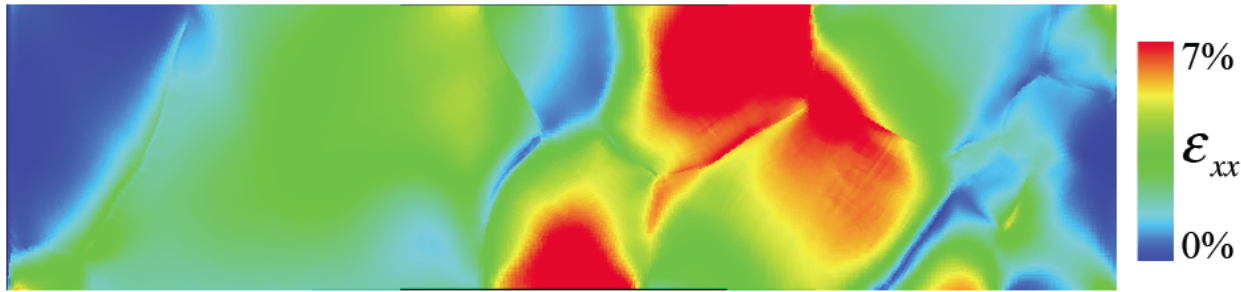
Medium Mesh

Coarse Mesh

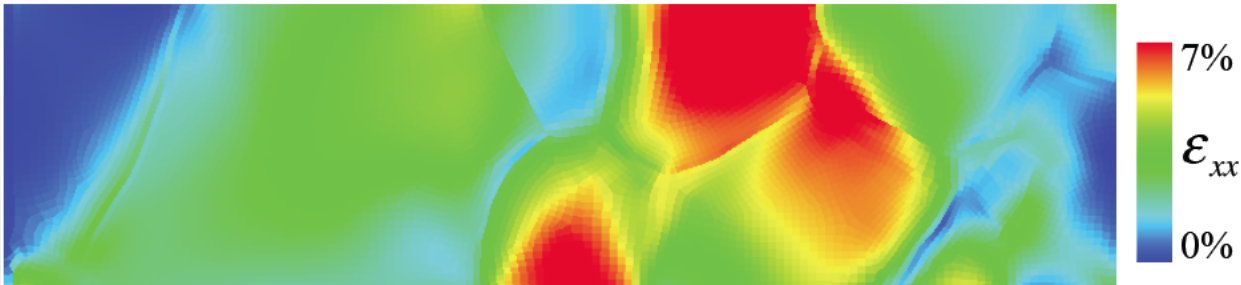
Fine Mesh
(Running)



