

Finite Element Predictions of Grainscale Behavior in BCC Metals

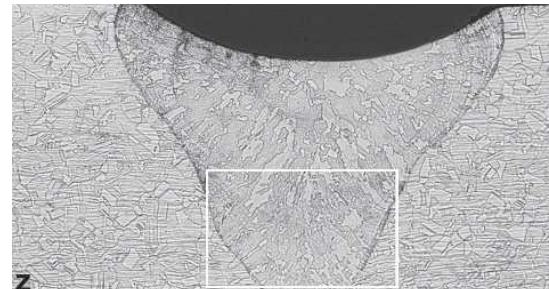
Christopher R. Weinberger

Hojun Lim, Jay Carroll, Thomas E. Buchheit,
Corbett C. Battaile and Brad L. Boyce

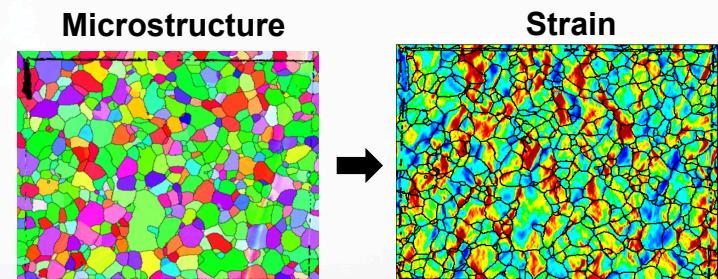
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Can We Predict Grainscale Deformation Behavior?

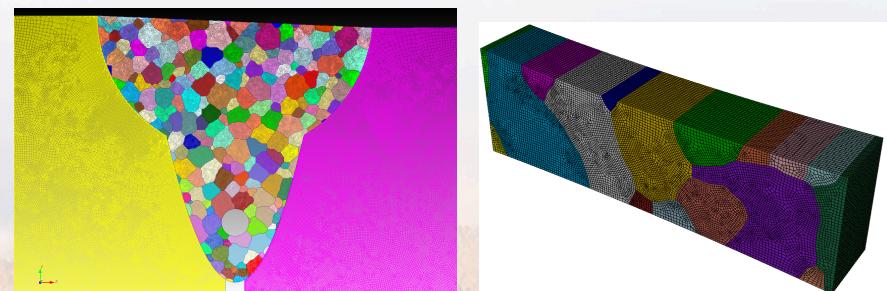
- Need the capability to model complex geometries with realistic microstructures.



- Deformation in polycrystals involves the networks of grains



- Need expedient models grainscale deformation
 - Can we use current CPFEM models?
 - How well do they work?
 - What needs improvement?

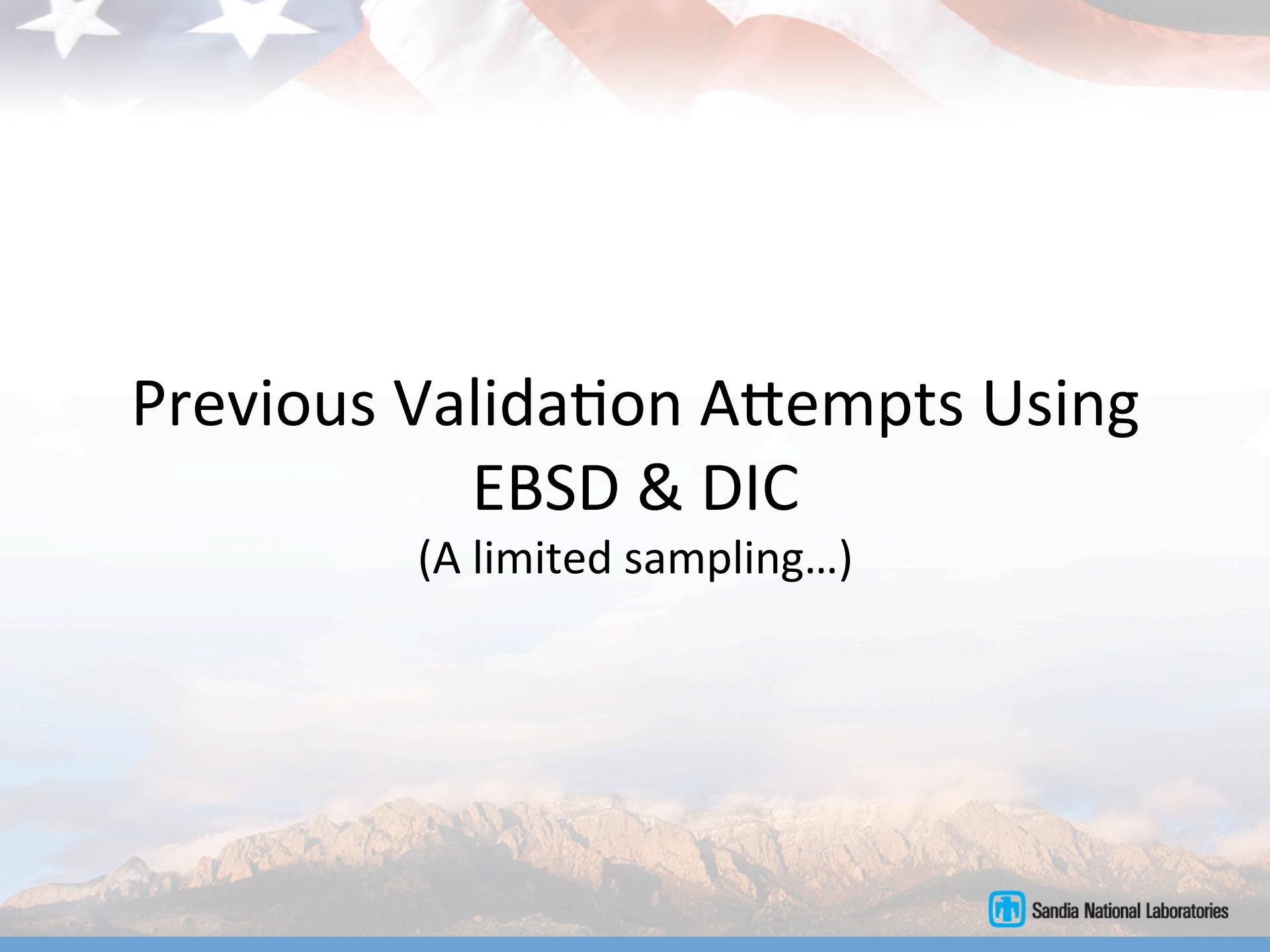


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Finite Element Predictions of Grainscale Behavior in BCC Metals

- Previous work in validating CPFEM models
 - Crystal Rotations (EBSD)
 - Surface Strains (DIC)
- Our Ongoing Validation of BCC Tantalum CPFEM Models
 - Our CPFEM model and experiments
 - Surface Strain Comparisons
 - Texture Evolution and Crystal Rotations
 - Failure
- Model Sensitivity
 - Mesh Sensitivity
 - Sensitivity to Initial Orientations
 - Sensitivity to Slip Planes





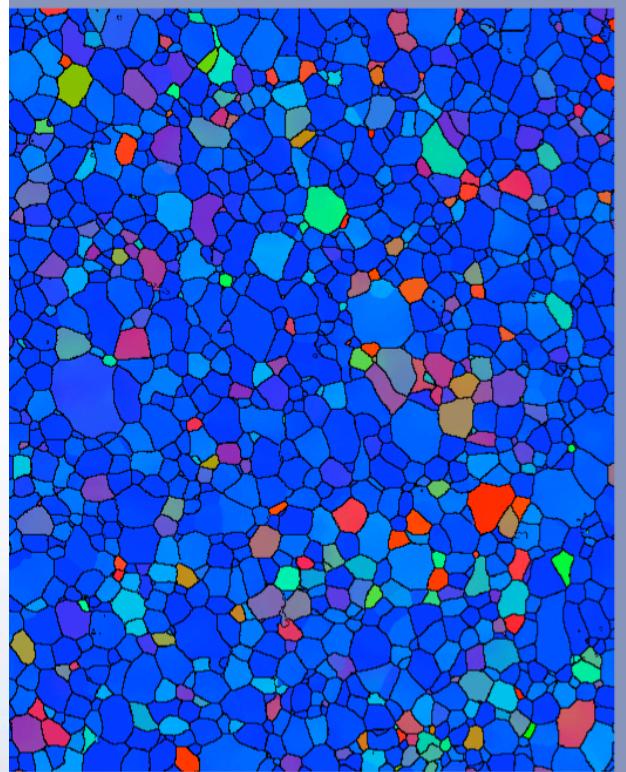
Previous Validation Attempts Using EBSD & DIC (A limited sampling...)



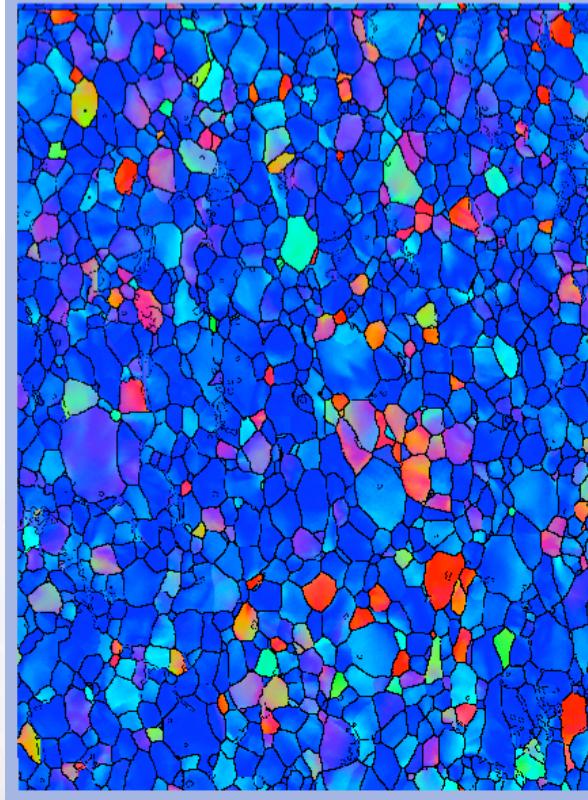
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EBSD of Interstitial-Free Steel Polycrystals

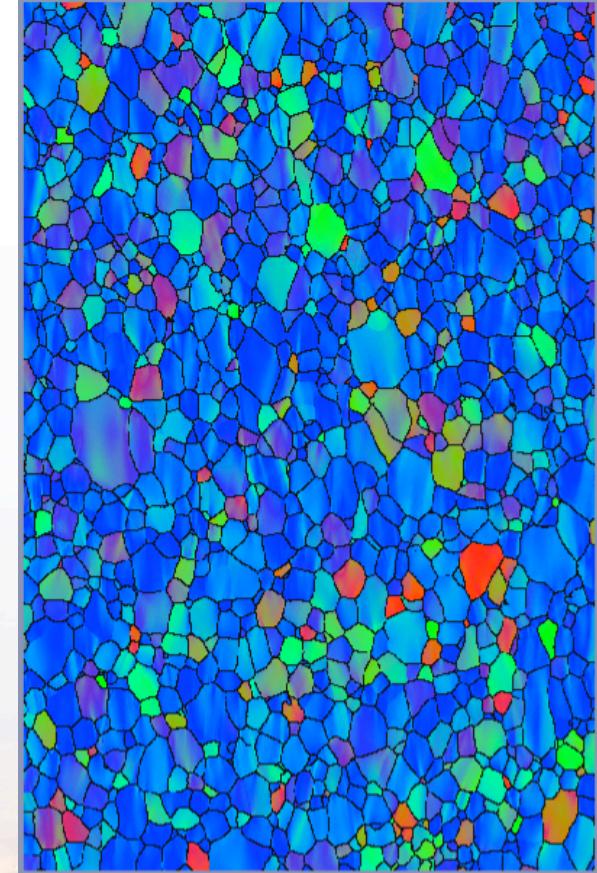
{110+112} only, surface layer



Experiment, 0% Strain



Experiment, 12% Strain



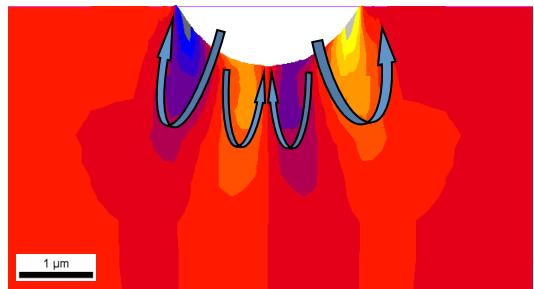
Simulation, 12% Strain



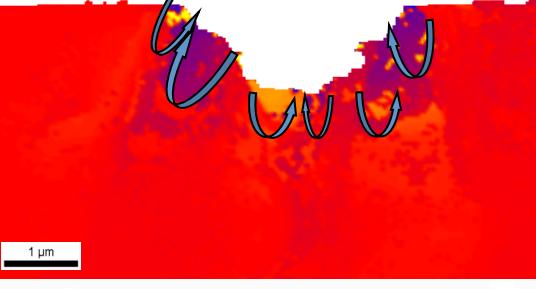
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EBSD of Indented Copper Single Crystals

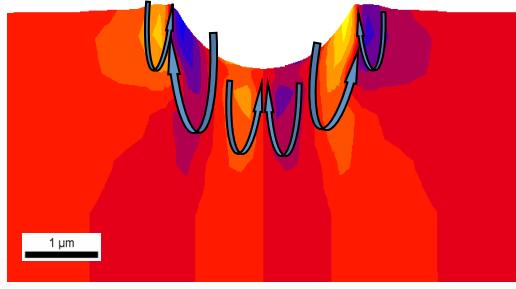
viscoplastic
CPFEM



experiment
3D EBSD



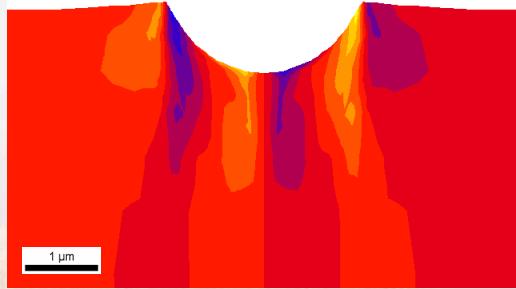
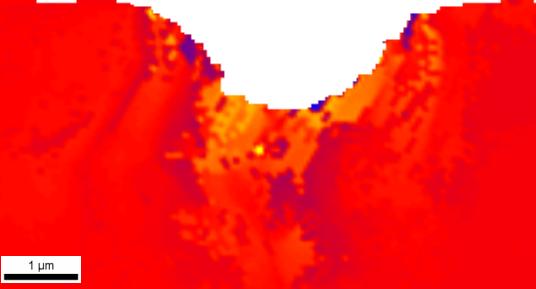
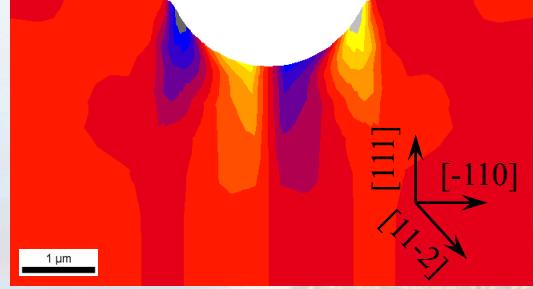
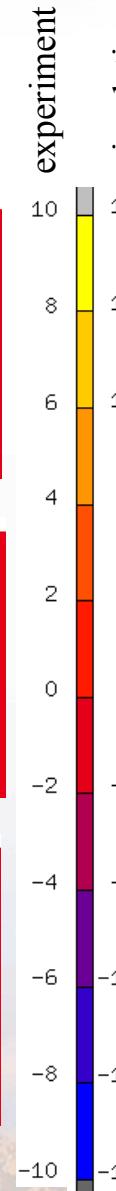
dislocation-based
CPFEM



scan 7

scan 8

scan 9



Courtesy of Dierk Raabe.

