

# Energy-Efficient Capacitive Deionization through Electrode Modification and Process Development

*Lauren Valentino<sup>a\*</sup>, Lily Callen<sup>a#</sup>, Samantha Kelly<sup>a\$</sup>, Sungjoon Kim<sup>a</sup>*

<sup>a</sup>Applied Materials Division, Argonne National Laboratory, Lemont, IL, 60439, USA

\*Corresponding author [phone: +1 630 252 5029; e-mail: lvalentino@anl.gov]

## ABSTRACT

Electrochemical separation technologies, such as capacitive deionization (CDI), are promising for addressing global energy and water challenges. However, there is a need to improve the performance, better understand property-performance relationships, and evaluate the longevity of CDI electrodes. This study explores the chemical modification of electrodes and the adjustment of CDI operating parameters. Results indicate that nitric acid (HNO<sub>3</sub>) conditioning of activated carbon cloth (ACC) electrodes removes metal oxides, introduces oxygen and nitrogen functionalities, and increases the specific capacitance (16% at 1 mV/s). Moreover, these changes in electrode properties positively impact device-level CDI performance. Through HNO<sub>3</sub>-conditioning of the ACC and tuning of the operational parameters, this work demonstrates higher electrosorption capacity (4.0×), greater charge efficiency (90% vs. 24%), and lower energy consumption (3.8×). Despite these enhancements, limitations of the HNO<sub>3</sub>-conditioned ACC include decreased desorption kinetics and a 32% loss in electrosorption capacity after 200 cycles. Overall, this work provides guidance on using oxidative pretreatment via HNO<sub>3</sub> to modify ACC electrodes for CDI and evaluates the tradeoffs associated with varying operational parameters.

## KEYWORDS

Separation, Electrosorption, Activated carbon, Pretreatment, Flow rate, Half-cycle time

## INTRODUCTION

Electrification has been recognized as a key strategy for mitigating greenhouse gas emissions and decarbonizing energy supply chains.<sup>1,2</sup> In line with this paradigm shift, electrochemical separation technologies, including capacitive deionization (CDI), that eliminate the need for high pressures or temperatures are gaining attention for low salinity deionization and resource recovery applications. CDI involves the application of an electric potential across a pair of porous electrodes, creating an electric field that drives ion transport towards oppositely charged electrodes as shown in Figure S1. Ions are electrostatically immobilized and stored capacitively in the electrical double layers (EDLs) within the electrode micropores, resulting in salt extraction from solution. The electrodes are then regenerated by removing or reversing the potential. CDI usually operates through a cyclic process to extract and recover ions. A typical CDI cell is comprised of porous carbon-based electrodes that are separated by a spacer, allowing feed solution to flow along the surface of the electrodes.<sup>3-5</sup> Activated carbon is particularly attractive for CDI because of its high specific surface area ( $S_{BET}$ ), porosity, commercial availability,<sup>6,7</sup> and low cost relative to other carbon materials.

Several strategies have been employed to further enhance the properties of activated carbon for CDI.<sup>8,9</sup> In particular, nitric acid ( $\text{HNO}_3$ ) has been used to modify the surface chemistry by introducing Oxygen (O)-containing functional groups,<sup>10-14</sup> which are known to increase the specific capacitance ( $C_s$ ).<sup>15-18</sup> Previous studies have employed  $\text{HNO}_3$ -conditioned electrodes in CDI and demonstrated enhanced ion extraction and kinetics; however, these studies lack

comprehensive evaluation of other performance metrics, *e.g.*, charge efficiency ( $A$ ) and specific energy consumption ( $E$ ).<sup>19,20</sup> Only two studies have reported data for adsorption capacity ( $\Gamma_A$ ) and  $A$ , but they are limited to a single  $\text{HNO}_3$  exposure.<sup>21,22</sup> While other studies have reported  $\text{HNO}_3$  pretreatment of CDI electrodes,<sup>23-26</sup> they do not specifically evaluate the impacts of pretreatment.

CDI performance is also largely dependent on the operating parameters. Using a constant voltage of 1.2 V provides a high driving force while minimizing undesirable side reactions.<sup>7,27</sup> In general,  $\Gamma_A$  increases as charging time increases, but the average salt adsorption rate ( $\bar{r}_A$ ) decreases.<sup>27,28</sup> Previous work has shown that increasing the flow rate can enhance  $\Gamma_A$ , but beyond a certain threshold, increases in flow rate reduce  $\Gamma_A$  due to insufficient contact time.<sup>29,30</sup> Since there is not a singular indicator of performance in CDI, assessing the optimal operating conditions is complex, and unoptimized operation can lead to inefficiencies that are detrimental to process feasibility. In addition, among the studies that implemented  $\text{HNO}_3$ -conditioned activated carbon for CDI, none have examined the impact of varying the flow rate or half-cycle time (HCT).

This study assesses the effects of  $\text{HNO}_3$  modification on the performance of activated carbon cloth (ACC) electrodes in CDI and further explores the synergy between these modifications and key operating parameters, specifically flow rate and HCT. Changes to the surface morphology and topology, chemical composition, electrochemical and porous properties, and CDI performance were assessed. Aside from the introduction of O- and Nitrogen (N)-containing groups, this study uniquely demonstrates the benefit of using  $\text{HNO}_3$  to remove residual metal oxides from the ACC along with a limitation regarding decreased desorption kinetics. After establishing the optimal pretreatment solution, CDI operating parameters were varied to further enhance separation performance. This is the first study to examine the combined effects of electrode modification and varying operating conditions on the performance of CDI. In addition,

the longevity of the electrodes was assessed by monitoring a range of CDI performance metrics over 200 continuous cycles. Through this investigation, this work advances both foundational and applied principles related to CDI for deionization applications and is essential for scaling up and transitioning CDI from bench-scale to larger-scale applications.

