

29 progressive dehydration, reducing the hydration shell and favoring stronger sulfate complexation.
30 At 200°C, sulfate ligands reorganize around Yb^{3+} , shifting its geometry to a capped octahedron
31 (CN = 7). At 300°C, sulfate binding dominates, leading to structural reorganization that parallels
32 the onset of sulfate mineral precipitation, consistent with the retrograde solubility of REE sulfates.
33 These findings provide direct molecular-scale evidence that sulfate acts as both a transport and
34 deposition ligand, critically influencing REE mobility in geochemical environments. Our results
35 can also help to refine thermodynamic models of REE speciation in high-temperature
36 hydrothermal fluids and improve our understanding of REE ore formation processes in nature.

37 **Key words:** Hydrothermal Fluid, Yb^{3+} , sulfate EXAFS, AIMD, REE Sulfate

38 **1. Introduction:**

39 The demand for rare earth elements (REEs, a group of 17 elements including Sc, Y, and La to Lu)
40 has surged in recent years due to their critical roles in cutting-edge technologies, including
41 renewable energy, electric mobility, and high-strength alloys. This growing industrial dependency
42 has spurred worldwide efforts to identify and exploit alternative geological sources of REEs
43 (Atwood, 2013; Balaram, 2019; Chen et al., 2024; Zhao et al., 2024). Hydrothermal fluids play a
44 crucial role in forming economically significant REE ore deposits, as observed in locations such
45 as the Gallinas Mountains in New Mexico (United States), the Snowbird Deposit in Montana
46 (United States), the Pea Ridge in Missouri (United States), and Bayan Obo in Inner Mongolia
47 (China) (Fan et al., 2016; Harlov et al., 2016; Metz et al., 1985; Smith et al., 2000; Williams-Jones
48 et al., 2000). Over the past two decades, experimental and computational studies have advanced
49 our understanding of REE speciation in hydrothermal systems under elevated temperature and
50 pressure. Techniques such as in situ X-ray absorption spectroscopy (XAS) and *ab initio* molecular

51 dynamics (AIMD) simulations have been instrumental in elucidating the molecular-level
52 coordination environment of REEs, particularly their interaction with ligands such as chloride,
53 sulfate, and fluoride. Previous studies, including those by Mayanovic *et al.* on REE aqua and
54 chloride complexes (Anderson et al., 2002; Mayanovic et al., 2007, 2009a), which demonstrated
55 the efficacy of *in situ* spectroscopic techniques for characterizing REE hydration and ligand
56 exchange processes. Subsequent studies extended this approach to sulfate and fluoride complexes,
57 highlighting how ligand type, temperature, and pressure affect REE mobility (Guan et al., 2022b;
58 Guan et al., 2020; Louvel et al., 2022b; Migdisov and Williams-Jones, 2008).

59 Ligands such as Cl^- , CO_3^{2-} and SO_4^{2-} have been considered mainly as transporting ligands in
60 nature, while F^- and PO_4^{3-} have been considered as depositional ligands (Louvel et al., 2022a;
61 Migdisov et al., 2016). Among these, SO_4^{2-} was considered as dominant transporting ligand in
62 many REE deposits, such as Bayan Obo (China) (Lai and Yang, 2013), Maoniuping (China)
63 (Zheng and Liu, 2019) Mountain Pass (United States) depositions (Verplanck et al., 2016) and so
64 on. In hydrothermal fluid, the solubility of REE sulfates undergoes a unique retrograde dissolving
65 behavior: the solubility decreases with increasing temperature, enabling precipitation at elevated
66 temperatures. However, in the presence of prograde soluble alkali sulfates (e.g., Li_2SO_4 and
67 Na_2SO_4) in the hydrothermal fluid ($>250^\circ\text{C}$ and 90 MPa), the increased alkali sulfate solubility
68 may enhance the stability of REE sulfate complexes, especially $\text{REE}(\text{SO}_4)^{2-}$ and REESO_4^+ (Wan et
69 al., 2023). Furthermore, the recent discovery of liquid–liquid immiscibility (LLI) in REE sulfates
70 revealed the potential for $\text{REE}^{3+}\text{-SO}_4$ and $\text{REE}^{3+}\text{-HSO}_4$ complexes to become concentrated into a
71 dense liquid phase (DLP) and transport effectively (Wan et al., 2021). These findings underscore
72 the significant role of sulfate in transporting and enriching REEs in hydrothermal environments.

73 Despite recent advances, the molecular-level structural dynamics of REE sulfates under high-
74 pressure and high-temperature (high P–T) conditions remains poorly understood. A molecular-
75 scale understanding is crucial for understanding and properly interpreting speciation and
76 consequently transport and deposition mechanisms in various hydrothermal environments.
77 Ytterbium (Yb) is selected as the focus of this study because it represents the heavy rare earth
78 element (HREE) group and exhibits a significantly smaller ionic radius compared to previously
79 studied REEs such as La(III), Nd(III), Sm(III) or Er(III) (Migdisov et al., 2006; Migdisov and
80 Williams-Jones, 2008). This allows us to investigate subtle speciation and structural variations
81 driven by lanthanide contraction induced ionic size effects, particularly under high P-T conditions
82 relevant to natural hydrothermal fluids. Furthermore, previous studies on REE–sulfate complexes
83 of Nd(III), Sm(III), and Er(III), have primarily relied on UV-Vis spectroscopy to infer coordination
84 environments (Migdisov et al., 2006; Migdisov and Williams-Jones, 2008). But the UV-Vis lacks
85 the spatial resolution needed to directly determine coordination numbers and bond distances. In
86 contrast, the present study employs synchrotron-based extended X-ray absorption fine structure
87 (EXAFS) spectroscopy, which provides direct, molecular-level insights into the coordination
88 geometry of Yb–SO₄ complexes. But to our knowledge, no prior EXAFS study has specifically
89 examined any Yb–sulfate complexes, despite EXAFS has been widely used to investigate REE
90 aqua, chloride, and carbonate complexes (Guan et al., 2022a; Guan et al., 2020; Louvel et al.,
91 2022a; Migdisov et al., 2019; Migdisov et al., 2016; Migdisov and Williams-Jones, 2008).
92 However, Yb is a strategically critical element due to its widespread applications in modern
93 technologies, including infrared lasers (Takasu et al., 2004), rechargeable batteries (Kim et al.,
94 2021), chemical reducing agents (Girard et al., 1980), and fiber optics (Pask et al., 1995). As such,
95 understanding the speciation and stability of Yb complexes in sulfate-rich environments is not only

96 of fundamental geochemical interest but also essential for guiding exploration, extraction, and
97 processing strategies for HREE resources.

98 In hydrothermal fluid, the primary reactions governing REE sulfate complexation in solution
99 are as follows (Migdisov and Williams-Jones, 2008):



103 Key question remains unanswered, which of these reactions is the dominating reaction in
104 HREE sulfate fluid, in other words, it is unclear whether Yb^{3+} preferentially binds mono-sulfate or
105 di-sulfate species. Additionally, how coordination parameters such as bond angles, bond distances,
106 and binding modes vary under high P–T conditions also remains poorly constrained. To address
107 these knowledge gaps, EXAFS spectroscopy coupled with hydrothermal diamond anvil cell
108 (HDAC). However, the integration of HDAC with energy-scanning spectroscopic techniques has
109 long presented challenges, including distortions in XAS spectra caused by Bragg reflections,
110 limited sample thickness, and significant x-ray absorption by the diamond at lower energies
111 (Ingalls et al., 1980b; Ohsumi et al., 1986; Sueno et al., 1986). These issues often lead to
112 overwhelming EXAFS signals into noises and limit the number of EXAFS studies on high P-T
113 hydrothermal fluids. While several artifacts mitigation approaches have been applied, but many of
114 these approaches require specialized equipment (e.g. laser drilled diamonds) (Mayanovic et al.,
115 2007, 2009b; Mayanovic et al., 2002, 2003) or involve hazardous materials (e.g., beryllium gaskets)
116 (Hu et al., 1994; Kaindl et al., 1988). In this study, we adopted a simple multi-angle scanning
117 approach to eliminate Bragg glitches. By rotating the liquid cell up to $\pm 4^\circ$ relative to the incident
118 beam direction, the Bragg peaks of diamonds in energy space will be shifted, allowing glitch-

119 containing regions in one scan to be replaced with glitch-free data from another (as illustrated in
120 **Figure. 1**). Besides the diffraction noise issue, the iterative shell-by-shell of EXAFS fitting
121 procedure can be prone to overfitting due to the large number of adjustable parameters, including
122 energy shift (ΔE), change in interatomic distances (ΔR), coordination number (N_i), absorber–back
123 scatterer distance (R_i), and Debye–Waller factor (σ^2), all of which contribute to the normalized
124 EXAFS spectrum as shown in Equation (4):

$$125 \quad \chi(k) = \sum_R \frac{S_0^2 CN}{kR^2} |f(k)| e^{-\frac{2R}{\lambda(k)}} e^{-2\sigma^2 k^2} \times \sin(2kR + \phi(k)) \quad (4)$$

126 To mitigate this limitation, we integrated AIMD simulations, which provide statistically
127 averaged coordination structures and allow for the generation of theoretical EXAFS spectra
128 (Kerisit and Prange, 2019; Prange et al., 2021). These AIMD-derived spectra are directly compared
129 with experimental data to validate the fitted models, thereby enhancing the reliability and
130 robustness of the structural interpretations. In addition to resolving potential overfitting issues,
131 AIMD offers complementary insights that EXAFS alone cannot provide. These include the
132 denticity of sulfate ligands—allowing for the distinction between mono- and bidentate
133 coordination—along with information on dynamic ligand exchange behavior and qualitative
134 assessments of bonding character, such as π -interactions and partial covalency.

