

Three-dimensional mapping of mineral in intact shark centra
with energy dispersive x-ray diffraction

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Abstract

The centra of shark vertebrae consist of cartilage mineralized by a bioapatite similar to bone's carbonated hydroxyapatite, and, without a repair mechanism analogous to remodeling in bone, these structures still survive millions of cycles of high-strain loading. The main structures of the centrum are an hourglass-shaped double cone and the intermedialia which supports the cones. Little is known about the nanostructure of shark centra, specifically the relationship between bioapatite and cartilage fibers, and this study uses energy dispersive diffraction (EDD) with polychromatic synchrotron x-radiation to study the spatial organization of the mineral phase and its crystallographic texture. The unique energy-sensitive detector array at beamline 6-BM-B, the Advanced Photon Source, enables EDD to quantify the texture within each sampling volume with one exposure while constructing 3D maps via specimen translation across the sampling volume. This study maps a centrum from two shark orders, a carcharhiniform and a lamniform, with different intermedialia structures. In the blue shark (*Prionace glauca*, Carcharhiniformes), the bioapatite's *c*-axes are oriented laterally within the centrum's cone walls but axially within the wide wedges of the intermedialia; the former is interpreted to resist lateral deformation, the latter to support axial loads. In the shortfin mako (*Isurus oxyrinchus*, Lamniformes), there is some tendency for *c*-axis variation with position, but the situation is unclear because one dimension of the sampling volume is considerably larger than the thickness and spacing of the intermedialia's radially-oriented lamellae. Because elastic modulus in collagen plus bioapatite mineralized tissues varies significantly with both volume fraction of bioapatite and crystallographic texture, the present 3D EDD-derived maps should inform future 3D numerical models of shark centra under applied load.

55 **Keywords:** energy dispersive diffraction; bioapatite (hydroxyapatite); shark; centrum;
56 mineralized cartilage; crystallographic texture
57

Introduction

Elasmobranchii, which include sharks, rays and skates, have cartilaginous skeletons, and relatively little is known (either experimentally or numerically) about how their subcranial axial skeletons function under in vivo loads. The sharks studied by Natanson et al. (2018) (elasmobranch orders Carcharhiniformes and Lamniformes) possess more than 80 vertebrae (Fig. 1a), each with a mostly unmineralized neural arch and a mineralized centrum, and these sharks' abdominal centra carry the loads generated during swimming (Porter and Long 2010). When a shark swims, its tail beats from left to right, compressing first one side then the other side of each centrum (Fig. 1b), and the magnitude increases from the juncture of the thorax and abdomen toward the tail. The structure and mineralization of the carcharhiniform and lamniform centra enables shark vertebrae to survive enormous compressive strains of 3-8% (Porter et al. 2014) for millions of cycles of loading (Watanabe et al. 2012), despite the absence of a repair mechanism like remodeling in bone.

Shark centra have complex 3D structures affecting mechanical performance, and there are significant similarities and important differences between carcharhiniform and lamniform centra. The centra of both orders contain an hourglass-shaped double cone of mineralized cartilage (Fig. 1c) termed the corpus calcarea. A fluid filled inter-vertebral capsule lies between the cone walls of adjacent vertebrae. Within a single centrum, the rostral and caudal cone walls are supported by the mineralized intermedialia whose structure differs between the two orders (Fig. 1c). In Carcharhiniformes, the intermedialia consists of four thick wedges (Fig. 1d) with unmineralized (cartilage) gaps separating the wedges. In Lamniformes, two dozen or more, relatively thin, radially-oriented plates (lamellae) comprise the intermedialia (Fig. 1d); these lamellae may be distinct from each other or may combine and separate from their neighbors, e.g.

the one o'clock position of the left side of Fig. 1d and Morse et al. (2022). The lamellae are grouped into four sectors separated by four gaps; two sectors are wider and contain more lamellae than the other two sectors. In both carcharhiniforms and lamniforms, the cartilage of one pair of gaps extends to form the neural arch and that of the other pair to form the hemal arch (if present).

During swimming, both the left and right sides of shark abdominal centra experience an alternating pattern of compression and tension. Some studies of the macroscopic patterns of strain in the shark vertebral column have appeared (Porter and Long 2010; Porter et al. 2014, Porter et al. 2016), but little is known about the 3D distributions of displacement or of strain within the complex structure of a shark centrum. Experimental measurement of these 3D strains under applied load is one way forward and is, in fact, an eventual goal of the authors, see the preliminary report of Park et al (2022c). An alternative to understanding centrum function is 3D numerical modeling which requires accurate geometrical and materials property inputs, and obtaining some of this information is the focus of the present report. Beyond the centra macrostructure (3D geometry at the $\sim 50\text{ }\mu\text{m}$ and higher scales), mineral density, microstructure (scales $1\text{ }\mu\text{m}$ and above) and nanostructure (scales 0.1 nm and higher) contribute to functionality. These centra structural levels are not independent, and, before introducing the energy-dispersive x-ray diffraction-based nanostructural studies of this paper, prior studies of centra macrostructure and microstructure are reviewed.

Macrostructure of entire lamniform and carcharhiniform centra have been studied with microComputed Tomography (microCT) with volume elements (voxels) $> 15\text{ }\mu\text{m}$, e.g. Geraghty et al. (2012), Natanson et al. (2018) and Morse et al. (2022). The last study found mineral levels in the cone wall were significantly greater than in the intermedialia. As elastic moduli increase

with increasing mineral content in collagen-based tissues (Currey 2002), such mineral content variation in centra should be incorporated into qualitative and/or quantitative models of response to applied loads. Microstructure in small blocks cut from centra have been studied with microCT with voxels $\sim 1 \mu\text{m}$ in size (Stock et al. 2022), these very limited data need to be supplemented before they can reliably guide understanding of mechanical properties. Nanostructure also affects properties, and the focus here is on the nanocrystals of the mineral phase and the crystallographic orientation of the mineral within the cartilage matrix. Data on bone suggests that mineralized tissues consisting of collagen reinforced by bioapatite often contain a mineral phase where certain crystal axes are strongly oriented relative to anatomical axes of principal in vivo stresses, i.e., contain crystallographic texture (Currey 2002), and that this texture is strong enough to produce significant variation in elastic constant as a function of orientation (Guo 2001). Failure to incorporate such elastic modulus variation, if present in shark centra, into interpretation of mechanical tests or in 3D modeling of shark centra may lead to inaccurate conclusions. Although microstructural, micromechanical and histological studies have appeared on the mineralized cartilage of shark tesserae, e.g. Chaumel et al. (2020) and Seidel et al. (2019, 2021), which is a rather different tissue from that of the centra, crystallographic data are sparse for tesserae. Some data, however, are available on shark bioapatite.

An early diffraction study (Urist 1961) showed that the centra's mineral is a bioapatite closely related to hydroxyapatite (hAp), and TEM showed a similar bioapatite in shark tesserae (Dean et al. 2005). Recently Park et al. (2022a) used monochromatic synchrotron x-radiation to collect diffraction patterns from small blocks cut from centra of four species and confirmed that the only crystalline phase visible was a bioapatite with lattice parameters slightly different from those of bone and stronger crystallographic texture than in mammalian long bones. This latter

study, however, was not designed to compare texture from different portions of the centra, and 3D mapping of centrum texture is the focus of the present study.

The present authors hypothesize that, like the collagen fibril (type I collagen) and bioapatite orientations in long bones, the cartilage (2/3 collagen type II, 1/3 collagen type I plus significant proteoglycans) fiber axes and bioapatite crystal *c*-axes in shark centra are aligned along the direction(s) of principal strain and that these directions may vary with position within the centrum. The study reported below, performed at beamline 6-BM-B of the Advanced Photon Source (APS), employs energy dispersive x-ray diffraction (EDD) and a unique array of energy sensitive detectors (Weidner et al. 2010) to obtain 3D position resolved diffraction maps of two shark centra. The maps are obtained by X, Y and Z translation of the specimen across the sampling volume and without sample rotation. Alternative EDD data collection strategies have been used in the past, e.g. to study cement paste degradation due to sulfate attack (Naik et al. 2006) and to study the cross-section of bones (Stock et al. 2017), but the present instrument allows direct quantification of the 3D variation of crystallographic quantities (intensity of diffraction peaks, lattice parameters, crystallite size/microstrain, texture) within specimens. Alternative 3D diffraction mapping methods using monochromatic x-ray are available and are contrasted with this EDD approach in the Discussion.

This paper reports 3D EDD-derived maps of a carcharhiniform shark and a lamniform shark. Maps of mineral content are already available for intact centra (Morse et al. 2022), and the focus here is on crystallographic texture, something the authors expect will be essential to accurate 3D numerical modeling. Incorporation of texture and of mineral content into 3D models is also discussed.

Materials and Methods

One abdominal vertebrae of a species of Carcharhiniformes (*Prionace glauca*, blue shark, from vertebra numbers 81-84) and one of a species of Lamniformes (*Isurus oxyrinchus*, shortfin mako, vertebra number 62) were examined in this study. Laboratory microCT (Morse et al. 2022) showed the diameter and height were diameter 24.3 mm and 12.7 mm, respectively, for the blue shark centrum and 21 mm and 12.4 mm, respectively, for the shortfin mako centrum. Figure 1c shows 3D schematics of the two types of centra with the material closest to the viewer rendered transparent so the inner structure could be seen. Figure 1d shows transverse sections through the axial midplane of each centrum with the mineralized tissue shown in black.

The experimental geometry for EDD at beamline 6-BM-B, APS, is shown in Fig. 2a (Weidner et al. 2010). A collimator forms a pencil beam of polychromatic radiation which passes through the specimen (a schematic transverse section near the middle of a shortfin mako centrum). The conical receiving slits block all radiation except that diffracted from the sampling volume “sv” at an angle $2\theta = 6.5^\circ$. Linear translators move the sample across the sampling volume along the three orthogonal axes X (horizontal, perpendicular to the incident beam), Y (vertical, perpendicular to the incident beam) and Z (horizontal, parallel to the incident beam).

