

1 **Formate-Induced Dissolution and Reprecipitation of a Copper Electrocatalyst**
2 **During Electrochemical CO₂ Reduction Reaction**

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10 **Abstract**

11 Catalyst size, morphology, and crystal structure play a crucial role in determining the activity and
12 selectivity of electrochemical CO₂ reduction reactions, which are known to change during the
13 reaction process. A comprehensive understanding of how, when, and why these parameters evolve
14 under operational conditions is essential for developing stable, efficient, and selective catalysts. In
15 this study, we reveal that formate, one of the reaction products, contributes to the degradation of
16 copper catalysts through a ligand-assisted dissolution mechanism. Utilizing in-situ electrochemical
17 atomic force microscopy and ex-situ scanning and transmission electron microscopy, we observed
18 a significant reduction in the size of copper nanoparticles, which decreased from over 30 nm to
19 less than 10 nm in diameter within 60 minutes of CO₂RR. The temporal production of formate
20 correlated with the particle size changes. Furthermore, analysis of the electrolyte using inductively
21 coupled plasma optical emission spectroscopy confirmed the dissolution of copper nanoparticles.
22 Control experiments involving various reaction products (H₂, CO, and HCOO⁻) demonstrated that
23 formate significantly promotes copper dissolution, thereby highlighting its role in the ligand-
24 assisted dissolution mechanism of copper electrocatalysts. Our findings provide critical insights

25 into copper catalyst behavior during electrochemical CO₂ reduction, facilitating the design of more
26 resilient and effective electrocatalysts.

27 **Introduction**

28 The electrochemical CO₂ reduction reaction (CO₂RR) with renewably-sourced electrons is
29 potentially a promising way to convert waste CO₂ into clean chemicals and value-added fuels such
30 as ethylene and ethanol.¹ As an electrocatalyst, copper has been widely recognized as the only
31 monometallic catalyst that can reduce CO₂ to multicarbon (C₂₊) products.² Great progress has been
32 made in understanding the parameters that govern its activity and selectivity,³⁻⁵ but a fundamental
33 understanding of its catalytic stability under reaction conditions is not complete, preventing the
34 development of robust copper-based CO₂RR catalysts.

35 Common consensus about the dynamic change of copper electrocatalysts is based on intrinsic
36 characteristic of copper such as its tendency of redox cycle and high surface mobility. Copper is
37 easily oxidized by electrolyte contact (or air exposure during sample preparation), however,
38 oxidized copper dissolves transiently and is reduced to metallic copper under a cathodic potential
39 (e.g., — 1.0 V vs. reversible hydrogen electrode, RHE). Its Pourbaix diagram supports that copper
40 exists in its metallic state across the entire pH window⁶ and metallic copper is supposed to be
41 stable at cathodic potentials used for CO₂RR.⁷⁻¹⁰

42 However, numerous studies have shown that the copper electrocatalyst undergo physical (e.g.,
43 morphology and crystal structure)¹¹⁻¹³ or chemical (e.g., oxidation or dissolution) change during
44 the reaction,^{14, 15} affecting product selectivity or distribution during the reaction. Several nanoscale
45 degradation mechanisms are proposed in the literature: detachment, dissolution¹⁶ and
46 reprecipitation^{14, 17, 18}, Ostwald ripening¹⁹, reshaping, fragmentation^{12, 20}, agglomeration²¹ and
47 poisoning²². Almost all degradation pathways result in morphological changes of copper catalysts
48 which often show reconstruction of the copper crystal structure. New facets can appear¹²; for
49 example, copper was found to undergo a transformation during CO₂RR from polycrystalline to
50 Cu(100)⁷.

51 Current consensus is that the morphological change of copper is related to its high surface mobility
52 and interaction with chemical species such as $*\text{CO}^{23-25}$ (*represents the adsorbed state) an
53 intermediate product formed during CO_2RR . Because of copper's low cohesive energy and high
54 surface mobility, the copper atoms at the electrode surface can easily migrate and the adsorption
55 of CO at copper sites with low coordination number can further lead to weakening of binding of
56 copper to its neighboring atoms, resulting in decomposition into smaller sizes.²⁴ However, since
57 the referenced studies employed single crystal copper and thin film structures of copper, it is
58 premature to conclude that CO is the sole chemical species that can lead to morphological changes
59 in all copper based catalysts. This is mainly because the binding energy of chemical species to
60 copper varies depending on crystal facet, particle size and morphology^{26,27}, which therefore could
61 lead to different degradation mechanisms of single crystal facets and multi-faceted particles.
62 Another common two-electron reduction product from CO_2RR is formate (HCOO^-) over copper
63 catalysts. However, its role on the morphological evolution of copper catalysts during CO_2RR has
64 been overlooked in the field.

65 In this study, we aim to reveal the mechanisms driving the copper catalyst degradation by tracking
66 reaction products including both CO and formate, dissolved copper concentration in the
67 electrolyte, and the size and morphology of the copper catalyst by combining in situ and ex situ
68 approaches (including in situ electrochemical (EC) atomic force microscopy (AFM) (EC-AFM),
69 in situ X-ray adsorption spectroscopy (XAS) and ex situ scanning electron microscopy (SEM) and
70 transmission electron microscopy (TEM)) under CO_2RR . Our combined results suggest that
71 formate, rather than CO, is the primary product that drives dissolution of the polycrystalline copper
72 oxide nanoparticles and redeposition of small copper nanoparticles on the carbon support under
73 our studied CO_2RR conditions.

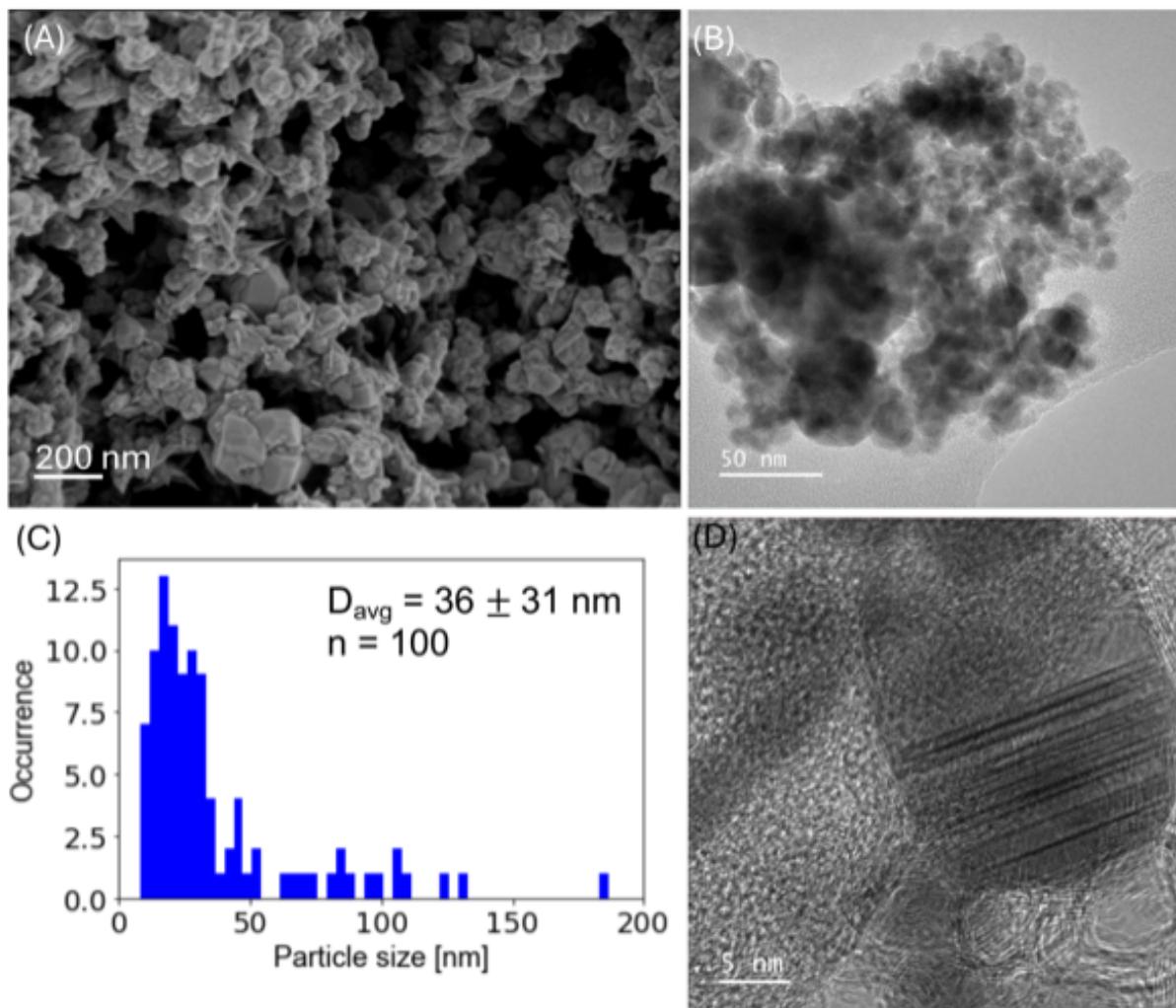
74 **Results and Discussion**

75 **1. Particle Morphology and Size Distribution**

76 **a. Catalyst Characterization before CO_2RR**

77 To assess morphological changes in the copper electrocatalyst during CO_2RR , the size distribution
78 of copper nanoparticle was characterized under different conditions: before any reaction and after

79 applying -1.0 V vs. RHE (V_{RHE}) for 1, 5, 10, and 60 minutes. Before the reaction, mean particle
80 size was $\sim 93 \pm 32$ nm measured by SEM image analysis (Figure 1 and 2A; images provided in
81 Figure S1-S26). Median particle size was $85 \text{ nm} \pm 32$ nm, with a minimum particle size of 35 nm
82 and a maximum particle size of 217 nm. TEM image analysis of the copper particles before coating
83 them on the electrode and reduction reaction showed a smaller mean particle size centered around
84 30 nm (n=100 particles) (Figure 1B and 1C). This indicates that the SEM analysis provides the
85 aggregate size and not primary particle size, whereas TEM analysis provides primary particle size.
86 High resolution TEM imaging showed twinning inside the copper particles, indicating
87 polycrystalline particles (1D).