135 By integrating *in situ* EXAFS measurements with AIMD simulations, this study provides first
136 molecular-level insights into the structural evolution of Yb^{3+} sulfate complexes under
137 hydrothermal conditions. The results reveal thermally induced dehydration, structural
138 reorganization, and systematic changes in how strongly sulfate binds to Yb^{3+} with temperature,
139 adding mechanistic insight into the role of sulfate in REE transport and deposition in geological
140 environments.

141 **2. Material and methods:**

142 **2.1. The hydrothermal diamond anvil cell**

143 The HDAC is a versatile instrument designed for high P–T studies of hydrothermal solutions
144 (Bassett et al., 1994). It features two opposing 1/8-carat diamond anvils mounted on tungsten
145 carbide seats, which are affixed to upper and lower stainless-steel platens. These platens are drawn
146 together by three screws and precisely aligned by three guide rods, ensuring stability and
147 reproducibility under extreme P–T conditions. Due to the small sample size and the exceptional
148 thermal conductivity of diamond, temperature gradients within the HDAC are minimal in both
149 vertical and horizontal orientations. As a result, temperature corrections during experiments are
150 typically within ± 5 °C. The HDAC is mounted on a rotation stage, allowing for rotation about two
151 orthogonal axes perpendicular to the incident X-ray beam to optimize experimental geometry. The
152 sample, comprising a fluid phase (solution and vapor bubble), is confined within a chamber created
153 by a 500 μm diameter hole in a 50 μm thick rhenium gasket. The diamond anvils, with 1 mm
154 diameter flat faces, compress the gasket to seal the sample chamber and maintain the desired
155 pressure.

156 **2.2. Sample preparation**

157 Ytterbium(III) sulfate octahydrate ($\text{Yb}_2(\text{SO}_4)_3 \cdot 8\text{H}_2\text{O}$, 99.9% purity) was purchased from Thermo
158 Fisher Scientific Inc. (United States). A 0.05 M solution was prepared on-site at the Advanced
159 Photon Source (APS) synchrotron facility by dissolving $\text{Yb}_2(\text{SO}_4)_3 \cdot 8\text{H}_2\text{O}$ in deionized water. The
160 solution samples were then loaded into the HDAC chamber using a micropipette under a binocular
161 microscope. While some overflow of the sample solution is often unavoidable during sample
162 loading, it does not affect the X-ray measurements.

163 2.3. XAS data acquisition and processing

164 XAS data at the Yb L₃-edge (8944 eV) was collected on the 20-BM-B beamline at APS, Argonne
165 National Laboratory (**Figure 1a**). The measurements were performed in transmission mode, with
166 the incident and transmitted X-ray intensities detected using N₂-gas-filled ionization chambers. To
167 minimize stray radiation around the focused X-ray beam, a pinhole aperture was positioned in
168 front of the incident-beam ionization chamber. The Bragg reflections from diamonds exceed the
169 amplitude of EXAFS oscillations, significantly degrading the quality of the data required for
170 accurate structural analysis (Ingalls et al., 1980a; Ohsumi et al., 1986; Sueno et al., 1986). Various
171 strategies have been proposed to mitigate these artifacts, including measuring multiple spectra at
172 different cell orientations (Freund et al., 1989; Ingalls et al., 1980a; Liu et al., 2020), drilling holes
173 in the diamond for measurements by fluorescence mode (Mayanovic et al., 2007, 2009b;
174 Mayanovic et al., 2002, 2003), using polycrystalline boron carbide (B₄C) anvils (Freund et al.,
175 1989; Sueno et al., 1986), employing low atomic number Beryllium gasket (Hu et al., 1994; Kaindl
176 et al., 1988), and using large-volume multianvil apparatus (Kaindl et al., 1988; Katayama et al.,
177 1997; Liu et al., 2020; Louvel et al., 2022a). In this study, we adopted a practical multi-angle
178 scanning approach to eliminate Bragg glitches and avoided involvement of specialized apparatus
179 and toxic gaskets. At each P–T set, up to eight scans were acquired at different angles up to $\pm 4^\circ$
180 relative to the incident beam direction, with each spectrum requiring approximately 30 minutes.
181 As shown in **Figure 1**, in instances where Bragg diffraction peaks were observed, the sample cell
182 was rotated by up to $\pm 4^\circ$ to shift the diffraction peaks in energy space, and the spectral regions
183 containing Bragg peaks were replaced with corresponding regions devoid of Bragg peaks, ensuring
184 high-quality data for analysis. After Bragg peak removal, the corrected spectra obtained at different
185 detector angles were merged into a single high-quality dataset. This merging process averaged out

186 angle-dependent variations and reduced noise, resulting in a robust and artifact-free spectrum
187 suitable for k -space and R -space analysis. To further validate the processed synchrotron data,
188 comparative XAFS measurements were performed using a benchtop X-ray spectrometer. The
189 agreement between the benchtop and synchrotron datasets confirmed the reliability of the
190 processed synchrotron data, enabling its use for detailed structural analysis of the Yb-SO₄ system
191 under high P-T conditions.

192 In previous studies, DFT-derived geometries were often employed to provide initial guesses for
193 EXAFS shell-by-shell fitting (Xu et al., 2024). This approach improves fitting reliability compared
194 to purely experimental fitting, as DFT-optimized structures generally offer a ground state stable
195 geometry to act as starting point. However, a few DFT-optimized geometry may not necessarily
196 reflect the true structural ensemble present under experimental conditions, especially in dynamic,
197 high-temperature aqueous systems. To address these limitations, a more rigorous and statistically
198 robust approach was adopted in this work. Instead of relying on static DFT-derived geometries, a
199 full ensemble of equilibrium configurations from AIMD trajectories was used to compute
200 theoretical EXAFS spectra (Kerisit et al., 2016; Kerisit et al., 2019; Kerisit and Prange, 2020; Latta
201 et al., 2025; McBriarty et al., 2018). This method captures thermal and structural disorder and
202 provides a direct, unbiased link between simulation and experiments. The theoretical spectra were
203 computed from these trajectories using ab initio multiple scattering calculations and applied a
204 systematic energy shift ($\Delta E = 0\text{--}10$ eV) to account for edge alignment uncertainties. The resulting
205 theoretical spectra were then directly compared to the experimental fitted data to assess the degree
206 of agreement. This effectively serves as a sanity check and provides a “double-blind” validation—
207 ensuring that the EXAFS signal could be reproduced by the underlying AIMD ensemble, which

208 enhances the physical meaning and reliability of the EXAFS interpretation and strengthens
209 confidence in the structural models derived from simulation.

210 Bench-top EXAFS measurements were collected at room temperature (RT) as a sanity check
211 using an easyXAFS300 instrument (easyXAFS, WA) located at Pacific Northwest National
212 Laboratory (PNNL). The spectra were obtained with a Si(422) spherically bent crystal analyzer
213 and a Mo anode X-ray tube operating at 500 W. Data were corrected for detector deadtime, and
214 the energy was calibrated using a Cr foil standard. Additional spectra were collected with a Ge(422)
215 spherically bent crystal analyzer and a W anode X-ray tube operating at 100 W. Details about these
216 instruments can be found in Seidler *et al.* (Seidler et al., 2014) The data processing and modeling
217 was performed in the Athena and Artemis programs (version 0.9.26).(Ravel and Newville, 2005)
218 The spectra were collected using a Si(551) spherically-bent crystal analyzer and Mo anode x-ray
219 tube operating at 30 kV voltage and 25 mA current. The silicon drift detector deadtime was kept
220 below 30%. The energy calibration of the instrument was done with Cu K-edge (8979 eV). A
221 spectrum collected on a copper foil standard (procured from Exafs materials, CA) was aligned to
222 the one collected at NSLS X11A and retrieved from the Hephaestus database (Ravel and Newville,
223 2005). Theta-to-energy correction was applied to all the spectra, using theta-to-energy nonlinear
224 shift of the copper standard foil. All the spectra were deadtime corrected. XAS data collection
225 was carried out using an alternating manner between the sample (I_t) and blank (I_0) to account for
226 possible effects of minor beam instabilities with time. A liquid cell, custom made for the
227 easyXAFS300 instrument, was used for the 0.05 M Yb L₃-edge XANES and EXAFS data
228 collection. A small amount of the sample was injected using a syringe attached to one of the PEEK
229 capillary tubes via a PEEK Lure-Lok adapter while the gas head-space was being released through
230 the other tube. The liquid cell had an aperture of 1 cm, flat and parallel x-ray windows (polyimide

231 or Kapton windows) for uniform pathlength that were sealed by a compression mechanism.
232 Absorption length was controlled by a spacer with a thickness of 1.5 mm. The data processing and
233 modeling was performed in the Athena and Artemis programs (version 0.9.26) (Ravel and
234 Newville, 2005).