Bragg’s law, $\lambda = 2 d_{hk.l} \sin \theta$, gives the angle θ at which peaks of diffracted intensity occur from crystalline material and shows that this angle depends on the x-ray wavelength (i.e., the inverse of the x-ray energy) and the crystal periodicity or d-spacing $d_{hk.l}$ for lattice planes $hk.l$.¹ With the Bragg angle fixed ($2\theta = 6.5^\circ$), a range of energies in the polychromatic beam and an array of differently oriented crystals with the sampling volume sv, different $hk.l$ select different energies and produce diffracted beams which reach one of the ten energy sensitive

¹ Here the abbreviated Miller-Bravais indexing system is used to emphasize the hexagonal crystal system of hAp (Cullity and Stock 2001).

detectors of the 6-BM-B array (Fig. 2b). For the bioapatite in the shark centra, each detector measures the intensity of 00.2 diffraction of ~32 keV x-rays as well as the intensity of the unresolved quadruplet of 21.1, 11.2, 30.0 and 20.2 of 38-42 keV x-rays and other reflections that might be intense enough to pick out from the background. Note that the orientation of the crystals diffracting into detector 5 is given by orientation of lattice plane normal $\mathbf{N}_5$ which makes an angle $(90 - \theta)^\circ$ from both the incident beam $\mathbf{S}_0$ and the diffracted beam direction $\mathbf{S}_5$ and which lies in the horizontal plane (Fig. 2b). The same is true each other detector, i.e., for detector 1, normal $\mathbf{N}_1$, $\mathbf{S}_0$ and the diffracted beam direction $\mathbf{S}_1$ are coplanar (vertical plane) and make the same angles. Therefore, comparison of 00.2 diffracted intensities between different detectors gives the relative fractions of crystalline bioapatite with different orientation defined by their lattice normal orientations $\mathbf{N}_i$, i.e., the crystallographic texture within the sampling volume sv.

Figure 2c shows two views of the shortfin mako centrum (the side view in the top panel and the central transverse section in the bottom panel). Figure 2d shows the same views of the blue shark centrum. The red squares indicate extent of the specimen covered by the sampling volumes; note that neither the spacing between sampling volumes nor their dimensions are to scale.

Table 1 gives the experimental parameters for the scans of the two sharks. For the blue shark and shortfin mako, the pencil beam dimensions were $\delta X = 0.1$ mm and $\delta Y = 0.2$ mm and $\delta X = \delta Y = 0.2$ mm, respectively. A ~0.9 mm thick powdered ceria standard (NIST SRM 674B) was scanned across the sampling volume, and the full-width at half-maximum of the diffracted peak intensities was used as the measure of the gage length δZ along the incident beam direction. The gage lengths for the different detectors varied slightly (see Supplemental Fig. S1, mean of $\sim 1.7 \pm 0.15$ mm) and sampled the same position within ~0.25 mm; these uncertainties are much

smaller than those from other sources and are hereafter ignored. For the beam dimensions used to map the blue shark centrum, δZ averaged 1.7 mm, and for the shortfin mako, δZ averaged 2.5 mm. The translation step sizes used to build up the 3D maps were $\Delta X = 2.0$ mm, $\Delta Y = 1.0$ mm and $\Delta Z = 1.9$ mm for the blue shark centrum and were $\Delta X = 1.5$ mm, $\Delta Y = 0.5$ mm and $\Delta Z = 1.0$ mm for the shortfin mako. Note that the entire volume of the blue shark centrum was scanned (diameter 24.3 mm, height 12.7 mm), but only a portion of the shortfin mako centrum (diameter ~21 mm, height 12.4 mm) was covered. For both centra, the 3D reconstruction on a 0.5 mm x 0.5 mm x 0.5 mm grid with interpolation of the neighboring volume elements (voxels) with smoothing.

As noted in the previous paragraph, data collection parameters (and fraction of centrum covered) differed for the two centra. Point by point mapping of 3D volumes is quite time consuming: Ignoring motion and detector readout overhead, collecting the $16 \times 10 \times 17 = 2,720$ patterns covering the blue shark centrum required 22.7 hr, and collecting the $3 \times 10 \times 47 = 1,410$ patterns covering a portion of the shortfin mako centrum required 19.6 hrs. Limited beam time (typically 4 days per scheduling cycle) dictated, therefore, that the sampling parameters were altered match centra characteristics while not exhausting available time. Given the blue shark intermedialia consisted of wedges, a relatively uniform sampling grid was selected at the cost of spatial resolution. Because the mako's intermedialia consisted of relatively thin lamellae compared to the sampling volume dimension along the incident beam direction Z, sampling along the beam direction was increased substantially compared to the blue shark, thereby improving the chances that individual lamellae could be resolved, at the cost of decreasing the volume scanned along X and Y.

The energy range of each detector was simultaneously calibrated using an array of ten, pre-aligned ^{57}Co sources. As mentioned above, a ceria standard was used to measure the sampling volume length along the beam direction. The d -spacings of the known ceria diffraction peaks were also used to confirm the linearity of the energy range.

The integrated intensity and position for each peak (00.2 and q) were determined by fitting a pseudoVoigt function (custom Matlab scripts). For each point-wise data set, integrated peak intensity for each peak (measured by each detector element in the detector array) was interpolated and smoothed using a $0.5\text{ mm} \times 0.5\text{ mm} \times 0.5\text{ mm}$ grid using Matlab *scatteredInterpolant* function with default parameters. The interpolated 3D maps were imported into ImageJ as a stack of slices and viewed with the orthogonal views function. Intensity maps from different detectors were combined using the color/merge function of ImageJ.

Results

Figure 3 shows a diffraction pattern (intensity vs x-ray energy) typical of a shark centra. Calibration lines from radioactive sources appear at high and low energies and are labeled “cal”. Two bioapatite reflections are labeled 00.2 and “ q ”, i.e., the closely spaced and unresolved quadruplet of 21.1, 11.2, 30.0 and 20.2. In Fig. 3, the maximum 00.2 and q peak intensities were ~150 cts and ~430 cts above background, respectively (compared to backgrounds of ~100 cts and ~120 cts, respectively). Two sections of the pattern are plotted in red; this is the energy range over which the 00.2 and quadruplet integrated intensities are calculated. The green peaks below the experimental data show peaks expected for a synthetic hAp reference pattern (Powder Diffraction File 86-1201, International Centre for Diffraction Data, Newtown, PA, USA), and there is a clear difference of in energy diffracted by each hkl (the green peak reference position

vs red highlighted experimental data from a centrum), and, hence, a small but significant difference in lattice parameters. The ratio of 00.2 to q peak intensities in Fig. 3 is 35%, somewhat lower than the 45% ratio expected for the crystallographic-texture-free powder of PDF 86-1201.

Blue shark

Figure 4 shows orthogonal sections through the 3D reconstructed volume of the blue shark centrum using the integrated intensity of reflection q (unresolved 21.1+11.2+03.0+20.2). Within each panel, the three sub-panels show orthogonal sections, the square sub-panels being the transverse section through the axial center of the centrum and the yellow lines and arrows indicating the sections' positions in the 3D volume. In the top row, the left panel (labeled 5) is the reconstruction using intensities from detector 5), and the middle panel is intensities from detector 10, diametrically opposite detector 5. The right panel of the top row of Fig. 4 shows a composite of the two reconstructions with detector 5 intensities shown in blue and detector 10 intensities shown in red. The reconstructed transverse cross-sections look like the lab microCT reconstruction in Fig. 1d, if one allows for the lower spatial resolution of the EDD data.

The left panel of the bottom row of Fig. 4 plots intensities from detector 1, the middle panel plots intensities of detector 9 (diametrically opposite detector 1) and the right panel shows a composite with detector 1 intensity shown in red and detector 9 in blue. Although there are regions of red in the 1+9 composite panels, most of the volume appears magenta, i.e., more or less even intensities of the two opposite orientation detectors. In the composite 5+10, most of the volume is magenta but the regions of red intensity are more prominent at the margins of the mineralized tissue.

At some positions in longitudinal sections (those parallel to the centrum's axis and running horizontally in Fig. 4), the integrated intensity of the q reflection is greater than in the neighboring volume (yellow arrowheads); the higher intensity corresponds to the location of the cone walls and the lower to intermedialia (Morse et al. 2022). In other portions of the same sections, there is little difference in q intensity across the section (orange arrowheads). Table 2 gives values of the mean integrated intensity (and its standard deviation given after the $\pm$ symbol) within the 3×3 voxel boxes within the cone wall and intermedialia. Inspection reveals that the intensities vary within the intermedialia, and boxes i and i' were measured and show $\sim 50\%$ variation (Table 2).

Crystallographic texture in mineralized tissues containing bioapatite typically manifests in differences in 00.2 diffracted intensity along different anatomical directions. Figure 5, also of the blue shark centrum, shows three orthogonal sections through the 00.2 intensity reconstructed volume. In Fig. 5a, diffracted intensity from detector 5 is shown in blue, that from detector 7 in green and that from detector 9 in red. The colored arrows in each panel indicate the direction of c -axis crystallites sampled by the respective detector. Before describing what the different colors within the sections reveal, it is useful to consider the magnitude of the integrated intensities (Table 2). The largest integrated intensity within this 00.2-reconstructed volume is 86 arbitrary units (a.u.) and the mean value within a 3×3 voxel box within fluid ("f" in Fig. 5a) is 2 a.u. Within the sampling box in the cone wall, the mean 00.2 integrated intensity was 46 ± 9 a.u. for detector 5 and 21 ± 5 a.u. for detector 1 (the direction equivalent to the detector 9 data plotted). Within the sampling box in the intermedialia, the detector 5 intensity is 5 ± 1 a.u. and detector 1 intensity is 35 ± 2 a.u.

The transverse section through the middle of the centrum (the square image, lower left panel of Fig. 5a) has an intensity distribution like the transverse section shown in Fig. 1d, and, over most of the cross-section, the crystallites have *c*-axes oriented axially (red color indicating bioapatite *c*-axes primarily normal to the transverse plane, see coordinate axes at the lower left of this panel). In the side-view cross-section (upper panel of Fig. 5a) the blue voxels represent hAp crystallites with their *c*-axes oriented to diffract along the X direction (blue arrow), i.e., laterally (radially) in relation to the backbone axis, and the red voxels show crystallites with *c*-axes aligned axially. In the section of the right-hand panel of Fig. 5a, little blue appears; in this section blue indicates *c*-axis orientations out of the plane of this section and along the hoop (circumferential) direction; therefore, indicating large fractions of the hAp crystallites in the cone wall are not correctly oriented to diffract in the directions viewed by the respective detectors. Figure 5b schematically indicates observations shown in Fig. 5a: lateral (radial) *c*-axis orientations in the cone walls and axial orientations in the wedge between the cone walls (intermedialia).

Shortfin mako

Figure 6 shows reconstructed transverse sections (diffracted intensity vs position) for all ten detectors; maps for reflection *q* appear in (a) and for 00.2 appear in (b). A segmented lab microCT section of the mako centrum, at approximately the same axial position as the EDD maps and with the same orientation, is inset in the middle of the detector images in Fig. 6a. Gaps *g*1 and *g*2 between sectors can be resolved in all of the EDD maps and can be matched to *g*1 and *g*2 in the lab microCT-derived image. A small part of a wide sector *s*1, which borders gap *g*1, appears at the top of the maps in Fig. 6; the scan covers the outer ~3 mm of the radial lamellae.

A significant portion of sector s3 appears below gap g2, and each panel shows 4 mm of sector s2, excluding the outer-most and inner-most portions of the lamellae.

