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Title: The Role of the Structure of Grain Boundary Interfaces during Shock Loading

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The Role of the Structure of Grain Boundary Interfaces During Shock Loading

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In order to understand the role of interfaces in the shock tolerance of metals, three specific copper bi-crystal boundaries have been studied under shock loading and incipient spall conditions. These boundaries, two 001/111 boundaries and a 001/001 tilt boundary and their structures have been characterized prior to deformation using both electron back scattered diffraction (EBSD) and transmission electron microscopy (TEM) to obtain their axis/angle pair relationship and the structure at the boundary. This characterization has been utilized as input for MD simulations to examine in-situ dislocation/grain boundary interactions. These boundaries were then shocked at 2.5 and 10GPa in an 80mm gas gun and soft recovered. Post-mortem characterization, EBSD and TEM, has revealed that typical grain boundaries readily form damage during shock loading but the special boundaries ($\Sigma 3$) are resistant to failure. This is linked to differences in slip transmissibility across these types of boundaries.

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Overview

- Background and Goals
- Experimental Procedures
 - Shock loading and Spall Experiments
 - Analysis Techniques
- Analysis of Shock Loaded Cu Grain Boundaries
 - Substructure Damage
 - Grain Boundary Structure
- Analysis of Damaged Cu Grain Boundaries
- Conclusions



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Background and Goals

■ Hypotheses:

1. The atomic structure of an interface controls the absorption, emission, storage and annihilation of defects at that interface.
2. The cohesive strength/mechanical damage evolution behavior of a given interface structure may change at high strain rates

■ Our Goal:

- To show that the response of grain boundaries during shock loading is dependent on structure and not merely upon grain orientation

■ Our Method:

- Shock Cu multicrystals at different peak shock pressures and under different shock conditions
- Investigate the structure of the grain boundaries and substructure of the grains via electron backscatter diffraction (EBSD) and transmission electron microscopy (TEM)



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Side 3



Background and Goals

■ Why Study the Role of Interface Structure in Copper?

- Copper is a relatively simple, well characterized system
- A controlled sample allowed for a tight study of the role of grain orientation versus boundary structures to be probed
- One of the boundaries easily produced in copper is a $\Sigma 3$ boundary – a boundary that has also been investigated in irradiation environments and has proved to be resistant to irradiation damage
- Such a controlled specimen also allowed for a tight study of generalized and special boundaries

■ Why Pick the Mechanical Extreme of Shock?