235 **2.4. Computational methods**

236 Density functional theory (DFT) (Becke, 2014) and *ab initio* molecular dynamics (AIMD)
237 simulations were performed with the pseudopotential plane-wave NWPW module (Bylaska et al.,
238 2011) implemented in the NWChem software package (Apra et al., 2020; Valiev et al., 2010). The
239 Perdew–Burke–Ernzerhof (PBE) (Perdew et al., 1996) functional was used to account for the
240 exchange correlation energy. Long-range dispersion forces were included in our calculations using
241 the Grimme DFT-D3 method (Grimme et al., 2011). In our plane-wave calculations, the valence
242 electron interactions with the atomic core were approximated with generalized norm-conserving
243 Hamann pseudopotentials (Hamann, 1989) for H, O, and S. These pseudopotentials were
244 constructed using the following core radii: $r_{cs} = r_{cp} = 0.8$ a.u. for H; $r_{cs} = r_{cp} = r_{cd} = 0.7$ a.u. for O;
245 and $r_{cs} = 0.843$ a.u., $r_{cp} = r_{cd} = 0.960$ a.u. for S. Norm-conserving Troullier-Martins
246 pseudopotentials (Troullier and Martins, 1991), which contain 4f, 5s, 5p, 5d, 6s and 6p projectors,
247 were applied for Yb. The following core radii were used to generate these pseudopotentials: $r_{cs} =$
248 1.618 a.u., $r_{cp} = 1.828$ a.u., $r_{cd} = 1.213$ a.u., and $r_{cf} = 1.403$ a.u.. All the pseudopotentials were
249 modified to the separable form suggested by Kleinman and Bylander (Kleinman and Bylander,
250 1982). Unrestricted calculations were performed since all the systems were open-shell. The
251 electronic wavefunctions were expanded using a plane-wave basis with periodic boundary
252 conditions at the Γ -point with a wavefunction cutoff energy of 544 eV and a density cutoff energy
253 of 1088 eV.

254 In AIMD, the system was propagated in time using the Car-Parrinello molecular dynamics
255 (CPMD) scheme (Car and Parrinello, 1985). The electronic density is described by Kohn-Sham
256 orbitals within the framework of density functional theory (DFT), and these orbitals are linear
257 combination of a plane-wave basis set. In this work, the simulations were conducted using the
258 canonical ensemble (NVT), with the density of the solution chosen according to the equation of
259 state for NaCl(aq) at similar chloride concentrations and the target temperature and pressure
260 (Driesner, 2007; Driesner and Heinrich, 2007). Each box is cubic in a size by $15\text{\AA} \times 15\text{\AA} \times 15\text{\AA}$,
261 and each box contains one Yb atom, two SO_4 ligands, 113 H_2O molecules, and it was found in
262 sulfate solution of Na_2SO_4 and K_2SO_4 that the density of the sulfate solution changes only very
263 little from RT to 450K (Obšil et al., 1997), hence the box size was not changed during the
264 simulation at different temperatures. The initial box was created by Packmol (Martínez et al.,
265 2009), by placing molecules in defined regions while avoiding atomic overlaps. This ensures a
266 physically reasonable starting configuration, especially for multicomponent or disordered systems
267 like solutions or confined fluids. The generated structure serves as a stable starting point for energy
268 minimization and equilibration in MD simulations. The trajectories of the simulation are shown in
269 Figure S1. Furthermore, in our simulations, we employed a fictitious orbital mass of 600.0 atomic
270 units (au), a standard choice that ensures numerical stability and minimizes artificial energy
271 exchange between the ionic and electronic subsystems. The equations of motion were integrated
272 using the position Verlet algorithm (Verlet, 1967) with a time step of 0.12 fs. All hydrogen atoms
273 were replaced by deuterium to facilitate integration. Temperature control was achieved using the
274 Nosé-Hoover chain thermostat (Blöchl and Parrinello, 1992; Hoover, 1985; Nosé, 1984) for both
275 ions (300 K, 373 K, and 473 K) and electrons (1200 K), and periodic boundary conditions were
276 applied to eliminate surface effects. Each MD simulation was run for 16–30 picoseconds (ps) to

277 ensure sufficient statistical accuracy. The time-averaged stoichiometric and geometric data were
278 obtained using the Visual Molecular Dynamics (VMD) software (Humphrey et al., 1996).
279 Coordination numbers were derived from the radial distribution functions (RDF), with errors
280 estimated from the differences in RDF integral values at distances of 2.75 Å and 3.25 Å for Yb-O
281 and 2.9 Å and 3.2 Å for Yb-S. The energy and temperature convergency of AIMD results can be
282 found at **Figure S2 - S4**.

283 **3. Results**

284 **3.1. Validation of synchrotron EXAFS data**

285 During EXAFS data processing, we employed a spectral correction approach to remove Bragg
286 peaks introduced by the diamond anvils during synchrotron experiments (as described in the
287 methodology section and shown in **Figure 1**). These Bragg peaks could potentially distort the
288 absorption spectra that are critical for EXAFS analysis. To validate our processed synchrotron data,
289 we conducted an RT experiment using the same sample in a liquid cell with a benchtop
290 easyXAFS300 instrument (easyXAFS, WA). Although this liquid cell cannot sustain the high
291 pressures achievable with HDACs, it effectively eliminates Bragg peaks associated with diamond
292 anvils. A comparison of the EXAFS spectra obtained from the synchrotron and the benchtop
293 instrument in k -space (**Figure 1C**) reveals consistent oscillation patterns across the k -range of 2
294 to 7 Å⁻¹, with minimal deviations in amplitude and phase. This consistency indicates that no
295 significant artifacts were introduced during the Bragg peak removal process. Slight variations in
296 intensity at higher k -values are likely due to differences in instrumental resolution or noise levels,
297 as the synchrotron provides higher energy resolution compared to the benchtop instrument.
298 Nevertheless, the overall agreement underscores the robustness of the data validation process and

299 confirms that the synchrotron EXAFS spectra can be reliably used for studying the Yb-SO₄ system
 300 under high-pressure and high-temperature conditions.

301

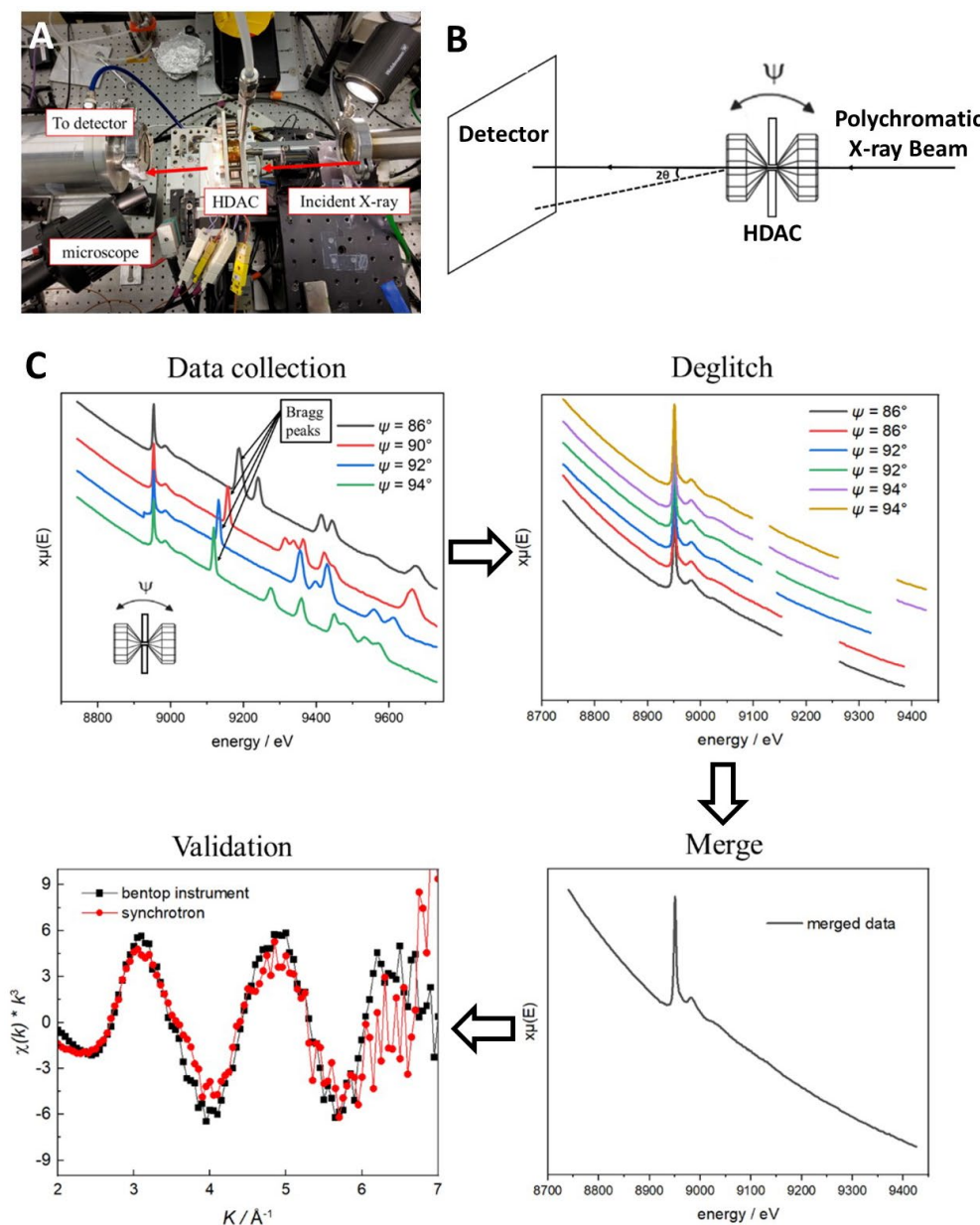


Figure 1. (A) Experimental setup for EXAFS measurements at APS. The HDAC is mounted on a rotation stage. The incident X-ray beam passes through the HDAC and the sample, and the transmitted beam is directed to the detector. A microscope is used for alignment and monitoring the sample within the HDAC. (B) Schematic of the HDAC and its rotation during EXAFS measurements. The detector angle (ψ) varies to minimize artifacts caused by Bragg peaks from

the diamond anvils. (C): Workflow for EXAFS data processing and validation. First step: Raw EXAFS spectra collected at various detector angles at RT ($\psi=86^\circ$ to 94°) showing Bragg peaks caused by diamond anvils. Second step: Corrected spectra after Bragg peak removal, demonstrating clean absorption profiles across the energy range. Third step: Merged EXAFS spectra combining data from multiple angles to reduce noise and ensure spectral consistency. Fourth step: Validation of synchrotron data by comparison with benchtop EXAFS spectra in k -space, showing consistent oscillation patterns across the k -range ($2-7 \text{ \AA}^{-1}$).