The spatial distribution of q reflection intensities is the same for all of the detectors (Fig. 6a). The magnitude of intensities differs somewhat; they are greater for detectors 1-4 than for 6-9 and are larger for detector 5 than for the diametrically opposite detector 10. The sectors' borders with the gaps have higher intensity than positions between the borders, consistent with the much closer spacing of lamellae in these locations seen in microCT, i.e., the image in the center of Fig. 6a and in Morse et al. (2022). The intensity map of sector s2, for example, shows two radial bands of increased intensity separated by a region of lower intensity. Away from its border with the gap, the reconstructed portion of sector s3 contains three distinct bands radiating from the middle of the centrum (indicated by the red arrows in the detector 8 map of Fig. 6a); the intensities of these bands are substantially lower those bordering the gaps.

Outside of the centrum, the q intensity within the fluid is mostly below 100 a.u. although there are scattered points where the measured intensity reaches 300 a.u. The maximum intensity for this volume for all detectors is 1,200 a.u. Within s1, the intensities approach about 900 a.u. for detector 1 and for 800 a.u. in detector 5. In the detector 1 map, the maximum intensities are somewhat larger (~1,000 a.u.) within the 10 o'clock and 9 o'clock oriented radial lamella(e) of s2 with the in-between volume having an intensity ~500 a.u. In the detector 5 map, the intensity of the 10 o'clock lamella(e) is about 800 a.u. and somewhat larger than that in the 9 o'clock band.

In many respects, the spatial distribution of 00.2 diffracted intensities mirrors that of the q reflection maps. The radial bands bordering the gaps have higher intensities than sectors' material between the gaps, but the three radial bands at the bottom of sector 3 are less clearly

defined for 00.2 than for q. The 00.2 intensities in the maps of detectors 4-6 are much greater than those collecting intensities at azimuths 90° away (detectors 1 and 2 and 8 and 9) and 180° away (detector 10). Further, the intensities in detector 1 and 2 maps are greater than those at corresponding positions in the 8 and 9 maps.

The maximum 00.2 integrated intensity for this volume for all detectors is 150 a.u., and the volumes containing only fluid (bottom area of the maps, below s3) are quite noisy intensity maps reflecting the especially poor counting statistics. Maximum intensity within the 10 o'clock orientated lamella(e) is about 90 a.u. in the detector 2 map and about 120 a.u. in the 9 and 10 o'clock lamella(e) with lower intensities between the two. The base of the s3 band (right hand side of the map, 8 o'clock orientation) has an intensity of 60 a.u. for detector 3 whereas the intensity is 70 a.u. in the detector 5 map.

Discussion

There are at least two contexts where the EDD results reported above would be of interest to readers of a journal concerned with mechanical behavior. First, the successful EDD mapping in this paper is a precursor to 3D mapping of internal strains arising from in situ loading of shark centra. Second, the 3D mapping of various bioapatite characteristics (notably crystallographic texture) is essential input for accurate modeling. Before discussing the present EDD data and considering how these might apply in these two contexts, it is important to summarize what is known about the tissue most closely related to that of the shark centra, i.e., tessellated cartilage of sharks and related elasmobranchs, and to review prior mechanical behavior studies of centra.

Tessellated cartilage of sharks and related species related to centra tissue

To date, tessellated cartilage has received much more attention than the mineralized cartilage of centra and, and these data may or may not guide understanding of centra mechanical functionality. In polished sections of dry stingray tesserae, Seidel et al. (2019) found different regions possess different mineral densities and nanoindentation hardnesses which they related to mechanical functionality. Seidel et al. (2019) correlated tesserae mineral density with nanoindentation hardness and found that stingray mineralized cartilage had a higher mineral content than reported by Gupta et al. (2005) for calcified human cartilage, but the relationship of Young's modulus to mineral content for tesserae was consistent with an extrapolation from data for human calcified cartilage.

Elasmobranch tesserae contain chondrocyte lacunae and canaliculi, but most of their volume is dense mineralized cartilage, at least at the one-micrometer level. In stingray tissue, for example, the lacunae average 6-7 vol%, and the canaliculi may add an additional couple vol% porosity (Chaumel et al. 2020). This tissue microstructure is quite different from that of shark centra, e.g. Fig. 5 of Stock et al. (2022): from this image, the present authors estimate ~76 vol% pores/soft tissue and 3-4.5 μm trabeculae thickness for the intermedialia volume and 59 vol% pores/soft tissue and ~8 μm trabeculae thickness for the cone wall volume. Even if the mineralized cartilage constitutive properties (mineral content, elastic moduli, yield stress, fracture toughness, ...) are the same at the 50-1,000 nm scale for both tissues, the very different microarchitectures of tesserae and centra suggest very different mechanical properties at scales one micrometer and above.

Prior studies of centra mechanical behavior

Before discussing prior studies of mechanical functionality of centra, it is useful to consider what is known about the mechanical properties of tessellated tissue, the tissue most closely related to that of centra. Liu et al. (2014) applied compressive loading normal to and parallel to the plane of blue shark jaw tesserae surrounded by uncalcified cartilage; their focus was measuring and modeling stress relaxation behavior during in vivo biting and their values of elastic moduli are informative for the unmineralized cartilage but not for mineralized tesserae. One way to isolate the calcified tissue is to produce polished sections of tesserae and perform nanoindentation, and, depending on the position within stingray tesserae, the resulting Young's moduli range from 15 to 35 GPa for tissue densities 1.5 to 2.7 g/cm³, respectively (Seidel et al. 2019), which on average are higher than in human cortical bone (Guo 2001).

Porter and Long (2010) tested vertebrae (neural arch plus centrum, centrum alone) from a carcharhiniform shark in uniaxial compression and determined that the arch did not carry appreciable load. The experimental Young's modulus determined was ~150 MPa which combines the tissue's intrinsic properties and deflection of the double cone and intermedialia structure, i.e., the continuum structural modulus.

Porter et al. (2014) studied in vivo and in situ strains of dogfish vertebrae using sonomicrometry. During low velocity swimming, compressive and tensile strains up to 2% were observed; post mortem ex vivo experiments imposing curvatures seen in steadily swimming dogfish revealed tensile strains approaching 4%. Ingle et al. (2018) performed cyclic compressive testing on vertebrae from carcharhiniforms and lamniforms and found anterior vertebrae had lower continuum moduli than posterior vertebrae; in all cases these moduli were less than 10 MPa. In three-point bending, more closely approximating loading in vivo than simple compression, Long et al. (2011) measured the apparent storage and loss moduli (E' and

E', respectively) of segments of multiple shark vertebrae. Depending on the amplitude of maximum curvature, the elastic portion of the modulus E' was between 0.3 and 1.3 MPa, for a fast swimming carcharhiniform species and 0.1 and 0.6 MPa for a carcharhiniform species known less for its speed and more for its maneuverability.

Blue shark

Interpretation of EDD intensity maps from the carcharhiniform centrum (blue shark) is much more straightforward than for the lamniform centrum (shortfin mako). Most of the centrum volume outside of the double cones of carcharhiniforms is occupied by four wedges (Fig. 1) with uniform microstructure down to the 15-25 μm level over millimeter lengths (Morse et al. 2022). Cone walls are multiple EDD voxels thick. Away from the edges of the cone walls, therefore, changes in diffracted intensity (greater than random variation) must result from differences in crystallographic texture.

As shown in Fig. 5 of the blue shark, bioapatite *c*-axes within the cone walls are strongly aligned laterally (i.e., along radii from the axis of the centrum), and the *c*-axes within the intermedialia are aligned axially. If the variation elastic properties with texture of sharks' mineralized cartilage resembles those of bone, something reasonable to assume for bioapatite-collagen based natural composites, then the strong *c*-axis directionality in the blue shark centrum (Fig. 5) corresponds to directional differences in elastic constants. In the cortices of human long bones, for example, bioapatite *c*-axes tend to be oriented along the axis of the bone, and elastic moduli for longitudinal vs transverse anatomical orientations are 1.8x greater for the former than the latter (Guo 2001). Park et al. (2022a) showed sharper *c*-axis texture in tissue cut from shark centra (both lamniforms and carcharhiniforms) than is observed in bone, and we hypothesize that

spatial/directional variation of moduli may be more pronounced in the mineralized shark centra than in long bones. If, like in bone, it is correct to infer increased stiffness along directions where *c*-axes are concentrated, then it appears that the mineral in the carcharhiniform cone walls is aligned to resist lateral deflections and that the mineral in the intermedialia is aligned to resist axial compression. 3D numerical models could confirm or refute this conjecture by incorporating elastic constants consistent with the 3D EDD maps with geometry derived from lab microCT of centra.

In Fig. 4 (reflection *q* of the blue shark centrum), comparison of the intensity map of detector 5 vs detector 1 (or 9) shows a difference of intensity only in the small sector oriented at 11 o'clock; however, the intensity map of the 11 o'clock section for detector 10 (diametrically opposite of detector 5) does not differ from those of detectors 1 (or 9). On balance, reflection *q* of the blue shark reveals weak texture, not unlike the weak texture seen for this set of reflections in mineralized tissues based on a type I collagen matrix (bone, dentin, cementum) reinforced with bioapatite, i.e., Almer and Stock (2005), Stock et al. (2014), Ryan et al (2020), Park et al. (2022b).

Shortfin mako

The EDD maps of the shortfin mako centrum (Fig. 6) cover only part of the intermedialia and none of the cone wall; because the lamellae were so narrow, this choice was made to obtain highest spatial resolution in the available beam time. One cannot compare, therefore, the texture of the cone walls with that of the lamellae for this data set. If the mako intermedialia texture were the same as in the blue shark intermedialia, the 00.2 intensities of detectors 1 and 9 would be substantially larger than those in detectors 5 and 10. The situation, however, appears more

complicated in the shortfin mako centrum: the 00.2 intensities of detector 5 and 9 are comparable, that of detector 1 is the largest of all detectors and that of detector 10 is substantially lower than any other detector. Note also that the detectors with an upward component (1-4) have higher intensities than those with a downward component (6-9).

The reduced axial orientation in the intermedialia of the shortfin mako compared to that of the blue shark tempts one to speculate that mechanical loading of the lamellae differs from that of the wedges and that this drives the difference in *c*-axis orientation (and in the underlying cartilage). Lamellae must resist out of plane bending produced by applied axial compression. The situation is further complicated by the fact that lamellae experience reactionary axial tensile loading when opposite side of the centrum is compressed. With multiaxial resistance required, one would expect the *c*-axes of growing mineralized tissue to be laid down in more isotropic orientations. In contrast, most of the wedge volume in carcharhiniforms has pronounced lateral constraint from the surrounding material, and one would expect axial reinforcement is the main requirement on the mineral phase. However attractive this speculation is, it needs to be confirmed by improved measurements.

For the shortfin mako centrum, the quadruplet reflections show some intensity variation between maps with different detectors (Fig. 6). Like that seen in the 00.2 maps, detector 5 intensity is greater than detector 10. In Fig. 6, detectors 4-6 also have greater intensities than detectors 1-3 and 7-9. The quadruplet reflection for the shortfin mako centrum, therefore, is more revealing of texture than the same reflection in bone. This is not unexpected because the matrix of shark centra is cartilage (and not type I collagen), and WAXS and SAXS of blocks of tissue cut from centra showed organization of the bioapatite nanoparticles a bit different from that in bone (Park et al. 2022a).