## **METHODS**

### Materials and Reagents

Commercially available ACC (Flexzorb FM10, Chemviron Carbon, Houghton le Spring, UK) was modified by immersing in one of 6 solutions: Milli-Q water (18.2 M $\Omega$ -cm resistivity), 1-4 M HNO<sub>3</sub>, or 3 M sodium nitrate (NaNO<sub>3</sub>) for 24 h. After thorough rinsing, ACC was cut to size, dried for 24 h at 100 °C, and weighed immediately after removal from the oven. The manufacturer specifications for FM10 and pretreatment conditions are provided in Tables S1 and S2. ACS reagent grade HNO<sub>3</sub> (70%) and sodium chloride (NaCl;  $\geq$ 99.0%) were obtained from Sigma Aldrich (St. Louis, MO) and Fisher Scientific (Waltham, MA), respectively, and diluted or dissolved using Milli-Q water.

### Analytical Characterization Techniques

X-ray photoelectron spectroscopy (XPS) was performed using a K-Alpha+ spectrometer (Thermo Fisher Scientific, Waltham, MA) with a monochromatic Al K $\alpha$  X-ray source (1487 eV) and a 400  $\mu$ m spot size. ACC was adhered to conductive Copper tape to minimize sample charging, survey spectra were collected using a pass energy of 200 eV and a step size of 1 eV, and high-resolution scans were collected with a pass energy of 50 eV and a step size of 0.1 eV. Data interpretation was performed by Thermo Fisher Advantage software with each measurement representing an average of 5 scans. All spectra were calibrated with the C1s peak at 284.8 eV.

Nitrogen (N<sub>2</sub>) gas sorption isotherms were measured at -196 °C using a surface area and porosity analyzer (3Flex, Micromeritics Instrument Corporation, Norcross, GA) after degassing at 100 °C for 24 h.  $S_{BET}$  and the pore size distribution were determined using Brunauer–Emmett–Teller (BET) theory and the Non-Local Density Functional Theory (NLDFT) model, respectively.  $S_{BET}$  was calculated using the adsorption data with Rouquerol criteria.<sup>31</sup> The total pore volume ( $V_{tot}$ ) was calculated using the adsorption data and the carbon slit-pore model for N<sub>2</sub> at 77 K.

Cyclic voltammetry (CV) was performed by scanning from -0.5 to 0.5 V using a range of scan rates ( $k$ ; 1-5 mV/s). The working electrode, a 3 cm<sup>2</sup> swatch of ACC, was mounted on a Platinum sheet, installed in a custom-made holder, and immersed in 0.5 M NaCl for 24 h prior to measurements. A graphite rod and Ag/AgCl electrode were used as the counter and reference electrode, respectively. The electrolyte was sparged with N<sub>2</sub> gas before testing, and a low flow of gas was continually flushed into the headspace during measurements.  $C_s$  was calculated using Equation S1.

#### CDI Performance Evaluation

CDI experiments were conducted using a custom-built electrochemical cell (14 cm<sup>2</sup> active area), consisting of a pair of electrodes separated by 1.4 mm. Solution flowed between the electrodes (Figure S2) at a rate of 4-16 mL/min. A LabVIEW (National Instruments, Austin, TX) program was used for control and data acquisition along with a DC power supply and flow-through conductivity probe (ET908, eDAQ, Colorado Springs, CO). CDI performance is described by several performance metrics (Equations S2-S8), which were calculated for each cycle.

## RESULTS AND DISCUSSION

### Elemental and Functional Group Analysis

XPS of the 6 ACC materials reveals key components: Carbon, N, and O, for which elemental analysis is presented in Table 1. The full compositional analysis, provided in Table S3, includes Aluminum and Zinc. These metals, likely a residual from the commercial manufacturing process, are leached from the ACC during HNO<sub>3</sub> conditioning. This is confirmed by scanning electron microscopy (SEM), energy dispersive X-ray spectroscopy (EDS), and inductively coupled plasma optical emission spectroscopy (ICP-OES), as described in detail in the Supplementary Information (SI; Table S3 and Figures S3-S6).

Table 1. Elemental analysis in atomic (at.) % characterized by XPS, curve fitting of the O1s and N1s regions from XPS, and  $S_{BET}$  and  $V_{tot}$  for ACC conditioned in 3 M NaNO<sub>3</sub> (control, C), Milli-Q water (0), or 1-4 M HNO<sub>3</sub>. XPS was performed using a 400  $\mu$ m spot size and by averaging 5 individual scans to account for spatial variability. The values for  $S_{BET}$  and  $V_{tot}$  represent 2 measurements and are presented as the average  $\pm$  standard deviation.