302

303 3.2. Local atomic structure and coordination dynamics of Yb^{3+} from experiments

304 The experimental results from RT to 200°C , fitting and simulated spectra are shown in **Figure 2**
 305 and **Figure 3**, and the fitted results are summarized in **Table 1**. At RT, the experimental k -space
 306 data agreed well with the simulated spectrum. The R -space analysis shows distinct peaks, with the
 307 first-shell Yb-O peak appearing at $\sim 2.33 \text{ \AA}$, corresponding to a coordination number of $\sim 8.4-8.5$.
 308 Indicated by the AIMD simulation, this shell consists of five water molecules and three sulfate
 309 oxygen atoms, with one sulfate ligand exhibited a bidentate mode. The sharpness of the peaks
 310 indicates minimal thermal disorder, supported by the low σ^2 value of 0.006 \AA^2 for the Yb-O shell.
 311 The Yb-S peak at $\sim 3.64 \text{ \AA}$ corresponds to a coordination number of ~ 2.5 , suggesting two sulfate
 312 ligand binding with Yb^{3+} . The σ^2 value for Yb-S at RT is also relatively low ($0.006-0.014 \text{ \AA}^2$),
 313 highlighting the rigidity of the sulfate coordination shell at ambient conditions.

314 **Table1.** Curvefit Parameters for Yb L-III edge EXAFS from RT to 200°C

Temperature	Path	CN #	R/ \AA	$\sigma^2/\text{\AA}^2$	Temperature	Path	CN #	R/ \AA	$\sigma^2/\text{\AA}^2$
RT (bench)	Yb-O	8.4	2.33(3)	0.006(1)	100°C	Yb-O	8.5	2.35(1)	0.010(3)
RT (bench)	Yb-S	2.5	3.66(1)	0.006(8)	100°C	Yb-S	2.5	3.68(16)	0.017(31)
RT (synchrotron)	Yb-O	8.5	2.32(1)	0.010(1)	200°C	Yb-O	7.8	2.34(2)	0.017(3)

RT (synchrotron)	Yb-S	2.6	3.64(2)	0.014(9)	200°C	Yb-S	1.7	3.64(1)	0.006(20)
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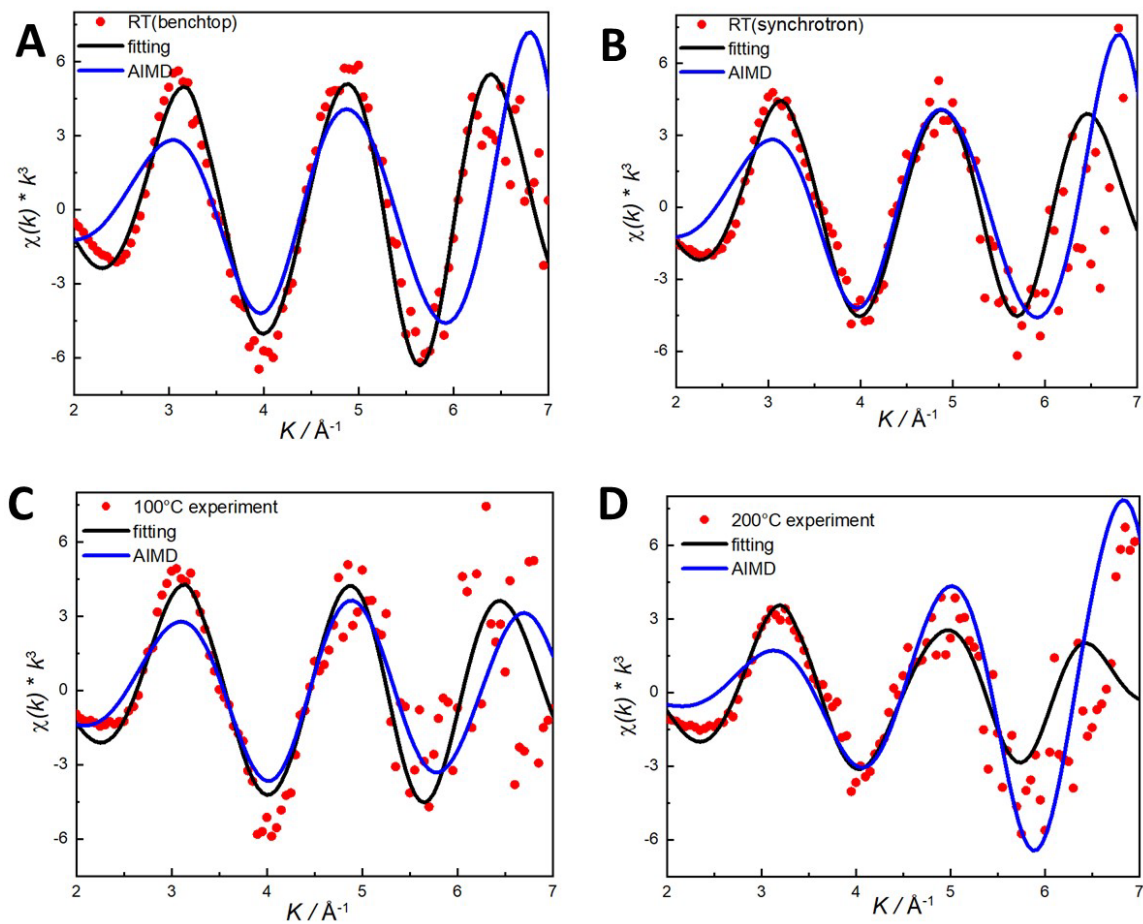


Figure 2. Yb L-III edge EXAFS for 0.05M $\text{Yb}_2(\text{SO}_4)_3$, shown in k^3 -weighted k -space from 2 to 7 \AA^{-1} . (A) RT data measured with a benchtop instrument (red points), theoretical curve fit (black line), and AIMD simulation (blue line). (B) RT data measured at a synchrotron (red points), theoretical curve fit (black line), and AIMD simulation (blue line). (C) Data collected at 100°C (red points), theoretical curve fit (black line), and AIMD simulation (blue line). (D) Data collected at 200°C (red points), theoretical curve fit (black line), and AIMD simulation (blue line). The consistency across experimental and simulated data highlights the accuracy of the modeling and validation processes.

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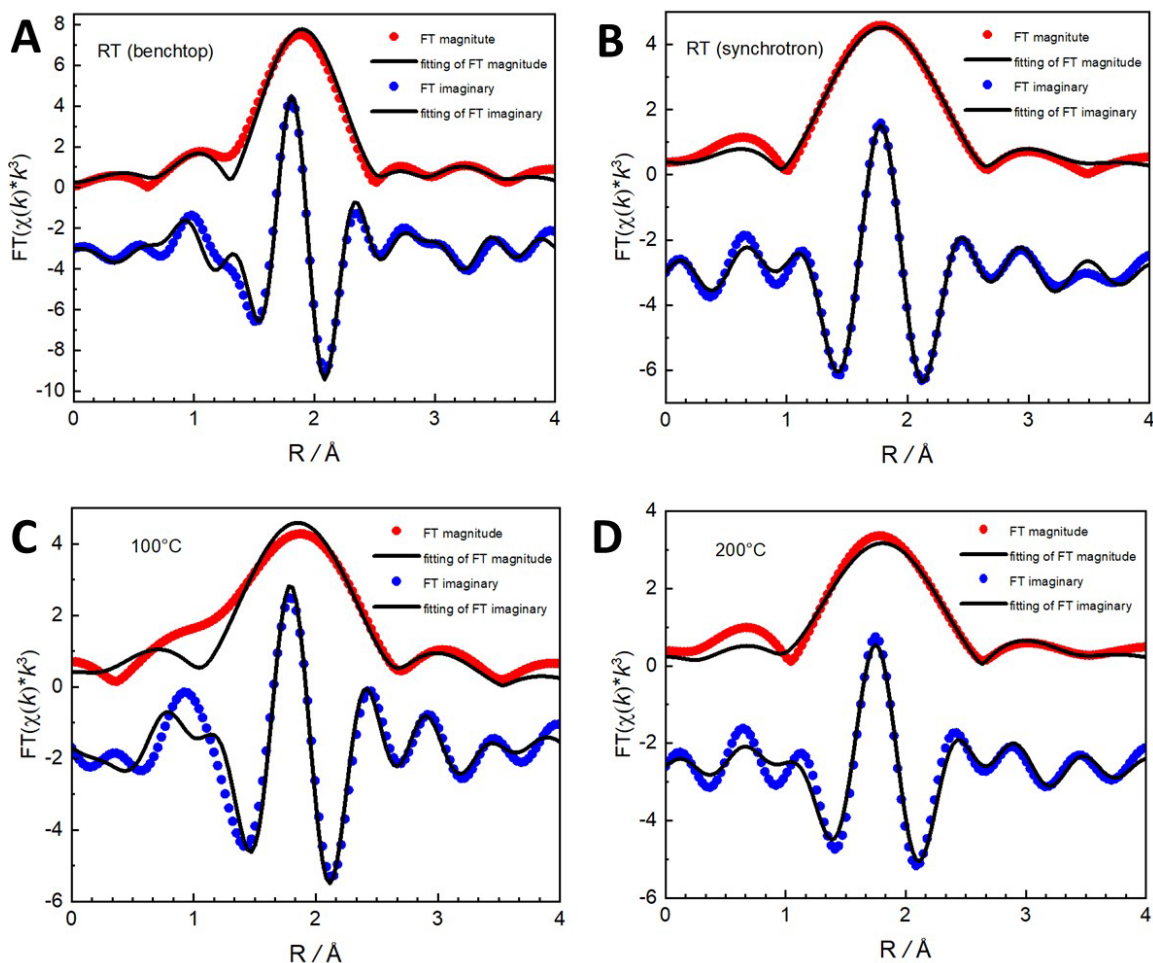