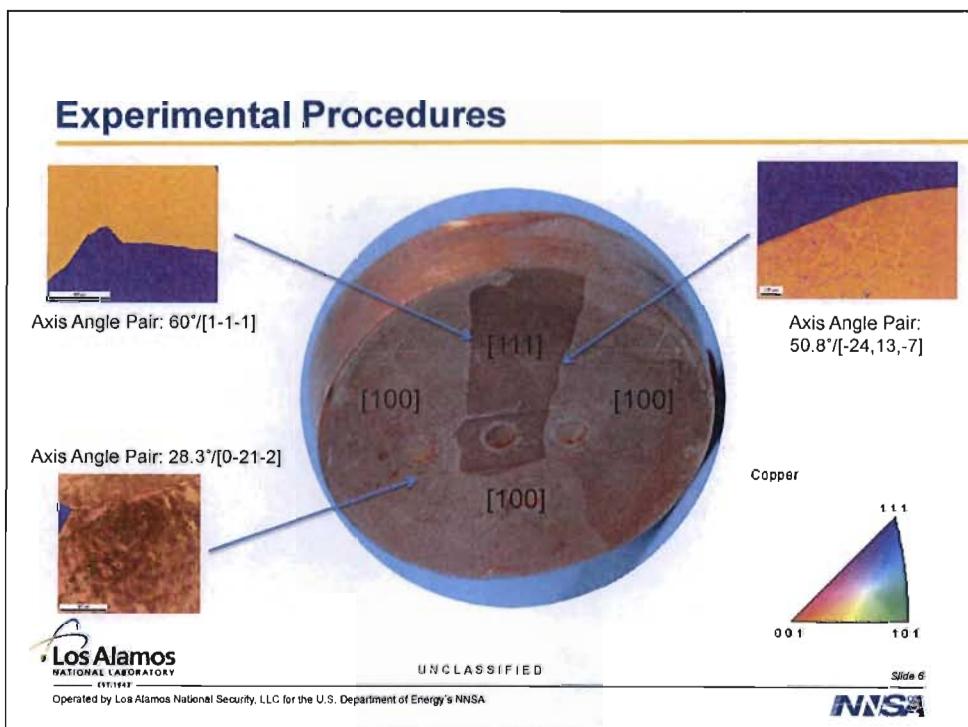
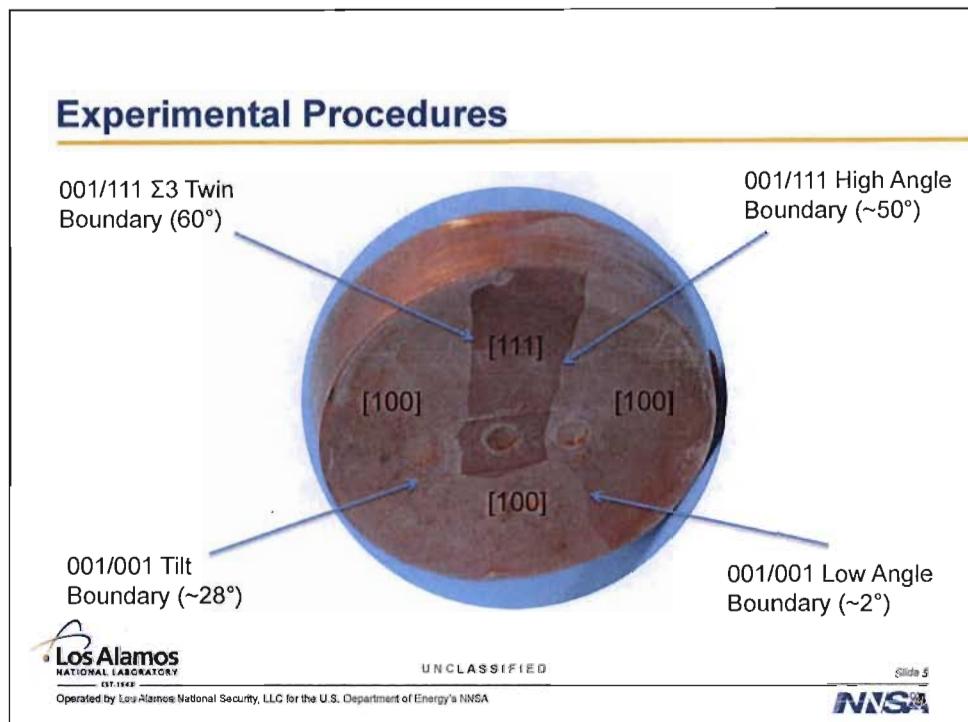
- Long history of shock physics research performed at LANL
- Produces large amounts of damage that can induce boundary failure
- Examines a strain rate that is directly relevant to MD



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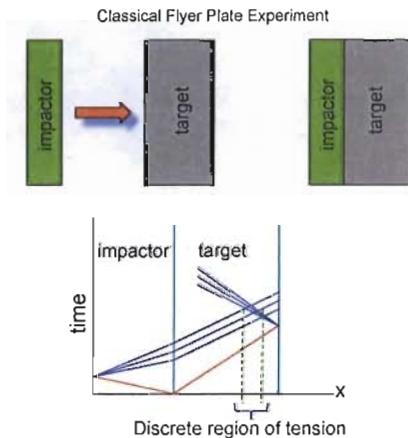
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Experimental Procedures

- Shock experiments performed using an 80 mm single stage gas gun
- Shock Loading Experiment**
 - Peak shock pressure of 10 GPa
 - Soft-recovered and sectioned for post-mortem TEM analysis
- Damage Experiment**
 - Peak shock pressure of 2.1 GPa
 - Pressure release waves interact within the sample
 - In situ pressure rise measurements taken
 - Soft-recovered and sectioned for post-mortem EBSD analysis