N. Zaafarani, D. Raabe, R.N. Singh, F. Roters, S. Zaafferer, *Acta Mater.* **54** (2006) 1707.

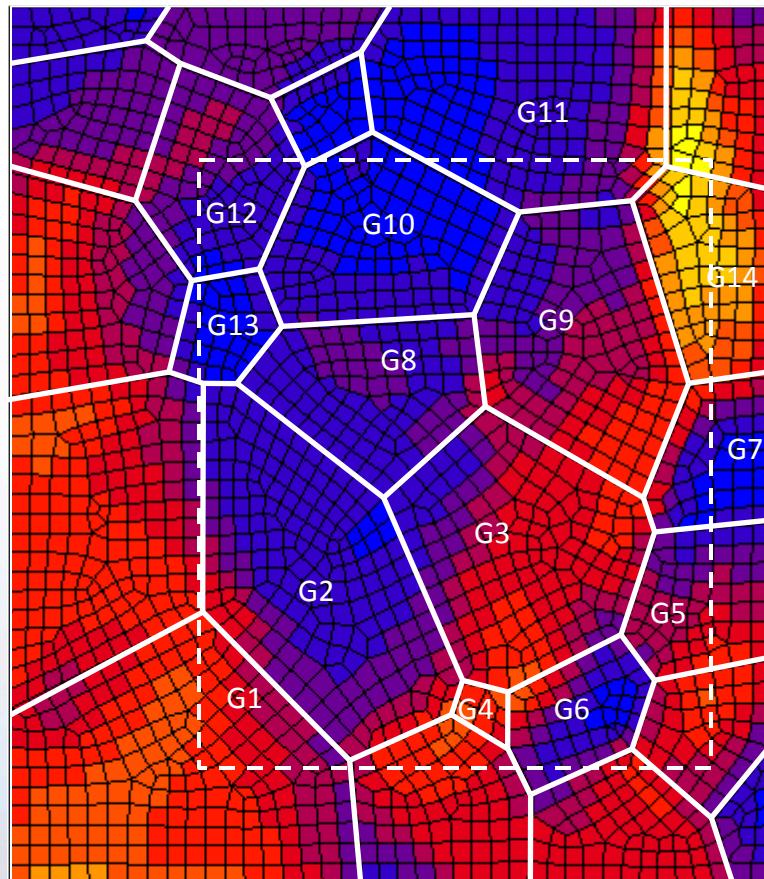


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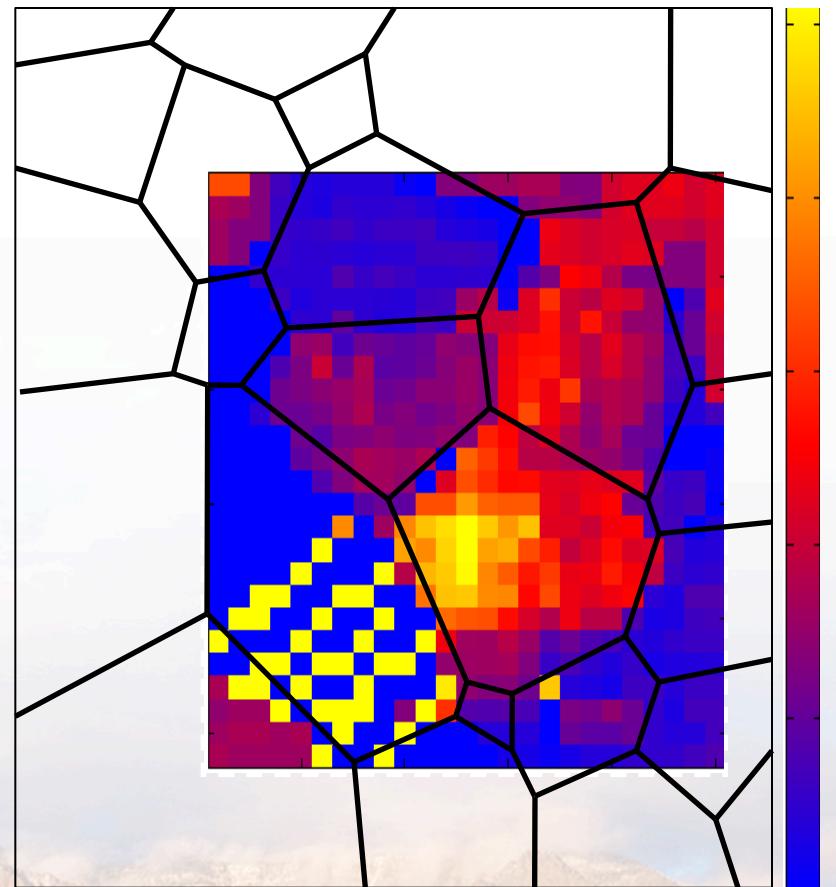
Strain Mapping in a Titanium Polycrystal

Total shear after 1.5% applied strain:

Simulation



Experiment



↔ Tensile loading

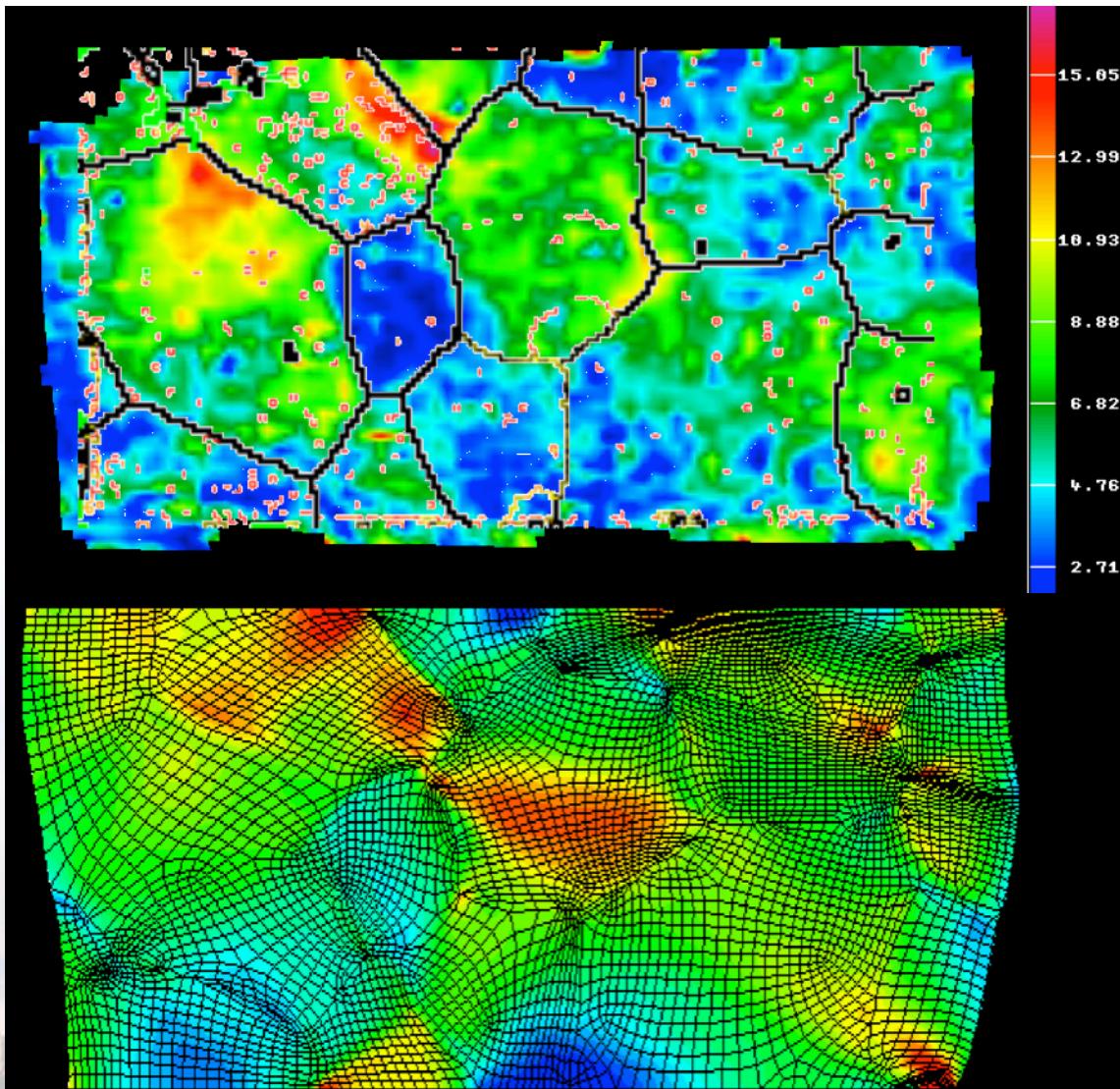
Courtesy of Marty Crimp and Tom Bieler.

Y. Yang, L. Wang, T.R. Bieler, P. Eisenlohr, M.A. Crimp, *Met. Trans. A* **42A** (2011) 636.



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Strains in an Al Oligocrystal



Experiment
(DIC, EBSD)
Mises strain

Simulation
(Viscoplastic
CP-FEM)
Mises strain

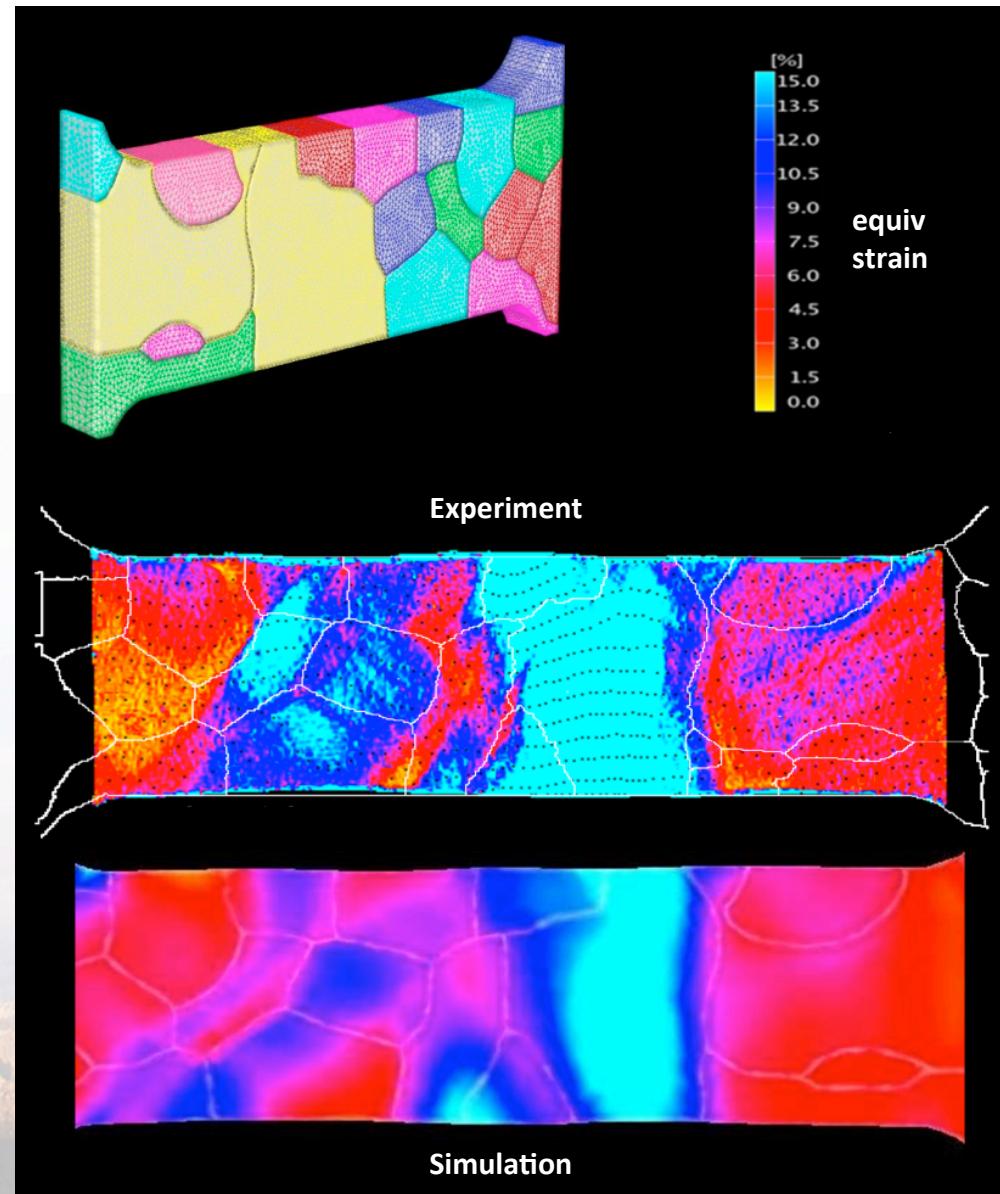
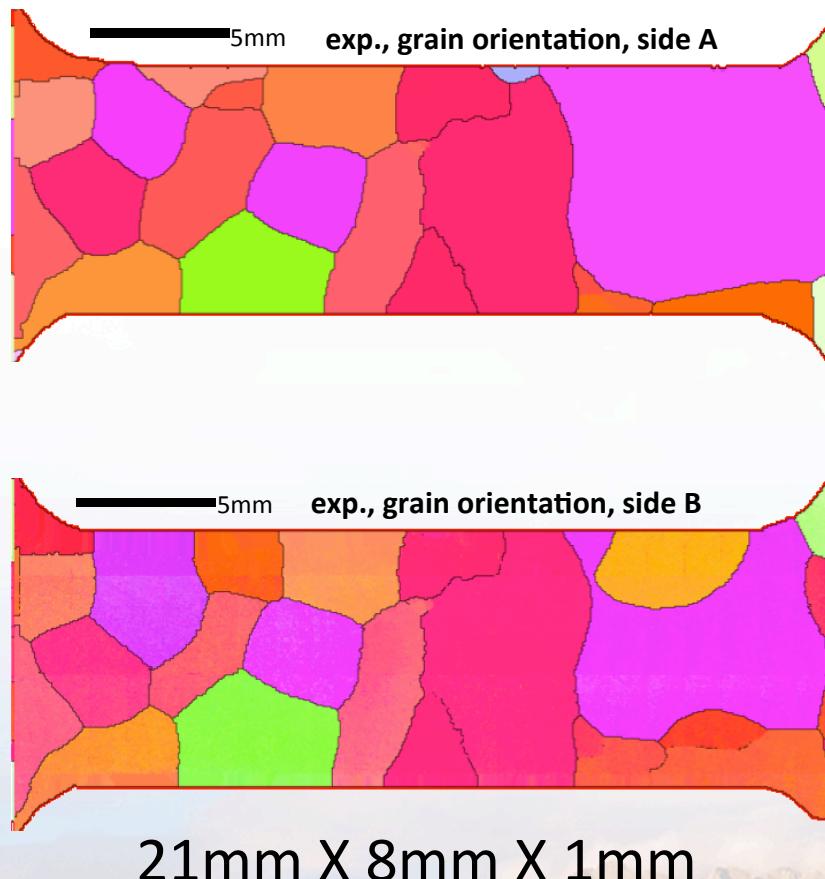
Courtesy of Dierk Raabe.