88

89 **Figure 1 Morphological characterization of copper catalyst before CO₂RR. (A) Scanning electron**
 90 **microscopy image of copper particles. Partial oxidation of particles is visible. (B) Bright field**
 91 **transmission electron microscope (TEM) image of typical particles. (C) Particle size distribution**
 92 **obtained from TEM image analysis, number of analyzed particles (n): 100, mean diameter:**
 93 **$36 \pm 31 \text{ nm}$. (D) Twinning in copper catalyst. Images analyzed to obtain particle size distribution are**
 94 **given in Figure S1-S26 and values in Table S1.**

95 XAS analysis of copper catalyst coated on carbon paper using a binder showed a coordination
 96 number of 4.8 ± 0.4 for Cu-Cu (measured bond distance is $2.542 \pm 0.002 \text{ \AA}$, which matches well
 97 with literature values of 2.56 \AA^{28}) and 1.4 ± 0.2 for Cu-O (measured bond distance is

1.90 ± 0.01 Å, which matches well with literature value of 1.86 Å for Cu₂O²⁹), indicating that some of the copper atoms were oxidized under ambient condition before the reaction.

b. Morphological Evolution During CO₂RR

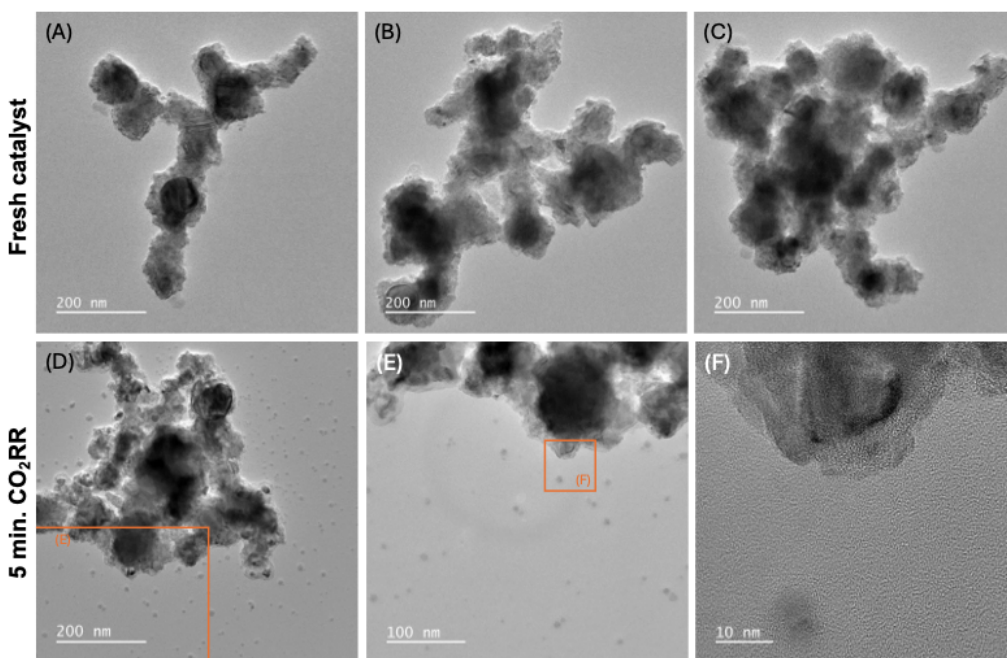
The morphology of the copper nanoparticles significantly changed under the reductive potential (−1.0 V_{RHE}) compared to pre-reaction (Figure 2A-E; additional images given in Figure S32-S36). Post-reaction particle sizes were mostly characterized by ex situ SEM due to difficulties to separate particles from binder post reaction. While SEM analysis does not provide the primary particle size but the aggregate size, as discussed above, it should still provide qualitative information for the size changes of copper nanoparticles.

After 1 min reaction, mean particle size measured in SEM image analysis shifted significantly to 34 ± 32 nm, which is equal to a 63% reduction in size. Copper particle size after 1 min CO₂RR was also characterized using TEM/STEM imaging, where the mean measured particle size was 6.1 ± 7.7 nm (n = 170), which is equal to 83% reduction in size (Images given in Figure S27-31, values in Table S2). So, the trend observed by SEM and TEM imaging on copper nanoparticle size change is in general agreement upon CO₂RR. The median particle size measured by SEM decreased to 23 nm, with a minimum size of 1 nm and a maximum particle size of 140 nm. The median particle size measured by TEM decreased to 3.6 nm, with a minimum size of 0.2 nm and a maximum size of 46.3 nm.

The mean particle size continued to decrease until 10 minutes had elapsed, per SEM measurements. The mean diameter of the copper nanoparticles was 20 ± 22 nm after 5 min and 17 ± 18 nm after 10 min. The mean diameter of the copper nanoparticles was 21 ± 11 nm after 60 minutes reaction time. Further aggregation was observed after 90 and 120 minutes of reaction time (Figure S37) with a mean diameter of copper nanoparticles of 31 ± 34 nm after 90 minutes and 62 ± 27 nm after 120 minutes. The median particle size was 14 nm after 5 minutes of reaction time, 11 nm after 10 minutes reaction rate time, 19 nm after 60 minutes and 55 nm after 120 minutes. There is also a distinct change in minimum and maximum. For 5, 10, 60 minutes, the minimum particle size are 1, 1 and 3 nm. The maximum particle size for 5, 10, 60 minutes is 137 nm, 88 nm and 60 nm. Because the standard deviation for the SEM particle size analysis is high, we also

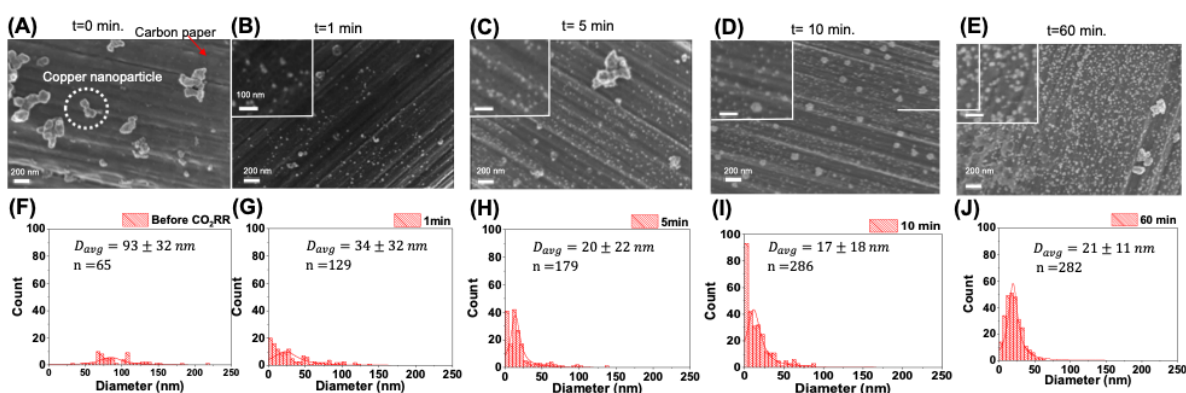
126 analyzed the particle size distribution (Figure 3F-J). The particle size distribution shows a clear
127 shift in size distribution from the before CO₂RR sample to the 10 min sample which contain a
128 larger proportion of particles <10 nm. All calculated values (mean, standard deviation, median,
129 maximum and minimum) are given in Table S3 of the supporting information.

130 To ensure that our ex-situ SEM and TEM observations were representative, we conducted near
131 identical location TEM. For these experiments, we utilized a pointer grid and analyzed copper
132 nanoparticles prior to CO₂RR. We then conducted CO₂RR using the TEM grid as working
133 electrode and analyzed copper nanoparticles after 5 minutes of CO₂RR. We observed typical
134 copper nanoparticle morphology in our fresh catalyst (Figure 2). After 5 minutes of CO₂RR, newly
135 formed small particles were visible (Figure 2), matching our SEM observations.



136
137 **Figure 2** Near identical location TEM results. (A), (B) and (C) show representative images of
138 fresh catalyst, (D) shows copper nanoparticles after 5 min of CO₂RR. Newly formed small particles
139 are visible around the copper catalyst. (E) provides a close-up view of region marked in (D) at the
140 edge of the original copper particle and newly formed particles. In (F) lattice is visible for both the
141 original copper particle and the newly formed particles.

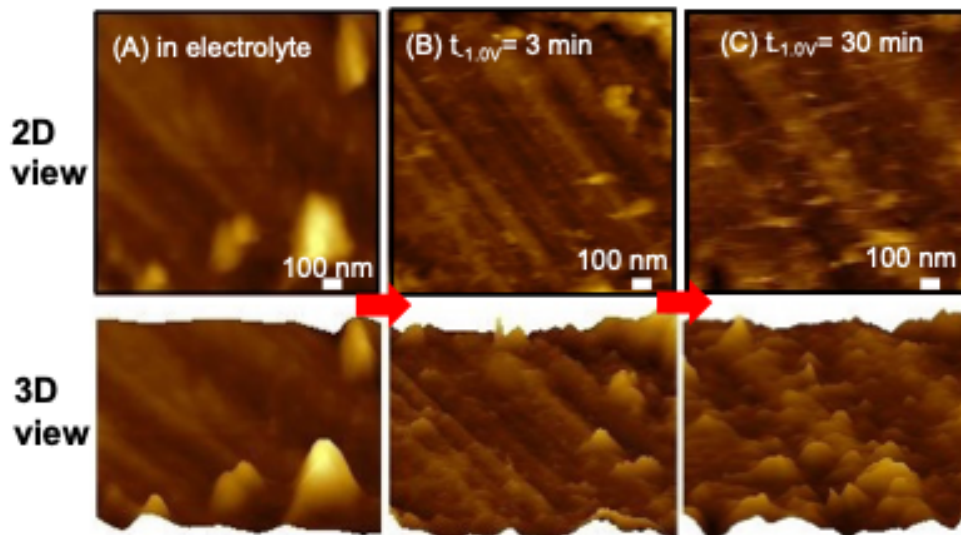
142 No morphological changes were observed when no electrical potential was applied (Figure S38a),
 143 suggesting that the alterations in morphology occurred solely under electrical bias and were not a
 144 result of merely immersing the copper-coated cathode in the electrolyte. EDS analysis also
 145 confirmed that the smaller nanoparticles formed belongs to copper and not an impurity (Figure
 146 S38b-c).



147
 148 **Figure 3 (A-E) Secondary electron (SE) contrast scanning electron microscopy (SEM) images of**
 149 **copper nanoparticles at different reaction times (t) from 0 to 60 minutes. Insets are showing a zoomed**
 150 **subset of the image showing the density of smaller copper particles increased over time. (F-J) Particle**
 151 **size distribution based on SEM image analysis. The mean particle diameter (D_{avg}) is given in (F) to**
 152 **(J) as well as the number of counted particles (n). Scale shown in (A) is the same for all images.**

153 It is possible that the morphological changes of the copper nanoparticles observed under the SEM
 154 was affected by oxidation caused by post-air and electrolyte exposure since copper is easily
 155 oxidized by air and liquid. To test whether this is the case, we monitored the morphological
 156 changes under in situ conditions by using electrochemical atomic force microscopy (EC-AFM),
 157 depicted in Figure 4. In EC-AFM, the similar general trend was observed: the initial mean particle
 158 size of the copper nanoparticles before CO₂RR was ~139 nm, the median particle size 137 nm,
 159 with minimum particle size of 93 nm and maximum particle size of 206 nm. After 3 minutes of
 160 CO₂RR, the mean and median particle size dropped to 68 nm with minimum particle size being
 161 61 nm and the maximum particle size 73 nm. After 30 minutes of CO₂RR, the mean particle size
 162 is 52 nm, the median particle size 54 nm, the minimum particle size 28 nm and the maximum
 163 73 nm. The in situ electrochemical AFM measurements confirm that the formation of the small

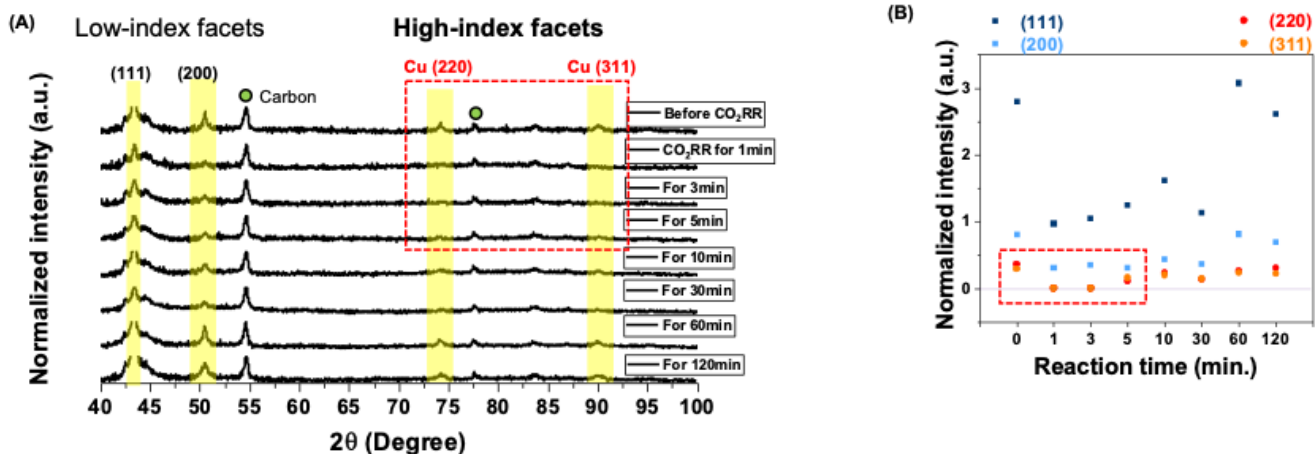
164 copper nanoparticles during the initial stage of reaction is the result of CO₂RR. The resulting
165 current data from EC-AFM is provided in supporting information (Figure S39) and a table of all
166 particle sizes is given (Table S4).