Figure 3. Fourier-transformed Yb L-III edge EXAFS for 0.05M $\text{Yb}_2(\text{SO}_4)_3$ in R -space and not phase-corrected. (A) Data collected at RT using a benchtop instrument: red points represent the Fourier-transformed magnitude, blue points the Fourier-transformed imaginary component, and black lines the curve fits for both. (B) Data collected at RT using a synchrotron instrument with the same labeling scheme. (C) Data collected at 100°C with matching Fourier-transformed components and curve fits. (D) Data collected at 200°C matching Fourier-transformed components and curve fits.

317

318 As the temperature increases to 100°C, the first-shell Yb-O peak in R -space shifts slightly to
 319 ~ 2.35 Å accompanied by peak broadening and a slight increase in the Debye-Waller factor to 0.010
 320 Å². These changes suggest an elongation of Yb–O bonds and increased structural disorder. Despite
 321 these changes, the CN remains stable at ~ 8.5 , with Yb^{3+} retaining its hydration shell of five water
 322 molecules and sulfate ligands. For Yb-S, the peak shifts slightly to ~ 3.68 Å, with a CN of ~ 2.5 and
 323 a σ^2 value of 0.017 Å². This indicates that sulfate ligands remain bound but exhibit slightly more

324 flexibility compared to RT. At 200°C, significant structural changes are observed, driven by
325 dehydration and reorganization of the coordination shell. The $\chi(k)*k^3$ oscillations are noticeably
326 broader, reflecting greater thermal disorder. In the *R*-space, the first-shell Yb-O peak shifts further
327 to ~ 2.34 Å with diminished intensity, indicating a reduction in the number of coordinating oxygen
328 atoms. The CN decreases to ~ 7.8 with one water molecule removed from the hydration shell
329 suggested by the AIMD simulation. The σ^2 value for the Yb-O shell increases to 0.017 Å², implying
330 greater thermal motion and bond variability. Meanwhile, the Yb-S peak becomes less intense, with
331 the CN decreasing to ~ 1.7 and the σ^2 value dropping to 0.006 Å². This suggests that sulfate groups
332 are still coordinating with Yb by a rigid bond, which is potentially due to stronger interactions with
333 the reduced number of sulfate ligands.

334 Overall, the structural evolution of Yb³⁺ from RT to 200°C reflects the interplay between
335 hydration, sulfate binding, and thermal effects. At RT, the Yb coordination environment is
336 characterized by a dense hydration shell and stable sulfate binding. As the temperature increases
337 to 100°C, the coordination shell remains intact, with only minor elongation of Yb-O bonds and
338 slight increases in disorder. However, at 200°C, the Yb-S bond length (~ 3.64 Å) remains stable,
339 and the decreased σ^2 value indicates that the remaining sulfate ligands bind more rigidly to Yb³⁺.
340 The loss of coordinating water molecules reduces competition for coordination sites and enhances
341 electrostatic interactions between Yb³⁺ and the divalent sulfate. Additionally, the bi-dentate
342 binding mode in sulfate can effectively chelate the Yb with a stable and rigid framework. The
343 structural reorganization at high temperatures further favors sulfate ligands occupying optimal
344 positions despite the reduced coordination number. These observations are consistent with the
345 picture that sulfate serves as a transport ligand at high temperature, at least until 200°C.

346 At 300°C, the EXAFS indicates a distinct precipitation condition for the Yb-SO₄ system. The
347 *k*-space data (**Figure 4A, Table 2**) shows dampened oscillations compared to lower temperatures,
348 indicating greater thermal disorder and possible structural reorganization. The *R*-space data
349 (**Figure 4B, Table 2**) demonstrates broadened peaks, reflecting increased variability in bond
350 lengths and reduced coordination numbers. More importantly, the total CN drops to ~7.2 and the
351 Yb-S CN increases to ~3.5, which suggests dehydration of the complex while yet retaining sulfate
352 ligands. These changes suggest that sulfate ligands bind more strongly with Yb³⁺ as the hydration
353 shell diminishes, likely driving the formation of more ordered sulfate-rich solids, which is
354 consistent with the retrograde solubility behavior of Yb-SO₄ (Cetiner et al., 2005; Cui et al., 2020;
355 Wan et al., 2023). Due to the resolution limitation of the optical scope in the experimental set-up,
356 the exact precipitation temperature could not be determined. Previous solubility data did not cover
357 temperature above 200°C (Judge et al., 2023), hence, we could only conclude that the precipitation
358 temperature is between 200°C and 300°C, further experiments are highly recommended to resolve
359 a precise precipitation temperature. Comparisons to diffraction data for crystalline
360 Yb₂(SO₄)₃·8H₂O indicate that the high-temperature coordination environment approaches that of
361 a crystalline solid, where Yb-S bonds are dominant (CN ~4, bond length ~3.60 Å). The reduced
362 CN for Yb-O at 300°C and the dominance of Yb-S coordination suggests the formation of a
363 partially dehydrated crystal.

364

365 **Table 2.** Curvefit Parameters for Yb L-III edge EXAFS at 300°C and bond lengths from
 366 diffraction data of $\text{Yb}_2(\text{SO}_4)_3 \cdot 8\text{H}_2\text{O}$ solid at RT.

system	Path	CN #	R/Å	$\sigma^2/\text{Å}^2$
HDAC (300°C)	Yb-O	7.2	2.33(8)	0.012(1)
HDAC (300°C)	Yb-S	3.5	3.50(12)	0.037(19)
$\text{Yb}_2(\text{SO}_4)_3 \cdot 8\text{H}_2\text{O}$ (RT)*	Yb-O	8	2.24 – 2.44	N/A
$\text{Yb}_2(\text{SO}_4)_3 \cdot 8\text{H}_2\text{O}$ (RT)*	Yb-S	4	3.60	N/A

367 * Diffraction data of crystal solid from X-ray diffraction by L Hiltunen et al. (Hiltunen and Niinisto, 1976)

368

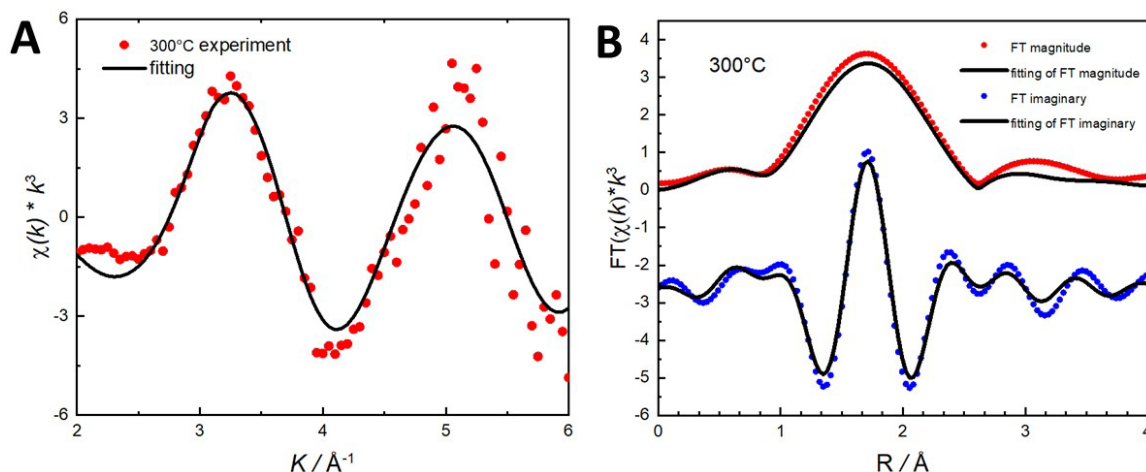


Figure 4. (A) Yb L-III edge EXAFS (red points) and the curvefit (line) for 0.05M $\text{Yb}_2(\text{SO}_4)_3$ at 300 °C, shown in k^3 weighted k -space. (B) Yb L-III edge EXAFS (red points for the Fourier-transformed magnitude, and blue points for the Fourier-transformed imaginary components) and the curvefit (line) for 0.05M $\text{Yb}_2(\text{SO}_4)_3$ at 300 °C, shown in k^3 weighted R -space and not phase-corrected.