D. Ma, F. Roters, D. Raabe, *Acta Mater.* **54** (2006) 2169 and 2181.



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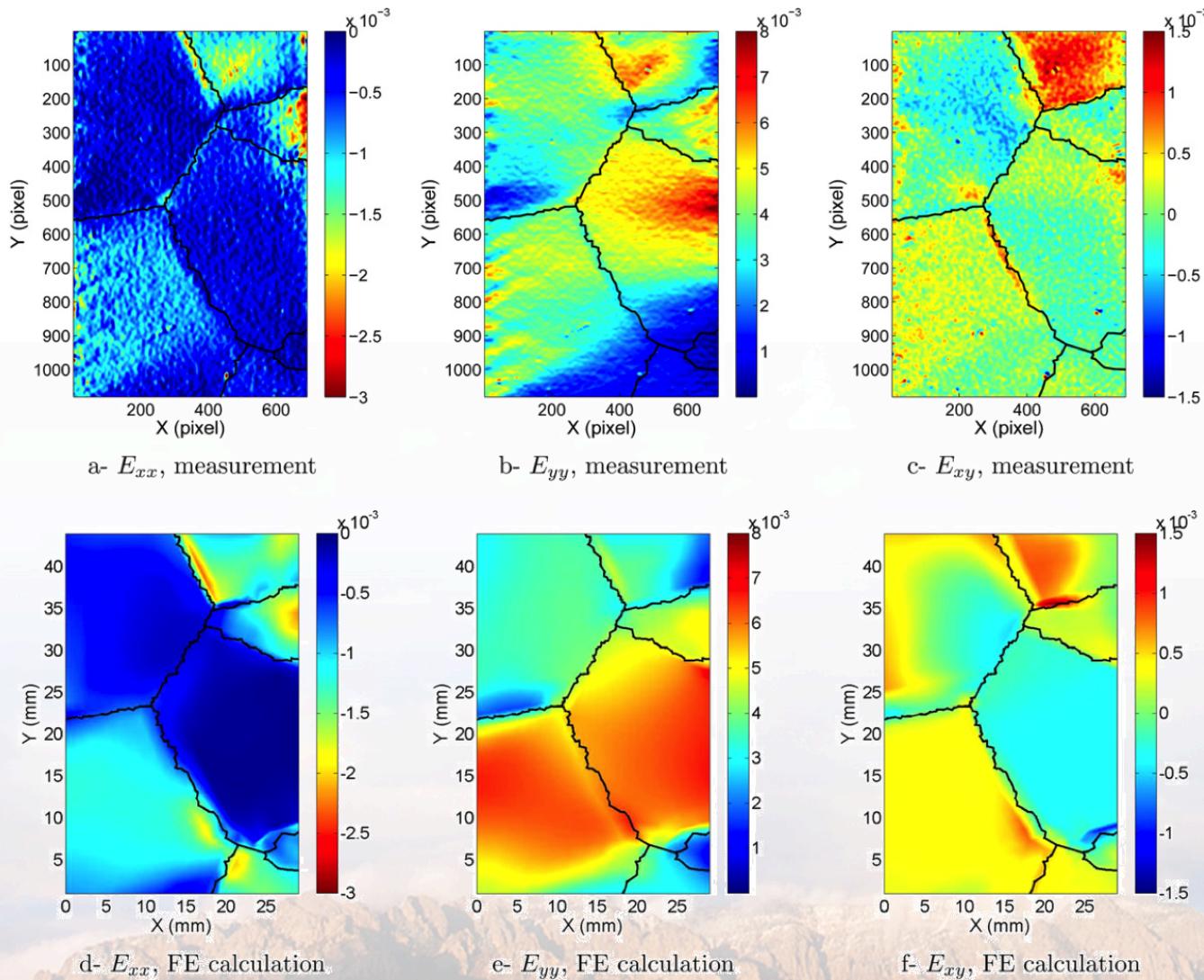
Strains in another Al Oligocrystal



Courtesy of Dierk Raabe.

Z. Zhao, M. Ramesh, D. Raabe, A.M. Cuitino, R. Radovitzky, *Int. J. Plast.* **24** (2008) 2278.

Strains in yet another Al Oligocrystal

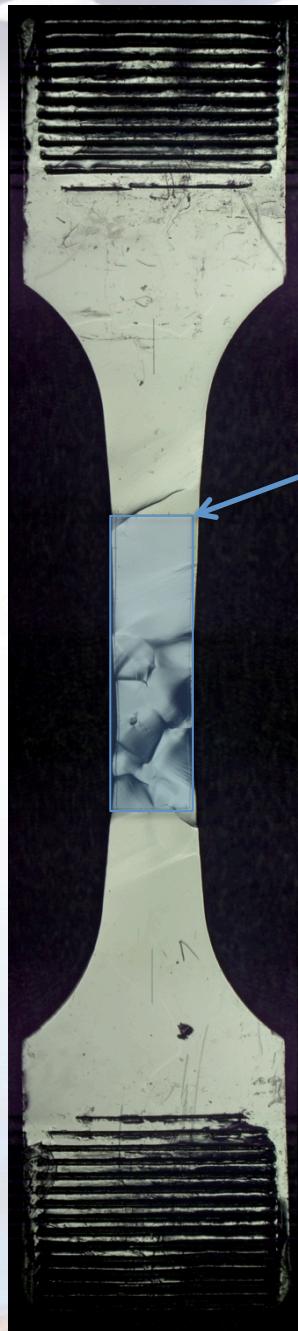


Validation of BCC Tantalum CPFEM Models

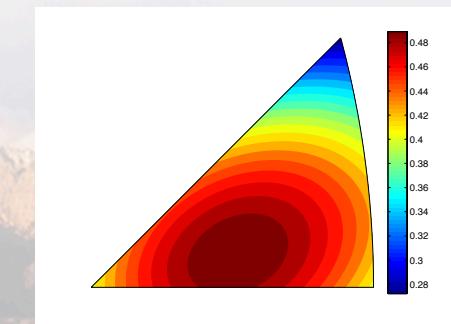
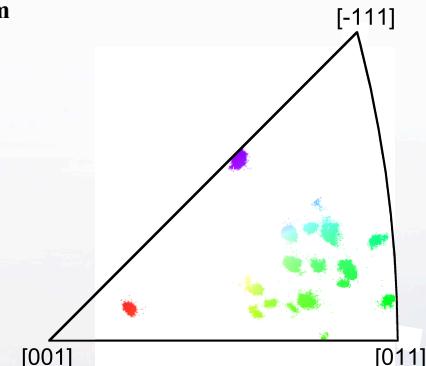
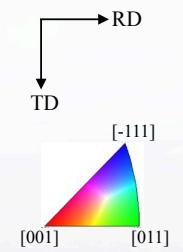
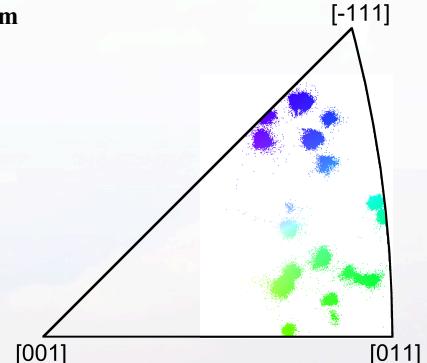
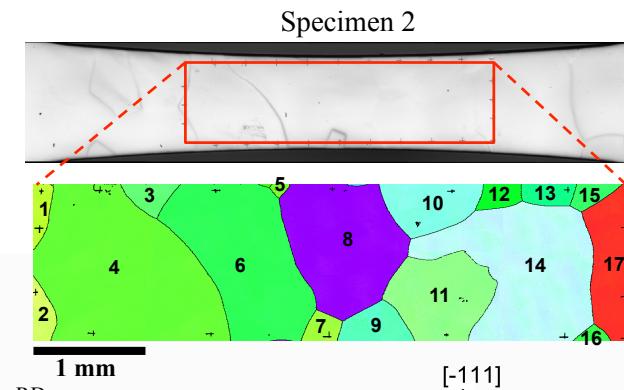
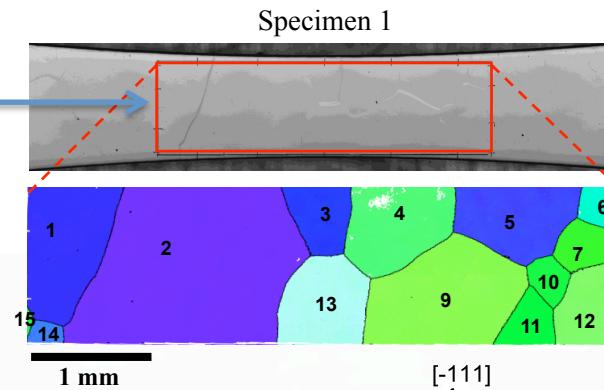


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Oligocrystals



Area
Of
Interest



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BCC CPFEM Formulation

Slip rate: $\dot{\gamma}^\alpha = \dot{\gamma}_0^\alpha \left(\frac{\tau^\alpha}{g^\alpha} \right)^{1/m}$ (Hutchinson, 1976)

Slip resistance: $g^\alpha = \max(\tau_{\text{cr}}^\alpha - \tau_{\text{ns}}^\alpha, 0) + \tau_{\text{obs}}^\alpha$ (Weinberger, 2012)

↳ Obstacle stress
↳ Lattice friction

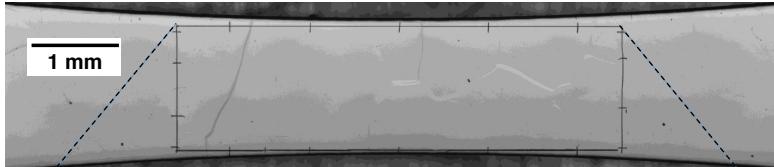
Obstacle stress: $\tau_{\text{obs}}^\alpha = \alpha \mu b \sqrt{\sum_{\beta=1}^{NS} \rho^\beta}$ (Taylor, 1934)

$$\rho^\alpha = \left(\kappa_1 \sqrt{\sum_{\beta=1}^{NS} \rho^\beta} - \kappa_2 \rho^\alpha \right) \cdot |\gamma^\alpha| \quad (\text{Kocks, 1976})$$

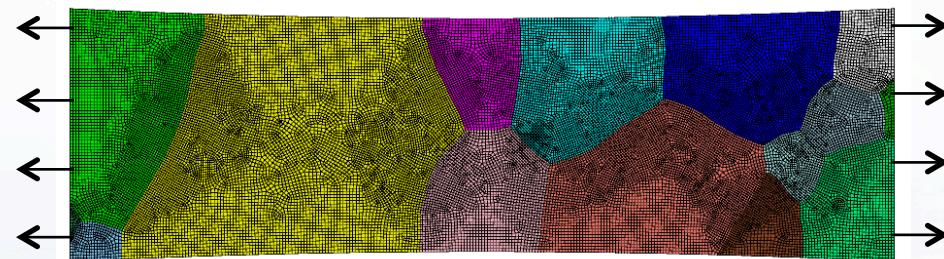
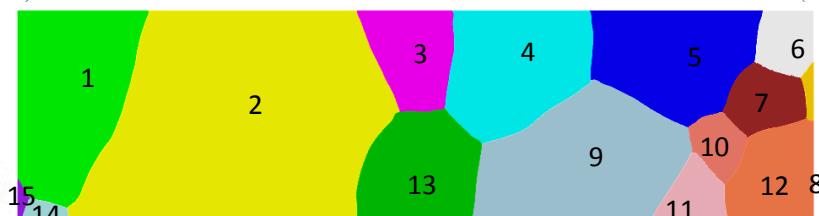


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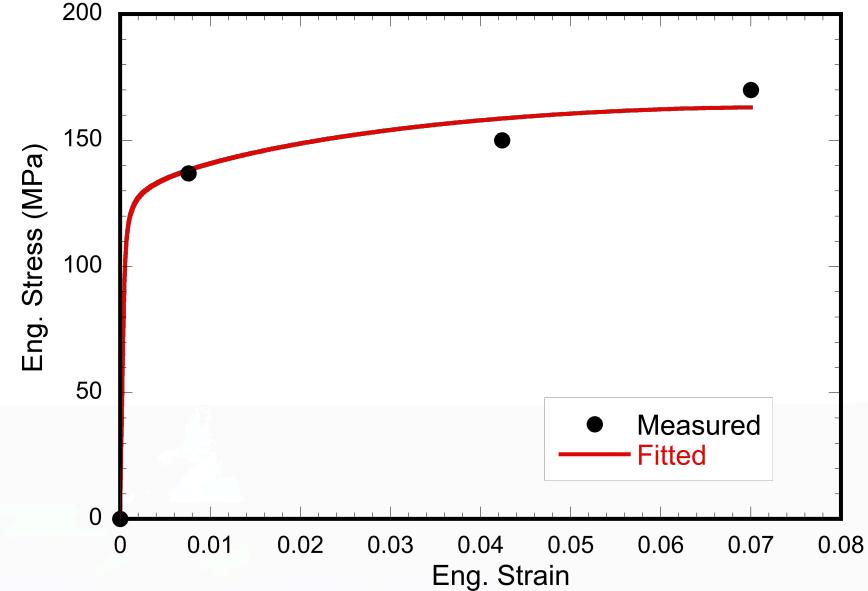
Tantalum Oligocrystal 1



Specimen



Finite Element Mesh



Elastic constants (GPa)

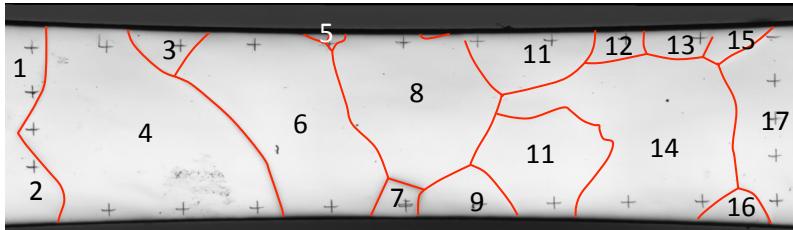
C_{11}	C_{12}	C_{44}	$k_1 (\text{m}^{-1})$	k_2
267	161	82.5	1.4×10^6	14

Hardening coefficients

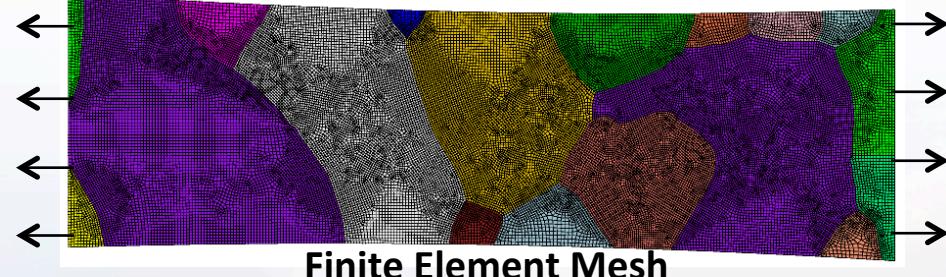
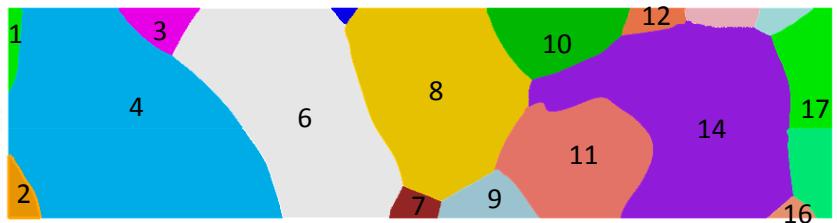