167
168 **Figure 4 (A) EC-AFM Height retrace images of copper nanoparticles on carbon paper (A) in 0.1 M**
169 **KHCO₃ solution before CO₂RR, (B-C) after CO₂RR for 3 and 30 minutes. The bottom row is the 3D**
170 **view.**

171 XRD analysis of copper catalyst after CO₂RR (Figure 5A) showed reconstruction of the copper
172 nanoparticles during the reaction as evident in the variation of the normalized intensity for different
173 facets vs. reaction time (Figure 5B). Two high-index facets (220) and (311) disappear during the
174 early stage of the reaction. High-index facets usually have an open surface structure featuring a
175 high density of atomic steps, kinks and lower-coordination atoms.^{30, 31} These surface defects can
176 lead to a change in adsorption energy for intermediates onto the surface of the catalyst.³² The (311)
177 high-index surface was previously determined to have a higher propensity to produce C₂₊
178 products.³³ It is likely that the higher reactivity of the (311) surface with higher reactivity
179 underwent reaction and reconstruction during the initial CO₂RR stage, such as dissolution of
180 surface copper species as shown in later ICP-OES analysis.

181



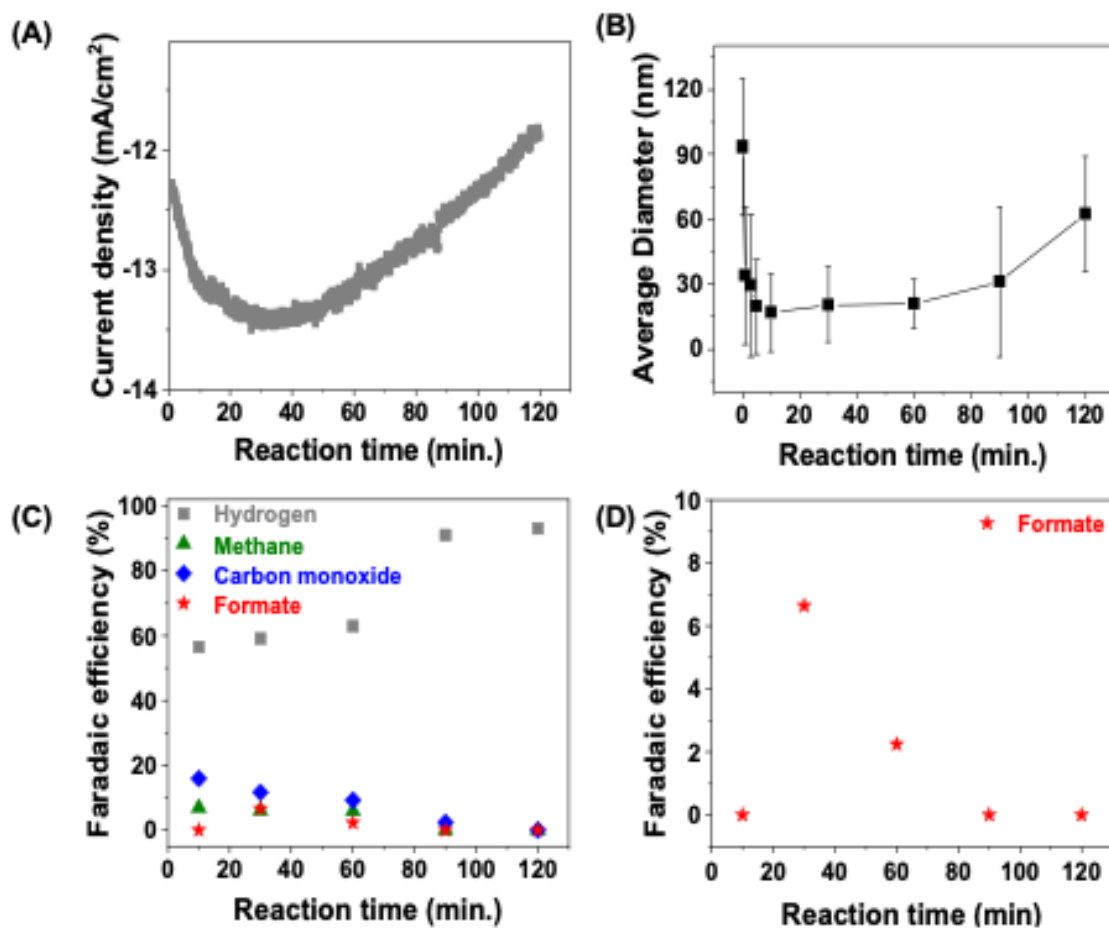
182

183 **Figure 5 Ex-situ XRD analysis of copper catalyst at different CO₂RR reaction times. (A) XRD**
 184 **diffraction and (B) normalized intensity of different copper facets.**

185 2. Product Evolution During CO₂RR

186 To understand how the smaller nanoparticles were formed and to correlate the morphological
 187 changes with CO₂RR reaction products, the evolution of products was studied with gas
 188 chromatography (GC) and ¹H nuclear magnetic resonance (NMR). The current density increased
 189 slightly as the smaller nanoparticles formed until 40-60 minutes of reaction time had elapsed, and
 190 then the current density decreased marginally as the mean copper nanoparticle size increased by
 191 aggregation (Figure 6A). During the reaction, three gaseous products (hydrogen (H₂), carbon
 192 monoxide (CO), methane (CH₄), and one liquid product (formate (HCOO⁻)) were produced
 193 (Figure 6C). However, their selectivity changed differently over time. The Faradaic efficiency (FE)
 194 of hydrogen production gradually increased whereas that of methane and carbon monoxide
 195 persistently decreased. Formate was temporarily produced coinciding with the occurrence of
 196 smaller nanoparticles until 60 minutes (Figure 6C-D). Formate concentrations in solution were
 197 observed between 0.08 mM (minimum) to 3.3 mM (maximum). It appears that formate production
 198 is inversely related to the copper particle size (Figure 6B-D) while other carbon products decrease
 199 continuously with reaction time (Figure 6C), at least qualitatively. There are several possibilities
 200 that can explain the temporary formate production. First, the reduction of oxidized copper to the
 201 zero-valence state could influence formate production in two possible ways: one is that only
 202 reduced copper produces formate, from which it would be expected that the formate production is

203 stable, which is not what we observe, nor has it been reported in any other studies to our
 204 knowledge. Second, the reduction of copper oxide could also take a certain amount of time
 205 (supported by in situ XAS result presented in a later section) and a mixed copper/copper oxide
 206 phase could favor formate production due to the transient presence of copper oxide, which could
 207 explain the temporary production of formate. This matches with previous studies showing the CuO
 208 favors formate production³⁴. A third possibility is that smaller copper particles favor formate
 209 production. Previous studies observed that nanoporous polycrystalline copper favored formate
 210 production at sizes smaller than 22 nm.³⁵ Similar observations were made for dendritic Cu
 211 catalysts.³⁶



212
 213 **Figure 6 (A) Total current density. (B) Mean diameter (D_{avg}) of copper as a function of reaction time**
 214 **as determined by SEM image analysis. (C) Faradaic efficiency (FE) of hydrogen, methane, carbon**

215 **monoxide, and formate as a function of time. (D) FE of formate. Formate was temporarily produced**
216 **until 40 minutes of CO₂RR.**

217 **3. Control Experiments to Identify Possible Effect of Intermediates**

218 Different reaction intermediates are proposed in the literature to cause copper degradation during
219 CO₂RR³⁷. One pathway described is that the degradation of copper into smaller nanoparticles leads
220 to the formation of isolated low-coordinated sites which favors hydrogen evolution reaction (HER)
221 over CO₂RR.³⁸ Another pathway is CO intermediate adsorption³⁷. Based on our experimental
222 results and analysis, we propose ligand-assisted dissolution as a possible degradation mechanism
223 where the ligand is the observed CO₂RR product, formate in this case. This type of ligand-assisted
224 dissolution (also commonly termed “ligand-promoted” or “ligand-induced”) is a commonly
225 observed phenomenon in geochemistry^{39, 40}, e.g., in the case of oxalate enhancing the dissolution
226 of Al₂O₃⁴¹ or salicylate enhancing gibbsite dissolution⁴². Recently, it has been studied for metal
227 dissolution⁴³ and also has been proposed to explain copper electrocatalyst degradation in the
228 presence of amines⁴⁴. Other ligands, e.g., humic acid, also have been shown to promote copper
229 dissolution⁴⁵.

230 To test our hypothesized role of formate species, we conducted three different control experiments
231 to identify which intermediate is causing the particle size changes: (1) *hydrogen evolution reaction*
232 *control experiment (HER)*, (2) *electrochemical CO reduction control experiment (CORR)*, (3)
233 *formate added to CORR control experiment (CORR with formate)* and (4) *SO₂ blocked formate*
234 *adsorption control experiment (CO₂RR with SO₂).*

235 Since the dominant reaction is HER as evidenced by high FE of H₂ (Figure 6C), we first tested
236 HER. In this first control experiment called *hydrogen evolution reaction control experiment*, only
237 argon gas was supplied to test whether hydrogen gas evolution and binding to copper catalyst
238 causes copper nanoparticle morphology changes. SEM image analysis did not show any reduction
239 in particle size as a function of reaction time (Figure S4). Since the pH of the *hydrogen evolution*
240 *reaction control experiment* is lower than the CO₂RR experiment, we also performed control
241 experiments in which we adjusted the pH to match the pH during CO₂RR (pH = 6.8). Particle size