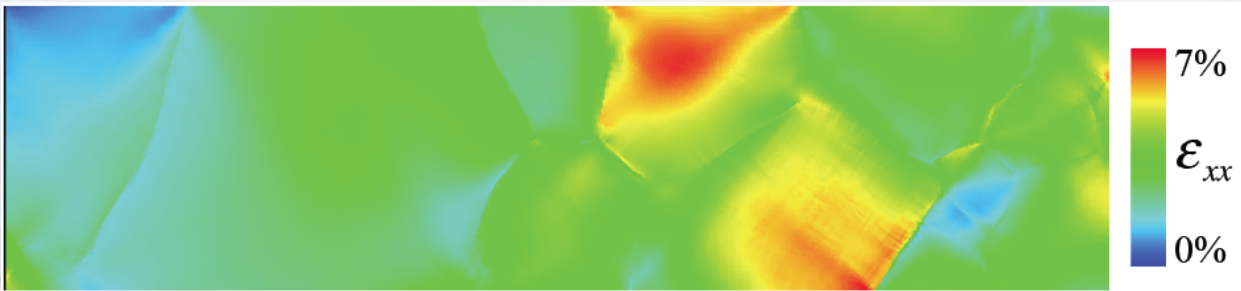
Mesh - Strain Predictions



Medium Mesh

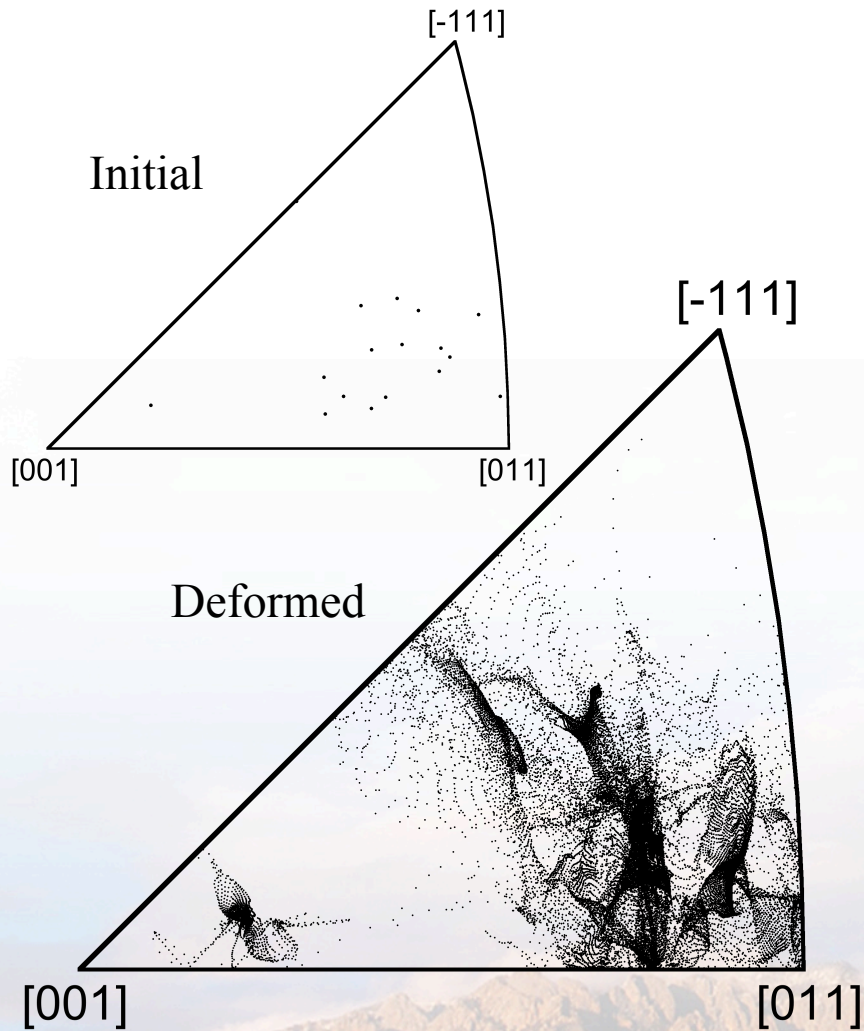
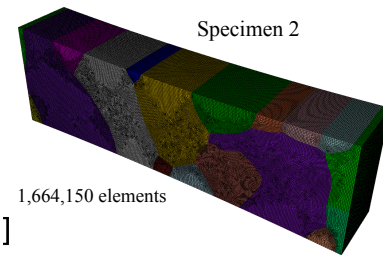