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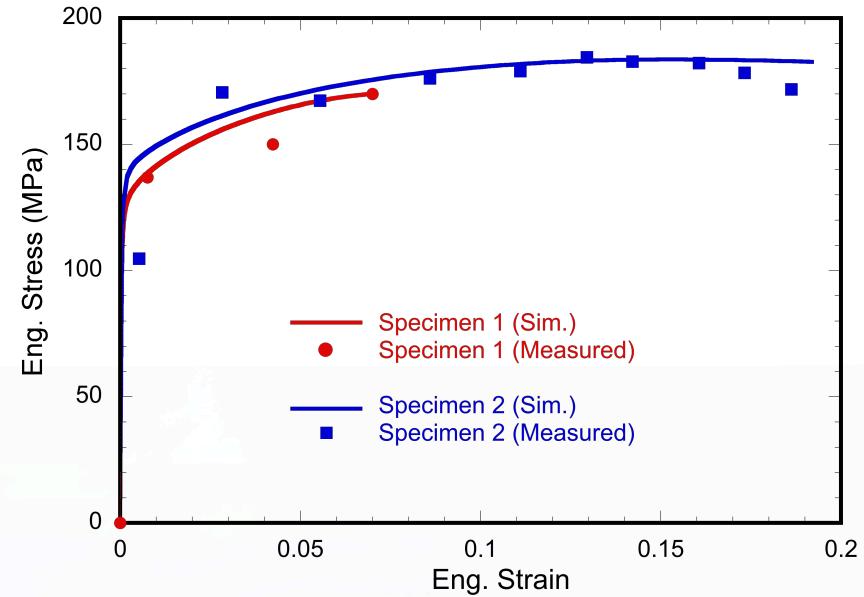
Tantalum Oligocrystal 2



Specimen



Finite Element Mesh



$$\bar{\tau} = 37 \text{ MPa}$$

Elastic constants (GPa)

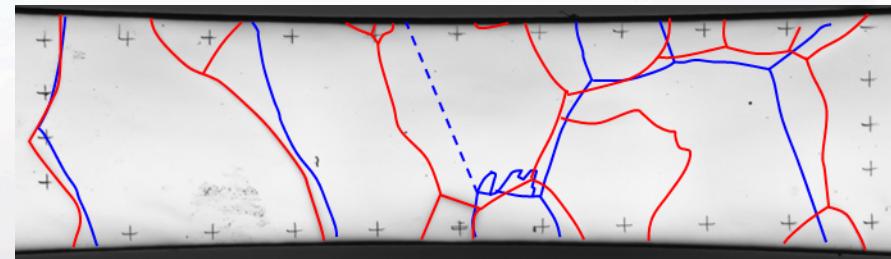
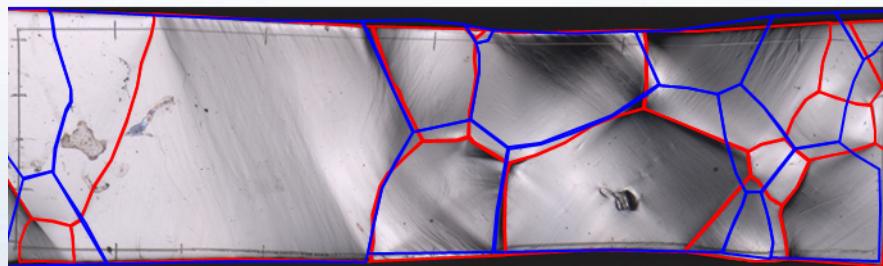
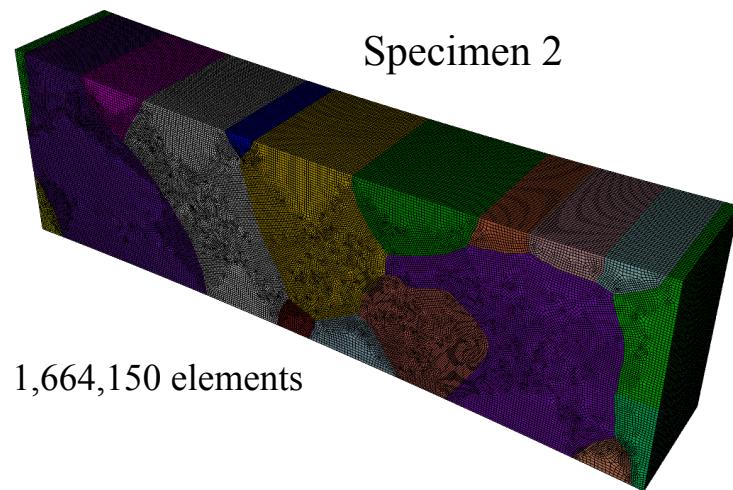
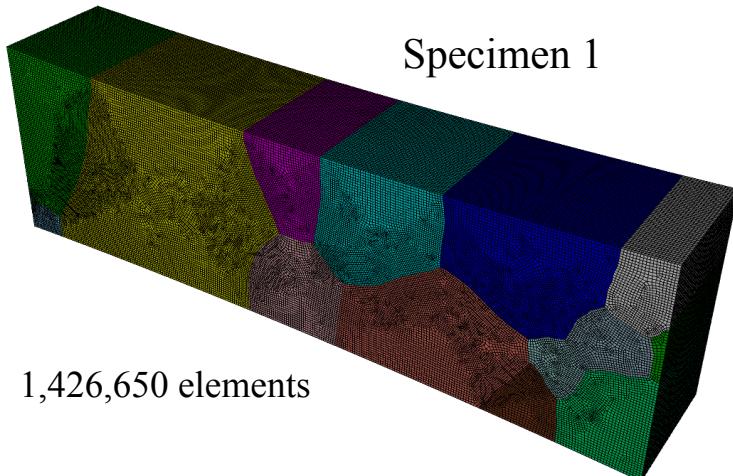
C_{11}	C_{12}	C_{44}	$k_1 (\text{m}^{-1})$	k_2
267	161	82.5	1.4×10^6	14

(Same as for Oligocrystal #1)



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Columnarity



Grain boundary (Front)

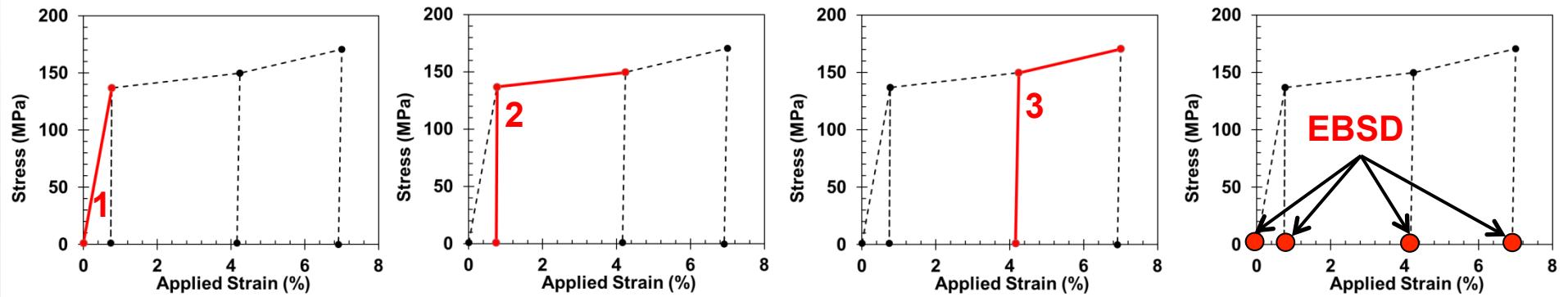
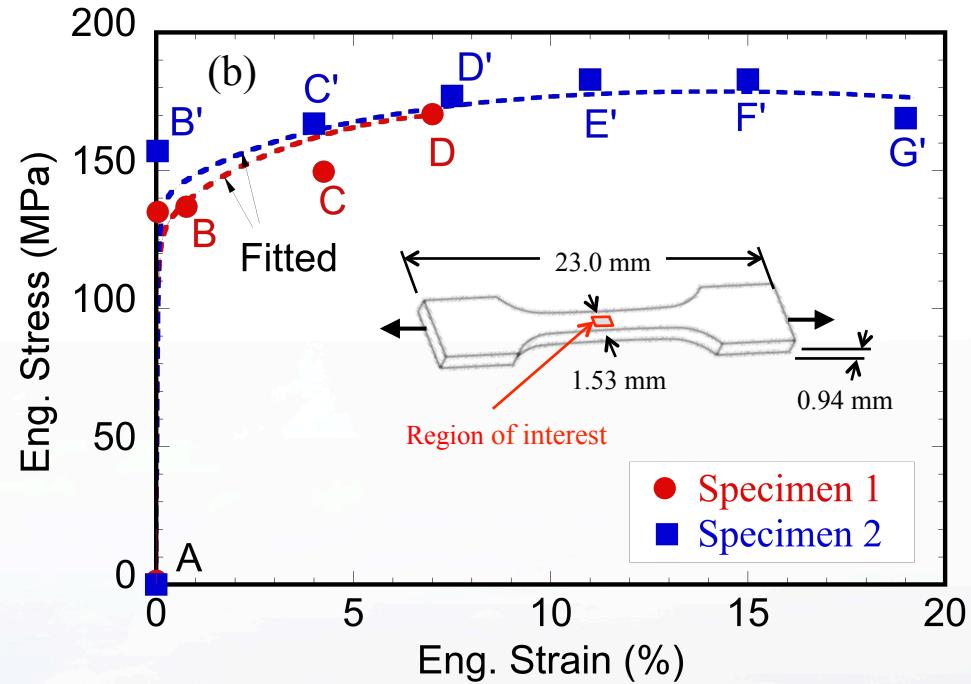
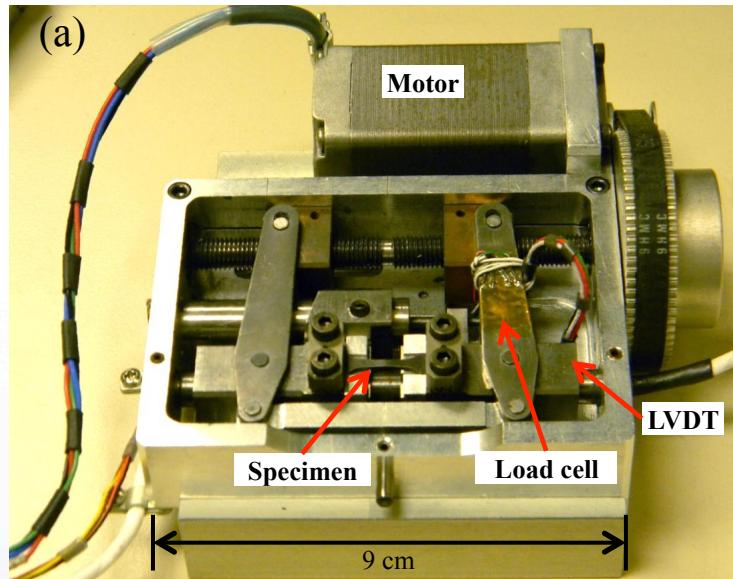
Grain boundary (Back)

1 mm



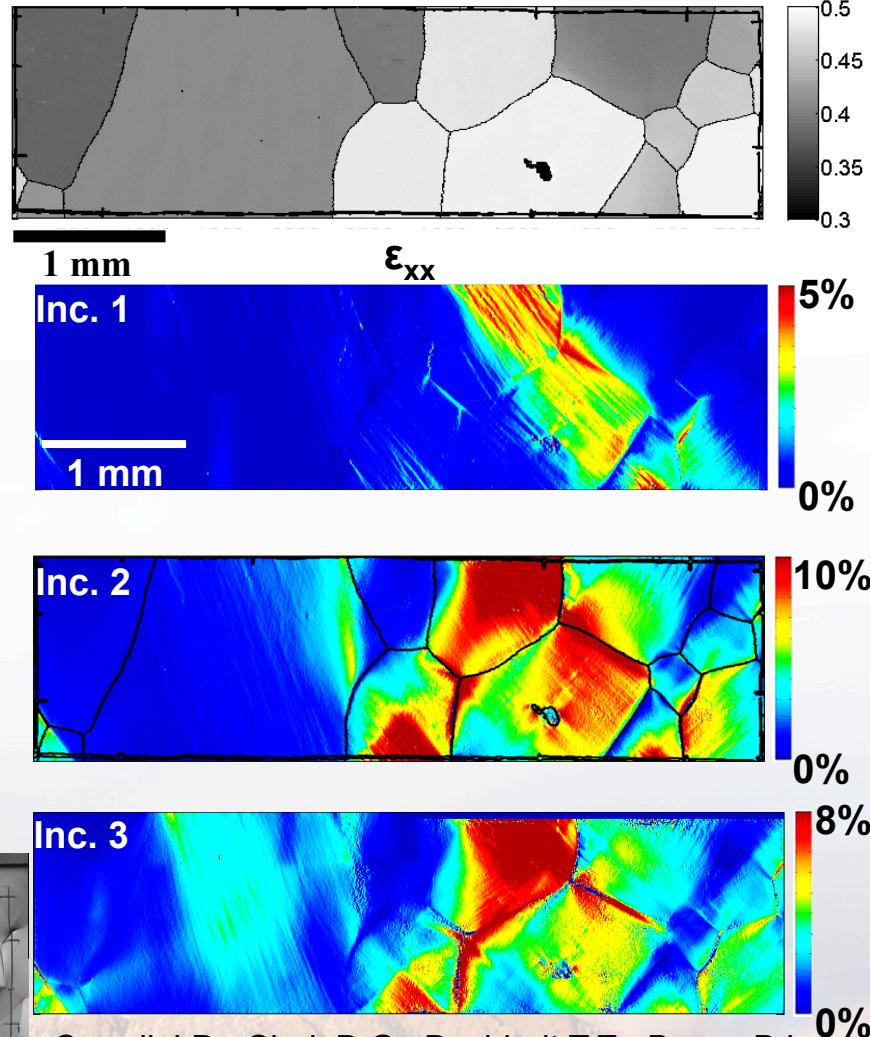
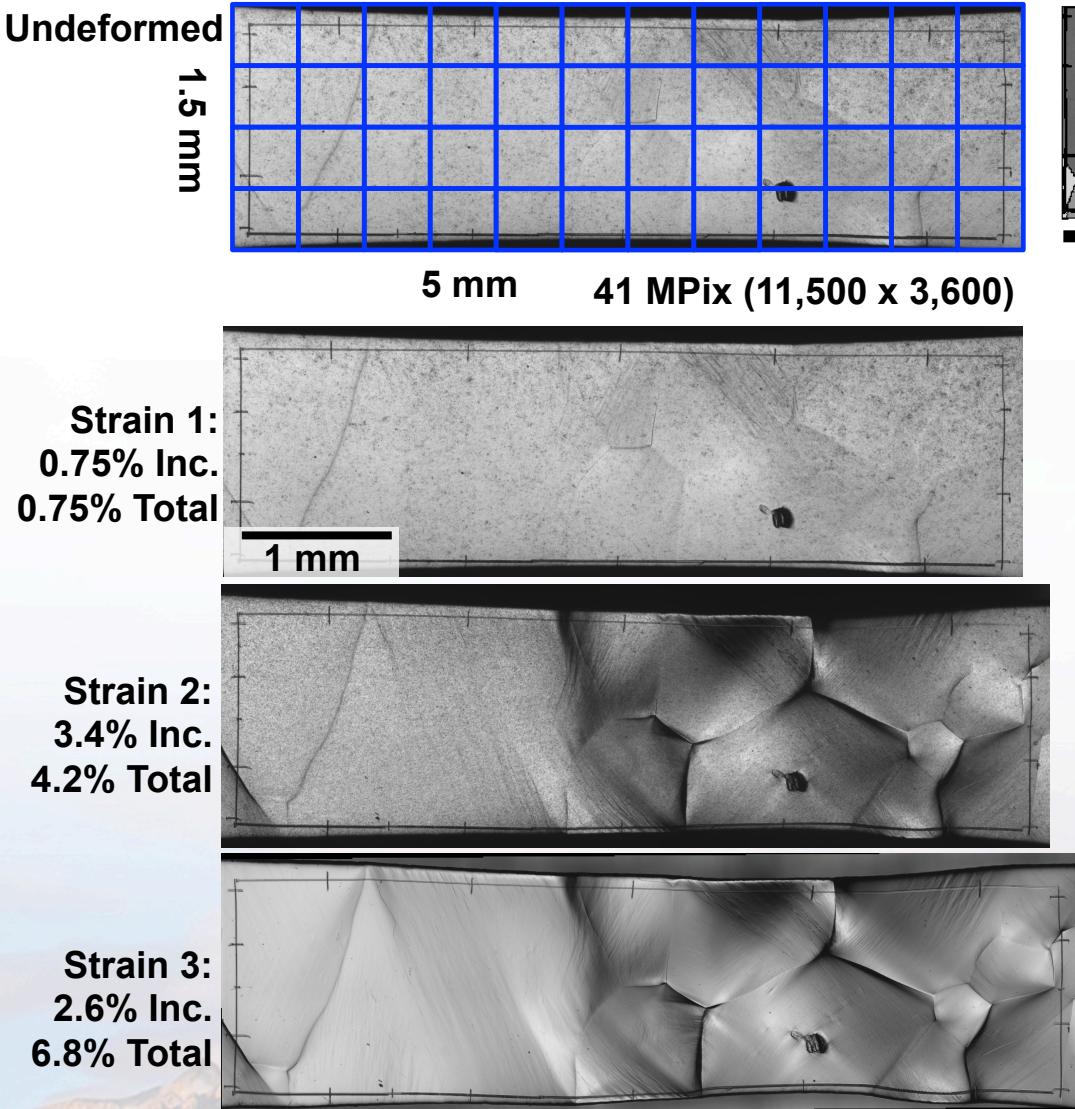
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Loading of Specimens and EBSD