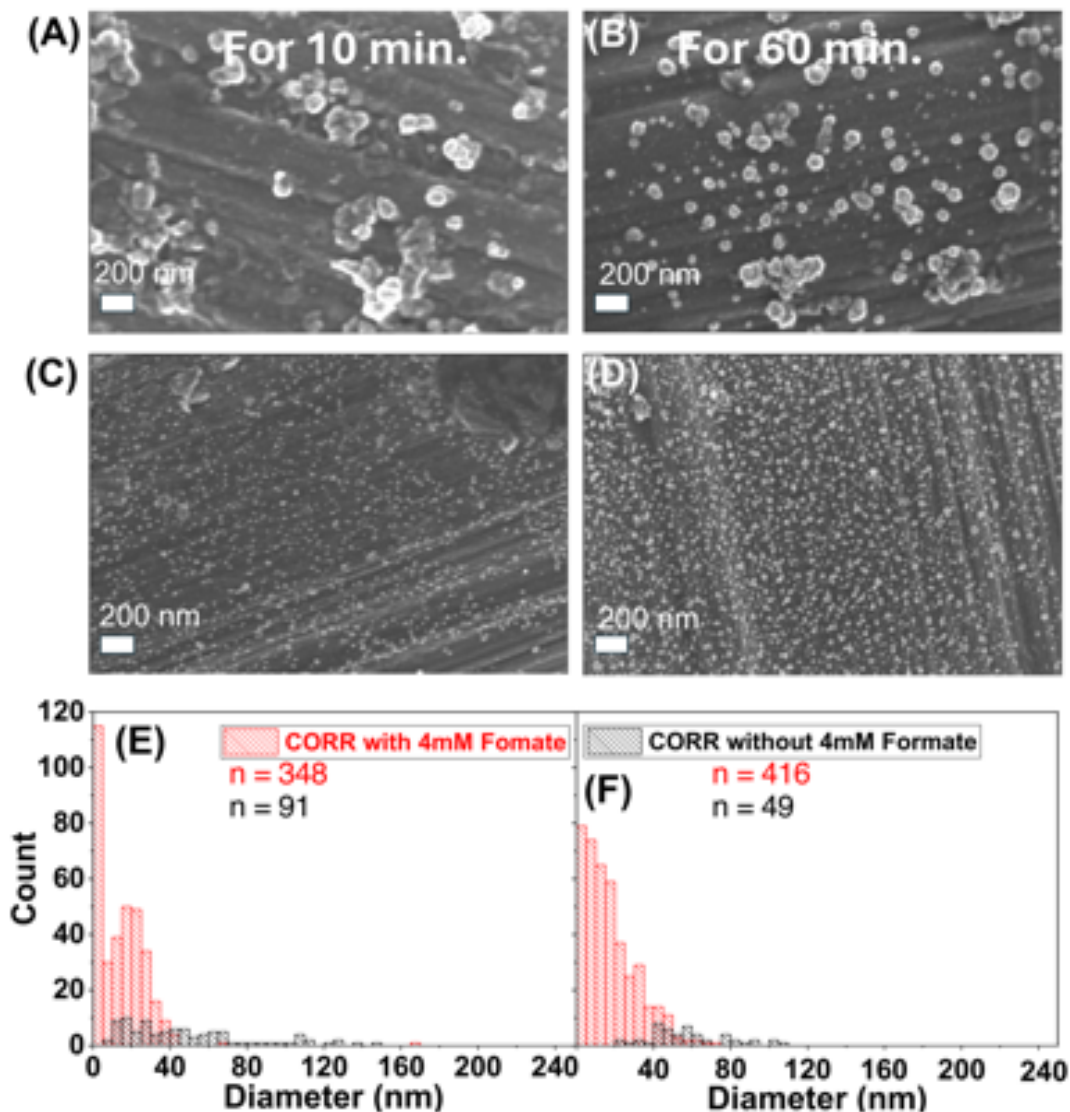
242 changes were much slower than during the CO₂RR, with a reduction of particle size only visible
243 after 60 min of reaction (Figure S40 + S41).

244 Second, CORR was tested since previous studies have reported that CO is a crucial intermediate.
245 Early work found that the reduction of CO yields very similar product distribution as CO₂RR on
246 copper, highlighting that CO is a key intermediate^{46, 47}. In the second control experiment, called
247 *CO reduction reaction control experiment*, CO was supplied instead of CO₂, to identify whether
248 the CO intermediate causes Cu size changes.⁴⁸ The size change occurs at a much slower pace in
249 this *control* experiment than CO₂RR as compared in Figure S42. Clearly, significant
250 morphological changes of the Cu catalyst only occur under CO₂RR. The main difference in
251 products between the HER/CO control experiments and the CO₂RR is that formate was
252 temporarily produced during CO₂RR, suggesting a major effect of formate in causing copper size
253 changes.

254 Therefore, we tested our hypothesis that ligand-assisted dissolution is causing copper particle size
255 changes with a third control experiment called *added formate CO reduction reaction control*
256 *experiment*. In this CORR experiment, we intentionally added a controlled concentration (1 mM
257 and 4 mM) of potassium formate, which was chosen to match formate concentration observed in
258 CO₂RR experiments. Smaller copper nanoparticles quickly (within 10 min) formed after adding
259 4 mM of potassium formate in the electrolyte even under CO supply condition (Figure 7A-F) and
260 the observed size distribution matched well with the one produced during CO₂RR (Figure 7A-F).
261 Results for 1 mM formate are given in Figure S43. We therefore conclude that the significant
262 morphological change of copper particles is mostly caused by formate during CO₂RR.

263 The findings suggest that copper dissolution during CO₂RR is specifically induced by the
264 combined presence of formate and an applied cathodic potential. Importantly, this mechanism is
265 fundamentally distinct from classical oxidative corrosion. While cathodic conditions typically
266 suppress metal dissolution, strong ligands such as formate can directly complex with copper even
267 under reducing conditions. Studies have shown that the formation constants for copper-ligand
268 complexes²⁵ particularly with chelating or π -acceptor ligands like amines⁴⁹, acetate and formate²⁵,
269 ⁵⁰ are sufficiently high to render the overall free energy change for dissolution favorable, provided

270 that the chemical stabilization by the ligand offsets the unfavorable redox potential. This ligand-
271 induced dissolution does not require an external oxidant and is driven by direct complexation, not
272 by potential-driven oxidation. Also, interfacial availability of formate at copper-based cathodic
273 active sites may boost this complexation.



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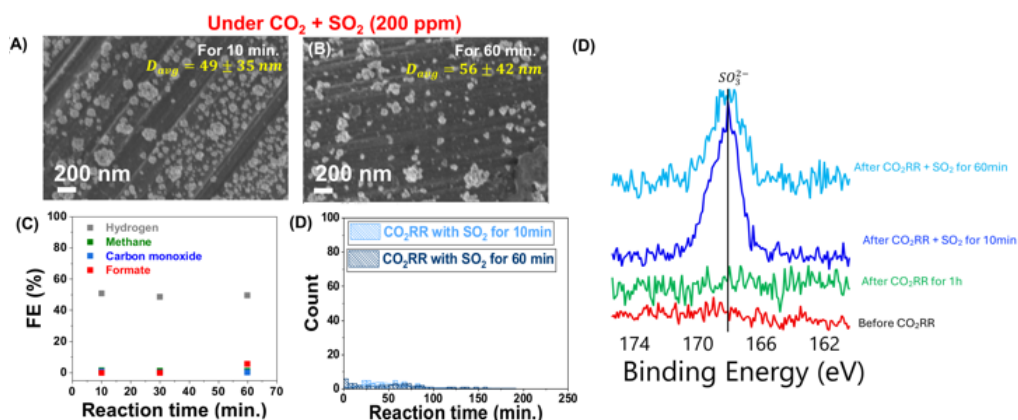
275 **Figure 7 Particle size analysis of control experiments. SE contrast SEM images after (A) *CO* reduction**
276 ***reaction control experiment* for 10 minutes and (B) 60 minutes. (C) *Added formate CO* reduction**
277 ***reaction control experiment* for 10 minutes and (D) 60 minutes. (E) Particle size distribution based on**
278 **SEM image analysis of the two control experiments for 10 minutes and (F) for 60 minutes.**

279 To test whether copper still changes its size once formate adsorption is blocked, we conducted SO₂
280 blocked formate adsorption control experiment in which we added SO₂ during CO₂RR reaction.
281 Particle size analysis of SEM images (Figure 8A+B) showed that after 10 min of CO₂RR in
282 presence of 200 ppm SO₂, the average particle size only slightly decreased, to 50 ± 35 nm. After
283 60 min of reaction, the average particle size was still 56 ± 42 nm. Product distribution also changed
284 in the presence of SO₂. Without SO₂, the FE of formate was 0 after 10 minutes, 6.64% after
285 30 minutes and 2.23% after 60 minutes (Figure 8D). In the presence of SO₂, no formate was
286 produced after 10 and 30 minutes, and FE was 5.48% after 60 minutes (Figure 8B). This control
287 experiment shows that when SO₂ is present, formate is formed later, and the copper particle size
288 does not significantly change within the first hour of reaction. One possible explanation for the
289 later onset of formate formation could be that its production is sensitive to the oxidation state. It
290 can be expected that the presence of SO₂ will affect the oxidation state of Cu.

291 The observed suppression in formate production is attributed to the strong adsorption of SO₂ on
292 copper surfaces⁵¹, which occurs preferentially at the same active sites responsible for CO₂
293 adsorption and activation. Although the precise atomic configuration of these active sites under
294 aqueous electrochemical CO₂RR remains elusive, surface science and theoretical studies
295 consistently indicate that SO₂ exhibits markedly stronger binding to copper than CO₂ or its
296 reduction intermediates⁵²⁻⁵⁴. Density functional theory (DFT) calculations further confirm the
297 higher adsorption energies of SO₂ relative to CO₂, which explains the competitive displacement of
298 CO₂ and consequent inhibition of formate⁵⁵. These findings agree with electrochemical
299 observations (Figure 8C) that SO₂ acts as a catalyst poison by occupying crucial copper sites, thus
300 reducing the available active area for CO₂ reduction to formate. While in-situ identification of the
301 SO₂ adsorption site under reaction conditions is still not available due to experimental challenges,
302 the collective evidence supports the conclusion that SO₂ hinders formate production primarily
303 through direct site competition and surface modification mechanisms.

304 Apart from the above control experiments, because previous studies⁵⁶ observed the appearance of
305 Cu 2S peaks in XPS analysis, we tested our samples using XPS, however we were not able to
306 observe this peak. We observed SO₃²⁻ peaks (Figure 8E). This peak could either be due to strong
307 interaction of copper with SO₂ or due to SO₂ oxidation in the electrolyte or interaction with surface

308 oxygen. Due to the applied reductive potential, SO₂ oxidation in the electrolyte is unlikely.⁵⁷ Since
 309 the SO₂ blocked formate adsorption control experiment did not show any formate production and
 310 no significant reduction in particle size, this further indicates that formate is accelerating the copper
 311 dissolution via a ligand-assisted dissolution mechanism.



312
 313 **Figure 8 Effect of SO₂ on CO₂RR. SEM images of electrode after 10 min (A) and 60 min (B) CO₂RR**
 314 **in presence of 200 ppm SO₂. (C) Faraday efficiency of reaction products as function of reaction time.**
 315 **(D) Particle size distribution of images shown in (A) and (B). (E) XPS data showing SO₂ binding.**

316 4. Analysis of the Dissolution of Copper Nanoparticles in the Electrolyte to Identify 317 Reaction Mechanism

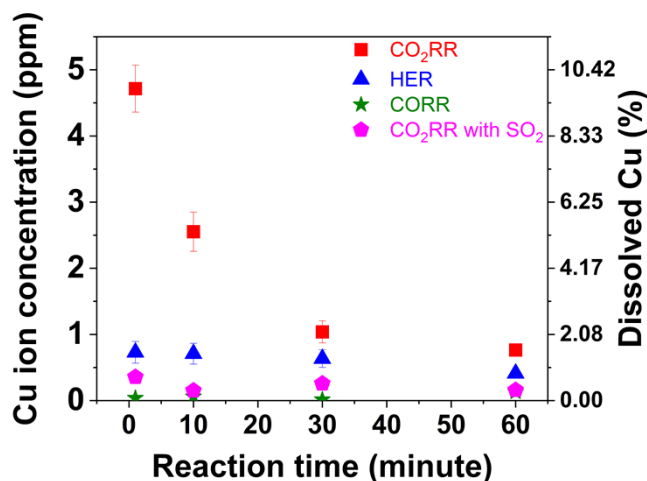
318 Due to the overestimation of particle size in the SEM analysis, it is not possible to distinguish
 319 based on SEM analysis whether the particle size reduction is due to the formation of smaller copper
 320 nanoparticles via a dissolution/reprecipitation mechanism or due to breaking up of agglomerates.
 321 One possible indication for the dissolution/reprecipitation mechanism would be the release of
 322 copper ions into the electrolyte solution.