|                                      |                                | [HNO <sub>3</sub> ] (M) |             |             |             |             |             |     |
|--------------------------------------|--------------------------------|-------------------------|-------------|-------------|-------------|-------------|-------------|-----|
|                                      |                                | C                       | 0           | 1           | 2           | 3           | 4           |     |
| Elemental composition by XPS (at. %) | C                              | 82.7                    | 79.9        | 87.0        | 82.0        | 84.0        | 81.0        |     |
|                                      | N                              | 1.6                     | 1.5         | 2.0         | 2.1         | 2.4         | 2.1         |     |
|                                      | O                              | 13.6                    | 14.8        | 9.8         | 13.2        | 11.9        | 14.2        |     |
|                                      | O/C                            | 0.16                    | 0.19        | 0.11        | 0.16        | 0.14        | 0.18        |     |
|                                      |                                |                         |             |             |             |             |             |     |
| Functional groups (at. %)            | O1s                            | O-C=O                   | 5.4         | 4.6         | 5.7         | 8.7         | 7.2         | 8.4 |
|                                      |                                | C=O                     | 7.5         | 9.6         | 3.6         | 3.9         | 4.1         | 4.9 |
|                                      |                                | C-O                     | 0.6         | 0.7         | 0.4         | 0.7         | 0.7         | 0.9 |
|                                      | N1s                            | N-6                     | 0.1         | 0.3         | 0.2         | 0.1         | 0.1         | 0.1 |
|                                      |                                | N-Q                     | 1.5         | 1.2         | 1.5         | 1.6         | 1.7         | 1.3 |
|                                      |                                | -NO2                    | -           | -           | 0.3         | 0.4         | 0.7         | 0.8 |
|                                      |                                |                         |             |             |             |             |             |     |
| Textural properties                  | $S_{BET}$ (m <sup>2</sup> /g)  | 1235 ± 29               | 1198 ± 0    | 1217 ± 4    | 1255 ± 45   | 1153 ± 1    | 1172 ± 13   |     |
|                                      | $V_{tot}$ (cm <sup>3</sup> /g) | 0.53 ± 0.01             | 0.53 ± 0.03 | 0.54 ± 0.03 | 0.57 ± 0.04 | 0.50 ± 0.01 | 0.52 ± 0.01 |     |

The data in Table 1 show that as [HNO<sub>3</sub>] increases from 1 to 4 M, there is an increase in surface O concentration. Further, the O/C ratio, which can be used as an indicator of surface oxidation, increases by 64%. Previous work has shown that the aliphatic chains are particularly susceptible to oxidation, producing carboxylic and ketone groups.<sup>10</sup> After curve fitting of the high-resolution O1s spectra (Figures 1 and S7), there are increases in carboxylic (O-C=O), carbonyl (C=O), and alcohol/ether (C-O) groups as [HNO<sub>3</sub>] increases from 1 M to 4 M. It is well established that the presence of these functional groups contributes to surface acidity.<sup>17,32</sup> These groups also increase the surface charge density and hydrophilicity, which can enhance  $\Gamma_A$ .<sup>8,14,33</sup> This is because EDLs, the primary mechanism through which ions are removed in CDI, can only be formed at solution-electrode interfaces,<sup>34</sup> so higher wettability facilitates greater infiltration of the electrolyte into the porous ACC.<sup>7,9</sup>

Table 1 also reveals a slight increase in N, which is consistent with literature,<sup>12,35-38</sup> although most studies focus on the incorporation of O-containing groups. After curve fitting, all high-resolution N1s spectra include a peak at ~401 eV that is attributed to aromatic amines (N-Q)<sup>39</sup> and a peak at ~398 eV that is associated with more reduced, pyridinic nitrogen (N-6).<sup>37,39-41</sup> In comparison to the ACC conditioned in water and NaNO<sub>3</sub>, the spectra for the ACC conditioned in 1-4 M HNO<sub>3</sub> exhibit a unique peak at ~406 eV, associated with nitro groups (-NO<sub>2</sub>).<sup>37,39-41</sup> The emergence of this peak is further illustrated in Figure S8a. The residuals from the spectral fitting are plotted in Figure S8b to demonstrate that the spectra for ACC conditioned with Milli-Q water and 3 M NaNO<sub>3</sub> are adequately represented without the nitro group peak at 406 eV. Both -NO<sub>2</sub> and N-6 functional groups can enhance the wettability and  $C_s$  of electrodes,<sup>42,43</sup> including activated carbon.<sup>40,44</sup> Overall, these results directly link HNO<sub>3</sub> conditioning to ACC oxidation, marking the first report connecting -NO<sub>2</sub> functionalization of ACC to enhanced CDI performance. Finally,

although Fourier transform infrared (FTIR) spectroscopy was performed, the analysis reveals no obvious differences. Additional detail about the FTIR methodology and results are presented in the SI (Figure S9).

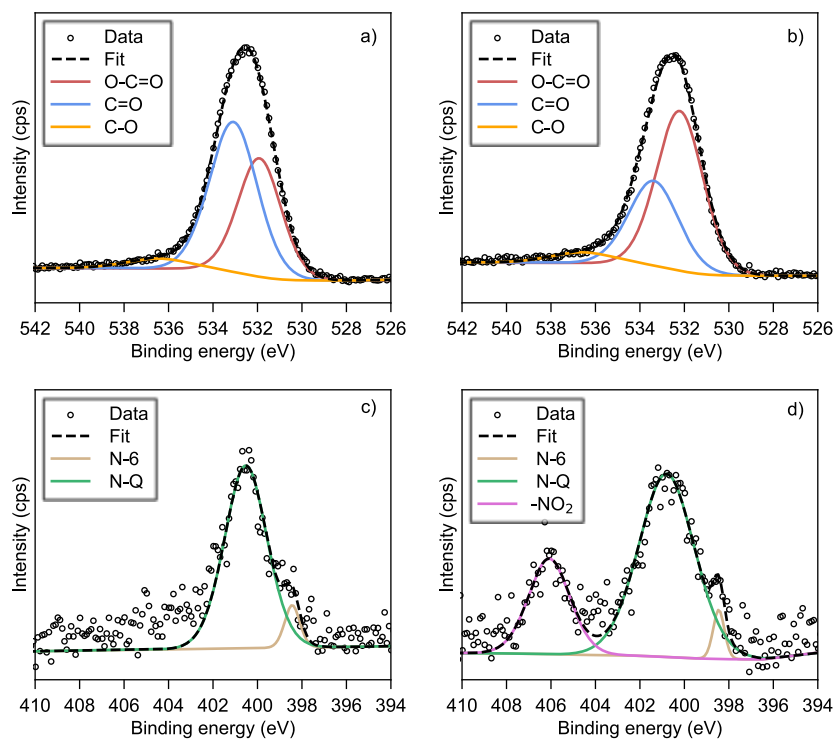


Figure 1. Representative O1s and N1s spectra for ACC conditioned with a and c) 3 M NaNO<sub>3</sub> and b and d) 3 M HNO<sub>3</sub>. Three components (O1s: O-C=O, C=O, and C-O and N1s: N-6, N-Q, and -NO<sub>2</sub>) were applied for O1s and N1s spectral fitting.