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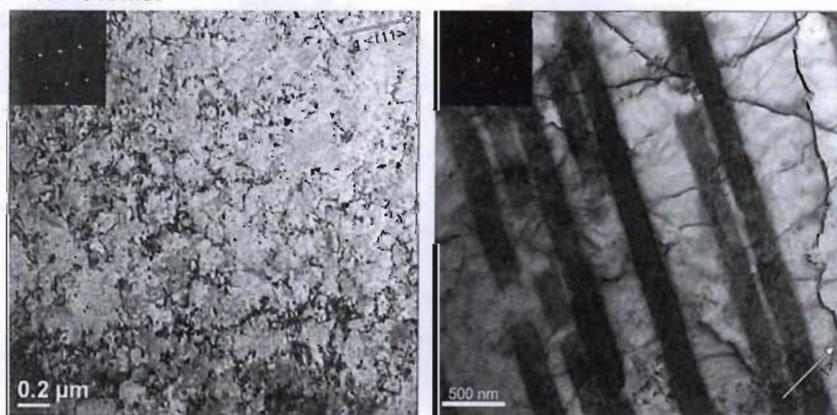
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Substructure of Shock Loaded Cu Grains

- 100 Grains:



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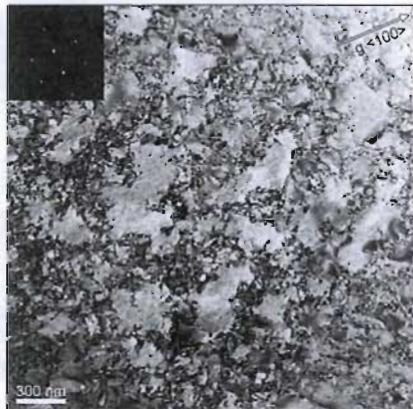
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Substructure of Shock Loaded Cu Grains

- 111 Grains:



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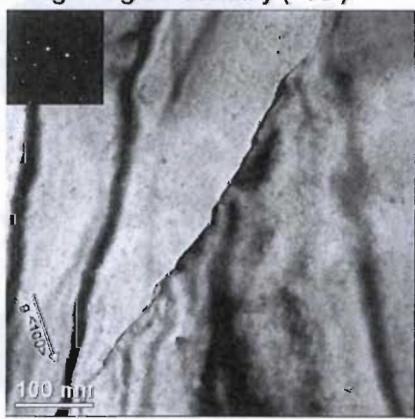
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Grain Boundary Structure of Shock Loaded Cu Grains

- High Angle Boundary (~50°)

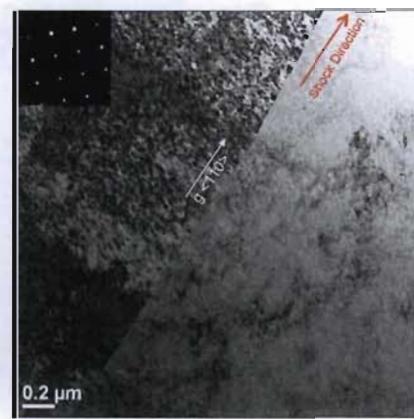


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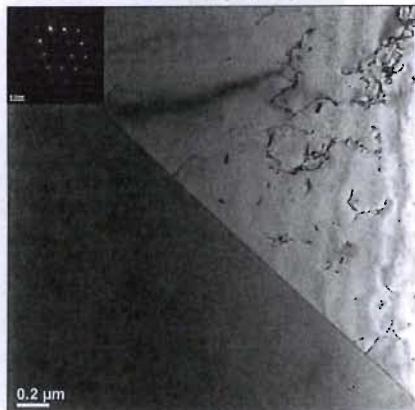
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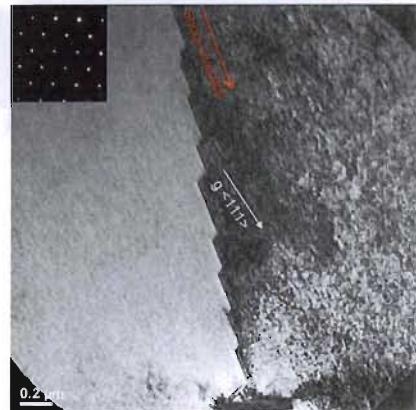