Coarse Mesh

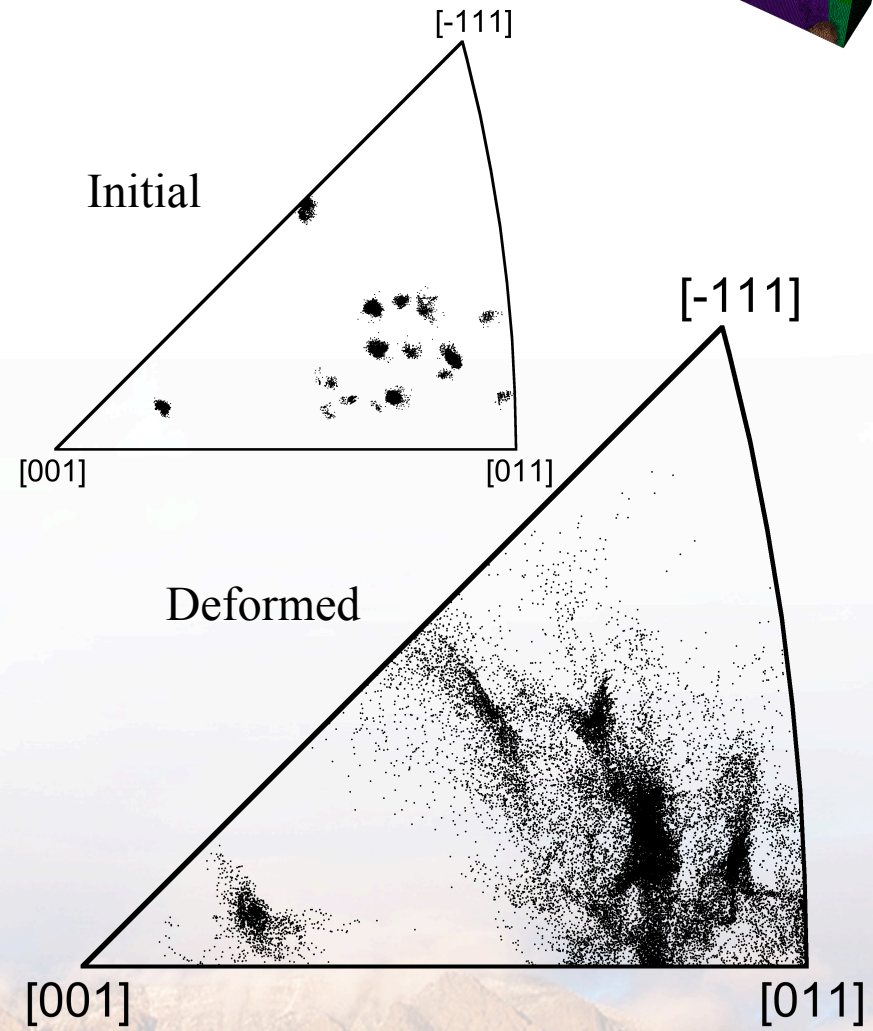


Medium Mesh
1 element through
thickness

Initial Orientations