### Electrochemical Characterization

Figure 2 shows the CV characterization results at various  $k$  (1-5 mV/s). The voltammograms collected using lower  $k$  have semi-rectangular shapes that are typically associated with EDL capacitors. In addition to EDL capacitance, small pseudocapacitive contributions are observed for all ACCs. These are more pronounced for ACC conditioned with 2-4 M HNO<sub>3</sub>

(Figures 2a and 2b), and the slight shift in the pseudocapacitive peak position is consistent with changes to the electrode surface observed by XPS. Similar pseudocapacitive peaks have been reported in literature for HNO<sub>3</sub>-treated activated carbon.<sup>20,40</sup> Specifically, the presence of N and O functional groups on the surface of carbon materials has been reported to induce pseudocapacitance or redox activity.<sup>40,42-44</sup> Additionally, as  $k$  increases, the current response, *i.e.*, current density ( $J$ ), increases. The CVs for ACCs conditioned with NaNO<sub>3</sub> and water maintain the same general shape. However, consistent with published literature,<sup>8,19,25</sup> the ACCs conditioned with 2-4 M HNO<sub>3</sub> appear more lens-shaped at 5 mV/s, suggesting greater deviation from ideal capacitive behavior.

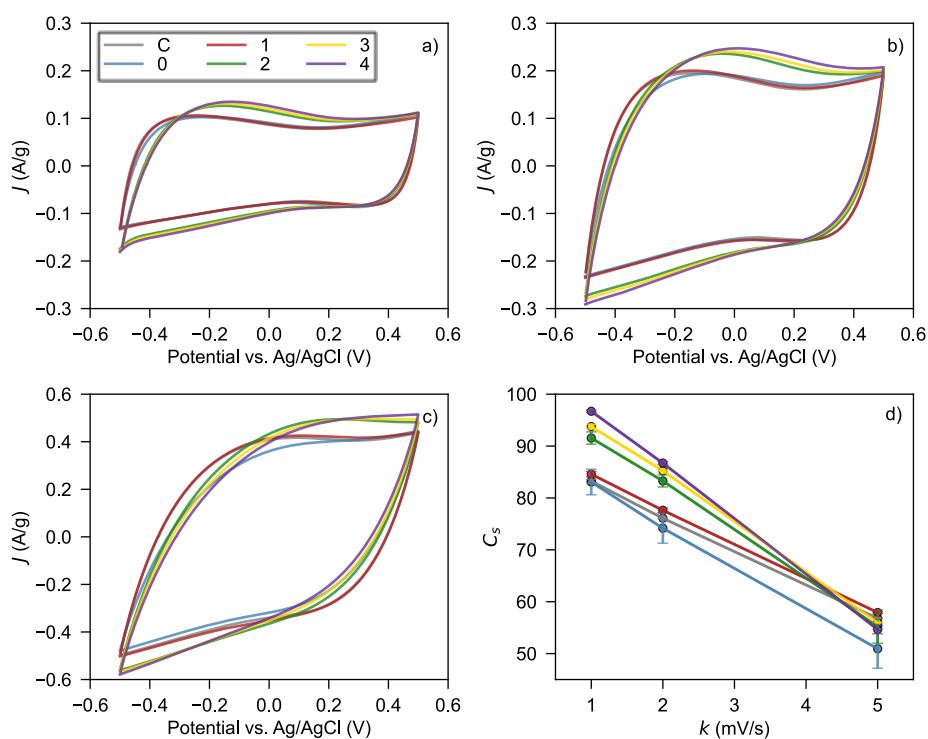


Figure 2. Representative voltammograms collected at a)  $k = 1$  mV/s, b)  $k = 2$  mV/s, and c)  $k = 5$  mV/s in addition to d)  $C_s$  vs.  $k$  for ACC conditioned in 3 M NaNO<sub>3</sub> (C), Milli-Q water (0), or 1-4

M HNO<sub>3</sub> (1-4). The error bars in panel (d) represent the standard deviation of 2-6 measurements for each condition.

The underlying motivation for electrode modification is increasing  $C_s$ , which is directly related to  $\Gamma_A$ ,<sup>45</sup> and Figure 2d shows an inverse relationship between  $C_s$  and  $k$ . Scanning slowly provides more time for diffusion, allowing enhanced contact between electrolyte ions and the internal surface of the electrode. As a result, greater charge storage leads to higher  $C_s$ . Conversely, when  $k$  increases, the time scale limits mass transfer into the smallest pores resulting in less charge storage and low  $C_s$ .<sup>25,46</sup> In other words, the microporosity limits the accessibility of electrolyte ions at high  $k$ . As shown in Figure 2d, at the highest  $k$ , the ACC conditioned in 1 M HNO<sub>3</sub> and Milli-Q water show the highest and lowest  $C_s$ , respectively. However, at the lowest  $k$  of 1 mV/s,  $C_s$  increases with [HNO<sub>3</sub>], with an overall 16% increase in the 0-4 M range. These measurements at different  $k$  provide insight into the contributions of pores of different sizes and suggest that micropore accessibility plays a key role in charge storage for these ACC materials. The change in slope for  $C_s$  vs.  $k$  when [HNO<sub>3</sub>] > 1 M also suggests a change in the charging mechanism, which is further supported by the introduction of N and O functionalities and the more pronounced pseudocapacitive peaks following ACC conditioning with 2-4 M HNO<sub>3</sub>.