369

370 3.3. Local atomic structure and coordination dynamics of Yb^{3+} from AIMD

371 At RT, the coordination environment of Yb^{3+} is defined by a dense hydration shell and strong
 372 sulfate binding. The radial distribution function (RDF) for Yb–O (**Figure 5A**) shows distinct peaks
 373 corresponding to five water molecules and three sulfate oxygen atoms, resulting in a total

374 coordination number (CN) of ~ 7.95 (**Figure 5B**). The Yb–O bond lengths vary with CN, while
375 mono-dentate sulfate oxygen forming the shortest bond (~ 2.38 Å) and bi-dentate sulfate oxygen
376 and water oxygen exhibit longer distances (as shown the broader peaks at ~ 2.65 Å and ~ 3.03 Å in
377 **Figure 5A**). Analysis of the AIMD trajectories revealed that the RDF feature near ~ 3.03 Å arises
378 primarily from the distal oxygen in bi-dentate sulfate ligands. In such configurations, one sulfate
379 oxygen binds strongly to Yb³⁺ at a short distance (~ 2.3 – 2.5 Å), while the second, more weakly
380 interacting oxygen, lies farther away (~ 3.0 – 3.3 Å), contributing to the longer-range RDF peak.
381 These two asymmetric binding modes yield a distorted square antiprism geometry due to the
382 uneven Coulombic repulsion between ligands and angular strain from bi-dentate sulfate ligands
383 (as illustrated in **Figure 6A-B**). Although the beading strain for the molecular structures could
384 destabilize the complex, the presence of π -bonding signature could compensate for these
385 distortions. As shown in molecular orbital visualizations in **Figure 6C-D**, the electron density is
386 always in a shoulder-to-shoulder shape, which suggests a synergistic overlap of ligand-based π
387 orbitals with Yb, which could introduce extra covalency to the bonding and stabilize the molecule
388 to exist in geometrically distorted forms without significant energetic penalties.

389 As the temperature increases to 100°C, the Yb–O RDF peaks broaden slightly, with minor
390 elongation in bond lengths (sharp peak at ~ 2.39 Å for mono-dentate sulfate oxygen and broad
391 peaks at ~ 2.67 Å and ~ 3.03 Å for bi-dentate sulfate and water oxygens, as in **Figure 5B**). Despite
392 these changes, the CN remains stable at ~ 7.90 (**Figure 5H**), indicating that the hydration shell is
393 still intact at this temperature. The coordination geometry remains distorted, as the bi-dentate
394 sulfate binding mode persists. The Yb–S RDF (**Figure 5E**) at 100°C shows that mono-dentate
395 sulfate coordination (3.04 Å) becomes more prominent, while bi-dentate sulfate coordination
396 exhibits a slight shift (~ 3.37 Å), suggesting increased thermal motion and dynamic exchange

397 between sulfate and water ligands. Significant structural reorganization occurs at 200°C due to
398 dehydration and increased thermal effects. The Yb–O CN decreases to ~7.19 (**Figure 5I**) and
399 exhibits a capped octahedron geometry (**Figure 6E-F**), reflecting the loss of one water molecule
400 from the hydration shell. The Yb–O RDF (**Figure 5C**) shows broader and more variable bond
401 lengths (ranging from 2.26 Å to 3.13 Å), indicative of greater structural disorder. However, the
402 Yb–S RDF (**Figure 5F**) demonstrates a pronounced peak at ~3.38 Å for bi-dentate sulfate
403 coordination, suggesting stronger and more rigid sulfate binding. The decrease in Yb–S CN to
404 ~1.54 (**Figure 5L**) highlights the reduced number of sulfate ligands directly coordinating with
405 Yb³⁺, yet less loss than the dehydration loss.

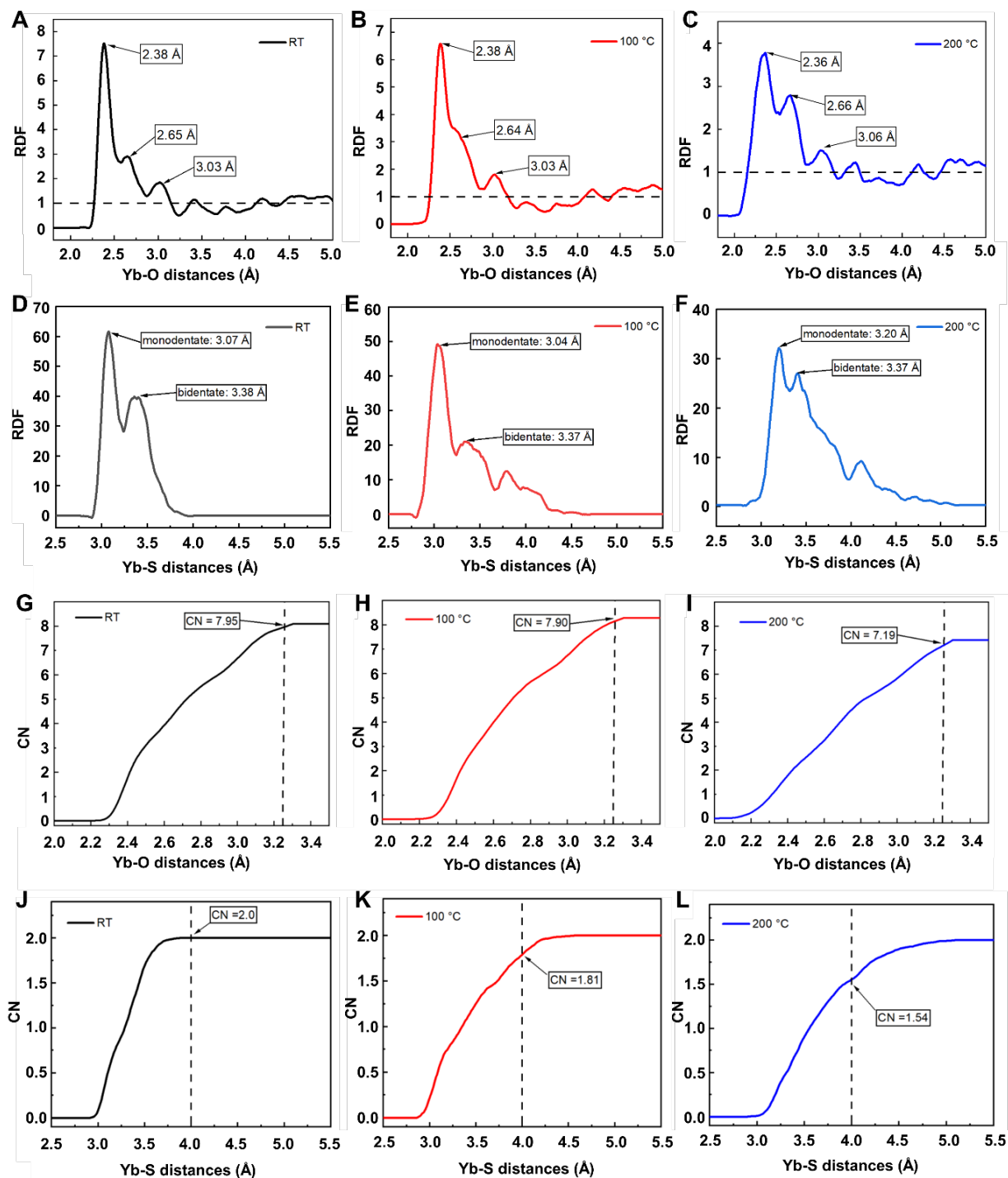


Figure 5. Illustration of the RDF (A--F) and total CN (G--L) for Yb-SO₄ at different temperatures: RT, 100 °C and 200 °C. In the Yb-S RDF illustration, the “monodentate” peak represents the SO₄²⁻ ligand who binds with Yb³⁺ via the monodentate mode, and the “bidentate” peak represents the SO₄²⁻ ligand who binds with Yb³⁺ via the bidentate mode. The Yb-S distance from the monodentate mode is always shorter than the Yb-S distance in the bidentate mode.