Grain Boundary Structure of Shock Loaded Cu Grains

- $\Sigma 3$ Twin Boundary (60°)



- Undeformed



- Shocked



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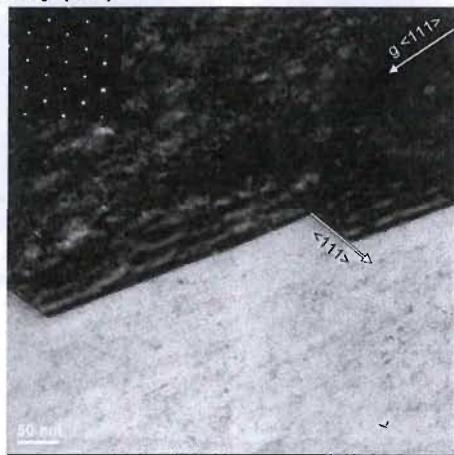
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Grain Boundary Structure of Shock Loaded Cu Grains

- $\Sigma 3$ Twin Boundary (60°)



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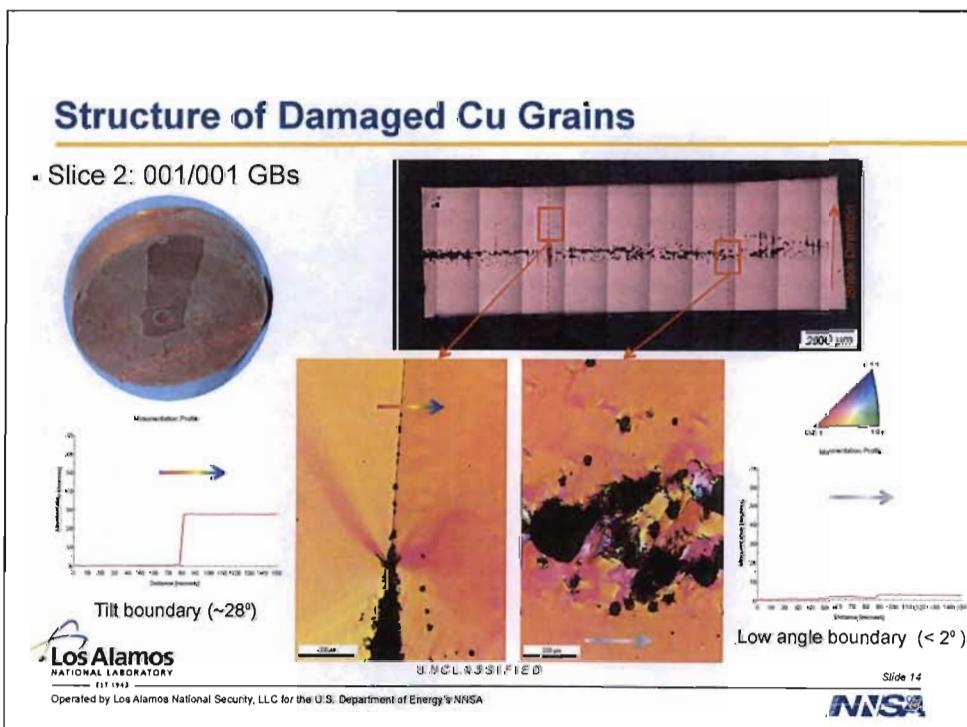
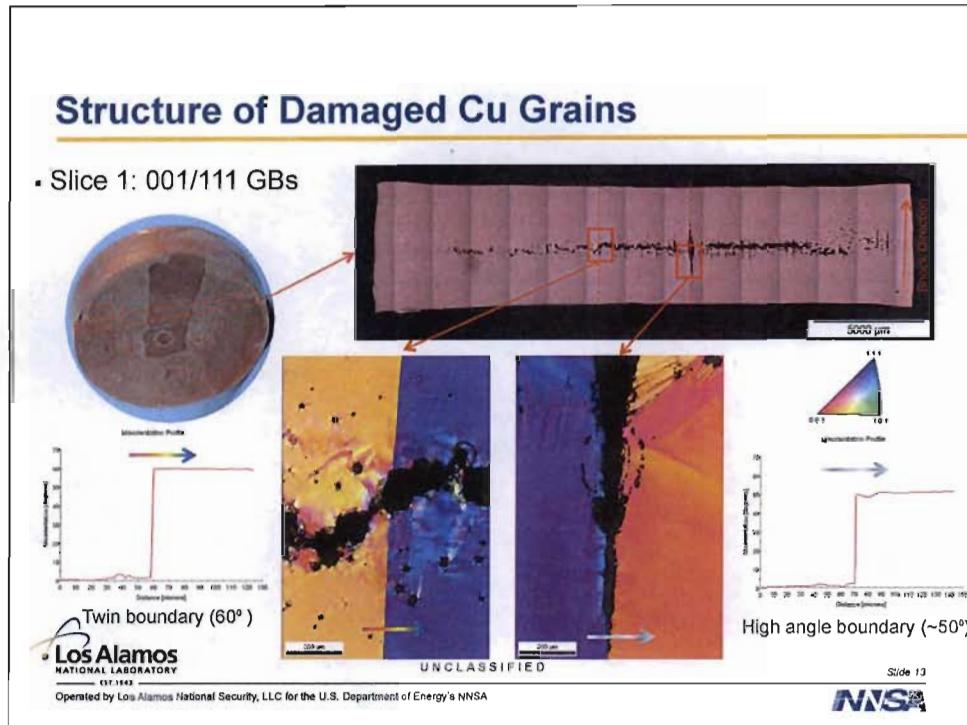
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Conclusions