Finally, although it is often assumed that the key to achieving high  $C_s$  through EDL charging is high  $S_{BET}$ , there is no clear relationship between these two properties for the ACC investigated in this study. A lack of correlation has also been observed by others.<sup>47,48</sup> While high  $S_{BET}$  generally indicates more surface area available for charge storage, the surface chemistry also plays a role in determining  $C_s$ . Table 1 shows a systematic increase in  $S_{BET}$  from 0 to 2 M HNO<sub>3</sub>, followed by a decrease after exposure to higher concentrations, reflecting the competing effects of surface functionalization and porous characteristics. After lower exposures, HNO<sub>3</sub> enhances  $S_{BET}$

by introducing functional groups and potentially etching micropores. At higher concentrations, excessive oxidation may cause structural damage, leading to a reduction in  $S_{BET}$ . Overall, for the ACC in this study,  $C_s$  is more sensitive to variations in surface chemistry than  $S_{BET}$ . In other words, the enhanced Faradaic activity and/or wettability provided by oxygenated and nitro functional groups outweigh the effects of reduced  $S_{BET}$ . In addition to  $S_{BET}$ , the results for  $V_{tot}$  are presented in Table 1; additional discussion and the isotherms are provided in the SI (Figure S10).

### CDI Performance

With increased surface functionality and  $C_s$ , this study incorporated the  $\text{HNO}_3$ -conditioned ACC in an electrochemical cell to evaluate  $E$  and  $A$ , along with other performance metrics. All CDI experiments were performed using NaCl solution because it is highly relevant for water treatment and energy production applications. Aside from being a major contributor to salinity in various aqueous environments, inorganic salts can have a detrimental effect on product recovery and catalytic upgrading in bioenergy conversion,<sup>49</sup> so techniques for extraction/recovery are key.

### *Effects of $\text{HNO}_3$ Conditioning*

To investigate the effects of  $\text{HNO}_3$  pretreatment on electrochemical separation performance, CDI experiments were carried out using a symmetric electrode configuration. Fifteen cycles were conducted to ensure dynamic equilibrium, and all experiments were conducted in duplicate using separate sets of electrodes. Figure S11a shows a representative effluent concentration ( $[\text{Cl}^-]_e$ ) profile for a single CDI cycle from each experiment. Each  $[\text{Cl}^-]_e$  data set was normalized by subtracting the influent concentration ( $[\text{Cl}^-]_0$ ) to obtain the net reduction or increase in ion concentration during adsorption and regeneration, respectively. Comparing the  $[\text{Cl}^-]_e - [\text{Cl}^-]_0$  profiles reveals notable differences in the change in concentration over the cycle. To further investigate this, concentration data were extracted for discrete time intervals, illustrated by the

dashed lines in Figure S11a. For each time interval, the slope of the  $[\text{Cl}^-]_e - [\text{Cl}^-]_0$  vs. time curve was calculated, providing a quantitative measure of the instantaneous adsorption or desorption rate ( $r_A$  or  $r_D$ ) within a given time interval. The results are presented in Figures S11b and S11c. At the beginning of the cycle,  $r_A$  is 10-24 $\times$  greater for the 2, 3 and 4 M  $\text{HNO}_3$ -pretreated electrodes in comparison to the electrodes conditioned in water, 3 M  $\text{NaNO}_3$ , and 1 M  $\text{HNO}_3$ . Looking at the concentration profiles in Figure S11a and  $r_D$  in Figure S11c, all data sets show a rapid concentration increase at the beginning of the discharging step, aligning with expectations for a purely electrostatic process. However, at  $\sim 730$  s, the rates for the 2 M, 3 M, and 4 M-conditioned electrodes slow. Overall, the differences in  $r_D$  suggest that other factors may be influencing the ion release dynamics. While other studies have reported enhanced adsorption kinetics using  $\text{HNO}_3$ -conditioned ACC,<sup>19,20</sup> this is the first report of slower discharge kinetics.

Equations S2-S8 were used to further evaluate the CDI performance. As shown in Figure 3,  $\Gamma_A$ ,  $\bar{r}_A$ , and  $\Delta$  increase while  $E$  decreases when electrodes are pretreated with  $\text{HNO}_3$ . In contrast, the control experiments, performed with  $\text{NaNO}_3$ -conditioned electrodes, show marginal improvements to the performance metrics. Initially, as  $[\text{HNO}_3]$  increases, the performance improves but then plateaus at 2 M. In other words, conditioning with 3 or 4 M  $\text{HNO}_3$  does not lead to further performance enhancements. This behavior indicates that there is an optimal exposure for  $\text{HNO}_3$  pretreatment beyond which higher concentrations do not provide additional improvements. Since 2 M  $\text{HNO}_3$  is equally as effective as 4 M  $\text{HNO}_3$ , there are several reasons why a lower acid concentration pretreatment is desirable. For example, using a lower acid concentration simplifies the conditioning and rinsing procedure, poses fewer safety risks, incurs less cost, minimizes waste generation, and facilitates scale up since handling and processing large volumes of lower concentration acids are more manageable. Overall, Figure 3 shows that  $\Gamma_A$  and  $\bar{r}_A$  increase by up

to 3.5 $\times$  following HNO<sub>3</sub> pretreatment. Similarly,  $E$  decreases by 3.5 $\times$ , and  $\Lambda$  increases from as low as 24% to 82%. Notably, these performance enhancements were achieved through electrode pretreatment alone, effectively scaling the technology without increasing the physical footprint of the CDI cell.