The authors anticipated that the relatively large sampling volume length along the beam direction and the narrow width of the lamellae in the shortfin mako intermedialia (Fig. 1) would complicate analysis and therefore concentrated available scanning time on sampling with a much finer grid over a much smaller volume than in the blue shark. Despite this, interpretation of the shortfin mako EDD data is still confounded by sub-voxel sampling: in some positions it appears two, three or more lamellae contribute intensities and nearby voxels only one lamella contributes. Three radial bands (red arrows at the bottom of Fig. 6a) are structures where individual lamellae or pairs of lamellae are sampled whereas the bands bordering the gaps are examples of signals from larger groups of lamellae combined in a single EDD voxel. As it is unlikely that EDD mapping with isotropic 100 μm voxels will be achieved in the near future, interpretation of data from lamniform centra will require care and be circumscribed with caveats whereas analysis of carcharhiniform centra should be straightforward. Accurate registration of lab microCT and EDD reconstructions could offer a way forward in studying lamniform intermedialia, but implementing, validating and applying such methodologies is apt to be extremely laborious.

In situ EDD strain mapping of centra

The studies cited above (Porter and Long 2010, Long et al. 2011, Porter et al. 2014, Ingle et al. 2018) demonstrated shark centra experience large continuum level strains in vivo but did not reveal (a) whether the mineral and not merely the type II cartilage was deforming nor (b) whether, if the mineral phase does carry strain, this strain varies as a function of position. Stock et al. (2022) used synchrotron microCT to show that cone walls and intermedialia consisted of thin, closely-spaced trabeculae with a substantial volume fraction of unmineralized tissue.

Concerning question (a), Stock et al. (2021) found that a polymer matrix composite containing a very large volume fraction of synthetic hAp particles and an open framework of struts experienced large structural distortions (engineering strains $> 20\%$) without hAp carrying appreciable strain. Regarding question (b), Park et al. (2022c) presented a preliminary report of EDD-derived strain maps in a shark centra under compression but did not report detailed analyses.

Use of purpose-built load frames for tomography studying specimens under load, e.g. Breunig et al. (1992, 1993), and position-resolved strain quantification under applied load for mineralized tissues, e.g. Almer and Stock (2010), Stock et al. (2011), are not new, but the approach of Park et al. (2022c) is novel because EDD is used and because these are the first positioned resolved and in situ strain measurements for shark mineralized tissue. In this proof of principle experiment, Park et al. used spacers within a plastic jar, between jar's lid and base, to compress the centra by known increments. The sampling volume was scanned across the middle transverse plane of the centra with no applied displacement and after one increment of compression. Changes in diffraction peak positions for the detector pair 5 and 10 and for pair 1 and 9 are the basis for the strain determination, a standard x-ray diffraction approach, and Park et al. (2022c) found shifts in bioapatite peak positions for different displacements, indicating the mineral phase was being strained. Work remains before 3D maps of strain distributions are extracted.

Modeling of centra response to loading

The authors are unaware of any 2D or 3D modeling of shark centra response to deformation. Mineralized shark cartilage, however, has been modeled numerically for several

interconnected tesserae (Seidel et al. 2019). The tesserae modeling was 2D and considered two structural units: stiffer, more highly-mineralized, radially-oriented spokes and inter-spoke volumes. Synchrotron microCT of centra (Stock et al. 2022) indicates the micrometer-level structure of centra is quite different from that of tesserae, so the results of Seidel et al. (2019) are not directly transferrable to centra. MicroCT-based, 3D models of trabecular bone and of antler are well developed, e.g., Kinney et al. (2000), Niebur et al. (2000), van Rietbergen (2001), Gupta et al. (2013), and use of these approaches might be very valuable with centra.

Modeling shark centra is outside of the areas in which the authors work. However, the available macrostructural, microstructural, mineral density and crystallographic texture data suggest differences, particularly between cone and wedge, evolved to provide improved mechanical functionality. The authors propose the following sequence of simple-to-complex models (Fig. 7) as an effective approach. For purposes of illustration, the present discussion is restricted to the more geometrically simple carcharhiniform centra (intermedialia consisting of a pair of large medio-lateral wedges and a pair of dorsal-ventral wedges) and ignores lamniform centra (intermedialia consisting of closely-spaced, diverging and merging lamellae), see Morse et al. (2022).

A first step could be 2D simplifications based on the diametral section through the center of medio-lateral wedges (locations at “xs” in the middle panels of Fig. 7a-c and sections shown in the bottom panels of Fig. 7a-c). This section is the one that would experience the largest range of in vivo strains. The simplest loading regime is uniaxial compression, applied load P , and simplest centrum model consists of three materials (Fig. 7a): unmineralized cartilage “uc” of the gap regions between wedges (“g” in Fig. 1), mineralized cartilage “mc” of the centrum (ignoring differences in mineralization levels between cone and intermedialia) and intervertebral fluid

“ivf” (incompressible, sealed in the volume between cones of adjoining centra). In the 2D model, one could investigate the effect of mineralized cartilage’s (isotropic) Young’s modulus on strain distribution. Different hourglass angles α , centrum heights h and diameters ϕ occur along the length of the shark vertebral column and for different carcharhiniform species (Ingle et al. 2018, Morse et al. 2022), and the effect of these on strain distributions can also be examined parametrically. Simple uniaxial compression of the same three-component model can be extended to a distribution of loads representing in vivo bending (bottom panel, Fig. 7b).

The next level of complexity for a 2D model is partitioning the mineralized cartilage into cone “c” with thickness δ and intermedialia “i” with isotropic Young’s moduli E_c and E_w , respectively. Although centra tissue moduli data do not presently exist, one expects $E_c > E_w$ because carcharhiniform cones contain more mineral than their intermedialia (Morse et al. 2022), and, as a zero-order approximation, one expects a relationship between tissue mineral level and Young’s modulus similar to that in bone (Currey 2002). Different combinations of α , h , ϕ and δ could be examined virtually and compared to what nature has evolved.

Centra are 3D structures, and the 2D models would not be expected to capture all aspects of deformation. The 3D macrostructure can be imported into a finite element or other numerical model using 3D microCT data, e.g. Morse et al. (2022). As mentioned above, this approach is widely applied for trabecular bone, and a sampling of such studies can be found elsewhere (Stock 2019). Following the steps outlined above in 2D, a three-component, two-material system with isotropic Young’s moduli would be modeled first in uniaxial compression (Fig. 7a, top) and then in bending (Fig. 7b, top). Note that $E_{mc} \gg E_{uc}$, where mc and uc denotes the mineralized and unmineralized cartilage, respectively. One expects a slight variation in strains between the

the wedge interiors to the volumes bordering the gaps containing unmineralized cartilage. It would also be interesting to see whether strain varied radially and axially within the wedge.

The next step might be partition of the mineralized tissue into cone and intermedialia with isotropic Young's moduli $E_c > E_w$. In addition to the spatial variations mentioned in the previous paragraph, strain gradients may occur between cone and intermedialia.

Based on the EDD observations reported above, namely, that strong bioapatite *c*-axis crystallographic textures occur in the cone and intermedialia, the authors believe that accurate modeling will require anisotropic moduli for both structures. In the centrum's coordinate system of tangential, radial and axial directions, denoted *t*, *r* and *a*, respectively (Fig. 7d), the anisotropic Young's moduli are denoted E_t , E_r and E_a , respectively. More experimental work is required to determine how much E_t , E_r and E_a actually differ, but this could be examined virtually via the 3D model.

Alternatives to EDD and future experiments

Alternatives to EDD exist which are suitable for samples containing nanocrystals (such as are present in bioapatites), which employ monochromatic x-radiation and which might allow diffraction from an individual lamella to be isolated from that of closely spaced neighbors. Use of conical slits (Park et al. 2013) or spiral slits (Martins et al. 2010) with monochromatic x-rays, for example, allows diffraction patterns to be collected with a simple area detector and would not require a specialized detector array. Translation similar to that employed here would allow 3D reconstruction of the volume, but the sampling volume would be elongated along the beam direction like that in the present study. Insertion devices such as that of 1-ID, APS, presently

offer orders of magnitude greater flux at any selected wavelength compared to 6-BM-B, and the greater flux could be traded for finer spatial sampling. Conical slits tailored for monochromatic x-rays and the hexagonal crystal system (i.e., for bioapatites) do not appear to exist, limiting the number of diffraction peaks which can be collected simultaneously, but spiral slits can capture all of the hexagonal reflections at the same time albeit with azimuthal gaps in the diffraction rings. It is, however, extremely difficult to obtain beam time at 1-ID, APS, so EDD at 6-BM-B, APS, remains an extremely attractive option.

An alternative to slit-based approaches is x-ray diffraction tomography with an incident monochromatic pencil beam, an open area detector and translation-rotation data acquisition, e.g. Stock et al. (2008), Birkbak et al. (2015). Reconstruction of cross-sectional variation of diffracted intensity employs back projection or other computed tomography algorithm. Compared to the translation-rotation method, 3D EDD mapping has the advantage of isolating diffraction from the sampling volume from all other scattering. Additionally, EDD mapping does not require specimen rotation, and specimens with complex surrounding tissue can be aligned so these structures do not affect the diffraction signal. For completeness, one should mention the 3D scattering approach termed tensor tomography which has been applied to bone, e.g. Maleki et al. (2014), but only to relatively small sections cut from larger specimens.

Accumulated x-ray damage affects mechanical properties of collagen-based mineralized tissues (Barth et al. 2011), these effects rise with increasing dose and become a concern in the context of in situ loading and strain measurements. While estimating dose is fairly straightforward with monochromatic high-energy x-rays, e.g., Stock et al. (2020), quantitative estimates for polychromatic x-radiation cannot currently be made mainly because incident x-ray spectra (intensity vs energy) are unavailable for the 6-BM-B bending magnet. Were the spectra

available, one could then calculate dose for each energy using tabulations such as that of Hubble and Seltzer (2004) and sum the individual contributions.

In qualitative terms, 3D mapping with EDD damages the tissue much more than the monochromatic, high-energy techniques described above; this may not be an issue for one-and-done mapping (like the data reported above) but might be for in situ loading experiments. First, photons in the 25-55 keV range (the EDD data) deposit much more energy per photon those at 70 keV and above. Second, in monochromatic methods, all of the photons traversing the specimen have the potential to contribute to the diffracted signal whereas in EDD, photons with energies between those diffracted into the detectors, deposit energy in the specimen and do not contribute to signal. Third, all points along the path of the incident beam are exposed when diffraction from one sampling volume is collected: If ten positions through the sample thickness are sampled, then the dose is ten time larger over the entire thickness.