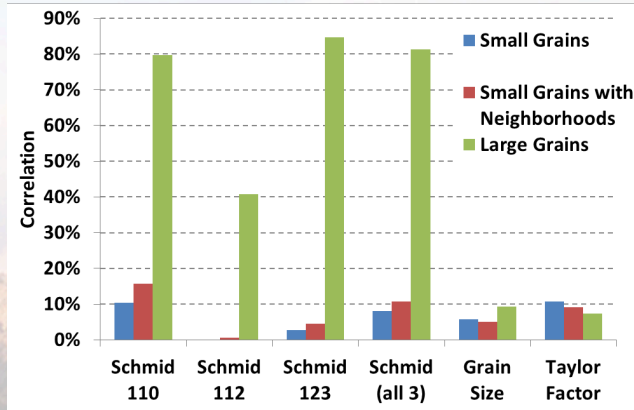
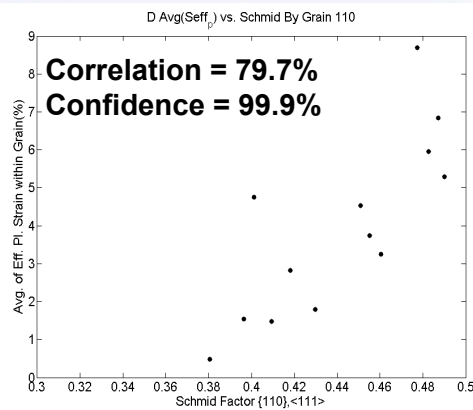
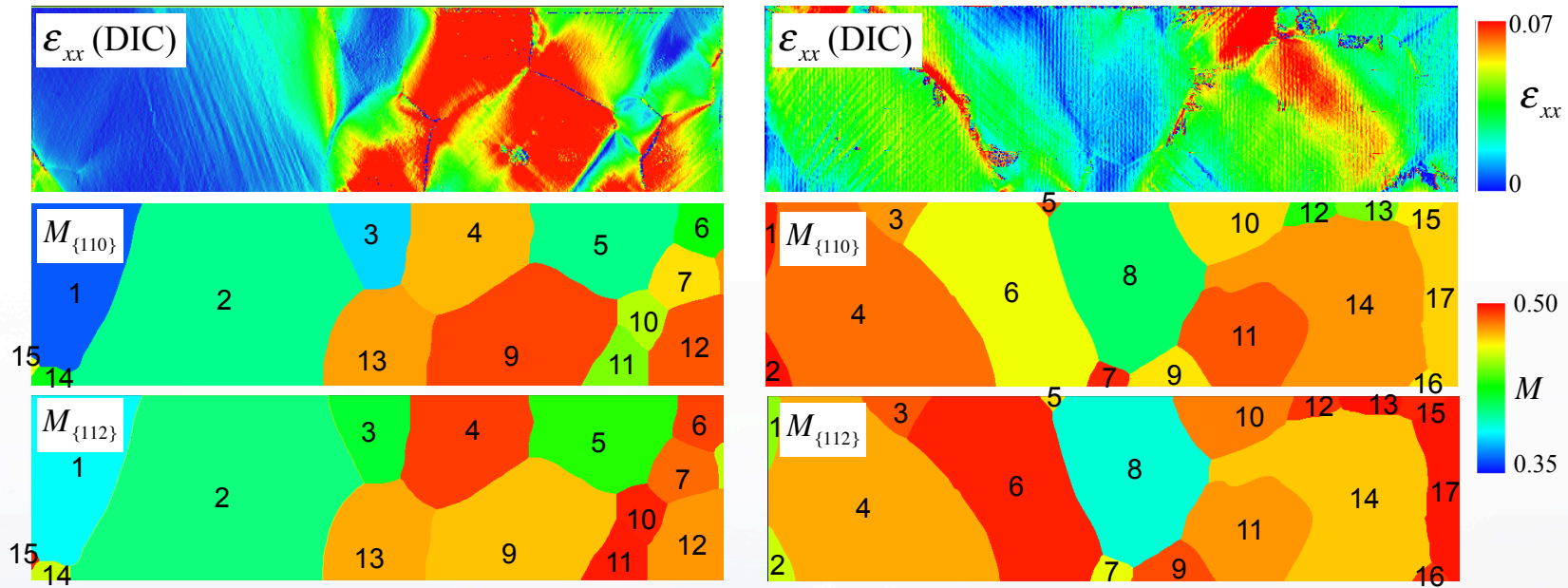