323 To test whether copper is dissolved into the electrolyte, we analyzed the copper concentration in
 324 the electrolyte using inductively coupled plasma optical emission spectroscopy (ICP-OES) at
 325 different stages of CO₂RR as well as control experiments. As shown in Figure 9, we observed that
 326 ~10% of the total copper was dissolved (equivalent to 4.5 ppm copper detected in the solution)
 327 into the electrolyte even after 1 min of CO₂RR. The dissolved copper concentration decreased

328 exponentially, with 1.8 ppm after 10-minute reaction, 1.0 ppm after 30-minute reaction and
329 0.76 ppm after 1 hour reaction. In contrast, the hydrogen evolution reaction control experiment
330 showed much lower copper concentration (0.7 ppm after 1-minute reaction, 0.7 ppm after 10-
331 minute reaction, 0.6 ppm after 30-minute reaction and 0.4 ppm after 1 hour reaction), indicating
332 that HER does not lead to significant copper dissolution. Similar trends were observed in control
333 experiments for CORR and CO₂RR with SO₂, which yielded negligible copper dissolution over
334 the 60-minute observation period. This further supports our hypothesis regarding the interaction
335 between formate and the copper catalyst under electrochemical potential, suggesting that formate
336 induces both the dissolution and reprecipitation of the copper electrocatalyst during CO₂RR. A
337 table with ICP-OES measurement data is given in the supporting information (Table S5).

338 The temporal discrepancy between the observation of formate using NMR (Figure 8D), where
339 more formate is present after 60 minutes compared to 10 minutes and the aqueous copper
340 concentration (Figure 9), where the copper concentration is higher after 10 minutes compared to
341 60 minutes, can be possible explained by the following reasoning: (1) unidentified complexation
342 of formate by copper or potassium and (2) changes in reactive sites on copper particles due to
343 dissolution and reprecipitation. The focus of our NMR measurements is the quantification of
344 formate, which proton NMR with water suppression and DMSO as internal standard is a suitable
345 method for⁵⁸ and not the dynamics of formate complex formation in the solution. It can therefore
346 be possible that there is formate present in solution either as potassium formate or copper formate
347 complex. Additionally, previous studies have shown that different crystal morphologies³² or
348 defects⁵⁹ stabilize reaction intermediates differently. It can therefore be expected that due to the
349 dissolution and reprecipitation of the copper catalyst, the reactive sites on the copper particles are
350 modified.

351 Based on the particle sizes analysis derived from SEM image of the control experiments which is
352 indeed qualitative and ICP-OES results which is quantitative, we can conclude that the dissolution
353 of copper during CO₂RR relative to HER, CORR, CO₂RR with SO₂ is due to the presence of
354 formate as well as applied electrochemical potential and the dissolved copper ion were redeposited
355 in the form of smaller nanoparticles under the cathodic potential only during CO₂RR.



356

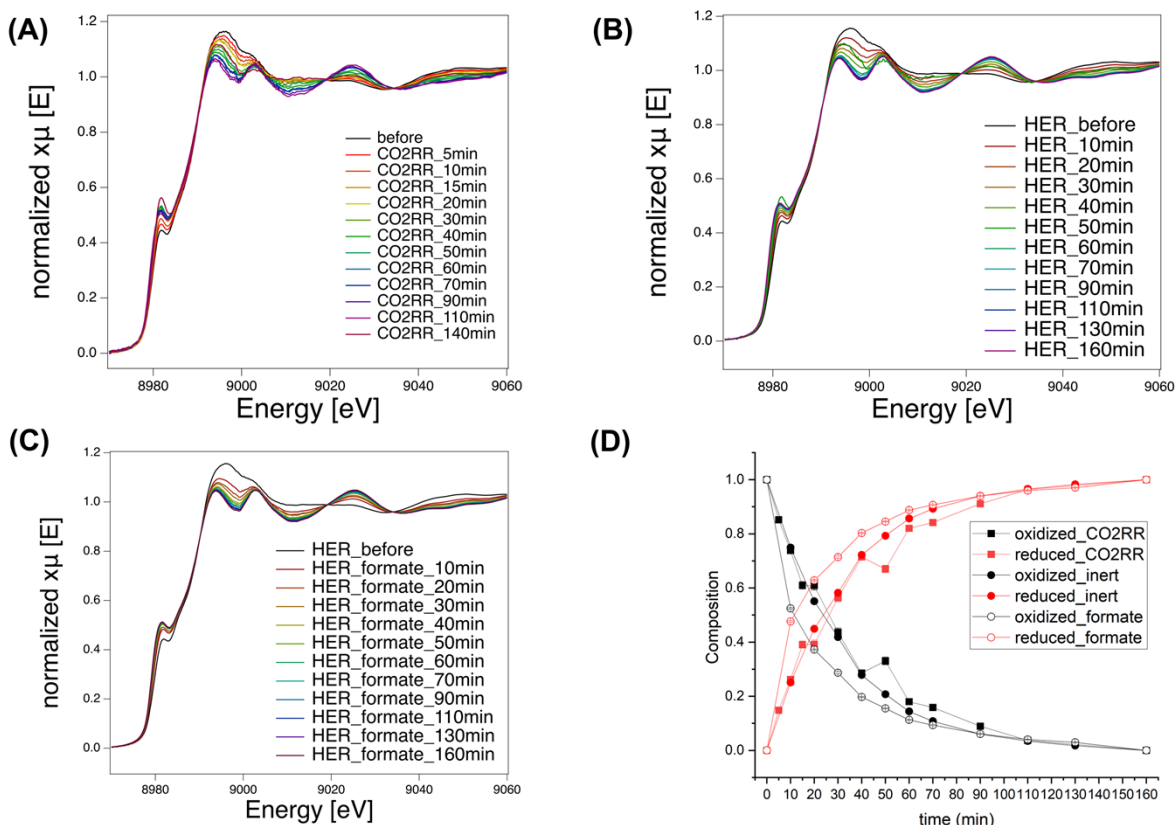
357 **Figure 9** Electrolyte copper concentration measured by inductively coupled plasma optical emission
 358 spectroscopy (ICP-OES) of electrolyte from CO₂RR, HER, CORR and CO₂RR with SO₂ reactions at
 359 different reaction times.

360 5. In-situ XAS

361 To monitor the structural evolution of copper in situ and further test our hypothesis that formate
 362 enhances copper dissolution, we collected Cu K edge XAS spectra in an in-situ electrochemistry
 363 setup (Figure S44). Three different experiments were conducted, (1) CO₂RR, (2) *HER control*
 364 experiment and (3) *added formate hydrogen evolution reduction (formate-HER) reaction control*
 365 *experiment*. As shown in Figure 10, before the reaction, oxidized copper can be detected. Under
 366 all reaction conditions, with the increase of the reaction time, the concentration of oxidized copper
 367 gradually decreased and that of metallic copper gradually increased. After ~20 minutes reaction,
 368 most of the copper is present in reduced form. After ~110 minutes, more than 90% of the copper
 369 are present in reduced state.

370 To compare the reduction rate of copper under the three different experimental conditions, linear
 371 combination fitting was performed. The fitting energy range is 8970 to 9030 eV. For the fitting,
 372 the two standards are the spectra collected before the reaction and the last scan collected at 160 min
 373 of reaction. As shown in Figure 10D, the reduction rate of copper follows the trend of *formate-*
 374 *HER* > *HER* > *CO₂RR* with *HER/CO₂RR* being very similar. By performing EXAFS analysis, it was
 375 found that after 160 min reaction, the Cu-Cu coordination number in the obtained Cu metallic

376 species is ~ 10 (Table 1), smaller than 12, suggesting that nanosized Cu dominates for all studied
 377 reactions.



378
 379 **Figure 10** In situ XAS results. Bias induced time-dependent change of Cu K-edge XANES spectra
 380 under (A) CO₂RR, (B) HER, (C) formate-HER. (D) The time-dependent change of copper composition
 381 (oxidized/reduced) in CO₂RR, HER, and formate-HER experiments.

382 **Table 1** The best fitting results obtained for the catalyst being exposed to CO₂RR for 160 min. The
 383 fitting k range is 2 - 11 Å⁻¹ and R range is 1 - 2.8 Å. The amplitude reduction factor (0.92) was
 384 obtained by fitting the spectrum of copper foil. Table provides N (coordination number), R (distance)
 385 (Å), σ^2 (variance) (Å²) and ΔE (energy shift) (eV).

Sample	Path	N	R (Å)	σ^2 (Å ²)	ΔE (eV)
Copper foil	Cu-Cu	12	2.544±0.002	0.0087±0.00 03	3.5±0.4

Before	Cu-O	1.4±0.2	1.90±0.01	0.004±0.002	10±1
	Cu-Cu	4.8±0.4	2.542±0.002	0.0075±0.00 06	5.1±0.7
HER_60min	Cu-Cu	9.2±0.5	2.540±0.002	0.0078±0.00 05	4.2±0.2
HER_70min	Cu-Cu	9.3±0.6	2.539±0.002	0.0079±0.00 05	4.2±0.2
HER_90min	Cu-Cu	9.4±0.5	2.540±0.002	0.0077±0.00 05	4.2±0.2
HER_110min	Cu-Cu	9.7±0.6	2.539±0.002	0.0077±0.00 05	4.2±0.2
HER_130min	Cu-Cu	9.7±0.5	2.540±0.002	0.0079±0.00 04	4.2±0.2
HER_160min	Cu-Cu	9.7±0.4	2.540±0.002	0.0077±0.00 04	4.2±0.2
Formate-HER- 60min	Cu-Cu	9.2±0.7	2.543±0.002	0.0080±0.00 06	4.7±0.2
Formate-HER- 70min	Cu-Cu	9.7±0.6	2.544±0.002	0.0083±0.00 06	4.7±0.2
Formate-HER- 90min	Cu-Cu	9.6±0.6	2.543±0.002	0.0081±0.00 05	4.7±0.2
Formate-HER- 110min	Cu-Cu	9.8±0.5	2.543±0.002	0.0081±0.00 04	4.7±0.2
Formate-HER- 130min	Cu-Cu	9.9±0.7	2.543±0.002	0.0081±0.00 06	4.7±0.2
Formate-HER- 160min	Cu-Cu	10.0±0. 5	2.543±0.002	0.0081±0.00 05	4.7±0.2
CO ₂ RR_60min	Cu-Cu	8.7±1.0	2.543±0.004	0.008±0.001	4.8±0.4

CO ₂ RR_70min	Cu-Cu	8.2±0.8	2.543±0.003	0.0073±0.0008	4.8±0.4
CO ₂ RR_90min	Cu-Cu	9.0±0.8	2.543±0.003	0.0078±0.0007	4.8±0.4
CO ₂ RR_110min	Cu-Cu	10.3±1.0	2.543±0.004	0.00791±0.0009	4.8±0.4
CO ₂ RR_160min	Cu-Cu	10.0±0.9	2.541±0.003	0.0086±0.0008	4.8±0.4

386 6. Comparison of In Situ XAS Results and H-Cell Results

387 When we compare the particle size changes measured by SEM/TEM and AFM analysis with in
388 situ XAS results, we observe contrasting qualitative results: results based on SEM/TEM and AFM
389 analysis show that the copper particle size decreases from larger particles (~30-90 nm) to single
390 digit nanometer size particles. XAS results show that the copper particle size is in the nanoparticle
391 size regime from the start of the experiment and does not significantly change, which is evident
392 from the coordination number of copper being close to 10 and not changing over the experiments.
393 One possibility that reconciles these seemingly contradicting results is that the copper particles
394 form agglomerates of oxidized particles due to oxidation during sample preparation. The particle
395 size observed in SEM, TEM and AFM therefore presents the agglomerate particle size. These
396 agglomerates break up during CO₂RR. The copper release into solution measured by ICP-OES
397 could be explained if we assume a shell of copper oxide forming around the reduced copper
398 particles, holding the agglomerates together, which then dissolved along boundaries between
399 particles. This would explain both why we are seeing oxidized particles being present in XAS
400 analysis, as well as different particle sizes observed in SEM/TEM/AFM compared with XAS.