Some damage to mineralized tissue samples is inevitable for x-ray-based methods such as diffraction, and damage is minimized if each sampled volume is exposed once for the minimum time required to obtain useful signal. In diffraction quantification of internal strains accompanying four-point bending, for example, Gallant et al. (2014) made use of the fact that the tensile to compressive strain gradients are nominally the same for all cross-sections within the inner span of the apparatus; they translated to a new cross-section before measuring strains for each increment of deflection. This approach could be adapted for EDD mapping of the intermedialia of carcharhiniform centra where strains should vary continuously. With reference to Fig. 2d, one could collect data at the red squares' positions for one increment of deformation and then at an adjacent position for the next.

The preliminary report of Park et al. (2022c) suggests in situ compression and EDD 3D mapping of strains in the bioapatite phase of shark centra will be very informative in describing how the centra function mechanically. It should be straightforward use EDD to apply x-ray diffraction strain quantification under four-point bending (e.g. Gallant et al. 2014) to segments of multiple shark vertebrae (e.g. Long et al. 2011), i.e., in a loading mode more closely approximating that in vivo.

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References

- Almer, J.D., Stock, S.R. 2005. Internal strains and stresses measured in cortical bone via high-energy x-ray diffraction. *J Struct Biol* **152**, 14-27.
- Almer, J.D., Stock, S.R. 2010. Loading-related strain gradients spanning the mature bovine dentinoenamel junction (DEJ): Quantification using high energy x-ray scattering. *J Biomech* **43**, 2294-2300.

653 Barth, H.D., Zimmermann, E.A., Schaible, E., Tang, S.Y., Alliston, T., Ritchie, R.O. 2011.
654 Characterization of the effects of x-ray irradiation on the hierarchical structure and
655 mechanical properties of human cortical bone. *Biomater* **32**, 8892-8904.

656 Birkbak, M.E., Leemreize, H., Frølich, S., Stock, S.R., Birkedal, H. 2015. Diffraction Scattering
657 Computed Tomography: A Window into the Structures of Complex Nanomaterials.
658 *Nanoscale* **7**, 18402 – 18410.

659 Breunig, T.M., Stock, S.R., Antolovich, S.D., Kinney, J.H., Massey, W.N, Nichols, M.C. 1992.
660 A Framework Relating Macroscopic Measures and Physical Processes of Crack Closure of
661 Al-Li Alloy 2090. in *Fracture Mechanics: Twenty-Second Symposium (Volume 1)*, ASTM
662 STP **1131** (1992) 749-761.

663 Breunig, T.M., Stock, S.R., Brown, R.C. 1993. Simple Load Frame for in situ Computed
664 Tomography and X-ray Tomographic Microscopy. *Mater Eval* **51**, 596-600.

665 Chaumel, J., Schotte, M., Bizzarro, J.J., Zaslansky, P., Fratzl, P., Baum, D., Dean, M.N. 2020.
666 Co-aligned chondrocytes: Zonal morphological variation and structures arrangement of cell
667 lacunae in tessellated cartilage. *Bone* **134**, 115624.

668 Cullity, B.D., Stock, S.R. 2001. *Elements of X-ray Diffraction*, 3rd Ed., Prentice-Hall: Upper
669 Saddle River.

670 Currey, J.D. 2002. *Bones – Structure and Mechanics* Princeton: Princeton Univ. Press. Dean,
671 M.N., Chiou, W.-A., Summers, A.P. 2005. Morphology and ultrastructure of prismatic
672 calcified cartilage. *Microsc Microanal* **11**, 1196-1197.

673 Gallant, M.A., Brown, D.M., Hammond, M., Wallace, J.M., Du, J., Deymier-Black, A.C.,
674 Almer, J.D., Stock, S.R., Allen, M.R., Burr, D.B. 2014. Bone cell-independent benefits of

675 raloxifene on the skeleton: A novel mechanism for improving bone material properties.
676 Bone **61**, 191-200.

677 Geraghty, P.T., Jones, A.S., Stewart, J., Macbeth, W.G. 2012. Micro-computed tomography: an
678 alternative method for shark aging. J Fish Biol **80**, 1292-1299.

679 Gupta, H.S., Krauss, S., Kerschnitzki, M., Karunaratne, A., Dunlop, J.W.C., Barber, A.H.,
680 Boesecke, P., Funari, S.S., Fratzl, P. 2013. Intrafibrillar plasticity through mineral/collagen
681 sliding is the dominant mechanism for the extreme toughness of antler bone. J Mech Behav
682 Biomed Mater **28**, 366-382.

683 Gupta, H.S., Schratter, S., Tesch, W., Roschger, P., Berzlanovich, A., Schoeberl, T., Klaushofer,
684 K., Fratzl, P. 2005. Two different correlations between nanoindentation modulus and mineral
685 content in the bone–cartilage interface. J Struct Biol **149**, 138-148.

686 Guo, X.E. 2001. Mechanical properties of cortical bone and cancellous bone tissue. Ch. 10 in
687 *Bone Mechanics Handbook*, 2nd Ed. Cowin, S.C., Ed. Boca Raton (FL), CRC Press, pp 10-1-
688 10-23.

689 Hubbell, J.H., Seltzer, S.M. 2004. Tables of X-Ray Mass Attenuation Coefficients and Mass
690 Energy-Absorption Coefficients from 1 keV to 20 MeV for Elements Z = 1 to 92 and 48
691 Additional Substances of Dosimetric Interest. NISTIR 5632, accessed Aug. 7, 2020.
692 <https://www.nist.gov/pml/x-ray-mass-attenuation-coefficients>

693 Ingle, D.I., Natanson, L.J., Porter, M.E. 2018. Mechanical behavior of shark vertebral centra at
694 biologically relevant strains. J Exp Biol **221**, 188318.

695 Kinney, J. H., Haupt, D.L., Balooch, M., Ladd, A. J. C., Ryaby, J. T. , Lane, N. E. 2000. Three-
696 dimensional morphometry of the L6 vertebra in the ovariectomized rat model of
697 osteoporosis: Biomechanical implications. J Bone Miner Res **15**, 1981-1991.

698 Liu, X., Dean, M.N., Youssefpour, H., Summers, A.P., Earthamn, J.C. 2014. Stress relaxation
 699 behavior of tessellated cartilage from the jaws of blue sharks. *J Mech Behav Biomed Mater*
 700 **29**, 68-80.

701 Long, J.H. Jr., Koob, T., Schaefer, J., Summers, A., Bantilan, K., Grotmol, S. Porter, M. 2011.
 702 Inspired by sharks: A biomimetic skeleton for the flapping, propulsive tail of an aquatic
 703 robot. *Marine Technol Soc J* **45**(4), 119-129.

704 Malecki, A., Potdevin, G., Biernath, T., Eggl, E., Willer, K., Lasser, T., Maisenbacher, J.,
 705 Gibmeier, J., Wanner, A., Pfeiffer, F. 2014. X-ray tensor tomography. *Europhys Lett* **105**,
 706 38002.

707 Martins, R.V., Ohms, C., and Decroos, K. 2010. Full 3D spatially resolved mapping of residual
 708 strain in a 316L austenitic stainless steel weld specimen. *Mater Sci Eng A* **527**, 4779–4787.

709 Morse, P.E., Stock, M.K., James, K.C., Natanson, L.J., Stock, S.R. 2022. Shark vertebral
 710 microanatomy and mineral density variation studied with laboratory microComputed
 711 Tomography. *J Struct Biol* **214**, 107831.

712 Naik, N.N., Jupe, A.C., Stock, S.R., Wilkinson, A.P., Lee, P.L., Kurtis, K.E. 2006. Sulfate attack
 713 monitored by microCT and EDXRD: Influence of cement type, water-to-cement ratio, and
 714 aggregate. *Cement Concrete Res* **36**, 144-159.

715 Natanson, L.J., Skomal, G.B., Hoffmann, S.L., Porter, M.E., Goldman, K.J., Serra, D. 2018. Age
 716 and growth of sharks: do vertebral band pairs record age? *Mar Freshwater Res* **69**, 1440-
 717 1452.

718 Niebur, G.L., Feldstein, M.J., Yuen, J.C., Chen, T.J., Keaveny, T.M. 2000. High-resolution finite
 719 element models with tissue strength asymmetry accurately predict failure of trabecular bone.
 720 *J Biomech* **33**, 1575-1583.

721 Park, J.S., Lienert, U., Dawson, P.R., and Miller, M.P. 2013. Quantifying Three-Dimensional
 722 Residual Stress Distributions Using Spatially-Resolved Diffraction Measurements and Finite
 723 Element Based Data Reduction. *Exp Mech* **53**, 1491–1507.

724 Park, J.-S. Almer, J.D., James, K.C., Natanson, L.J., Stock, S.R. 2022a. Mineral in shark
 725 vertebrae studied by wide angle and by small angle x-ray scattering. Under revision.

726 Park, J.-S., Laugesen, M., Mays, S., Birkedal, H., Almer, J.D., Stock, S.R. 2022b. Intact
 727 archeological human bones studied with transmission x-ray diffraction and small angle x-ray
 728 scattering. *Int J Osteoarchaeol* **32**, 170-181.

729 Park, J.S., Chuang, A.C., Okasinski, J., Chen, H., Shade, P., Stock, S.R., Almer, J. A new
 730 residual strain mapping program using energy dispersive x-ray diffraction at the Advanced
 731 Photon Source. 2022c. *Exp Mech*, accepted.

732 Porter, M.E., Long, J.H. Jr. 2010. Vertebrae in compression: Mechanical behavior of arches and
 733 centra in the gray smooth-hound shark (*Mustelus californicus*). *J Morphol* **271**, 366-375.

734 Porter, M.E., Diaz, C., Sturm, J.J., Grotmol, S., Summers, A.P., Long, J.H., Jr. 2014. Built for
 735 speed: strain in the cartilaginous vertebral columns of sharks. *Zool* **117**, 19-27.

736 Porter, M.E., Ewoldt, R.H., Long, J.H. Jr. 2016. Automatic control: the vertebral column of
 737 dogfish sharks behaves as a continuously variable transmission with smoothly shifting
 738 functions. *J Exp Biol* **219**, 2908-2919.

739 Ryan, J., Stulajter, M.M., Okasinski, J.S., Cai, Z., Gonzalez, G.B., Stock, S.R. 2020. Carbonated
 740 apatite lattice parameter variation across incremental growth lines in teeth. *Materialia* **14**,
 741 100935.

742 Seidel, R., Roschger, A., Li, L., Bizzarro, J.J., Zhang, Q., Yin, J., Yang, T., Weaver, J.C., Fratzl,
 743 P. Roschger, P., Dean, M.N. 2019. Mechanical properties of stingray tesserae: High-

744 resolution correlative analysis of mineral density and indentation moduli in tessellated
745 cartilage. *Acta Biomater* **96**, 421-435.

746 Seidel, R., Jayasankar, A.K., Dean, M.N. 2021. The multiscale architecture of tessellated
747 cartilage and its relation to function. *J Fish Biol* **98**, 942-955.