(a) Orientation per grain



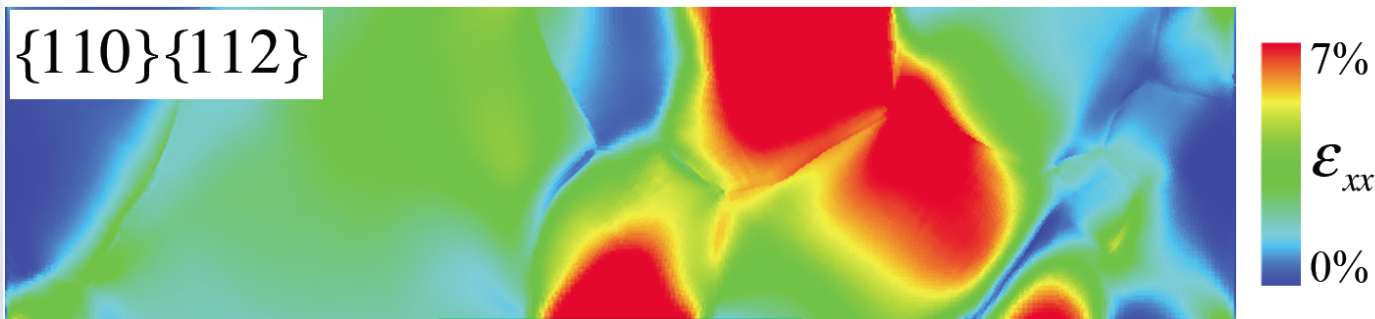
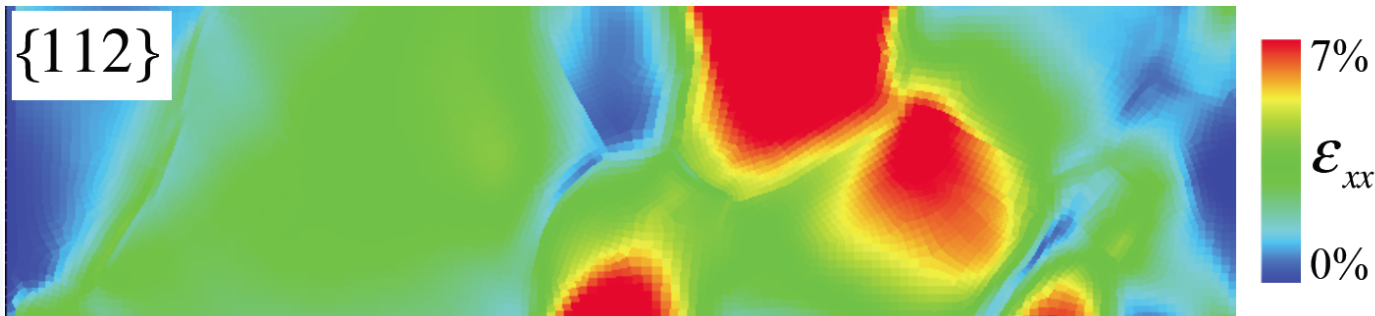
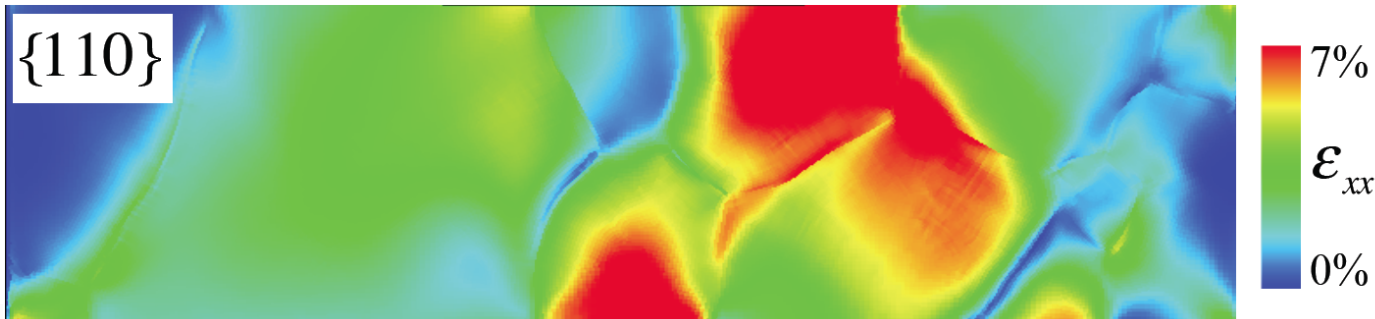
(b) Orientation per element