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Digital Image Correlation

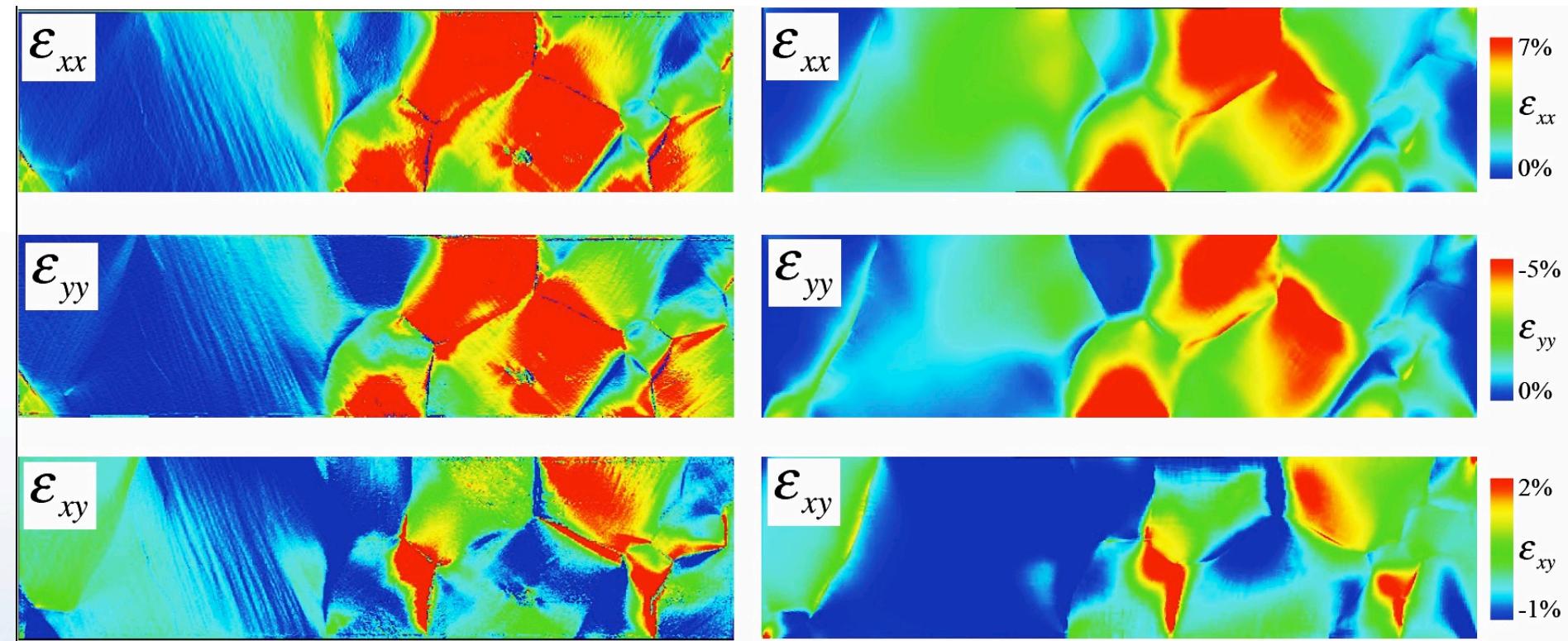


Carroll J.D., Clark B.G., Buchheit T.E., Boyce B.L., Weinberger C.R., Mater. Sci. Eng. A (2013)

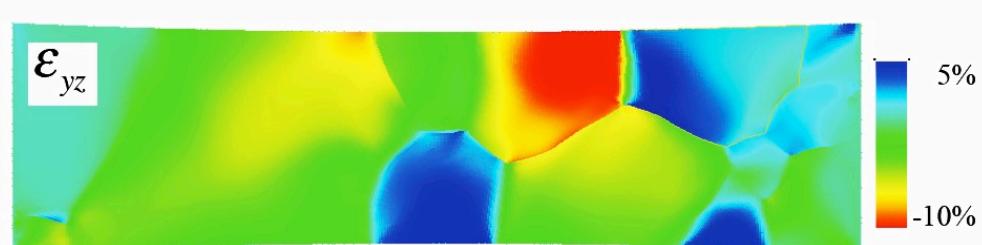
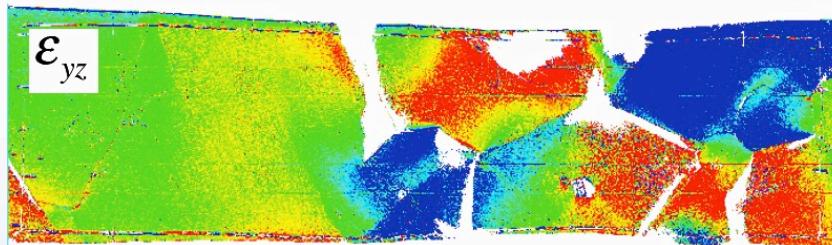
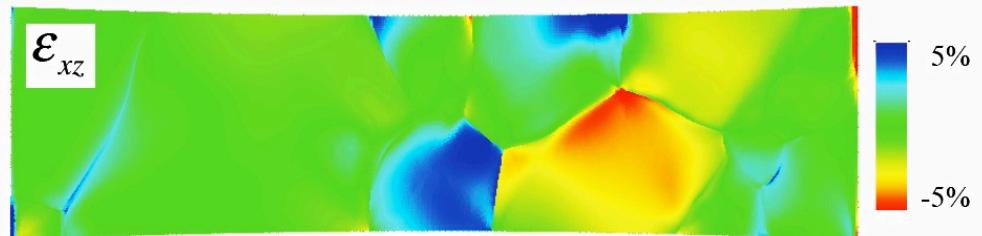
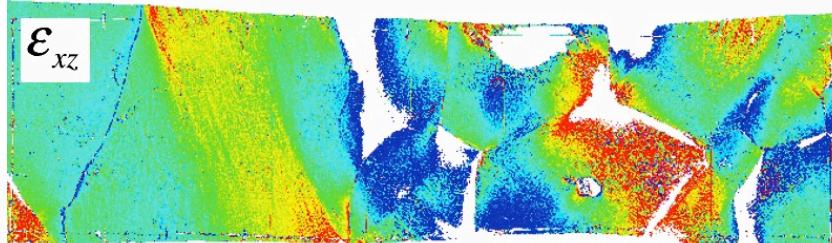


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Predicted In-Plane Strains

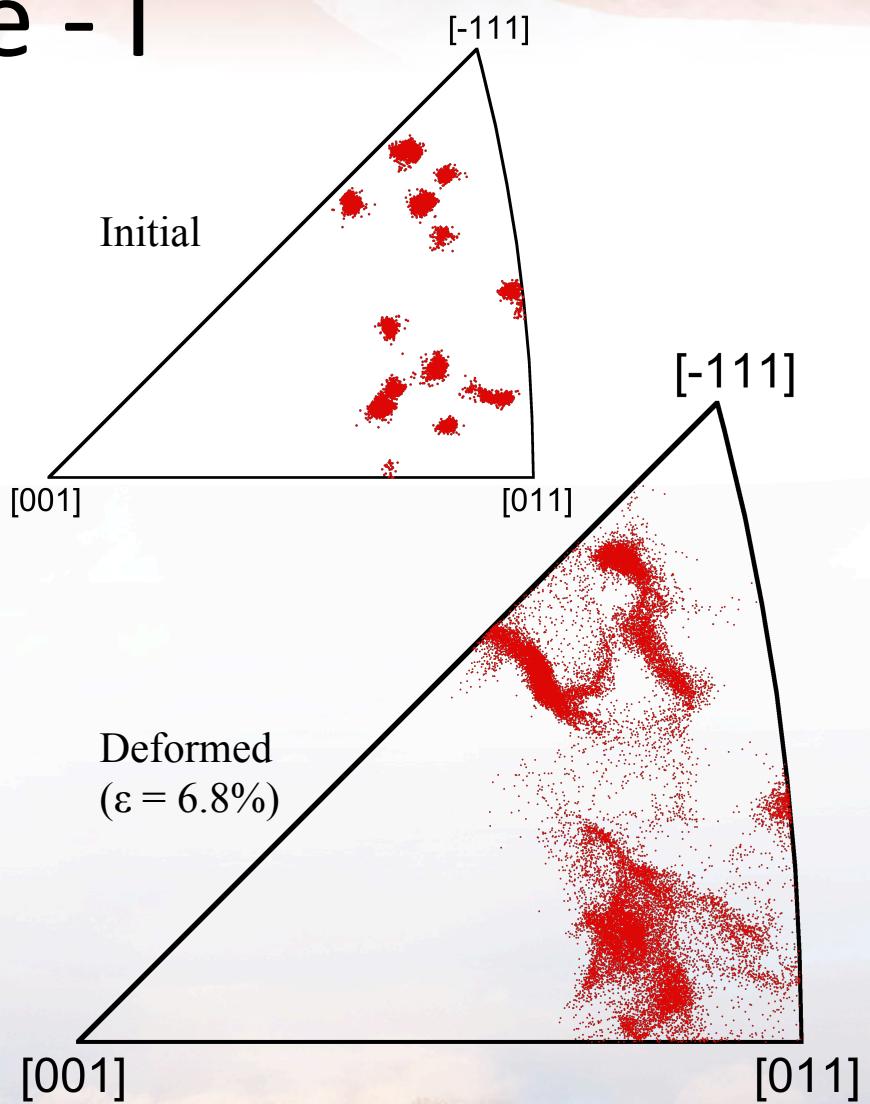
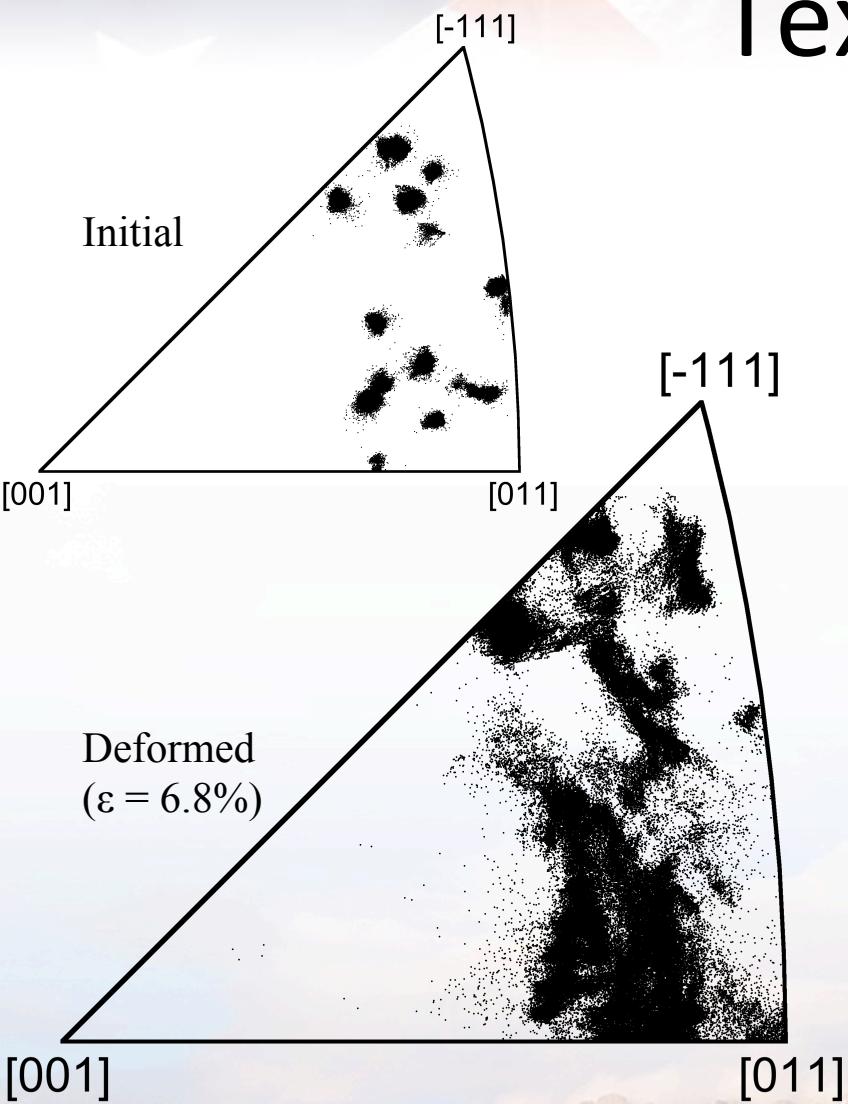