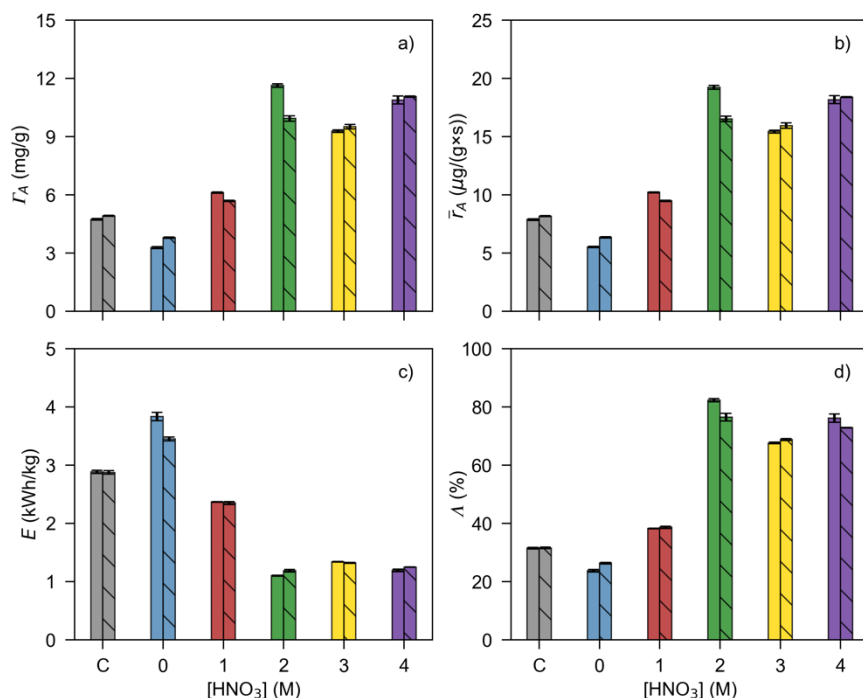


Figure 3. a)  $r_A$ , b)  $\bar{r}_A$ , c)  $E$ , and d)  $\Lambda$  for each CDI experiment conducted using electrodes conditioned in 3 M NaNO<sub>3</sub> (C), water (0), or 1-4 M HNO<sub>3</sub>. Each bar represents one of two duplicate experiments (solid and hatched), and the error bars represent the standard deviation of each metric over the last 5 cycles of operation.

### Effects of Operating Parameters

Aside from electrode material, CDI performance is sensitive to operating variables, *e.g.*, voltage, flow rate, and time. Previous studies have well-established that 1.2 V is optimal for effective ion removal while minimizing side reactions.<sup>7,27</sup> As such, this study does not investigate

multiple voltage conditions. Rather, this work evaluates the effects of flow rate and HCT. Focusing on HCT (the duration of either the charging or discharging phase) rather than total cycle time (the sum of both the charging and discharging steps) emphasizes the charging phase, which is the key period for monitoring adsorption/extraction in CDI.

Having established 2 M HNO<sub>3</sub> as the optimal conditioning solution, experiments were conducted with a range of flow rates (4, 8, and 16 mL/min) and HCTs (5, 10, and 15 min). Note that in this study, the charging and discharging steps are of the same duration, so HCT is half of the total cycle time. The results are provided in Figure 4 and Table S4. For a given flow rate, increasing the HCT leads to greater  $\Gamma_A$  until the maximum capacity is reached (*i.e.*, 12.5-13 mg/g) and lower  $\bar{r}_A$  as shown in Figure 4a. Alternatively, for a fixed HCT, increasing the flow rate enables the system to reach equilibrium more quickly. With a short HCT of 5 min, increasing the flow rate from 4 to 16 mL/min increases  $\Gamma_A$  by 26%, but coupling longer HCT with higher flow rates does not provide additional gains because the electrodes saturate. As shown in Figure 4b, the combination of 16 mL/min and 5 min HCT provides the lowest  $E$  (1.0 kWh/kg) and highest  $A$  (90%), but  $\Gamma_A$  is <70% of the maximum value. Doubling HCT increases  $\Gamma_A$  to 97% of the maximum value, but  $E$  increases by 10-15%. In general, longer HCT results in higher  $E$  because the electrodes are charged for a longer period of time. As the electrodes approach equilibrium, extended charging becomes increasingly inefficient. The ion adsorption kinetics slow until no additional ions can be extracted. Balancing this tradeoff is critical; shorter cycles can save energy but may leave  $\Gamma_A$  underutilized, while longer cycles maximize  $\Gamma_A$  but risk inefficient energy use. Overall, understanding these tradeoffs is key to the scale up and implementation of CDI systems because  $E$  is directly related to the operational cost and environmental impacts.