Slip Planes

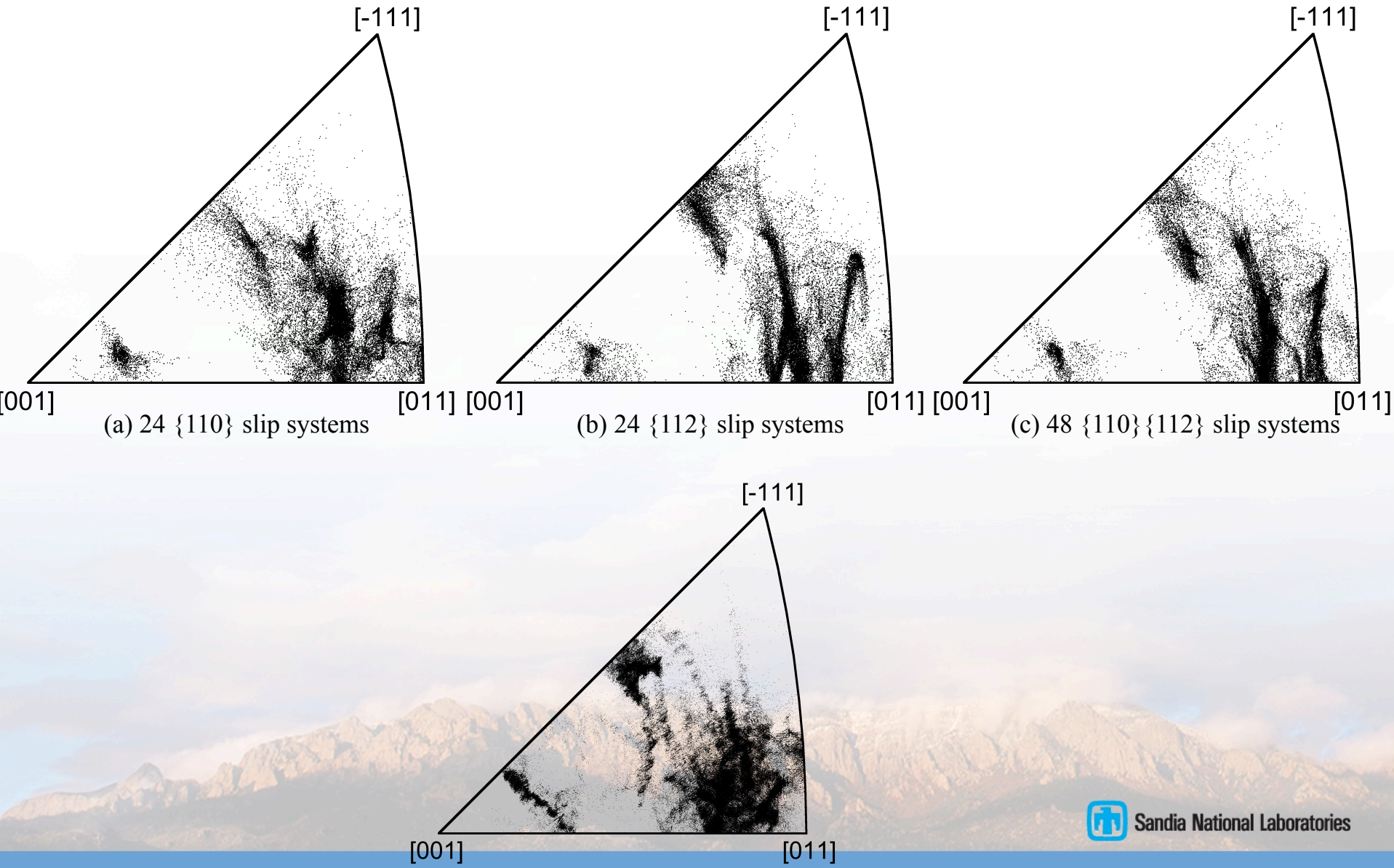


Carroll J.D., Clark B.G., Buchheit T.E., Boyce B.L., Weinberger C.R., Mater. Sci. Eng. A (2013)