Out-Of-Plane Strains



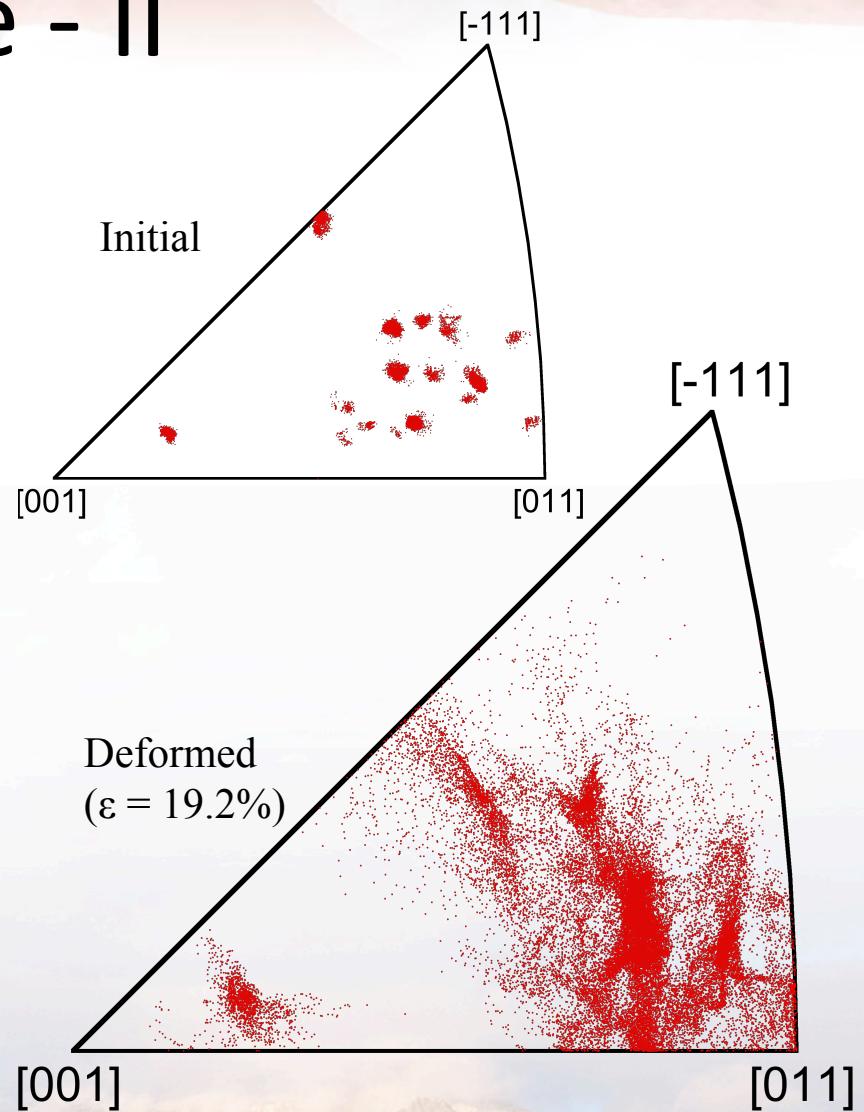
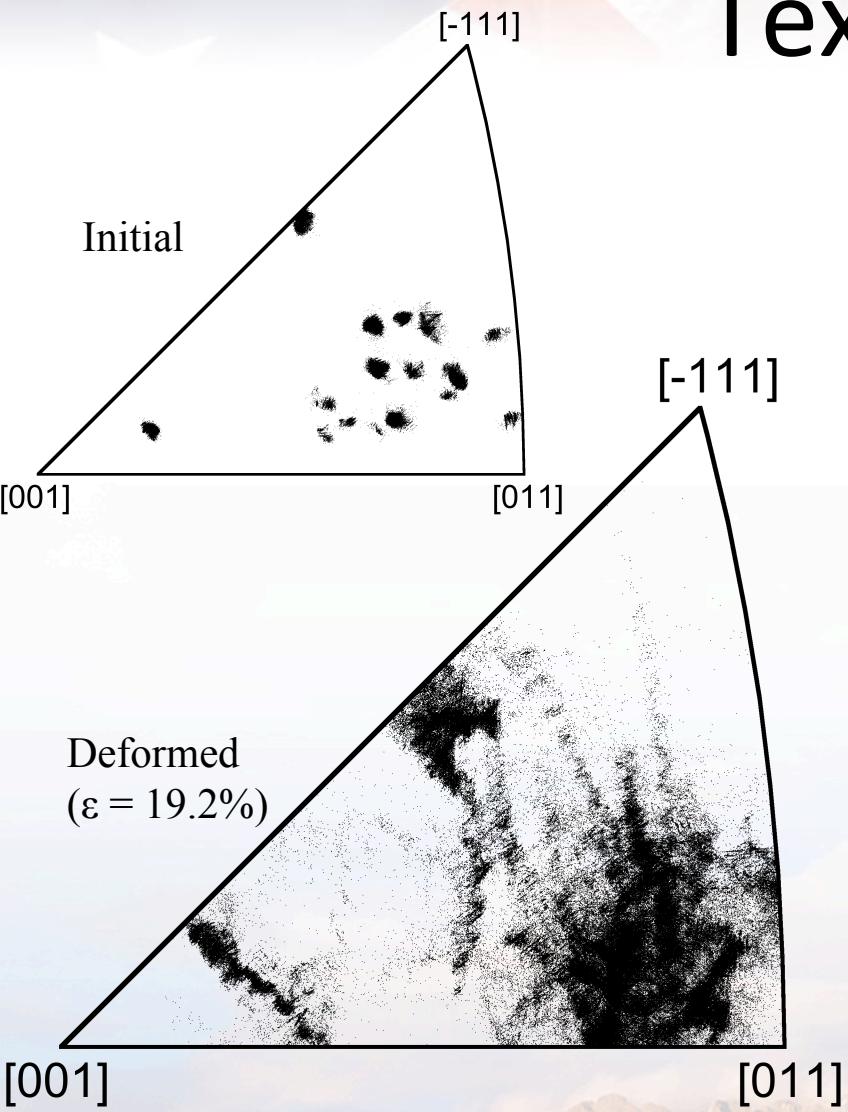
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Texture - I



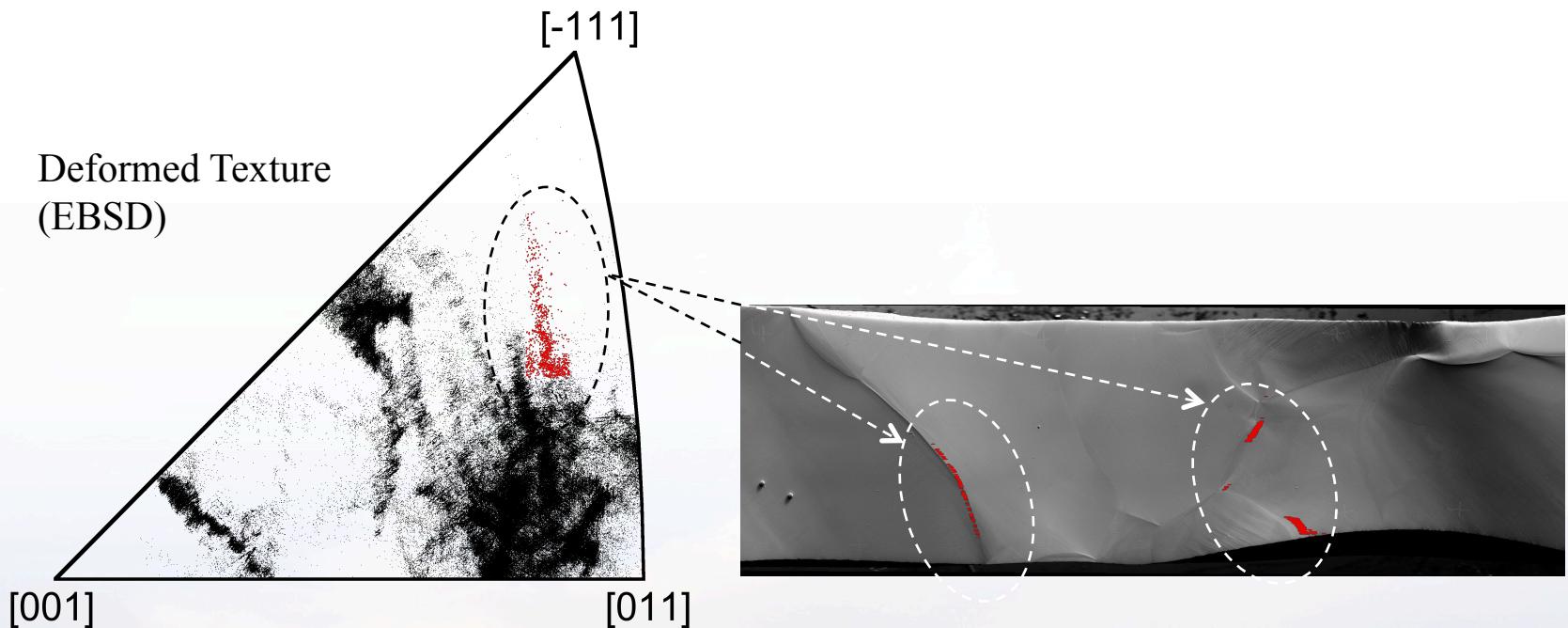
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Texture - II

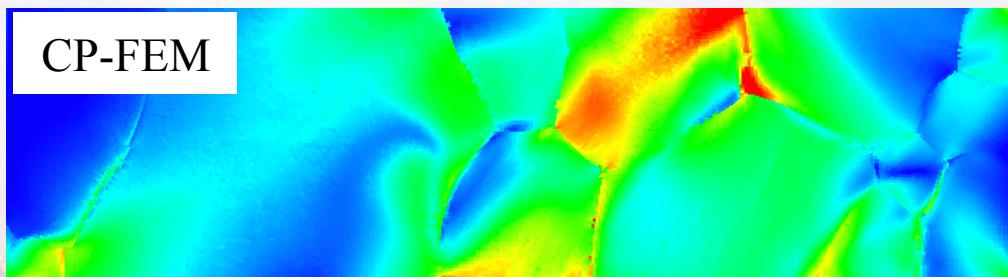
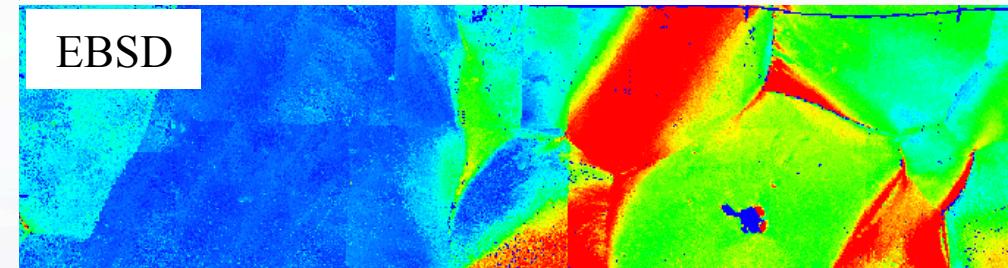
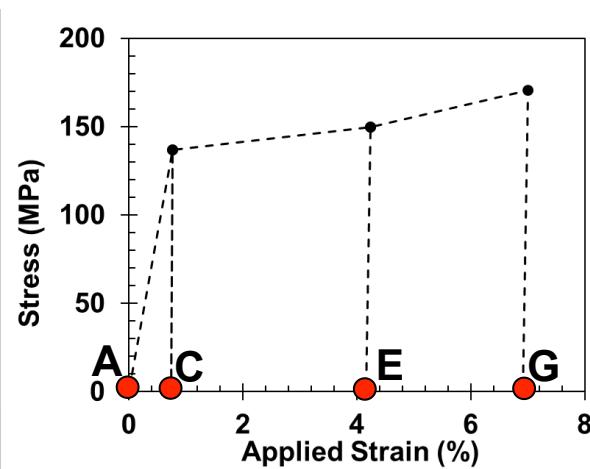
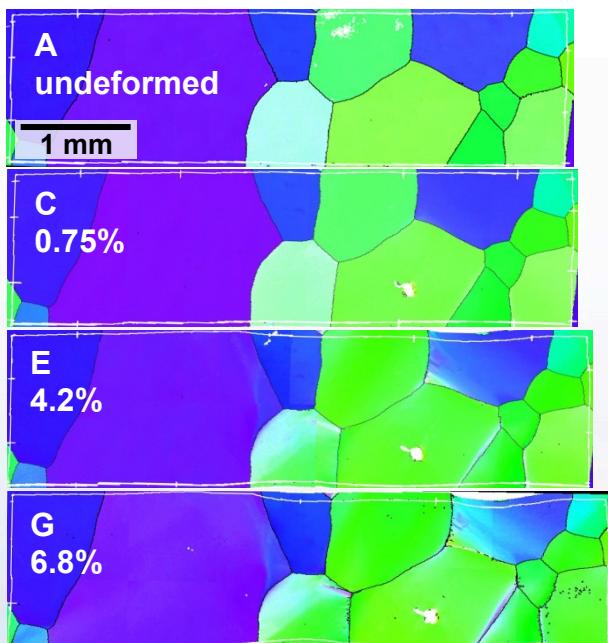


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Texture Issues Occur Near GBs

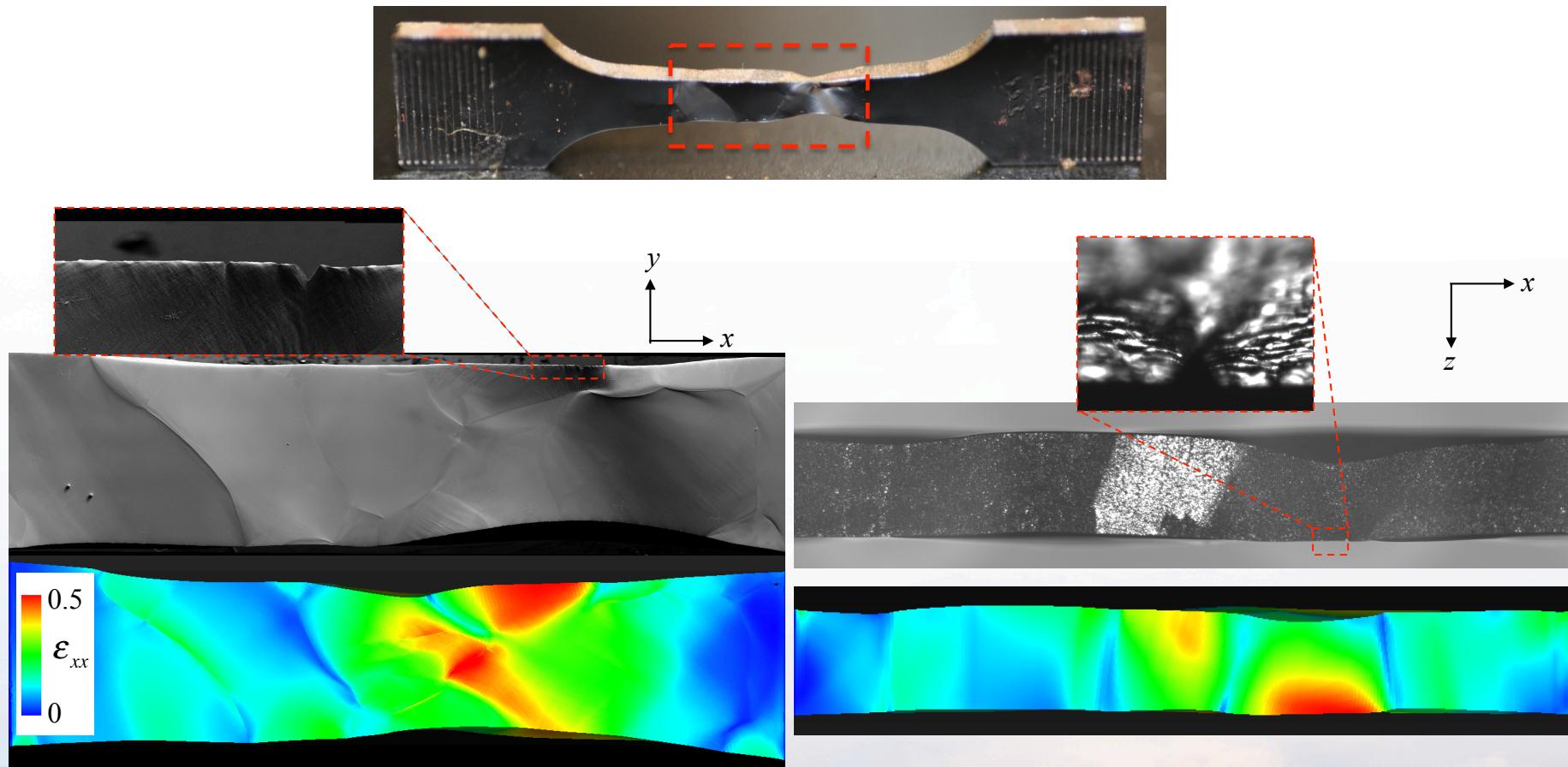


Misorientation



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Failure Predictions



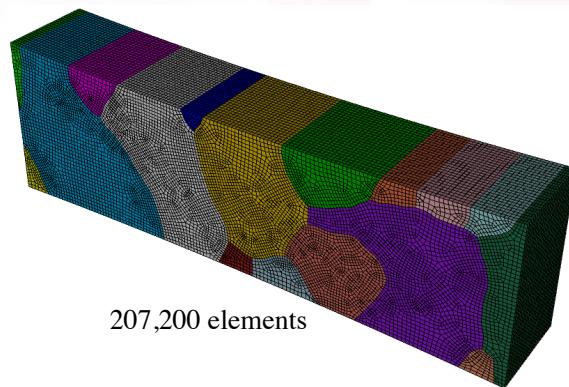
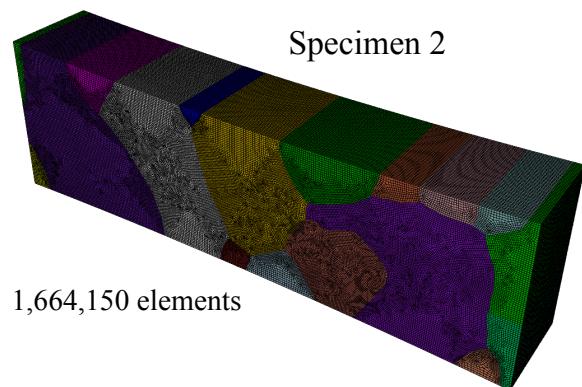
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Model Sensitivity