401 Another potential difference between in situ XAS result and the lab electrocatalysis CO₂RR result
402 is the kinetics of the reduction of oxidized Cu to metallic Cu during CO₂RR. The oxidation state
403 of Cu could impact the dissolution rate of Cu during the reaction, which seems fast (less than 10
404 min) according to the formate production trend, the SEM/AFM analysis of Cu size and the ICP
405 analysis of dissolved Cu species. This contrasts with the in situ XAS result showing a rather slow

406 reduction rate. For example, about half of the copper is still in oxidized form for the first
407 20 minutes of reaction. Such a discrepancy in Cu oxidation state could be due to the difference in
408 the design of the electrochemical cell between in situ XAS study and the rest experiments. The
409 XAS electrochemical cell is a flow-through cell, whereas the rest of experiments used the H-cell
410 configuration. It can therefore be assumed that the reduction of copper in the H-cell is faster than
411 20 minutes, but we cannot state that with certainty since we do not have any data on copper
412 oxidation state during the reaction.

413 **Conclusion**

414 Morphological evolution of copper nanoparticles was investigated by combining in- and ex-situ
415 characterization to ascertain which intermediate is affecting the structural and morphological
416 change of the copper nanoparticles during CO₂RR. We controlled gas (Ar, CO, and CO₂) and
417 liquid (with/without formate) conditions. We find that the morphology as well as dissolution of
418 the copper nanoparticles in the electrolyte during electrical bias was significantly changed by
419 formate whereas *H and *CO did not significantly affect morphology and dissolution of the copper
420 nanoparticles. Our results indicate that the strong binding of formate with copper nanoparticles
421 causes ligand-assisted dissolution which drives the morphological and structural change of Cu
422 nanoparticles under CO₂RR.

423 **Experimental Section**

424 **7. Preparation of catalyst**

425 Copper nanoparticles were purchased from US research. 0.8 mg of catalyst was dispersed in a
426 mixture of 1.25 mL IPA (Isopropyl alcohol) and 7 μ L of ionomer (XA-9 Alkaline Ionomer 5% in
427 ethanol, Dioxide Materials). The mixture was sonicated for at least 20 minutes to produce the
428 copper nanoparticle ink and then the spray-casted on a carbon paper (Sigracet 39AA, Fuel Cell
429 store). For the anode preparation, IrO₂ powder (99%, IrO₂ powder, ThermoScientific) was affixed
430 on a titanium mesh. 24 mg of the IrO₂ were mixed with IPA and 12.5 μ L nafion (D520, Ion Power).
431 The mixture was sonicated for 20 minutes and then spray-casted on the titanium screen (Platinized

432 Titanium screen, Fuel Cell store). SEM imaging of the Cu catalyst showed multiple facets (Fig.
433 S1).

434 8. Electrochemical experiments

435 Electrochemical measurements were carried out with an H-Type electrochemical cell and SP-
436 200 potentiostat (BioLogic). All potentials were measured in volts against an Ag/AgCl reference
437 electrode (filled with 4 M KCl) and converted to the RHE reference scale using:

$$E \text{ (vs RHE)} = E \text{ (vs Ag/AgCl)} + 0.199 + 0.0591 \times pH, \quad (1)$$

438 where E (vs. RHE) represents the potential (V) of the reversible hydrogen electrode (RHE), E (vs
439 Ag/AgCl) represents the experimentally measured potential (V) against the Ag/AgCl reference,
440 and pH is the pH of the electrolyte (pH 7). The cathode and anode were separated by anion
441 exchange membrane (X37-50, Dioxide Materials). Electrolyte solution was 0.1 M KHCO₃
442 solution, which was prepared from 99.7% purity (237205-500G, Sigma Aldrich) salt with
443 deionized water (18 mΩ·cm). CO₂ gas was bubbled through the electrolyte to prepare CO₂-
444 saturated electrolyte for at least 30 minutes before starting the experiment. Then, CO₂ was
445 constantly bubbled with a flow rate of 10 mL/min through the electrolyte to prevent depletion of
446 CO₂ in the electrolyte and allow continuous analysis of gaseous products via a gas chromatograph.
447 For all electrochemical experiments, the experiments were started immediately after setup, and no
448 open circuit potential measurements were performed. Previous studies²³ have shown that the
449 duration of open circuit potential measurements can affect the copper degradation and we therefore
450 controlled for this variable. For hydrogen evolution reaction and carbon monoxide reduction
451 reaction control experiments, either Ar or CO gas (flow rate 10 mL/min) was supplied to the cell
452 after bubbling Ar or CO in the electrolyte for at least 30 minutes. Before conducting the control
453 hydrogen evolution reaction, a small volume of hydrochloric acid (50 μL) was added to the
454 electrolyte to lower the pH of the electrolyte to match the pH of CO₂ reduction condition. For the
455 potassium formate-added control experiment, 1 mM and 4 mM of potassium formate (BioUltra,
456 >99.0%, Sigma Aldrich) was added to 0.1 M KHCO₃ and measured with NMR to confirm the
457 formate peak.

458 In the SO₂ contaminant study, 10 mL/min of SO₂/Ar (200 ppm SO₂ balanced Ar, Airgas) gas was
 459 bubbled through the electrolyte for at least two hours and then mixed with 10 mL/min of CO₂ gas
 460 to initiate the CO₂RR reaction. pH measurements were conducted with Denver Instrument, Model
 461 250 using a three-point calibration.

462 9. Liquid and gas products analysis

463 545 μL of the reacted electrolyte was mixed with 50 μL of D₂O (Sigma Aldrich) and 5 μL of
 464 dimethyl sulfoxide (DMSO, 10 mM as an internal standard, >99.7%, Fisher)] for NMR (400 MHz
 465 NMR Spectrometer) measurements. The one-dimensional H spectrum was measured under water
 466 suppression mode. Figure S9 shows the NMR spectra of formate and Figure S10 gives the raw
 467 data. The Faradaic efficiency (FE) of liquid products was calculated as follows. First, the molar
 468 concentration of each liquid product (C_x) was calculated using the following formula:

$$C_x = \frac{I_{liquid}}{I_{DMSO}} \times \frac{N_{DMSO}}{N_{liquid}} \times C_{DMSO}, \quad (2)$$

469 where I, N, and C are the integral area of the NMR signal, the number of protons, and the
 470 concentration of the compound of interest (liquid product x) or the calibrant (DMSO), respectively.
 471 Then, the FE of liquid products was calculated using the following formula:

	$FE = \frac{q_1}{q_{tot}} = f \times c_x \times V \times z_1 / q_{tot} \times 100,$	(3)
--	-------------------------------------------------------------------------------------	-----

472 where q_1 , f , V , z_1 , q_{tot} are the charge consumed for generation of a particular liquid product, the
 473 faraday constant (C/mol), the volume of the reacted electrolyte (liters), the number of electrons to
 474 produce the liquid product, and the total charge ($q_{tot} = \int I dt$), respectively. The amount for each
 475 gaseous product, G_x , was quantified by gas chromatography (GC) equipped with both flame
 476 ionization detector (FID) and thermal conductivity detector (TCD) (SRI 8160C). The FE of the
 477 gas products were determined via the following formula:

	$FE = \frac{q_1 (output)}{q_{tot} (input)} = f \times G_x \times z_g / q_{tot} \times 100,$	(4)
--	---------------------------------------------------------------------------------------------	-----

478 where G_x is the amount of gas product in moles produced during a target reaction section, z_g is the
479 number of electrons to produce a gas product, and q_{tot} is the total charge determined by the
480 integration of current during the target reaction time session ($\int I dt$).

481 **10. Catalyst characterization**

482 The morphology of the samples was analyzed using scanning electron microscopy (SEM,
483 MERLIN SEM, Zeiss) at 1 kV, transmission electron microscopy (TEM) at 300 kV using a FEI
484 Titan TEM, and electrochemical atomic force microscopy (EC-AFM, MFD-3D). The tip was PNP-
485 TR-50. For EC-AFM, the leakless Ag/AgCl reference was used with customized titanium plate as
486 an anode. The bias and current data were collected by using a SP-200 potentiostat. The
487 compositions of the samples were studied using energy-dispersive X-ray spectroscopy (EDS)
488 system (MERLIN SEM, Zeiss) coupled to the SEM. X-ray diffraction (XRD) patterns were
489 collected using an PANalytical X-ray Diffractometer with a $\text{CuK}\alpha$ X-ray source ($\lambda = 1.540598 \text{ \AA}$),
490 soller slit (0.04 rad), and 0.026 steps. The peaks were analyzed with single/multi gauss fits from
491 Origin 2018 version software.

492 **11. ICP-OES**

493 Copper concentration in the electrolyte were analyzed using a ThermoFisher Scientific iCAP 7400
494 inductively coupled plasma - optical emission spectrometer (ICP-OES). 2% nitric acid for dilution
495 was prepared by mixing deionized water purified with a MilliQ system (Millipore) and Ultrapure
496 II nitric acid (J.T. Baker). Solutions were filtered with 0.22 μm syringe filters and then 1:10
497 dilutions of solutions were prepared in 2% HNO_3 . Copper emissions were measured in radial mode
498 at 324.754 and 327.396 nm. Internal standard was 5 ppm yttrium. A commercial elemental
499 standard for copper (High Purity Standards, Charleston, South Carolina, USA) was used to prepare
500 a 50 ppm stock solution. Intelligent dilution by the automatic sample loader (Elemental Scientific,
501 Omaha, Nebraska) was used to generate a seven-point standard curve with concentrations of 0.5,
502 1, 2, 5, 10, 20, and 50 ppm.

503 **12. XPS**

504 X-ray photoelectron spectroscopy (XPS) was performed using a Thermo Scientific (Waltham,
505 MA, USA) Model K-Alpha XPS instrument. The instrument utilizes monochromated, micro-
506 focused, Al K_α X-rays (1486.6 eV) with a variable spot size (i.e., 30-400 μm). Analyses of the
507 sample was performed with the 400 μm X-ray spot size for maximum signal and to obtain an
508 average surface composition over the largest possible area. The instrument has a hemispherical
509 electron energy analyzer equipped with a 128-channel detector system. Base pressure in the
510 analysis chamber is typically 2 x 10⁻⁹ mbar or lower. Samples were prepared for analysis by
511 attaching the samples directly to the XPS holder using metal clips. Survey spectra (pass
512 energy = 200 eV) were acquired for qualitative and quantitative analysis and high-resolution core
513 level spectra (pass energy = 50 eV) were acquired for detailed chemical state analysis. All spectra
514 were acquired with the charge neutralization flood gun turned on to maintain a stable analysis
515 condition. The flood gun uses a combination of low energy electrons and argon ions for optimum
516 charge compensation. The typical pressure in the analysis chamber with the flood gun operating
517 is 2 x 10⁻⁷ mbar.

518 **13. In situ X-ray absorption spectroscopy (XAS)**

519 In situ Cu K edge XAS data were collected at the QAS beamline (7-BM) of National Synchrotron
520 Light Source II (NSLS II), Brookhaven National Laboratory. All data were collected in
521 fluorescence mode. A three-electrode arrangement (the same setup as those described in
522 electrochemical measurements) was used for the in situ measurements. The measurements were
523 performed in a half-cell of a commercial in situ electrochemical cell (GDE XRD EC 1.75 ml, MSE
524 Supplies) with a Kapton tape window, allowing X-ray transmission. A picture of the setup is given
525 in Figure S8. Electrolyte preparation was the same as for H-type cell experiment. 0.8 mg of copper
526 was dispersed in a mixture of 1.25 mL IPA (Isopropyl alcohol) and 7 μL of ionomer (XA-9
527 Alkaline Ionomer 5% in ethanol, Dioxide Materials). The mixture was sonicated for at least
528 30 minutes to produce the copper nanoparticle ink and then spray-casted on a carbon paper (30%
529 hydrophobicity). Current data was monitored using a SP-300 potentiostat. The electrolyte was
530 circulated using a peristaltic pump at 10 mL/h. Three different experimental conditions were

531 tested: 1) CO₂RR, 2) HER 3) HER with added formate. For HER, Ar gas was bubbled through the
532 electrolyte for 1 hour. For HER with added formate, 4 mM potassium formate (BioUltra, >99.0%,
533 Sigma Aldrich) were added into the electrolyte before bubbling the electrolyte with Ar gas for
534 1 hour. XAS data were processed and analyzed by using the IFEFFIT package⁶⁰.