748 Stock, S.R. 2019. *MicroComputed Tomography: Methodology and Applications*, 2nd Ed., Taylor
749 and Francis.

750 Stock, S.R., DeCarlo, F., Almer, J.D. 2008. High energy x-ray scattering tomography applied to
751 bone. *J Struct Biol* **161**, 144-150.

752 Stock, S.R., Deymier-Black, A.C., Veis, A., Telser, A., Lux, E., Cai, Z. 2014. Bovine and equine
753 peritubular and intertubular dentin. *Acta Biomater* **10**, 3969-3977.

754 Stock, S.R., Yuan, Fang, Brinson, L.C., Almer, J.D. 2011. High-energy x-ray scattering
755 quantification of internal strains and their gradients in bone under load. *J Biomech* **44**, 291-
756 296.

757 Stock, S.R., Stock, M.K., Almer, J.D. 2020. Combined Computed Tomography and position-
758 resolved x-ray diffraction of an intact Roman-era Egyptian portrait mummy. *J Royal Soc*
759 *Interface* **17**, 20200686.

760 Stock, S.R., Park, J.-S., Jakus, A., Birkbak, M., Frølich, S., Birkedal, H., Shah, R., Almer, J.D.
761 2021. “*In situ* loading and x-ray diffraction quantification of strains in hydroxyapatite
762 particles within a 3D printed scaffold.” *Materialia* **18**, 101174.

763 Stock, S.R., Morse, P.E., Stock, M.K., James, K.C., Natanson, L.J., Chen, Haiyan, Shevchenko,
764 P.D., Maxey, E.R., Antipova, O.A., Park, J.-S. 2022. “Microstructure and energy dispersive
765 diffraction reconstruction of 3D patterns of crystallographic texture in a shark centrum.” *J*
766 *Medical Imaging* **9**, 031504.

Urist, M.R. 1961. Calcium and phosphorus in the blood and skeleton of the Elasmobranchii. *Endocrinol* **69**, 778-801.

van Rietbergen, B. 2001. Micro-FE analyses of bone - State of the art. in Noninvasive Assessment of Trabecular Bone Architecture and the Competence of Bone. Bay, B.K., Majumdar, S. Eds. New York, Kluwer/Plenum. **Adv Exp Med Biol Vol 496**: 21-30.

Watanabe, Y.Y., Lydersen, C., Fisk, A.T., Kovacs, K.M. 2012. The slowest fish: Swim speed and tail-beat frequency of Greenland sharks. *J Exp Mar Biol Ecol* **426/427**, 5-11.

Weidner, D.J., Vaughan, M.T., Wang, L., Long, H., Li, L., Dixon, N.A., Durham, W.B. 2010. Precise stress measurements with white synchrotron x-rays. *Rev Sci Instrum* **81**, 013903.

Figure captions

Fig. 1. Schematics of sharks. (a) Side view of a generalized shark. The blocks within the silhouette represent the vertebrae: gray for cervical and thoracic vertebrae and black for abdominal vertebrae. The arrow indicates very approximately the position of the vertebrae in this study. (b) Schematic from above of vertebral compression when a shark swims, illustrated schematically by three positions of the tail. When the tail swings to the right (right tail diagram), the right side of the vertebra is compressed (right trapezoid with arrows indicating compression). As the left side of the schematic indicates, the vertebra's opposite (left) side compresses when the tail moves to the left. Panels (c) and (d) show lamniform (left column) and carcharhiniform (right column) abdominal centra. (c) 3D representation of a lamniform and a carcharhiniform centrum with some material shown transparent so that the structures' cross-sections can be seen. c - hourglass-shaped cone walls, i – intermedialia, L – lamella, W – wedge. (d) Thresholded slices (black in these transverse sections perpendicular to the vertebral column axis and near the centra's axial midplane) of a mako centrum (lamniform shark mapped in this study) showing the

radial lamellae and of a sandbar shark centrum (carcharhiniform shark similar to the blue shark mapped in this study). The dorsal-ventral plane is vertical (double arrowed line), and the intersection of the frontal (coronal) plane with each slice is shown as a horizontal red dotted line. Gaps “g” between wedges (sandbar shark) and between groups of closely spaced lamellae (mako shark) are labeled.

Fig. 2. Schematic of the EDD apparatus with the arrangement of ten detector elements and of the sampling scheme illustrated for a lamniform and a carcharhiniform shark. (a) Energy dispersive diffraction apparatus illustrated with a schematic of a lamniform centrum. A set of three orthogonal linear translators (along X, Y and Z) scan the specimen across the sampling volume “sv”. The other main components are: detector elements “de”, conical slits “cs” and incident beam collimator “col”. (b) Arrangement of detector elements 1-10 and the orientation of different lattice plane normals N_1 and N_5 for the crystallites producing diffracted intensities measured in detectors 1 and 5. (c) Schematic of shortfin mako and (d) diagram of blue shark sampling grids illustrated by side views (top) and transverse views (bottom) of the two centra. The labels are: cone wall “cw”, lamellae “L”, wedge “W” and vertebral column axis “vca”. Note that only a small portion of the shortfin mako centrum was covered, but the entire volume of the blue shark centrum was scanned.

Fig. 3. Typical energy dispersive diffraction pattern from a shark centrum. The 00.2 and “q” (unresolved 21.1, 11.2, 30.0 and 20.2) peaks from hAp are labeled as well as peaks “cal” from the calibration source. The green peaks below the data are for a synthetic hAp reference pattern (Powder Diffraction File 86-1201).

Fig. 4. Orthogonal sections through the blue shark centrum reconstructed with the integrated intensity of the quadruplet reflection (21.1+11.2+30.0+20.2). The brighter the pixel, the higher

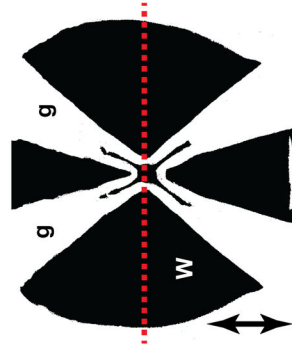
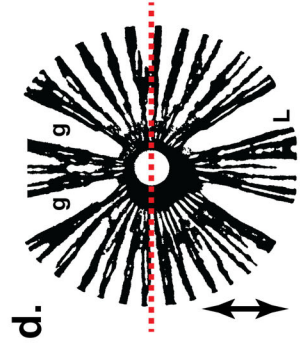
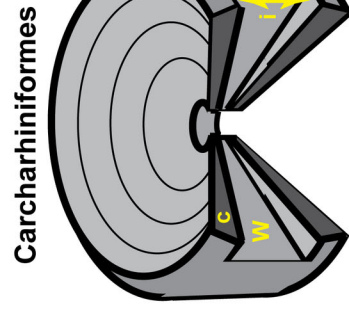
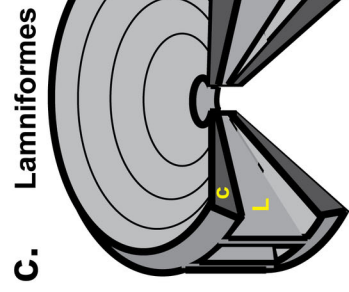
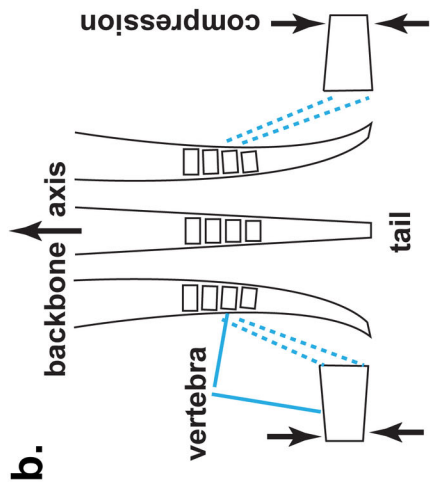
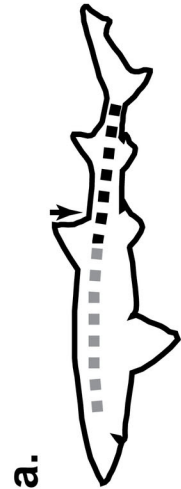
the intensity in the voxel. The number in the upper left of each panel identifies the detector number. The gray scale images of the top row show intensities recorded with detector 5 (left) and the diametrically opposite detector 10 (middle) and of the bottom row intensities from detectors 1 (left) and diametrically opposite 9 (right). The color panels (right column) show the combined intensities of detectors 5 and 10 (top) and 1 and 9 (bottom) with 5 and 9 in blue and 10 and 1 in red. The yellow arrowheads identify the cone wall diffracting greater intensity than the adjacent intermedialia. The orange arrowheads show portions of the maps where intermedialia and cone wall intensities are comparable. The yellow (and one black) boxes in the detector 5 and 1 panels indicate the positions of 3 voxel x 3 voxel x 1 voxel regions used to measure mean diffracted intensity within the cone wall (cw) and intermedialia (i, i'); these values are reported in Table 2.