- Sensitivity of Mesh
- Sensitivity of Orientation Choices
- Sensitivity of Slip plane Choice

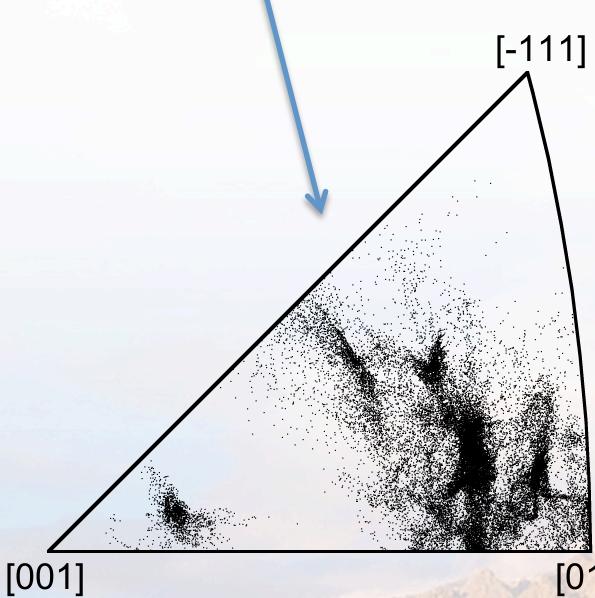


Mesh



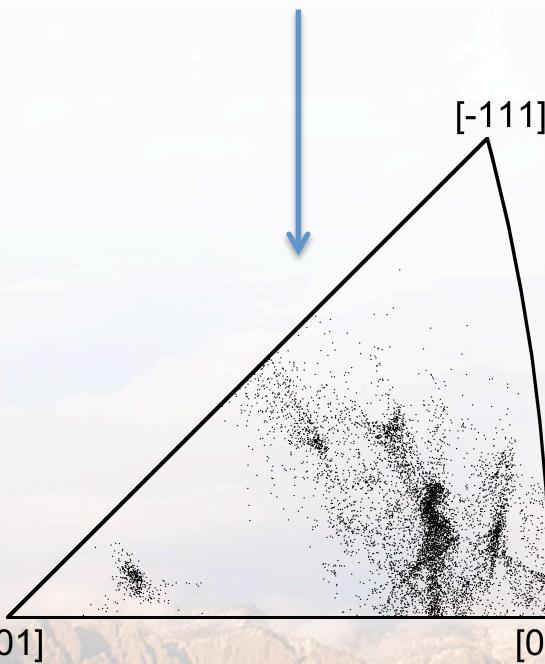
Fine Mesh
(Running)

Medium Mesh

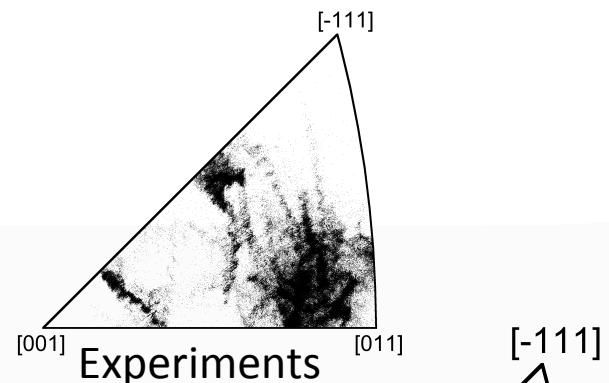


(a) $33,283 \times 50$ elements

Coarse Mesh



(b) $8,288 \times 25$ elements

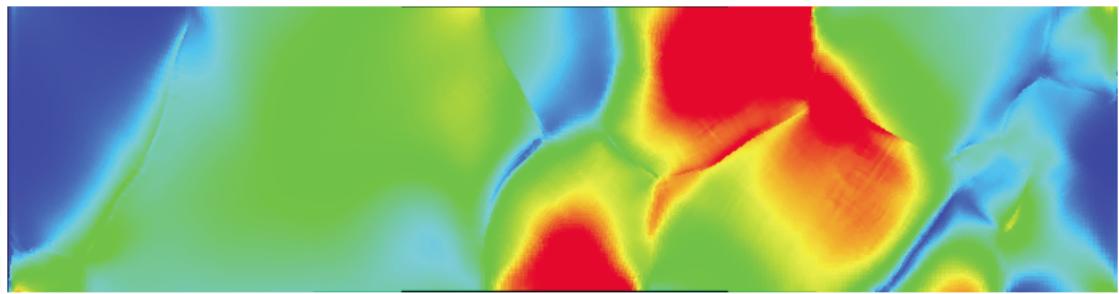


(c) $33,283 \times 1$ elements

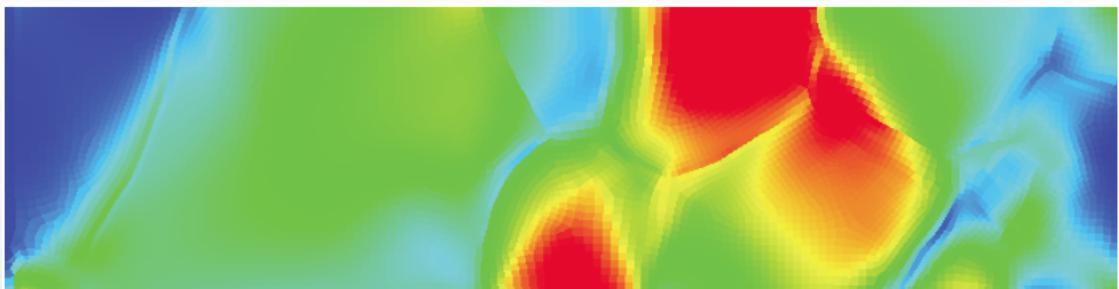


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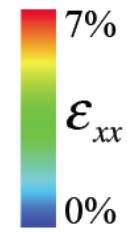
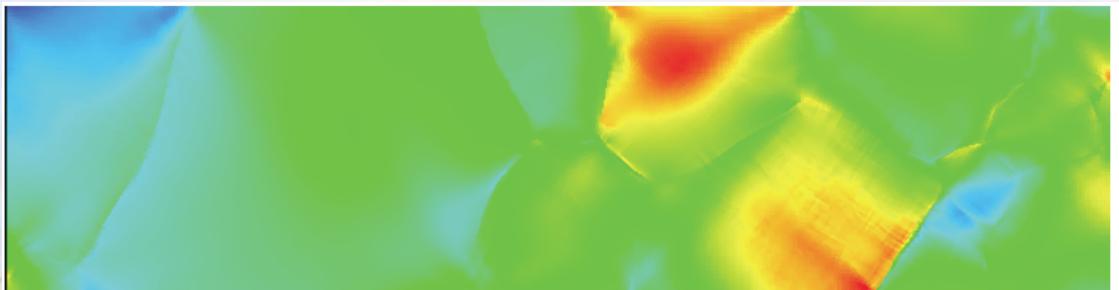
Mesh - Strain Predictions