535 **14. Near Identical Location TEM**

536 Near Identical location TEM was performed using PELCO Pinpointer Grids. Fresh copper catalyst
537 was mixed with isopropanol, sonicated for 30 minutes and then drop casted onto the grid. A FEI
538 Titan instrument was used at 300 kV to collect TEM images before electrochemical experiments.
539 Electrochemical experiments were conducted in the H-cell setup described previously. After
540 5 minutes of CO₂RR, the grid was removed from solution and air dried prior to re-examination
541 using the TEM.

542 ASSOCIATED CONTENT

543 Supplementary PDF document containing: Transmission Electron Microscopy and Scanning
544 Electron Microscopy images analyzed for particle size distributions, tables with measurement data,
545 additional information on in-situ electrochemical atomic force microscopy experiments, additional
546 information on X-ray adsorption spectroscopy setup, Nuclear Magnetic Resonance data, gas
547 chromatography data, ICP-OES results

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569 Notes

570 The authors declare no competing financial interest.

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586 **References**

587 (1) Lin, Y.; Wang, T.; Zhang, L.; Zhang, G.; Li, L.; Chang, Q.; Pang, Z.; Gao, H.; Huang, K.;
588 Zhang, P.; et al. Tunable CO₂ electroreduction to ethanol and ethylene with controllable interfacial
589 wettability. *Nature Communications* **2023**, *14* (1), 3575. DOI: 10.1038/s41467-023-39351-2.

590 (2) Popović, S.; Smiljanić, M.; Jovanović, P.; Vavra, J.; Buonsanti, R.; Hodnik, N. Stability and
591 degradation mechanisms of copper-based catalysts for electrochemical CO₂ reduction.
592 *Angewandte Chemie* **2020**, *132* (35), 14844-14854.

593 (3) Nitopi, S.; Bertheussen, E.; Scott, S. B.; Liu, X.; Engstfeld, A. K.; Horch, S.; Seger, B.;
594 Stephens, I. E. L.; Chan, K.; Hahn, C.; et al. Progress and Perspectives of Electrochemical CO₂
595 Reduction on Copper in Aqueous Electrolyte. *Chemical Reviews* **2019**, *119* (12), 7610-7672. DOI:
596 10.1021/acs.chemrev.8b00705.

597 (4) Birdja, Y. Y.; Pérez-Gallent, E.; Figueiredo, M. C.; Göttle, A. J.; Calle-Vallejo, F.; Koper, M.
598 T. M. Advances and challenges in understanding the electrocatalytic conversion of carbon dioxide
599 to fuels. *Nature Energy* **2019**, *4* (9), 732-745. DOI: 10.1038/s41560-019-0450-y.

600 (5) Ummireddi, A. K.; Li, Z.; Wu, J. Copper defects for CO₂ electrocatalysis toward a specific
601 multicarbon product. *Trends in Chemistry* **2023**, *5* (3), 170-173.

602 (6) McCafferty, E. Thermodynamics of Corrosion: Pourbaix Diagrams. In *Introduction to*
603 *Corrosion Science*, 2010; pp 95-117.

604 (7) Kim, Y.-G.; Baricuatro, J. H.; Javier, A.; Gregoire, J. M.; Soriaga, M. P. The Evolution of the
605 Polycrystalline Copper Surface, First to Cu(111) and Then to Cu(100), at a Fixed CO₂RR
606 Potential: A Study by Operando EC-STM. *Langmuir* **2014**, *30* (50), 15053-15056. DOI:
607 10.1021/la504445g.

608 (8) Manthiram, K.; Beberwyck, B. J.; Alivisatos, A. P. Enhanced electrochemical methanation of
609 carbon dioxide with a dispersible nanoscale copper catalyst. *Journal of the American Chemical*
610 *Society* **2014**, *136* (38), 13319-13325.

- 611 (9) Gunathunge, C. M.; Li, X.; Li, J.; Hicks, R. P.; Ovalle, V. J.; Waegele, M. M. Spectroscopic
612 observation of reversible surface reconstruction of copper electrodes under CO₂ reduction. *The*
613 *Journal of Physical Chemistry C* **2017**, *121* (22), 12337-12344.
- 614 (10) Kim, D.; Kley, C. S.; Li, Y.; Yang, P. Copper nanoparticle ensembles for selective
615 electroreduction of CO₂ to C₂-C₃ products. *Proceedings of the National Academy of Sciences*
616 **2017**, *114* (40), 10560-10565.
- 617 (11) Grosse, P.; Gao, D.; Scholten, F.; Sinev, I.; Mistry, H.; Roldan Cuenya, B. Dynamic Changes
618 in the Structure, Chemical State and Catalytic Selectivity of Cu Nanocubes during CO₂
619 Electroreduction: Size and Support Effects. *Angew Chem Int Ed Engl* **2018**, *57* (21), 6192-6197.
620 DOI: 10.1002/anie.201802083 From NLM PubMed-not-MEDLINE.
- 621 (12) Grosse, P.; Yoon, A.; Rettenmaier, C.; Herzog, A.; Chee, S. W.; Roldan Cuenya, B. Dynamic
622 transformation of cubic copper catalysts during CO₂ electroreduction and its impact on catalytic
623 selectivity. *Nat Commun* **2021**, *12* (1), 6736. DOI: 10.1038/s41467-021-26743-5 From NLM
624 PubMed-not-MEDLINE.
- 625 (13) Simon, G. H.; Kley, C. S.; Roldan Cuenya, B. Potential-Dependent Morphology of Copper
626 Catalysts During CO₂ Electroreduction Revealed by In Situ Atomic Force Microscopy.
627 *Angewandte Chemie International Edition* **2021**, *60* (5), 2561-2568. DOI:
628 <https://doi.org/10.1002/anie.202010449>.
- 629 (14) Vavra, J.; Shen, T. H.; Stoian, D.; Tileli, V.; Buonsanti, R. Real-time Monitoring Reveals
630 Dissolution/Redeposition Mechanism in Copper Nanocatalysts during the Initial Stages of the CO₂
631 Reduction Reaction. *Angew Chem Int Ed Engl* **2021**, *60* (3), 1347-1354. DOI:
632 10.1002/anie.202011137 From NLM PubMed-not-MEDLINE.
- 633 (15) Lee, S. H.; Lin, J. C.; Farmand, M.; Landers, A. T.; Feaster, J. T.; Avilés Acosta, J. E.;
634 Beeman, J. W.; Ye, Y.; Yano, J.; Mehta, A.; et al. Oxidation State and Surface Reconstruction of
635 Cu under CO₂ Reduction Conditions from In Situ X-ray Characterization. *Journal of the American*
636 *Chemical Society* **2021**, *143* (2), 588-592. DOI: 10.1021/jacs.0c10017.
- 637 (16) Cherevko, F. D. S. S. Electrochemical copper dissolution: A benchmark for stable CO₂
638 reduction on copper electrocatalysts. *Electrochemistry Communications* **2020/06/01**, *115*, 106739.
639 DOI: 10.1016/j.elecom.2020.106739.
- 640 (17) Takamatsu, D.; Fukatani, N.; Yoneyama, A.; Hirano, T.; Hirai, K.; Yabuuchi, S.; Watanabe,
641 K.; Kamiya, K.; Nakanishi, S. Dynamic Relocation of Copper Catalysts in Gas Diffusion
642 Electrodes during CO₂ Electroreduction. *Journal of the American Chemical Society* **June 25**,
643 **2025**. DOI: 10.1021/jacs.5c07944.
- 644 (18) Tomc, B.; Tomc, B.; Bele, M.; Bele, M.; Nazrulla, M. A.; Nazrulla, M. A.; Šket, P.; Šket, P.;
645 Finšgar, M.; Finšgar, M.; et al. Deactivation of copper electrocatalysts during CO₂ reduction

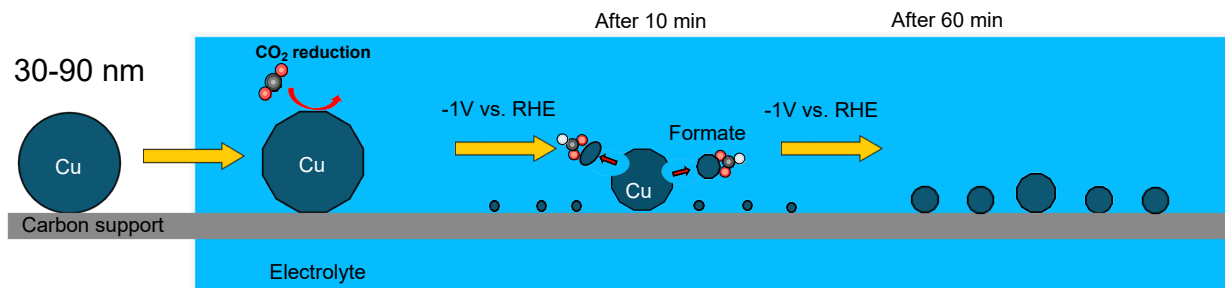
- 646 occurs via dissolution and selective redeposition mechanism. *Journal of Materials Chemistry A*
647 **2025/02/04**, 13 (6). DOI: 10.1039/D4TA06466F.
- 648 (19) Lee, S. H.; Avilés Acosta, J. E.; Lee, D.; Larson, D. M.; Li, H.; Chen, J.; Lee, J.; Erdem, E.;
649 Lee, D. U.; Blair, S. J. Structural Transformation and Degradation of Cu Oxide Nanocatalysts
650 during Electrochemical CO₂ Reduction. *Journal of the American Chemical Society* **2025**.
- 651 (20) Jung, H.; Lee, S. Y.; Lee, C. W.; Cho, M. K.; Won, D. H.; Kim, C.; Oh, H.-S.; Min, B. K.;
652 Hwang, Y. J. Electrochemical fragmentation of Cu₂O nanoparticles enhancing selective C–C
653 coupling from CO₂ reduction reaction. *Journal of the American Chemical Society* **2019**, 141 (11),
654 4624-4633.
- 655 (21) Huang, J.; Hörmann, N.; Oveisi, E.; Loiudice, A.; De Gregorio, G. L.; Andreussi, O.; Marzari,
656 N.; Buonsanti, R. Potential-induced nanoclustering of metallic catalysts during electrochemical
657 CO₂ reduction. *Nature communications* **2018**, 9 (1), 3117.
- 658 (22) Hori, Y.; Konishi, H.; Futamura, T.; Murata, A.; Koga, O.; Sakurai, H.; Oguma, K.
659 “Deactivation of copper electrode” in electrochemical reduction of CO₂. *Electrochimica acta*
660 **2005**, 50 (27), 5354-5369.
- 661 (23) Vavra, J.; Ramona, G. P. L.; Dattila, F.; Kormányos, A.; Priamushko, T.; Albertini, P. P.;
662 Loiudice, A.; Cherevko, S.; Lopéz, N.; Buonsanti, R. Solution-based Cu⁺ transient species mediate
663 the reconstruction of copper electrocatalysts for CO₂ reduction. *Nature Catalysis* **2024**, 7 (1), 89-
664 97. DOI: 10.1038/s41929-023-01070-8.
- 665 (24) Eren, B.; Zhrebetsky, D.; Patera, L. L.; Wu, C. H.; Bluhm, H.; Africh, C.; Wang, L.-W.;
666 Somorjai, G. A.; Salmeron, M. Activation of Cu(111) surface by decomposition into nanoclusters
667 driven by CO adsorption. *Science* **2016**, 351 (6272), 475-478. DOI: doi:10.1126/science.aad8868.
- 668 (25) Amirbeigi, R.; Tian, J.; Herzog, A.; Qiu, C.; Bergmann, A.; Roldan Cuenya, B.;
669 Magnussen, O. M. Atomic-scale surface restructuring of copper electrodes under CO₂
670 electroreduction conditions. *Nature Catalysis* **2023**, 6 (9), 837-846.
- 671 (26) Xu, L.; Lin, J.; Bai, Y.; Mavrikakis, M. Atomic and molecular adsorption on Cu (111). *Topics*
672 *in Catalysis* **2018**, 61, 736-750.
- 673 (27) Vollmer, S.; Witte, G.; Wöll, C. Determination of site specific adsorption energies of CO on
674 copper. *Catalysis letters* **2001**, 77, 97-101.
- 675 (28) Materials Data on Cu by Materials Project. LBNL Materials Project; Lawrence Berkeley
676 National Laboratory (LBNL), B., CA (United States), Ed.; 2020.
- 677 (29) Materials Data on Cu₂O by Materials Project. LBNL Materials Project; Lawrence Berkeley
678 National Laboratory (LBNL), B., CA (United States), Ed.; 2020.