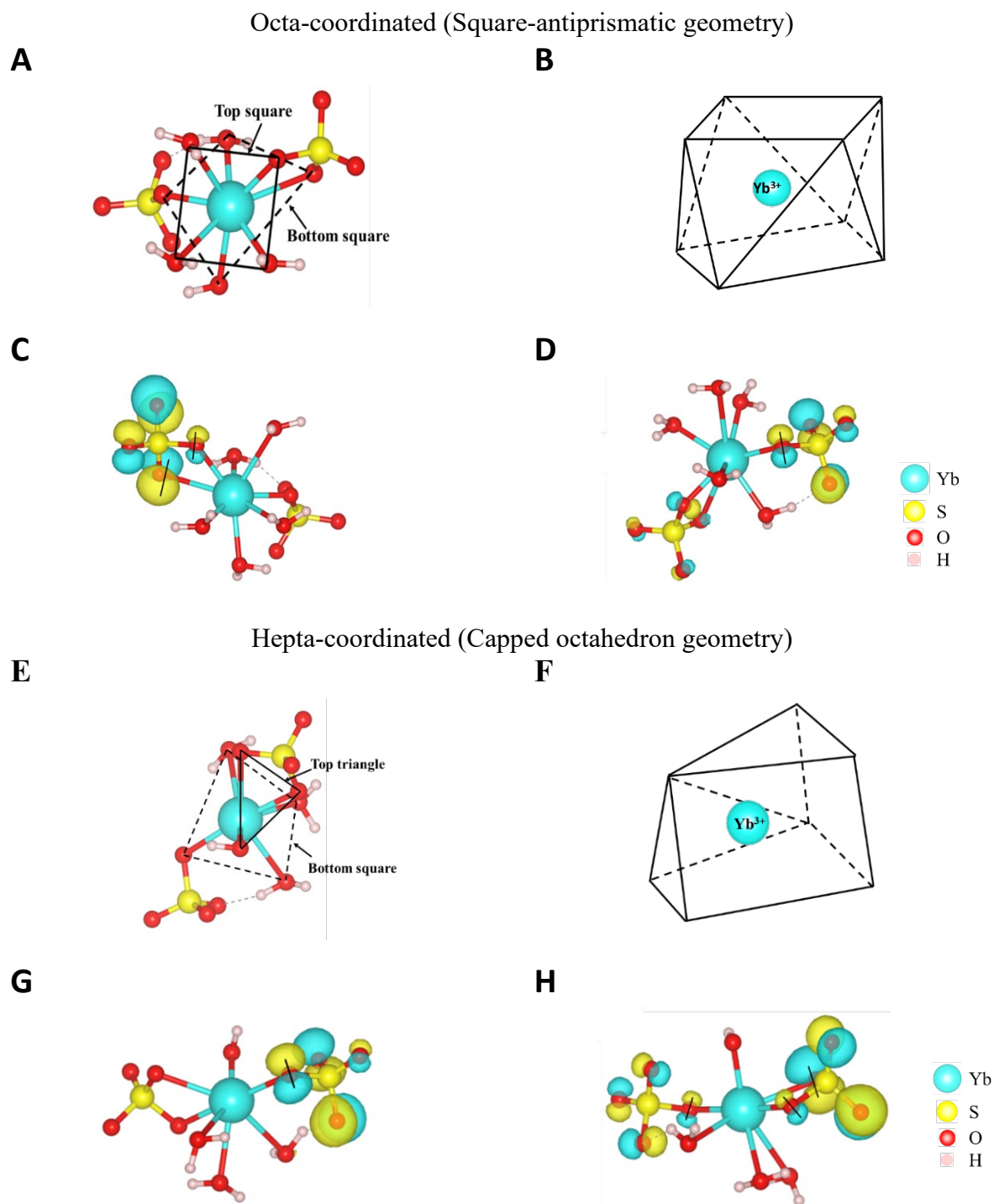


Figure 6. Display of octa- and hepta-coordinated Yb(III)-SO₄ complex from AIMD simulation. The snapshots are respectively from simulations at RT (octa-coordinated) and 200 °C. (A) Molecular structure of the octa-coordinated Yb complex with a square-antiprismatic geometry. (B) Polyhedral representation of the square-antiprismatic coordination geometry of Yb, showing the eight coordinating atoms forming two parallel squares. (C) Electron density visualization of a bonding orbital for the octa-coordinated Yb complex. (D) Electron density visualization of

another bonding orbital for the octa-coordinated Yb complex. (E) Molecular structure of the hepta-coordinated Yb complex with a capped octahedron geometry. (F) Polyhedral representation of the capped octahedron coordination geometry of Yb. (G) Electron density visualization of a bonding orbital for the hepta-coordinated Yb complex. (H) Electron density visualization of another bonding orbital for the hepta-coordinated Yb complex. Cyan atoms represent Yb, yellow atoms represent S, red atoms represent O, and pale pink atoms represent H.

407

408 **4. Discussion**

409 **4.1 Geometry evolution of Yb³⁺ sulfate complexes**

410 The structural dynamics of Yb³⁺-sulfate complexes show significant temperature-dependent
411 evolution. At RT, Yb³⁺ is coordinated by five water molecules and two sulfate ligands, resulting
412 in a total CN of ~8.5. The coordination geometry corresponds to a distorted square antiprism, with
413 one sulfate ligand binding in a bidentate mode and the other in a monodentate mode. This
414 configuration produces a range of Yb–O bond lengths, with monodentate sulfate bonds at shorter
415 distances (~2.36 Å) and bidentate bonds at longer distances (2.33–3.06 Å). The low Debye-Waller
416 factors (σ^2) for Yb–O and Yb–S bonds indicate minimal thermal disorder, reflecting a stable and
417 ordered coordination environment. These results are consistent across experimental and simulated
418 datasets, underscoring the accuracy of AIMD-derived structural models. At 100°C, the
419 coordination shell of Yb³⁺ remains largely intact. Experimental data suggests slight elongation in
420 Yb–O (~2.35 Å) and Yb–S (~3.68 Å) bond lengths, while the CN stable at ~8.5. However, the
421 increase in σ^2 values for both Yb–O and Yb–S shells suggest enhanced thermal motion and
422 dynamic ligand exchange. AIMD simulations further illustrate this dynamic behavior, showing
423 increased flexibility in sulfate ligands and greater interaction with water molecules. Despite these
424 fluctuations, the overall geometry remains stable, preserving the distorted square antiprism
425 structure.

426 At 200°C, significant structural reorganization occurs due to thermal effects and partial
427 dehydration. The experimental CN for Yb–O decreases to ~7.8, indicating the displacement of one
428 water molecule from the hydration shell. AIMD simulations confirm this dehydration process and
429 show that the remaining sulfate ligands bind more rigidly to Yb³⁺. The Yb–S bond length remains
430 stable (~3.64 Å), while the σ^2 for the sulfate shell decreases, suggesting reduced thermal motion
431 and stronger interactions. This transition marks a shift from hydration-dominated coordination to
432 sulfate-dominated coordination, driven by the chelating effect of bidentate sulfate ligands and
433 reduced competition from water molecules.

434 At 300°C, the coordination environment evolves toward a structure resembling crystalline
435 Yb₂(SO₄)₃·8H₂O. Experimental EXAFS data show reduced Yb–O coordination (CN ~7.2) and
436 increased Yb–S coordination (CN ~3.5), with shorter Yb–S bond lengths (~3.50 Å). These
437 observations suggest a further decrease in hydration and a dominance of sulfate ligands in the
438 coordination shell. This transition aligns with the retrograde solubility behavior of REE sulfates
439 and hints at the onset of solid-phase formation.

440 **4.2 Geological implications and geological modeling**

441 The coordination environment of Yb³⁺ observed in this study transitions from a hydration-
442 dominated structure at RT to a sulfate-dominated structure at elevated temperatures, contrasting
443 with previous studies on chloride (Cl⁻) and fluoride (F⁻) complexes. Chloride ligands exhibit weak
444 electrostatic interactions with REEs, leading to more flexible ligand exchange and dynamic
445 hydration shells. This results in a predominantly hydration-controlled transport mechanism,
446 enabling REEs to remain mobile across a broad range of hydrothermal conditions (Mayanovic et
447 al., 2007; Guan et al., 2020). In contrast, fluoride forms strongly bound inner-sphere complexes
448 that restrict REE mobility and promote selective precipitation at relatively lower

449 temperatures(Migdisov and Williams-Jones, 2008). Sulfate ligands, however, exhibit a dual
450 behavior depending on temperature. At RT and 100°C, EXAFS and AIMD results confirm that
451 sulfate coordinates with Yb³⁺ in either a bidentate or monodentate mode, forming stable Yb³⁺-SO₄
452 complexes in solution. At 200°C, the presence of two mono-dentate sulfate coordination observed
453 suggests that sulfate strengthens its bond with Yb³⁺ as the hydration shell diminishes—unlike
454 chloride complexes, which tend to weaken at high temperatures. Above 200°C, sulfate may start
455 to act as a deposition ligand, given that at 300°C the EXAFS data and AIMD simulations reveal a
456 shift toward sulfate-dominated coordination, resembling the solid-phase Yb₂(SO₄)₃·8H₂O
457 observed in diffraction studies (Hiltunen and Niinistö, 1976). The reduction in Yb-O coordination
458 number (CN ~7.2) and increased Yb-S interaction (CN ~3.5) indicates the collapse of the hydration
459 shell, favoring sulfate coordination over water. This suggests that sulfate facilitates REE
460 precipitation under high-temperature conditions, consistent with the retrograde solubility of REE
461 sulfates reported in natural hydrothermal deposits (Wan et al., 2023).

462 Unlike Cl⁻, which remains a transport ligand across a wider P-T range, sulfate transitions from
463 a stabilizing ligand in solution to a precipitating agent at high temperatures. This temperature-
464 dependent transformation actively regulates REE solubility in hydrothermal fluids, ultimately
465 influencing mineral deposition. The progressive dehydration observed from 100°C to 300°C
466 enhances sulfate complexation strength, reducing REE solubility and triggering precipitation at
467 higher temperatures. Geological evidence indicates that many REE deposits, such as Wicheeda
468 (Canada) and Dalucao (China), formed due to abrupt T-P drops in sulfate-rich fluids (Trofanenko
469 et al., 2016; Zhang et al., 2022). Fluid immiscibility further enhances REE concentration by
470 partitioning REEs into sulfate-rich dense phases, a phenomenon observed in high-grade bastnäsite
471 and monazite deposits (Wan et al., 2021). The increased rigidity of sulfate-bound REEs at high

472 temperatures, as suggested by lower Debye-Waller factors (σ^2) in EXAFS fits, indicates that sulfate
473 complexes play a stabilizing role at moderate temperatures but promote crystallization at higher
474 temperatures. Overall, this temperature-dependent hydration-to-sulfate transition demonstrates
475 that, in the absence of perturbations from other ions, sulfate can function both as a transport agent
476 at lower temperatures and as a deposition trigger at higher temperatures—behavior distinct from
477 chloride or fluoride systems.