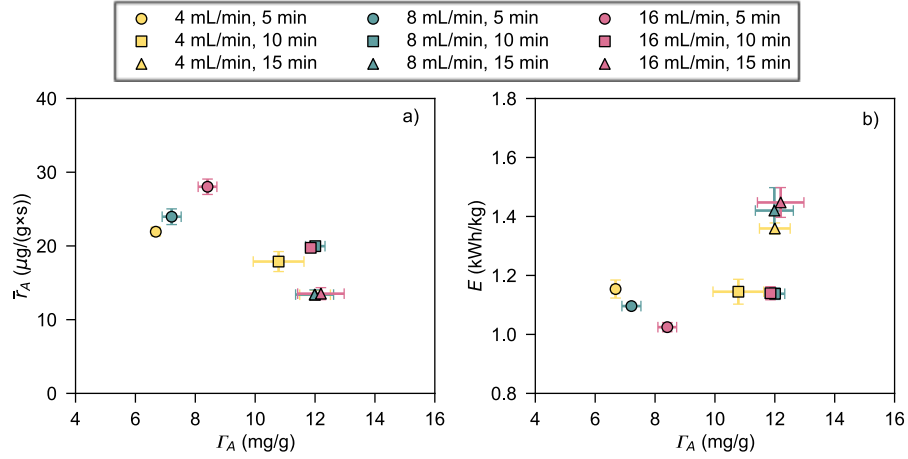


Figure 4. Effects of flow rate and HCT on CDI performance showing the tradeoff between a)  $\Gamma_A$  and  $\bar{r}_A$  and b)  $\Gamma_A$  and  $E$ . Each experiment was replicated, and the symbols and error bars represent the average and standard deviation, respectively, of duplicates.

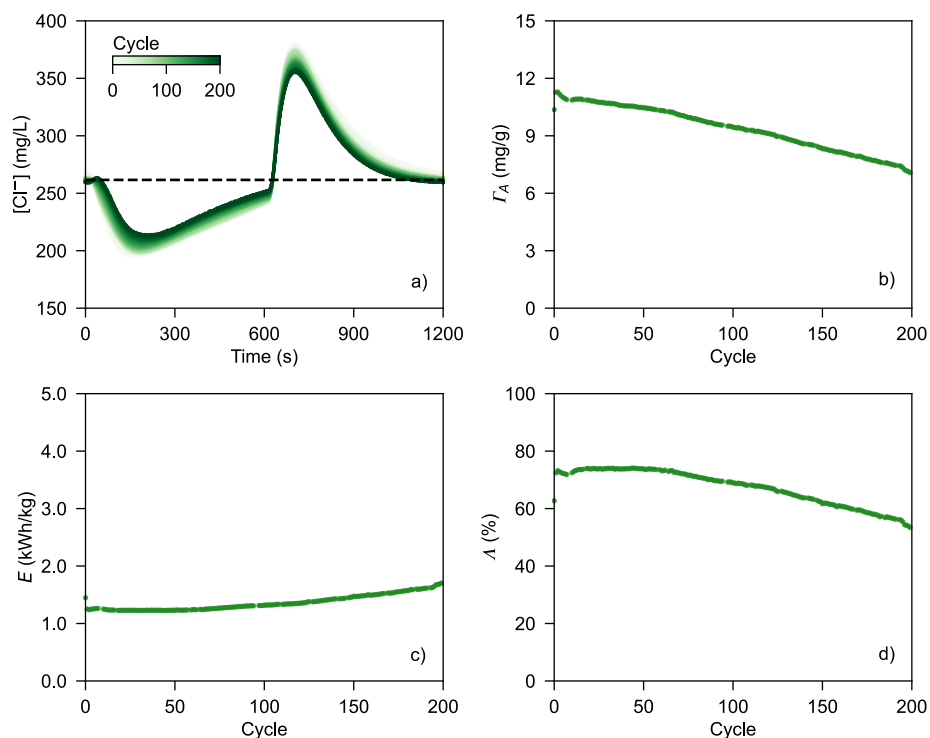
To tailor CDI systems for specific applications beyond NaCl extraction, experimental designs should account for whether maximizing  $\Gamma_A$  or minimizing  $E$  is the primary objective. For applications prioritizing  $\Gamma_A$ , longer HCT can be combined with moderate to high flow rate. For applications focused on minimizing  $E$ , shorter cycles with a higher flow rate will utilize the most efficient phase of adsorption. In many cases, the goal is to simultaneously maximize  $\Gamma_A$  while minimizing  $E$ , making the energy-capacity tradeoff plot (Figure 4b) an especially useful and practical tool for optimizing CDI systems. Finally, in this study, the highest flow rate (16 mL/min) yielded favorable results, but it is important to note that excessively high flow rates may reduce contact time and limit ion removal efficiency.

#### *Electrode longevity*

Despite beneficial short-term improvements in this work and a previous study concluding that CDI with HNO<sub>3</sub>-conditioned ACC electrodes shows better cyclability than with untreated

ACC,<sup>21</sup> others have shown that oxygenated functional groups are detrimental to the cyclability of carbon-based supercapacitors.<sup>50</sup> To investigate the impacts on the long-term stability in CDI, electrosorption was tracked over 200 cycles using 2 M HNO<sub>3</sub>-conditioned ACC, a flow rate of 4 mL/min, and HCT of 10 min. The results are presented in Figure 5. The overlay of effluent concentration profiles shown in Figure 5a illustrates the time evolution of ion removal and recovery. Figures 5b and S12 show that  $\Gamma_A$  and desorption decrease as the number of cycles increases despite maintaining ~100% recovery. Note that data from the initial cycles are not representative because the system has not yet reached dynamic equilibrium. Over the 200 cycles,  $\Gamma_A$  decreases by 32% from 10.4 mg/g to 7.1 mg/g. This is comparable to previous work that observed a 25% decrease in  $\Gamma_A$  after 200 CDI cycles using water-washed ACC.<sup>51</sup> However, it is important to note that the ACC used in that study was different, the cycles were shorter (4 vs. 20 min), and the flow rate was slightly higher (7 vs 4 mL/min).