- 679 (30) Ma, D.; Zhi, C.; Zhang, Y.; Chen, J.; Zhang, Y.; Shi, J.-W. A Review on the Influence of
680 Crystal Facets on the Product Selectivity of CO₂RR over Cu Metal Catalysts. *ACS nano* **2024**, *18*
681 (33), 21714-21746.
- 682 (31) Quan, Z.; Wang, Y.; Fang, J. High-index faceted noble metal nanocrystals. *Accounts of*
683 *Chemical Research* **2013**, *46* (2), 191-202.
- 684 (32) Durand, W. J.; Peterson, A. A.; Studt, F.; Abild-Pedersen, F.; Nørskov, J. K. Structure effects
685 on the energetics of the electrochemical reduction of CO₂ by copper surfaces. *Surface Science*
686 **2011**, *605* (15-16), 1354-1359.
- 687 (33) Han, C.; Kundi, V.; Ma, Z.; Toe, C. Y.; Kumar, P.; Tsounis, C.; Jiang, J.; Xi, S.; Han, Z.; Lu,
688 X. Differentiating the Impacts of Cu₂O Initial Low-and High-Index Facets on Their
689 Reconstruction and Catalytic Performance in Electrochemical CO₂ Reduction Reaction. *Advanced*
690 *Functional Materials* **2023**, *33* (12), 2210938.
- 691 (34) Huang, H.; Yue, K.; Liu, C.; Zhan, K.; Dong, H.; Yan, Y. CuO (111) Microcrystalline evoked
692 indium–organic framework for efficient electroreduction of CO₂ to formate. *Small* **2024**, *20* (34),
693 2400441.
- 694 (35) Zhong, W.; Chi, Y.; Yu, R.; Kong, C.; Zhou, S.; Han, C.; Vongsvivut, J.; Mao, G.;
695 Kalantar-Zadeh, K.; Amal, R. Liquid Metal-Enabled Tunable Synthesis of Nanoporous
696 Polycrystalline Copper for Selective CO₂-to-Formate Electrochemical Conversion. *Small* **2024**, *20*
697 (49), 2403939.
- 698 (36) Rahaman, M.; Dutta, A.; Zanetti, A.; Broekmann, P. Electrochemical reduction of CO₂ into
699 multicarbon alcohols on activated Cu mesh catalysts: an identical location (IL) study. *ACS*
700 *Catalysis* **2017**, *7* (11), 7946-7956.
- 701 (37) Scott, S. B.; Hogg, T. V.; Landers, A. T.; Maagaard, T.; Bertheussen, E.; Lin, J. C.; Davis, R.
702 C.; Beeman, J. W.; Higgins, D.; Drisdell, W. S. Absence of oxidized phases in Cu under CO
703 reduction conditions. *ACS Energy Letters* **2019**, *4* (3), 803-804.
- 704 (38) Reske, R.; Mistry, H.; Behafarid, F.; Roldan Cuenya, B.; Strasser, P. Particle size effects in
705 the catalytic electroreduction of CO₂ on Cu nanoparticles. *Journal of the American Chemical*
706 *Society* **2014**, *136* (19), 6978-6986.
- 707 (39) Ludwig, C.; Casey, W. H.; Rock, P. A. Prediction of ligand-promoted dissolution rates from
708 the reactivities of aqueous complexes. *Nature* **1995**, *375* (6526), 44-47.
- 709 (40) Holmén, B. A.; Casey, W. H. Hydroxamate ligands, surface chemistry, and the mechanism of
710 ligand-promoted dissolution of goethite [α -FeOOH (s)]. *Geochimica et Cosmochimica Acta* **1996**,
711 *60* (22), 4403-4416.

- 712 (41) Furrer, G.; Stumm, W. The coordination chemistry of weathering: I. Dissolution kinetics of
713 δ -Al₂O₃ and BeO. *Geochimica et Cosmochimica Acta* **1986**, *50* (9), 1847-1860.
- 714 (42) Molis, E.; Barres, O.; Marchand, H.; Sauzéat, E.; Humbert, B.; Thomas, F. Initial steps of
715 ligand-promoted dissolution of gibbsite. *Colloids and Surfaces A: Physicochemical and*
716 *Engineering Aspects* **2000**, *163* (2-3), 283-292.
- 717 (43) Heliövaara, E.; Liljeqvist, H.; Muuronen, M.; Eronen, A.; Moslova, K.; Repo, T. Cooperative
718 Ligands in Dissolution of Gold. *Chemistry—A European Journal* **2021**, *27* (34), 8668-8672.
- 719 (44) Guan, Y.; Kümper, J.; Mürtz, S. D.; Kumari, S.; Hausoul, P. J.; Palkovits, R.; Sautet, P. Origin
720 of copper dissolution under electrocatalytic reduction conditions involving amines. *Chemical*
721 *Science* **2024**, *15* (35), 14485-14496.
- 722 (45) Liu, S.; Liu, Y.; Pan, B.; He, Y.; Li, B.; Zhou, D.; Xiao, Y.; Qiu, H.; Vijver, M. G.;
723 Peijnenburg, W. J. The promoted dissolution of copper oxide nanoparticles by dissolved humic
724 acid: Copper complexation over particle dispersion. *Chemosphere* **2020**, *245*, 125612.
- 725 (46) Hori, Y.; Murata, A.; Takahashi, R. Formation of hydrocarbons in the electrochemical
726 reduction of carbon dioxide at a copper electrode in aqueous solution. *Journal of the Chemical*
727 *Society, Faraday Transactions 1: Physical Chemistry in Condensed Phases* **1989**, *85* (8), 2309-
728 2326.
- 729 (47) DeWulf, D. W.; Jin, T.; Bard, A. J. Electrochemical and surface studies of carbon dioxide
730 reduction to methane and ethylene at copper electrodes in aqueous solutions. *Journal of the*
731 *Electrochemical Society* **1989**, *136* (6), 1686.
- 732 (48) Feaster, J. T.; Shi, C.; Cave, E. R.; Hatsukade, T.; Abram, D. N.; Kuhl, K. P.; Hahn, C.;
733 Nørskov, J. K.; Jaramillo, T. F. Understanding Selectivity for the Electrochemical Reduction of
734 Carbon Dioxide to Formic Acid and Carbon Monoxide on Metal Electrodes. *ACS Catalysis* **2017**,
735 *7* (7), 4822-4827. DOI: 10.1021/acscatal.7b00687.
- 736 (49) Guan, Y.; Guan, Y.; Kümper, J.; Kümper, J.; Mürtz, S. D.; Mürtz, S. D.; Kumari, S.; Kumari,
737 S.; Hausoul, P. J. C.; Hausoul, P. J. C.; et al. Origin of copper dissolution under electrocatalytic
738 reduction conditions involving amines. *Chemical Science* **2024/09/11**, *15* (35). DOI:
739 10.1039/D4SC01944J.
- 740 (50) V.N. Krasil'nikov, V. P. Z., E.V. Chulkov, I.V. Baklanova, D.G. Kellerman, O.I. Gyrdasova,
741 ; T.V. Dyachkova, A. P. T. Novel method for the production of copper(II) formates, their thermal,
742 spectral and magnetic properties. *Journal of Alloys and Compounds* **2020/12/10**, *845*. DOI:
743 10.1016/j.jallcom.2020.156208.
- 744 (51) C. M. Pradier*, a.; Dubot‡, P. Adsorption and Reactivity of Sulfur Dioxide on Cu(110).
745 Influence of Oxidation and Hydroxylation of the Surface. **June 6, 1998**. DOI: 10.1021/jp981000.

- 746 (52) Polčik, M.; Wilde, L.; Haase, J. $\{\mathrm{SO}\}_2$ -induced surface reconstruction of
747 Cu(111): An x-ray-absorption fine-structure study. *Physical Review B* **1998-01-15**, 57 (3). DOI:
748 10.1103/PhysRevB.57.1868.
- 749 (53) Leung, K. T.; Zhang, X. S.; Shirley, D. A. Adsorption and surface reactions of hydrogen
750 sulfide and sulfur dioxide on copper(100) studied by electron energy loss spectroscopy. **May 1,**
751 **2002**. DOI: 10.1021/j100353.
- 752 (54) Alemozafar, A. R.; , X.-C. G., Robert J. Madix. Topographic nano-restructuring: sulfur
753 dioxide adsorption on Cu(110). *Surface Science* **2003/02/01**, 524 (1-3). DOI: 10.1016/S0039-
754 6028(02)02539-6.
- 755 (55) Behjatmanesh-Ardakani, R.; Magkoev, T. T. SO₂ adsorption and its direct proportional
756 dissociation on the Cu(100), Cu(110), and Cu(111) surfaces: a periodic DFT study. *Chemical*
757 *Review and Letters* **2024/11/01**, 7 (5). DOI: 10.22034/crl.2024.467740.1380.
- 758 (56) Luc, W.; Ko, B. H.; Kattel, S.; Li, S.; Su, D.; Chen, J. G.; Jiao, F. SO₂-induced selectivity
759 change in CO₂ electroreduction. *Journal of the American Chemical Society* **2019**, 141 (25), 9902-
760 9909.
- 761 (57) Giovannelli, G.; Natali, S.; Zortea, L.; Bozzini, B. An investigation into the surface layers
762 formed on oxidised copper exposed to SO₂ in humid air under hypoxic conditions. *Corrosion*
763 *science* **2012**, 57, 104-113.
- 764 (58) Hatab, A. S. A.; Ahmad, Y. H.; Mady, M. F.; Hassan, Y.; Zkria, A.; Sinopoli, A.; Abdullah,
765 A. M.; Al-Qaradawi, S. Y.; Yoshitake, T.; Khaled, M. Nuclear magnetic resonance spectroscopy:
766 A comprehensive tool for analyzing liquid products in electrochemical CO₂ reduction. *Journal of*
767 *Electroanalytical Chemistry* **2025**, 986, 119097.
- 768 (59) Cheng, D.; Nguyen, K.-L. C.; Sumaria, V.; Wei, Z.; Zhang, Z.; Gee, W.; Li, Y.; Morales-
769 Guio, C. G.; Heyde, M.; Roldan Cuenya, B. Structure sensitivity and catalyst restructuring for CO₂
770 electro-reduction on copper. *Nature Communications* **2025**, 16 (1), 4064.
- 771 (60) Ravel, B.; Newville, M. ATHENA, ARTEMIS, HEPHAESTUS: data analysis for X-ray
772 absorption spectroscopy using IFEFFIT. *Synchrotron Radiation* **2005**, 12 (4), 537-541.

773 **Table of Content Graphic**



(2) Strong binding of formic acid intermediate

(3) Small growth of the small and new NPs (20-60 nm)

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