- EBSD analysis shows that certain grain boundaries are resistant to damage and failure
 - Low angle ($<5^\circ$) and special GBs resist void nucleation under spall conditions
- TEM of shock loaded Cu specimens shows distinct jogging of the $\Sigma 3$ grain boundary along slip planes
 - May be an alternate stress release method, possibly explaining resistance of the $\Sigma 3$ boundary to failure
- Overall, results show that grain boundary structure, and not just grain orientation, strongly affect strength and damage evolution at grain boundary interfaces



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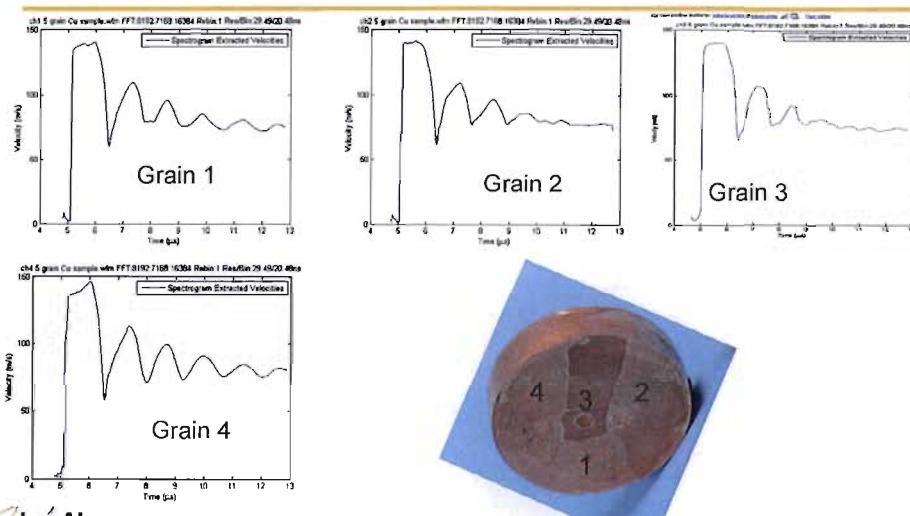
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Damage Experiments Measure Shock Velocity in Each Grain



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Simulation of the $\Sigma 3$ Boundary Reveals Details of the Propagating Shock Front and Boundary Motion



- Common neighbor analysis:
- Suggest partial dislocation mediated GB motion
- Particle velocity (shock direction):
- Two shock fronts have different propagation speeds
- Shock front in grain 1 is inclined to the piston velocity direction, whereas in grain 2 the shock front stays in the same direction



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