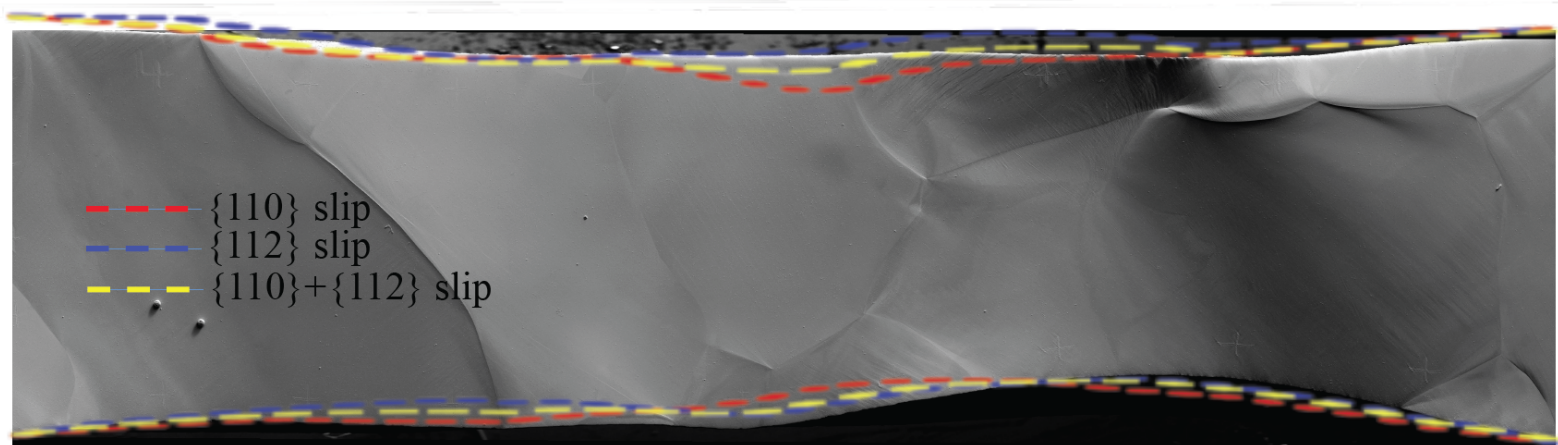
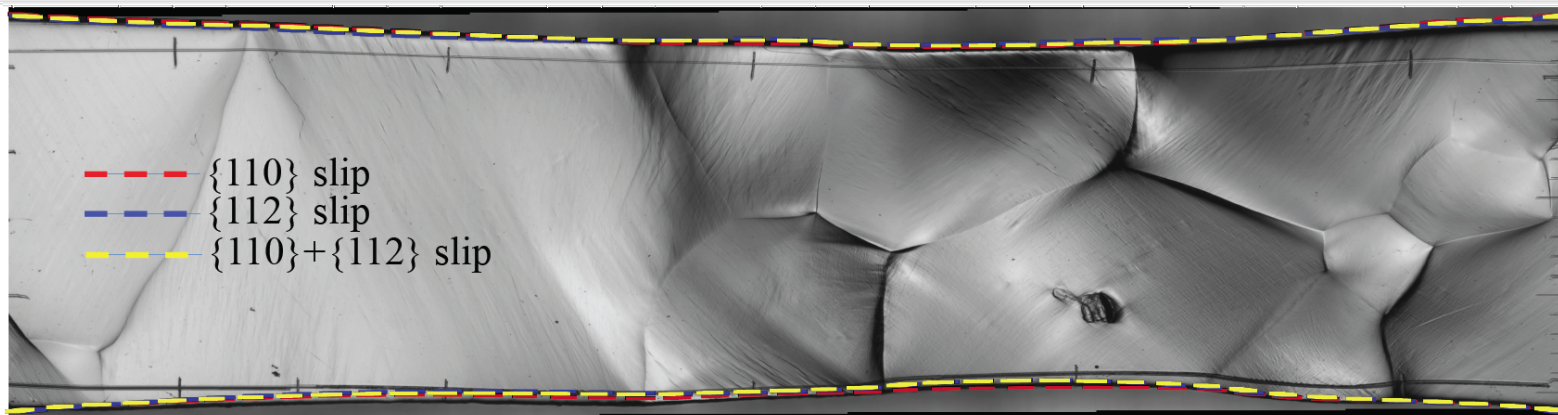
Slip Planes – Strain Distributions



Texture Evolution



Deformed Shapes



Conclusions

- Numerous studies indicate good qualitative agreement – the “trends” are correct – but the quantitative details are often missing.
- CPFEM simulations appear to predict grain-scale strains accurately (relatively)
 - Relatively insensitive to mesh size & slip planes
 - Does depend on through thickness
- CPFEM simulations predict texture evolution moderately well
 - Missing some details
 - More mesh sensitive than strains
- Sub-grain deformations depend on many factors. Thus the details of deformation depend on the details of the experimentation and simulation.
- What level of agreement / validation can we reasonably expect from models of this class?
- More work is needed to “dot i’s and cross t’s.”