Lower  $\Gamma_A$  results in higher  $E$  and lower  $A$ , as shown in Figures 5c and 5d. Essentially, more energy is consumed for less effective ion capture, leading to lower process efficiency. These observations suggest that the findings from supercapacitor research can be extended to CDI. This study highlights the need for further optimization to balance the initial performance with the longevity required for practical deionization applications. Finally, although the primary focus of this study is to investigate the effects of electrode conditioning on performance, including the potential for sustaining the improvement observed in the short-term over hundreds of cycles, future research should focus on further characterizing the degradation mechanisms associated with cycling.



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317 Figure 5. a) Overlay of effluent concentration profiles from a continuous CDI run of 200 cycles  
 318 using 2 M  $HNO_3$ -pretreated electrodes. The dashed line shows the influent concentration, and the  
 319 color gradient represents cycle progression. Performance metrics b)  $\Gamma_A$ , c)  $E$ , and d)  $\Delta$  for each of  
 320 the 200 cycles.

## CONCLUSIONS

This study presents a comprehensive analysis that simultaneously evaluates the effects of HNO<sub>3</sub> conditioning and varying operating conditions on the performance of ACC in CDI systems. After relatively low HNO<sub>3</sub> exposure (2 M for 24 h), charge-discharge cycling experiments demonstrate significant improvements in  $\Gamma_A$ ,  $A$ , and  $E$ , attributed to the introduction of O and N-containing functional groups and increased  $C_s$ . With the exception of the desorption rate, the changes in surface chemistry and  $C_s$  from HNO<sub>3</sub> conditioning are linked to positive improvements in CDI performance, including higher  $\Gamma_A$ , greater  $A$ , and reduced  $E$ . This approach highlights the potential for optimizing existing CDI systems for better performance while maintaining original device size and design, thus facilitating integration into current applications. In addition, this study fine-tuned the operating conditions to further improve  $E$  and  $A$ , surpassing those reported in previous studies. Finally, this work investigated a critical aspect that has been largely overlooked in CDI literature: the long-term performance of HNO<sub>3</sub>-conditioned ACC electrodes. Initially appearing to stabilize after ~10 cycles, monitoring of the effluent over 200 continuous CDI cycles reveals a gradual decline of all performance metrics, *e.g.*, decreasing  $\Gamma_A$  and  $A$  and increasing  $E$ . Future research should focus on understanding the mechanisms of degradation during cycling and developing strategies to mitigate these effects. In summary, by investigating the effects of HNO<sub>3</sub> pretreatment and operational variables, this study provides valuable insights that can facilitate the scale up and transition of CDI technology into commercial settings. The findings offer guidance for design decisions and operational strategies, ultimately leading to enhanced efficiency and effectiveness in chemical separation processes.

## **SUPPORTING INFORMATION**

Electrode charging and discharging; Manufacturer specifications for the ACC; Pretreatment, rinsing, and drying conditions; Calculation of  $C_s$ ; Electrochemical cell configuration; Calculation of CDI performance metrics; Elemental concentrations for ACC; SEM and EDS methods and results; ICP-OES method and results; XPS curve fitting and stacked N1s spectra; FTIR method and results; Adsorption/desorption isotherms; Normalized CDI effluent concentrations; CDI performance at different flow rates and HCTs; Long-term CDI performance

## **ACKNOWLEDGEMENTS**

This work was sponsored by the U.S. Department of Energy (DOE) Energy Efficiency and Renewable Energy (EERE) Bioenergy Technologies Office (BETO) through the Bioprocessing Separations Consortium under Contract DE-AC02-06CH11357. SEM and EDS were performed at the Center for Nanoscale Materials, a U.S. Department of Energy Office of Science User Facility, supported by the U.S. DOE, Office of Basic Energy Sciences under Contract No. DE-AC02-06CH11357.

## **PRESENT ADDRESSES**

<sup>#</sup>L.C.: Department of Chemical and Biological Engineering, University of Wisconsin-Madison, Madison, WI, 53706, USA

<sup>\$</sup>S.K.: Wisconsin Energy Institute, University of Wisconsin-Madison, Madison, WI, 53706, USA

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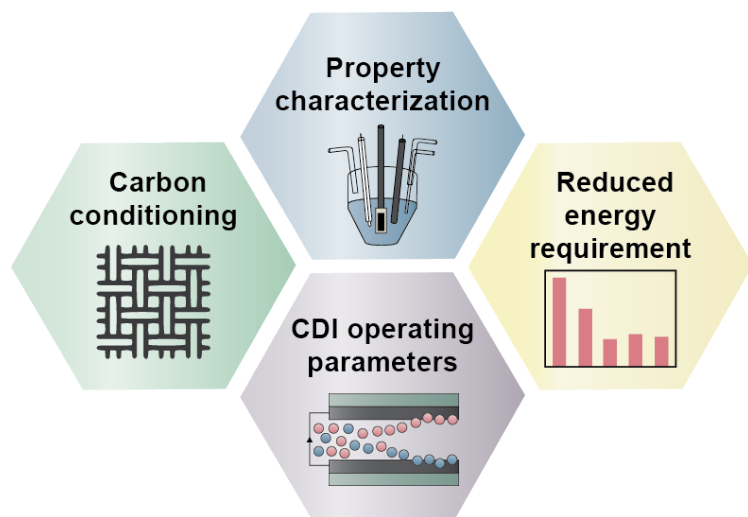
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## 522 TOC/ABSTRACT GRAHPIC



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## 524 SYNOPSIS

525 This study reduces CDI energy consumption by  $3.8\times$  through a combination of electrode  
526 pretreatment and tuning of operating parameters.