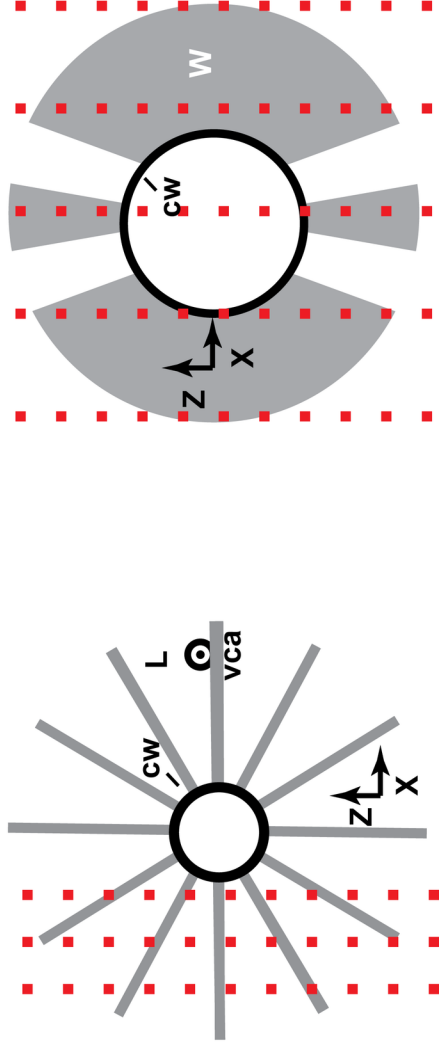
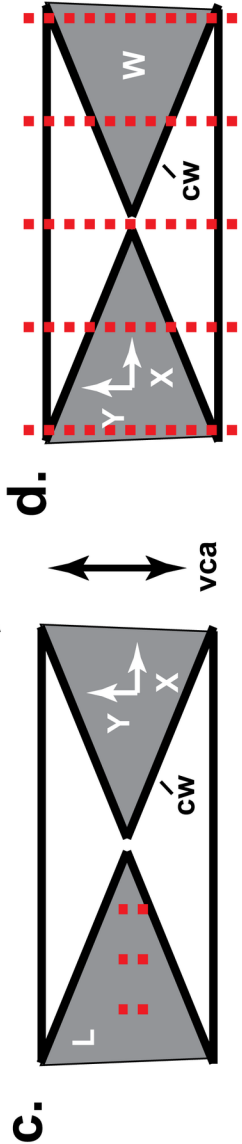
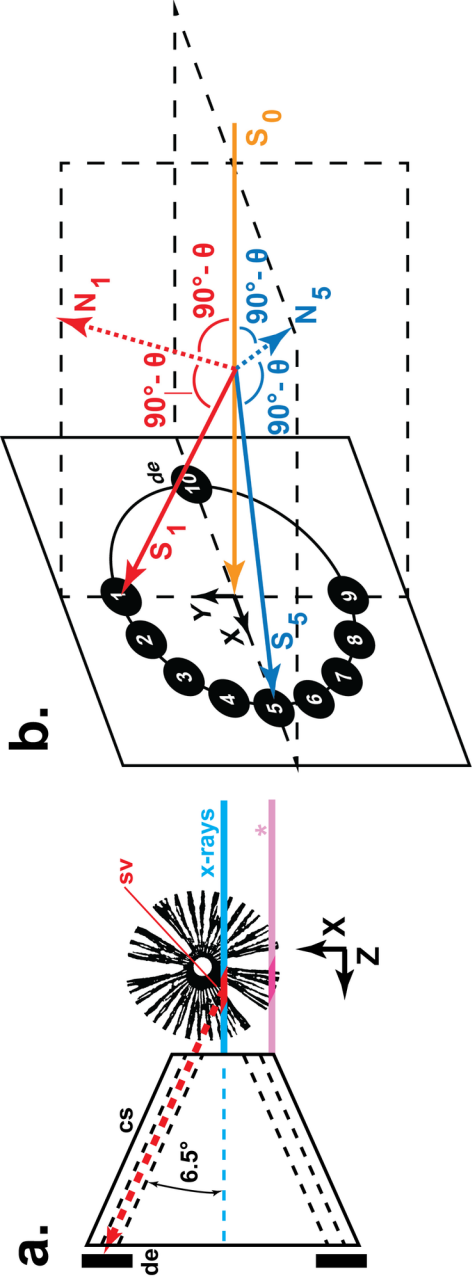
Fig. 5. (a) Orthogonal sections through the blue shark centrum reconstructed with the 00.2 integrated intensity. The brighter the pixel, the higher the intensity in the voxel. Intensities from detectors 5 (blue), 7 (green) and 9 (red) are combined. The different colored arrows show the c -axis orientation measured by each detector. (b) Schematic of the bioapatite c -axis orientations revealed in panel a. The white boxes indicate the positions of 3 voxel x 3 voxel x 1 voxel regions used to measure mean diffracted intensity within the cone wall (cw), intermedialia (i, i') and fluid (f) for detectors 1 and 5; these values are reported in Table 2.

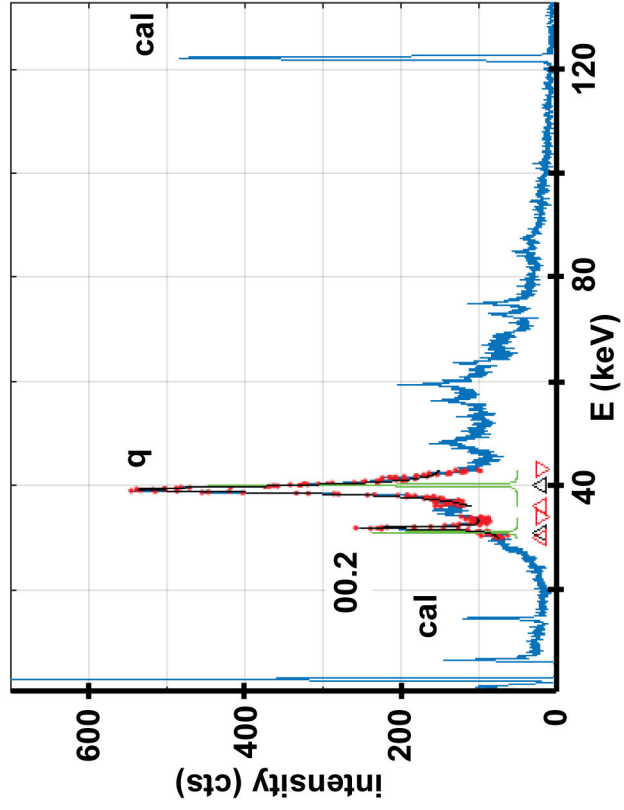
Fig. 6. Shortfin mako's spatial distribution of diffracted intensity measured with the ten detectors for the quadruplet q (a) and 00.2 (b) reflections. The color bar between the panels gives the intensity range which was 0 to 1,200 a.u. for q and 0-150 for 00.2. The detector data are placed in their correct relative orientations. The transverse section is from near the centrum's axial center and covers only part of the cross-section. Portions of two gaps ($g1$ and $g2$) and of three sectors ($s1, s2$ and $s3$) are labeled in the detector 1 map of (a) as is the 2 mm scale bar

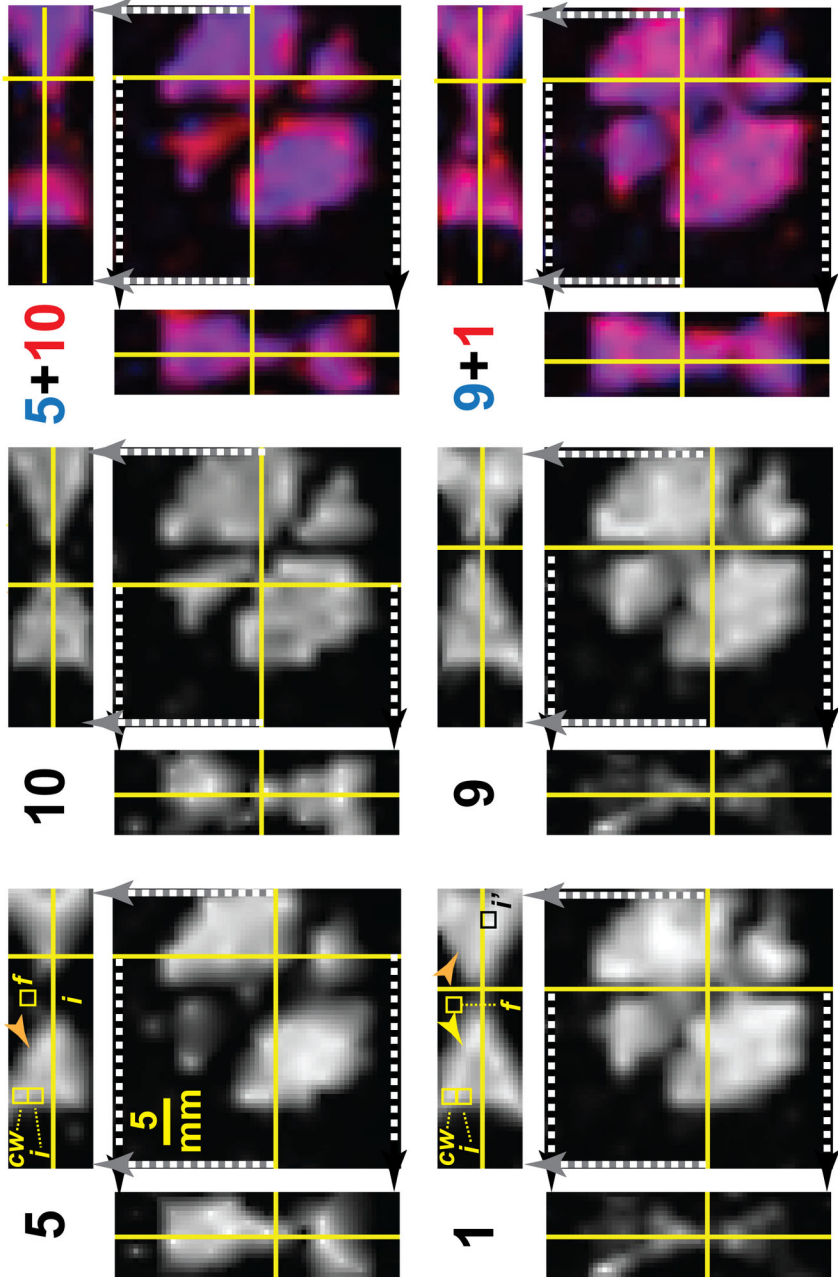
(white bar at the bottom). The segmented image inset in the middle of (a) is a lab microCT slice of the same centrum (from Morse et al. 2022).

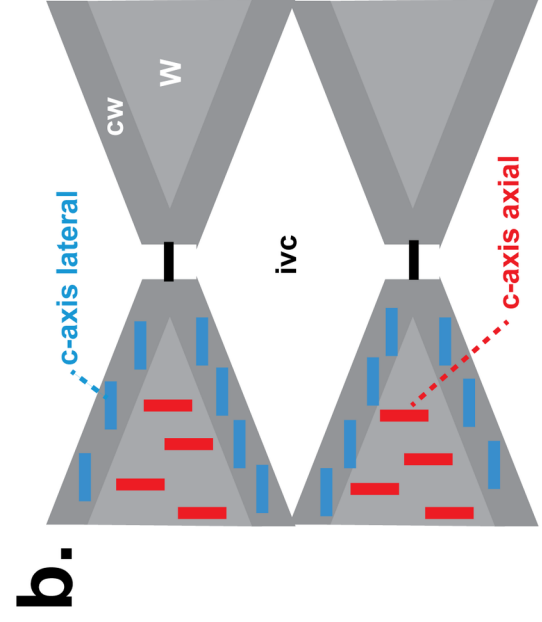
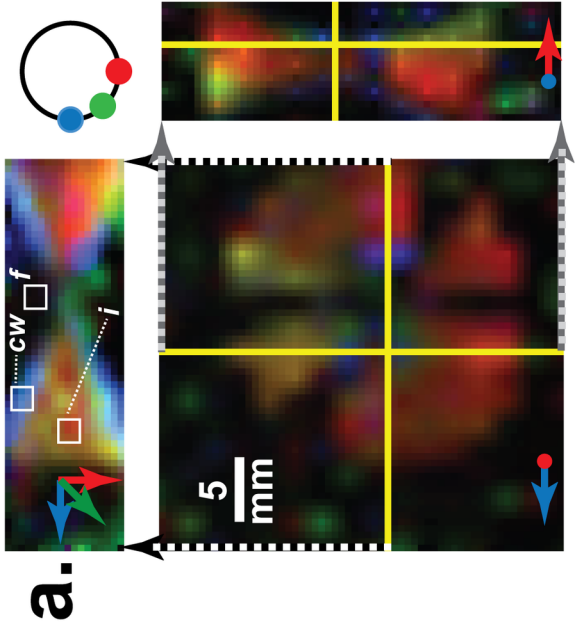
Fig. 7. Illustration of how carcharhiniform centra data could be incorporated into numerical models of mechanical response to loading. Centra 3D macroarchitecture from lab microCT (e.g. Morse et al. 2022) would be converted into a 3D model of the structure. Incompressible intervertebral fluid (ivf) would fill the space between the cones and the rigid platens applying load P . (a) Simple uniaxial compression of centra simplified to contain two isotropic phases: mineralized cartilage (mc, dark gray) and unmineralized cartilage (uc, light gray). The magenta arrows (xs) mark the position of the cross-section shown at the bottom of the panel. The centrum geometry is defined by diameter ϕ , height h and hourglass angle α . (b) Distribution of loads for simulated bending. The location of two large and two small wedges beneath the cone are shown in yellow, and the largest load occurs at the center of the outer edge of the large wedges, mimicking in vivo bending. Here the wedge and cone materials are not differentiated. (c) Non-uniform loading and partition of the mineralized cartilage into cone “c” (with thickness δ , shown in black) and intermedialia “i” (bottom panel). The corresponding 3D model (top panel) includes unmineralized cartilage “uc”, the intermedialia is in the form of wedges “W” and isotropic Young’s moduli $E_c > E_W \gg E_{uc}$. (d) Transverse centrum cross-section at the axial position between the cone apex and its axial end. The radial r , tangential t and axial a directions are indicated as are the potentially differing Young’s moduli along these three directions. The text at the bottom of the panel indicates expected relative magnitudes of moduli along different directions for the wedge and cone; this is based on the c -axis crystallographic textures reported in this paper.