478 The molecular-level insights gained from this study also have significant implications for
479 geochemical modeling of REE behavior in hydrothermal systems. Accurate structural data on
480 REE-sulfate complexes at high temperatures and pressures are essential for developing robust
481 thermodynamic models that can predict REE speciation, solubility, and transport in natural
482 hydrothermal fluids. These models are critical for understanding the formation of REE deposits
483 and for guiding exploration and resource assessment strategies. For example, the retrograde
484 solubility behavior of REE sulfates, as observed in this study, can be incorporated into
485 thermodynamic databases to improve the accuracy of predictive models for REE transport and
486 deposition.

487 The temperature-dependent transition from hydration-dominated to sulfate-dominated
488 coordination provides a mechanistic basis for understanding the precipitation of REE-rich minerals
489 in hydrothermal systems. At lower temperatures, sulfate ligands stabilize REEs in solution,
490 facilitating their transport through hydrothermal fluids. However, as temperatures rise, the
491 dehydration of REE complexes and the strengthening of sulfate binding led to the formation of
492 more ordered, sulfate-rich solids. This transition is particularly relevant in high-temperature
493 hydrothermal environments, where sulfate ligands can act as both transport and deposition agents.
494 By integrating these insights into geochemical models, we can better predict the conditions under

495 which REEs are likely to be transported or deposited, aiding in the identification of potential REE
496 resources. While the present study focuses on the bulk fluid conditions, which are representative
497 of hydrothermal veins in natural geological systems. We also recommend further investigation into
498 exploring the behavior of hydrothermal microfluid or fluid under nano confinements. Such
499 conditions are relevant to understanding fluid–rock interactions within mineral nanopores,
500 interlayer spaces of clay minerals, or other restricted geometries commonly found in subsurface
501 environments. Microfluidic and nanoconfinement coupled spectroscopic experiments could
502 complexation, nucleation, and mineral transformation processes of hydrothermal fluids under
503 confinement.

504 Moreover, the structural and stoichiometric information obtained in this study enables the very
505 first and most essential input required to build up reliable geological and thermodynamic models.
506 Accurate coordination environment and stoichiometry of REE–ligand complexes govern how
507 REEs behave in hydrothermal systems, including their solubility, mobility, and eventual
508 deposition. In particular, the long-standing ambiguity over whether Yb in sulfate-rich fluids forms
509 predominantly mono-sulfate (YbSO_4^+) or di-sulfate ($\text{Yb}(\text{SO}_4)_2^-$) complexes has limited the
510 precision of thermodynamic databases. Our AIMD simulations resolve this puzzle by showing that
511 Yb favors the formation of a stable di-sulfate complex up to at least 200 °C, supporting the
512 dominance of the following reaction:



514 while reactions (1) and (3), involving mono-sulfate species, appear to be less relevant under these
515 conditions. At temperatures above 200 °C, complex destabilization and dehydration was evidenced,
516 ultimately leading to precipitation, consistent with REE mineralization. These results not only
517 clarify a long-standing question regarding Yb speciation in sulfate-bearing fluids but also provide

518 robust molecular-level constraints for thermodynamic models, which can be applied to natural
519 systems, such as carbonatite-related REE deposits or hydrothermal veins, to better understand the
520 processes controlling REE enrichment and mineralization. Additionally, the role of fluid
521 immiscibility in concentrating REEs into sulfate-rich dense liquid phases can be incorporated into
522 models to explain the formation of high-grade REE deposits, such as bastnäsite and monazite.

523 The integration of molecular-level insights into geochemical models also has practical
524 applications in resource exploration. By understanding the thermal conditions under which REE
525 sulfates are stable or prone to precipitation, exploration efforts can be targeted toward areas with
526 specific temperature, pressure, and fluid chemistry conditions that favor REE enrichment.
527 Furthermore, the molecular information can inform strategies for the sustainable extraction and
528 recovery of REEs from hydrothermal ore deposits, contributing to the development of more
529 efficient and environmentally friendly mining practices.

530 In conclusion, the molecular-level understanding of Yb^{3+} -sulfate complexes provided by this
531 study offers valuable insights for geochemical modeling of REE behavior in hydrothermal systems.
532 By incorporating these findings into thermodynamic models, we can improve our ability to predict
533 REE transport and deposition, ultimately enhancing our understanding of REE ore formation and
534 guiding the exploration and exploitation of these critical resources.

535 **5. Summary**

536 This study provides a detailed molecular-level understanding of Yb^{3+} coordination in sulfate-rich
537 hydrothermal fluids under high-pressure and high-temperature conditions. By integrating in situ
538 EXAFS spectroscopy with AIMD simulations, we have characterized the structural evolution of
539 Yb^{3+} complexes in hydrothermal fluid up to 300°C. The results indicate a temperature-dependent

540 transition in the Yb coordination environment, with hydration shell depletion and increasing
541 sulfate interaction playing key roles in structural reorganization. At room temperature, Yb³⁺ is
542 coordinated by five water molecules and two sulfate ligands, forming a distorted square antiprism
543 geometry with a coordination number of ~8. As the temperature rises, progressive dehydration
544 occurs, reducing water coordination and strengthening sulfate binding. By 200°C, Yb³⁺ adopts a
545 capped octahedron shape with a coordination number of ~7, and sulfate ligands become the
546 dominant coordinating species. At 300°C, further dehydration and precipitation formed due to
547 retrograde solubility of Yb₂(SO₄)₃, which also enhanced sulfate binding. These findings have
548 significant geological implications in understanding REE transport and deposition in hydrothermal
549 environments, which reveals sulfate acts as a critical ligand in REE mobility, stabilizing Yb³⁺ in
550 solution at lower temperatures while facilitating precipitation at higher temperatures due to
551 stronger sulfate interactions. The transition from hydration-dominated to sulfate-dominated
552 coordination provides new insights into the mechanisms driving REE mineralization, which could
553 aid in refining predictive geochemical models for REE ore formation under extreme conditions.

554 **CRedit authorship contribution statement**

555 **Xiaodong Zhao:** Experimental data curation, Calculation, Formal analysis, Investigation,
556 Methodology, Writing – original draft, Writing – review & editing. **Duo Song:** Calculation,
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558 Investigation, Methodology, Writing – review & editing. **Micah Prange:** Formal analysis,
559 Investigation, Methodology, Writing – review & editing. **Daria Boglaienko:** Experimental data
560 curation, Methodology, Writing – review & editing. **Zihua Zhu:** Formal analysis, Writing –
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562 **Pearce:** Methodology, Resources, Supervision, Writing – review & editing. **Chengjun Sun:**

563 Methodology, Resources, Writing – review & editing. **Kevin M. Rosso:** Writing – review &
564 editing, Supervision, Resources, Funding acquisition. **Xiaofeng Guo:** Conceptualization, Data
565 curation, Formal analysis, Investigation, Methodology, Funding acquisition, Writing – review &
566 editing. **Xin Zhang:** Conceptualization, Investigation, Methodology, Funding acquisition,
567 Writing – review & editing.

568 **Declaration of competing interest**

569 The authors declare that they have no known competing financial interests or personal
570 relationships that could have appeared to influence the work reported in this paper.

571 **Data availability**

572 Data is made available through Mendeley at <https://doi.org/10.17632/8sfrxcdmew.1>

573 **Acknowledgements**

574 This work was supported by the U.S. Department of Energy (DOE), Office of Science, Basic
575 Energy Sciences, Chemical Sciences, Geosciences, and Biosciences Division through its
576 Geosciences Program at Pacific Northwest National Laboratory (PNNL) (FWP #56674). XZ and
577 XZ also acknowledge the support from the Laboratory Directed Research and Development
578 (LDRD) program at PNNL, under the Non-Equilibrium Transport Driven Separations Initiative
579 (NETS) and the Advanced Research Projects Agency-Energy's (ARPA-E) Mining Innovations for
580 Negative Emissions Resource Recovery (MINER) program with award number 22CJ0000901,
581 “Re-Mining Red Mud Waste for CO₂ Capture and Storage and Critical Element Recovery
582 (RMCCS-CER)”. CIP acknowledge support from IDREAM (Interfacial Dynamics in Radioactive
583 Environments and Materials), an Energy Frontier Research Center funded by the U.S. Department

584 of Energy (DOE), Office of Science, Basic Energy Sciences (FWP 68932). X.G. acknowledges
585 the support of this work by the National Science Foundation (NSF), Division of Earth Sciences,
586 under award No. 2149848. Part of the research was performed with a user proposal #61223 (Award
587 DOI: 10.46936/lser.proj.2024.61223/60012698) and 51922 (Award DOI:
588 10.46936/lser.proj.2021.51922/60000373) at the William R. Wiley Environmental Molecular
589 Sciences Laboratory (EMSL), a national scientific user facility sponsored by the U.S. DOE's
590 Office of Biological and Environmental Research and located at PNNL in Richland, WA.
591 Simulations were performed using the National Energy Research Scientific Computing Center
592 (NERSC) supported by the Office of Science of the U.S. DOE operating under Contract No. DE-
593 AC02-05CH11231. PNNL is a multiprogram national laboratory operated by Battelle Memorial
594 Institute under contract no. DE-AC05-76RL01830 for the DOE. This research used resources of
595 the Advanced Photo Source (APS); a US Department of Energy (DOE) Office of Science user
596 facility operated for the DOE Office of Science by Argonne National Laboratory under contract
597 no. DE-AC02-06CH113577.

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