Medium Mesh



Coarse Mesh

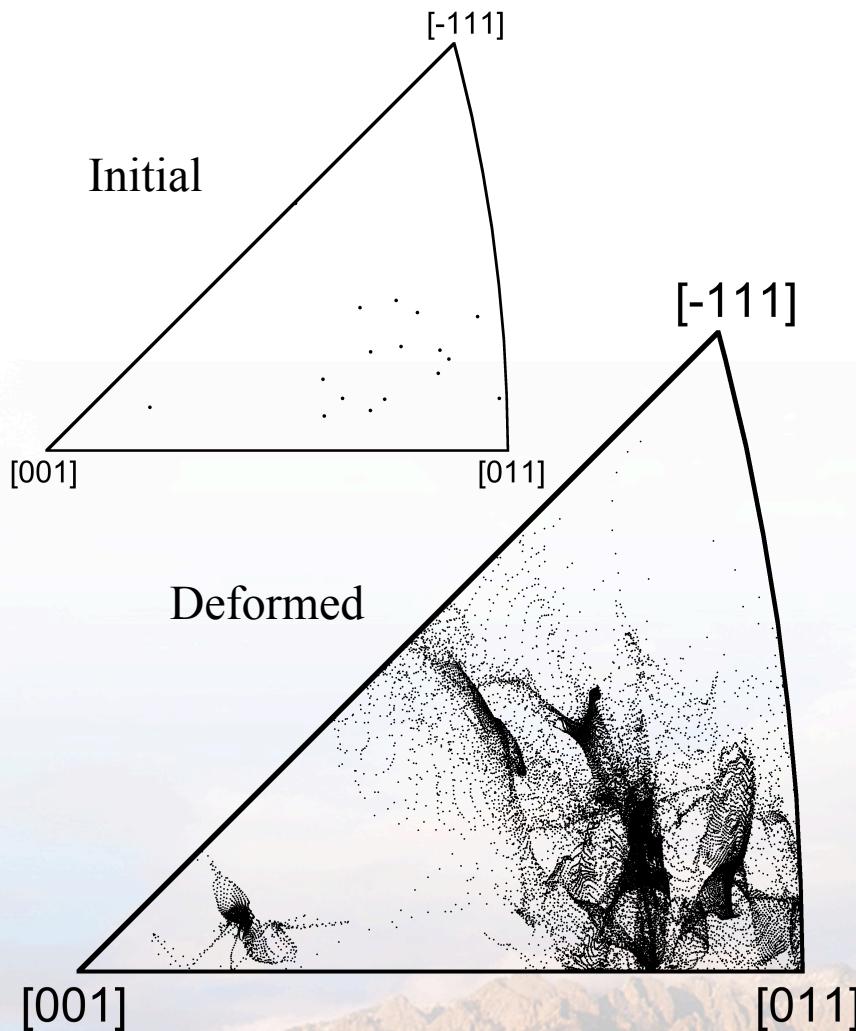


Medium Mesh
1 element through
thickness

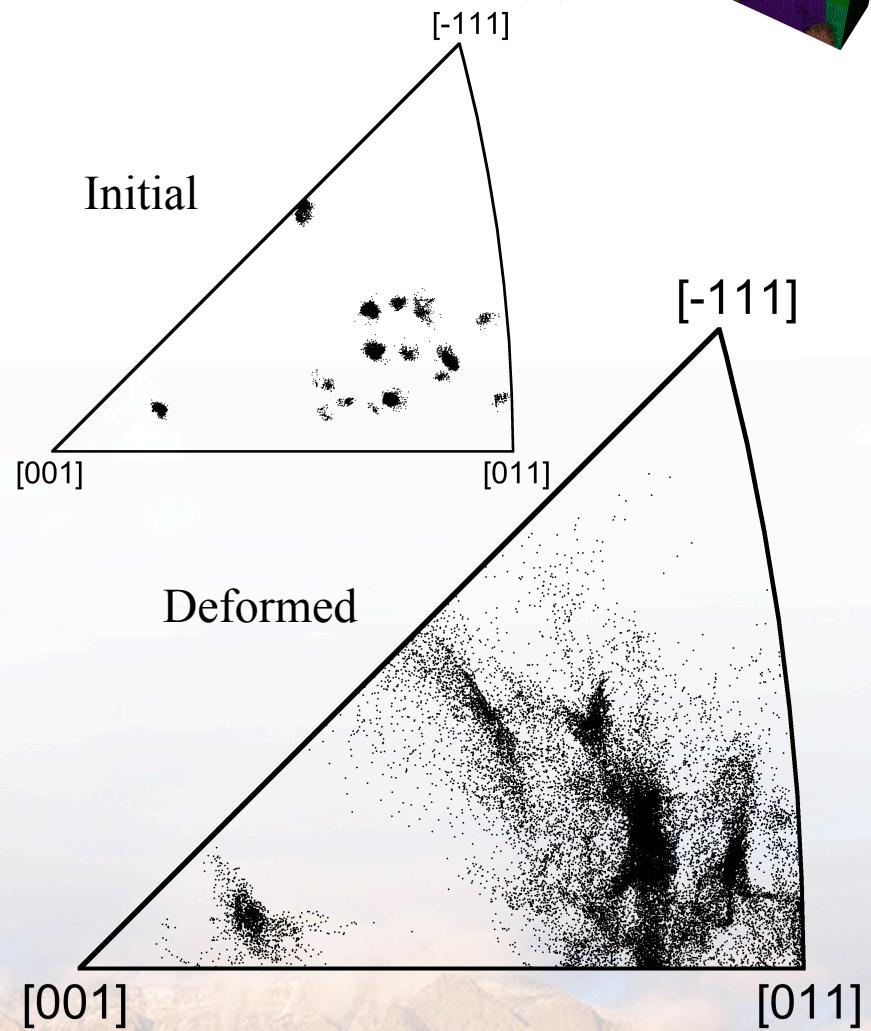


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Initial Orientations



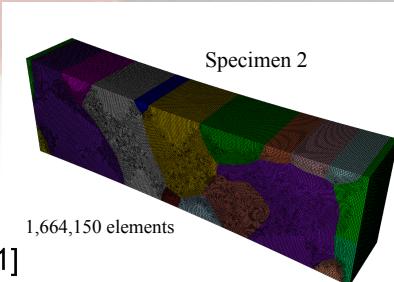
(a) Orientation per grain



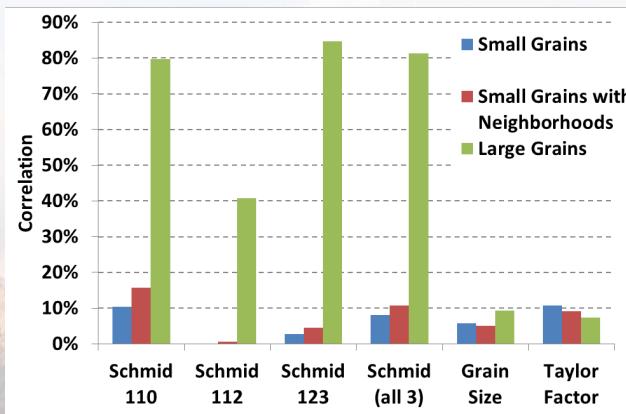
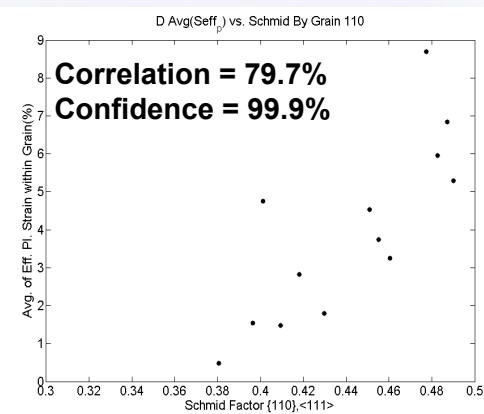
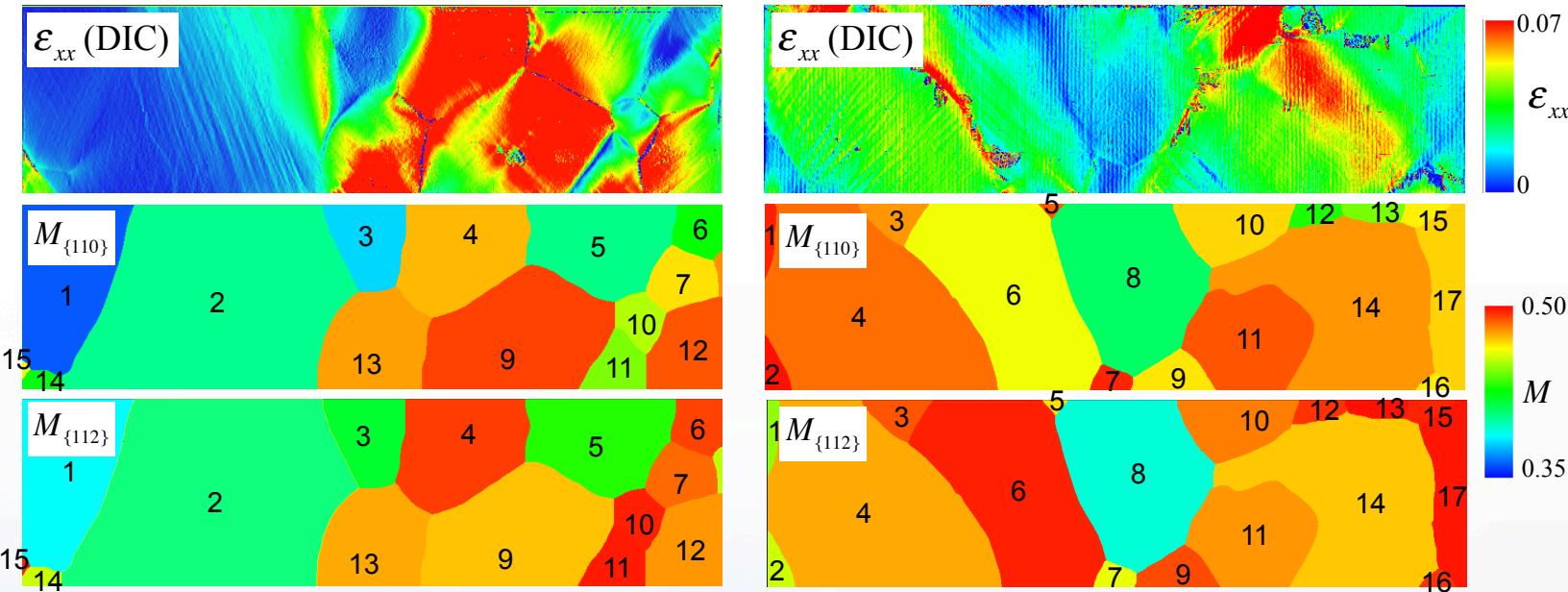
(b) Orientation per element



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Slip Planes

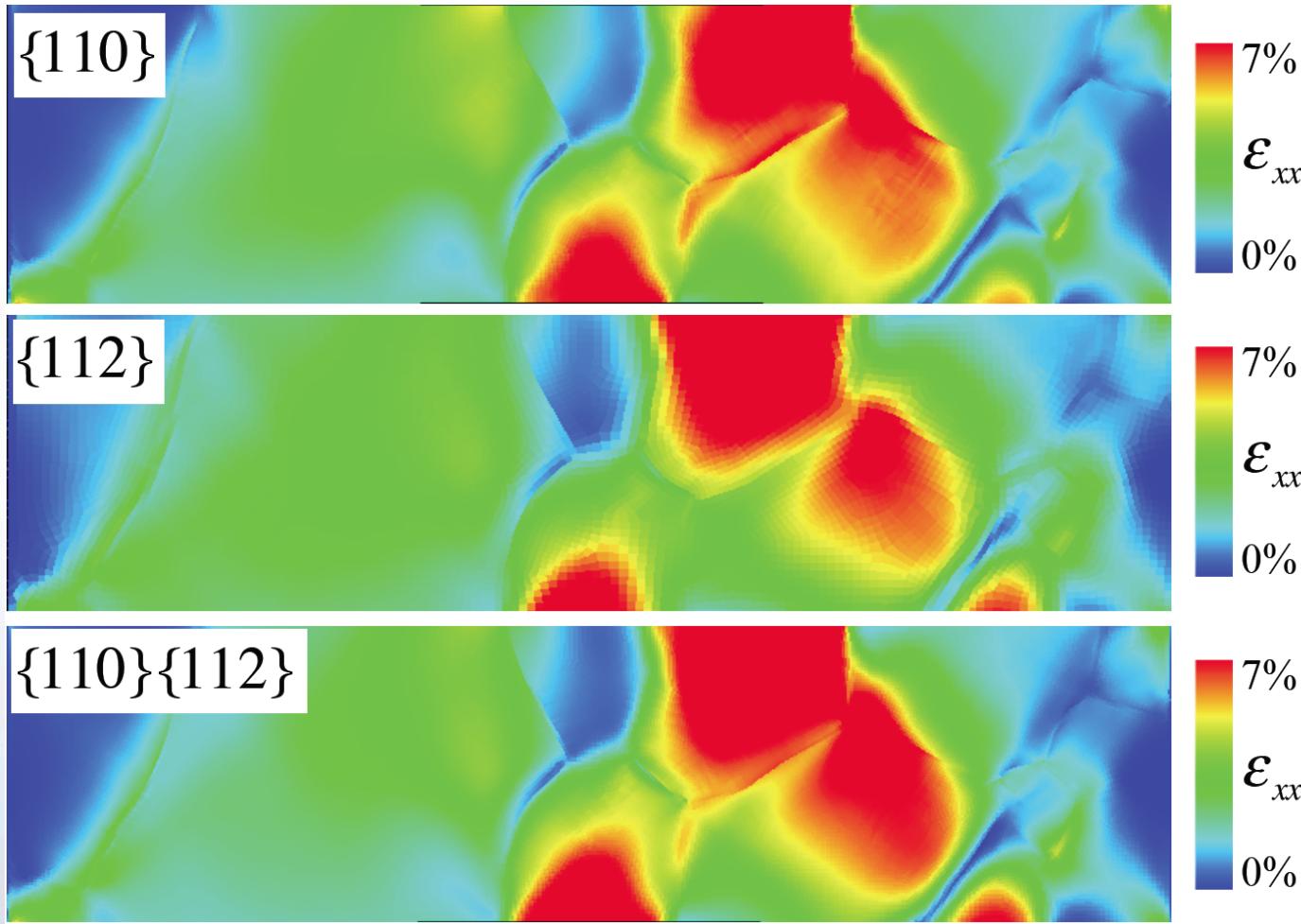


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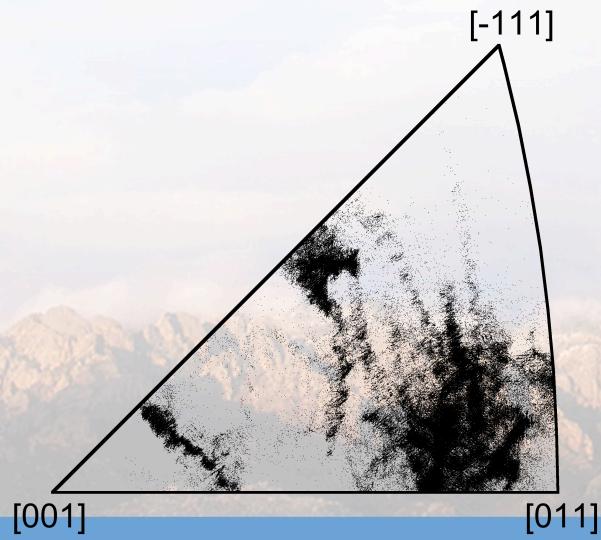
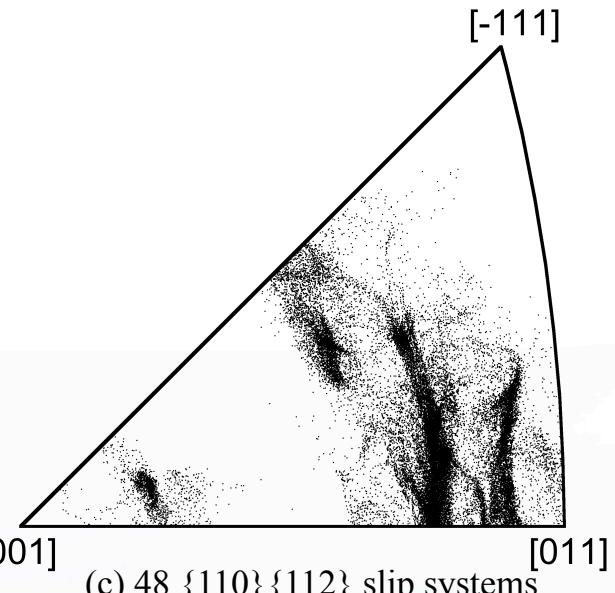
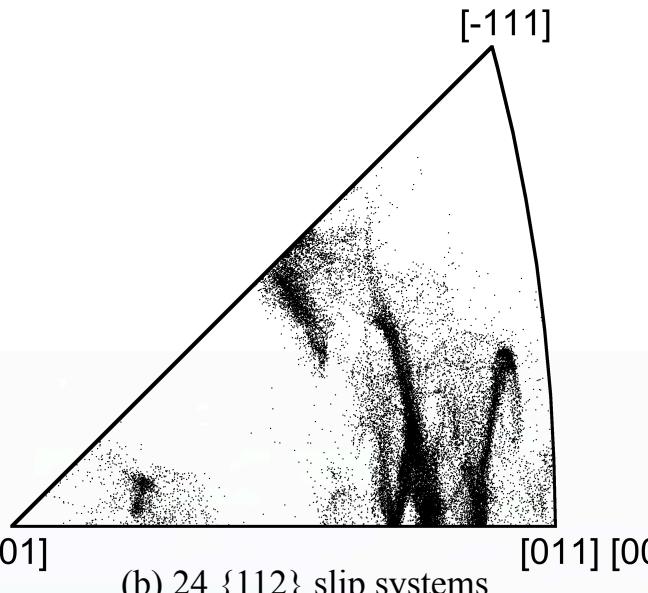
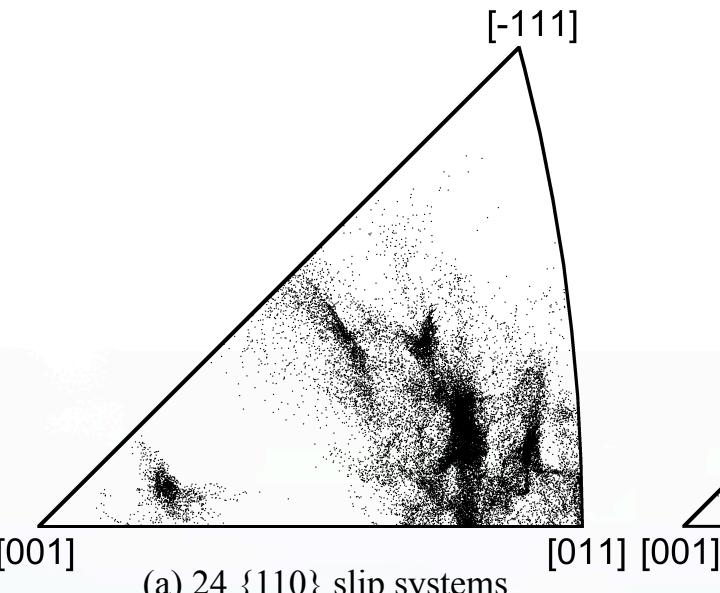


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Slip Planes – Strain Distributions

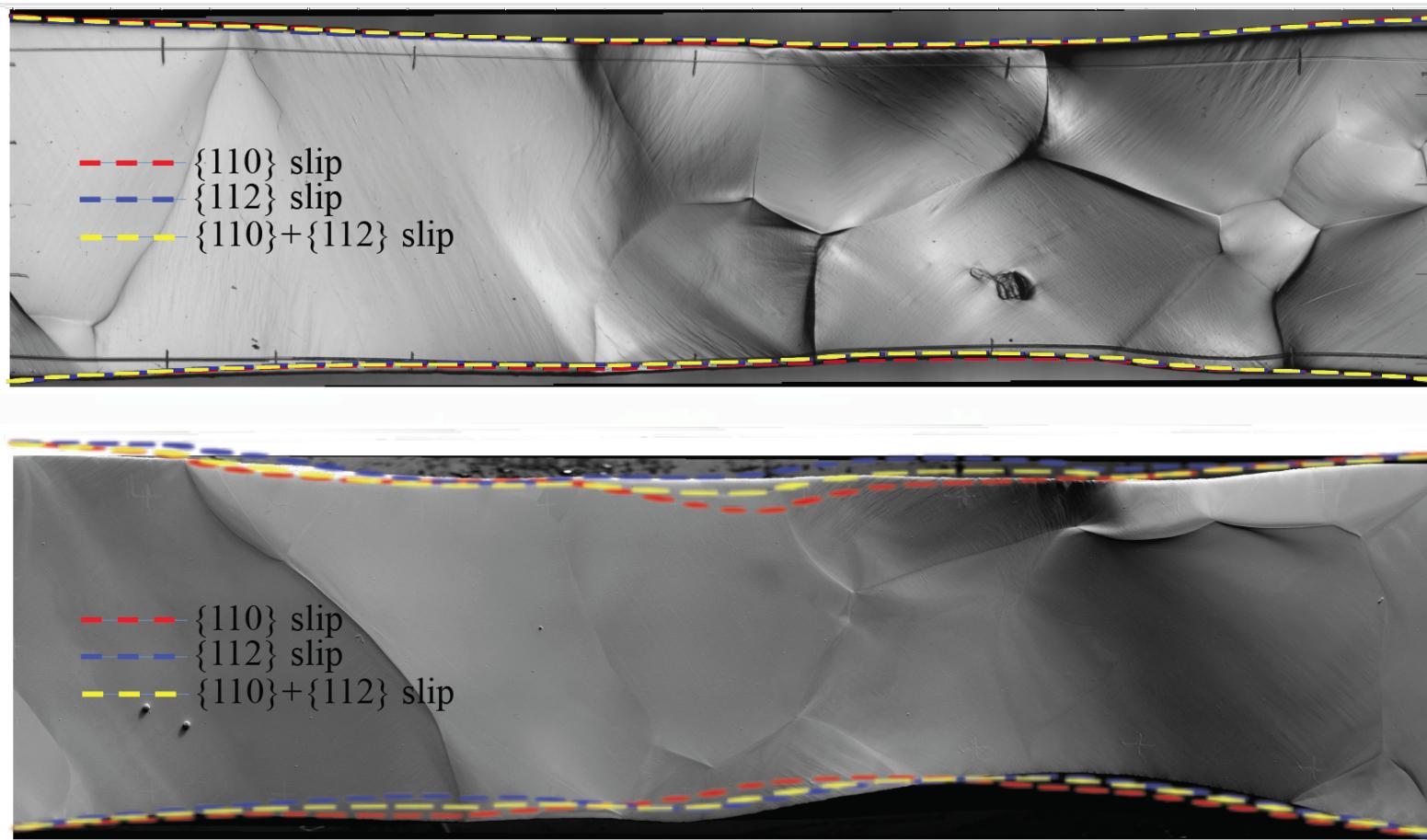


Texture Evolution



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Deformed Shapes



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Conclusions

- Numerous studies indicate good qualitative agreement – the “trends” are correct – but the quantitative details are often missing.
- CPFEM simulations appear to predict grain-scale strains accurately (relatively)
 - Relatively insensitive to mesh size & slip planes
 - Does depend on through thickness
- CPFEM simulations predict texture evolution moderately well
 - Missing some details
 - More mesh sensitive than strains
- Sub-grain deformations depend on many factors. Thus the details of deformation depend on the details of the experimentation and simulation.
- What level of agreement / validation can we reasonably expect from models of this class?
- More work is needed to “dot i’s and cross t’s.”