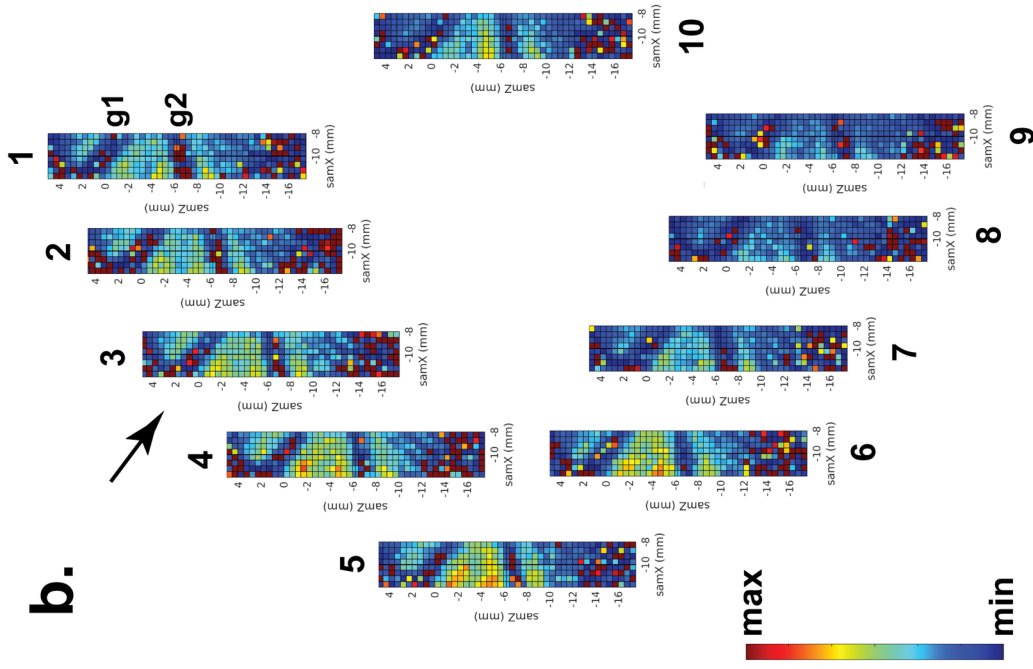
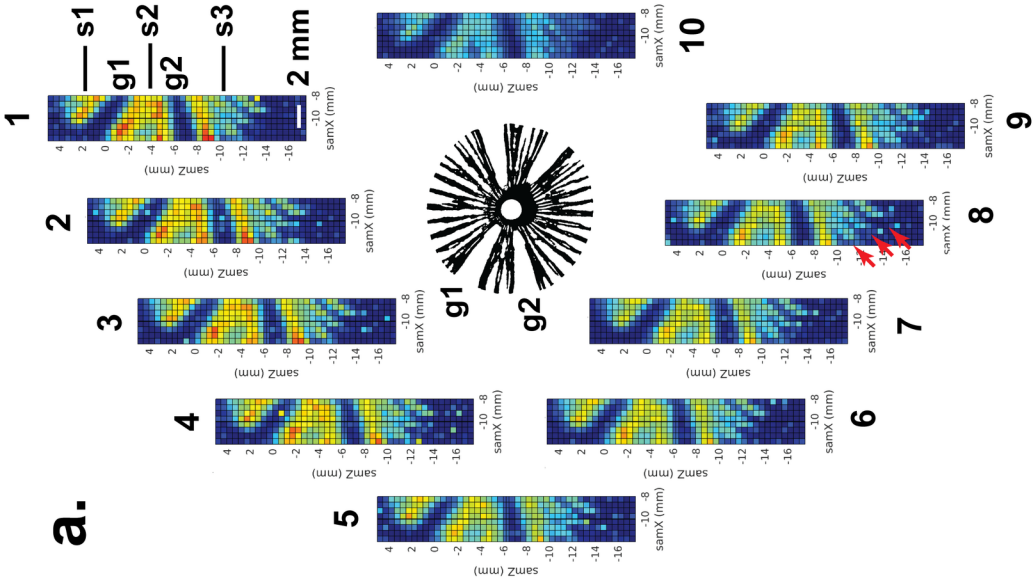












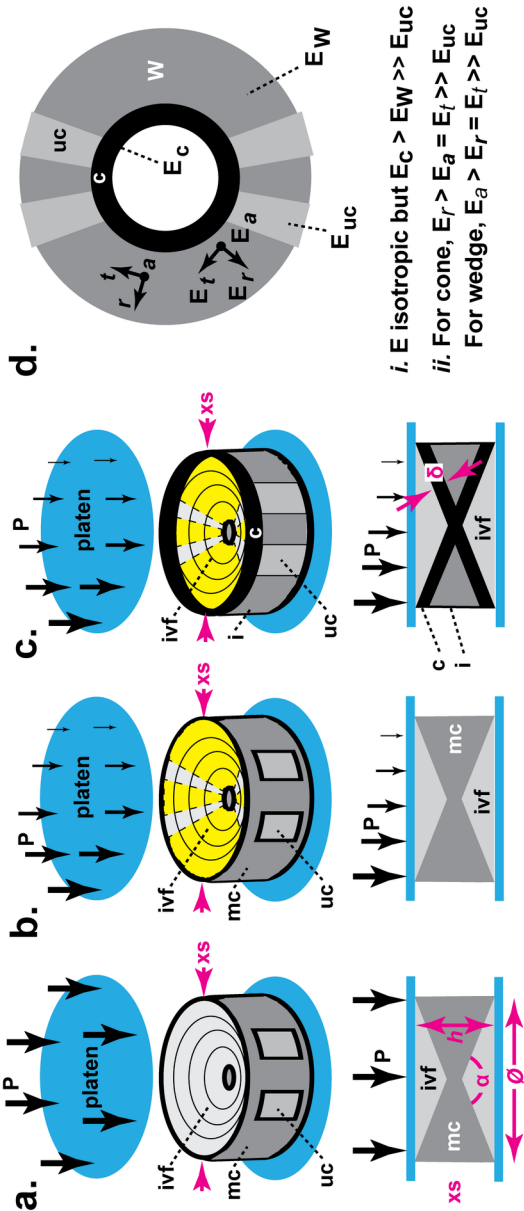
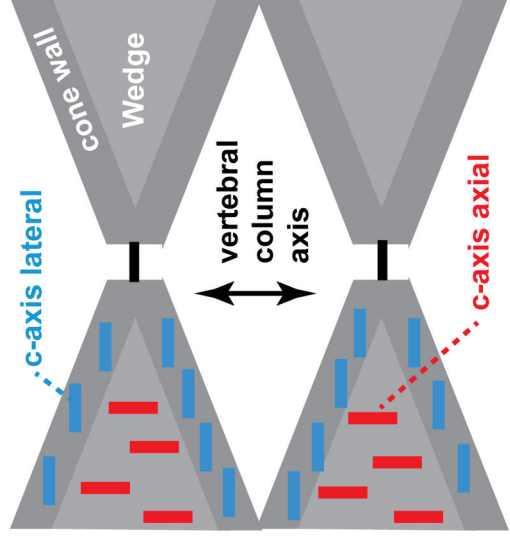
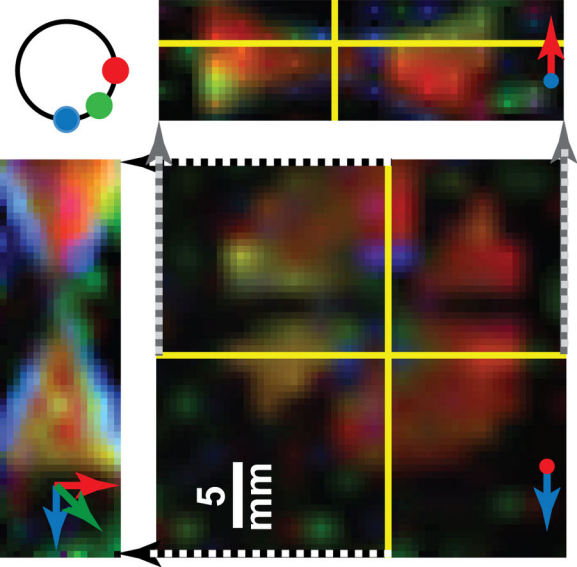


Table 1. Experimental parameters

	<u>Gage volume (mm)</u>			<u>Step size (mm)</u>			<u>Number of steps</u>			exposure (s)
	δX	δY	δZ	ΔX	ΔY	ΔZ	N_X	N_Y	N_Z	
Mako	0.2	0.2	2.5	1.5	0.5	1.0	3	10	47	50
Blue	0.1	0.2	1.7	2.0	1.0	1.9	16	10	17	30

Table 2. Integrated intensities in the cone wall (cw) and intermedialia (i) of the q and 00.2 reflections for the blue shark centrum. Mean values are for the 3 voxel x 3 voxel regions within the yellow squares of Fig. 4 and Fig. 5. The standard deviation of each mean is the number following $\pm$. The maximum integrated intensity of the 3D reconstructed volume with q reflection was 98 (arbitrary units) and with the 00.2 reflection was 86 (arbitrary units, a.u.). For 00.2, the integrated intensity recorded within fluid-filled volumes were 2 a.u. for detectors 1 and 5, and, for q, values were 2 and 4 a.u., respectively.

Reflection	detector 5		detector 1		
	<i>cw</i>	<i>i</i>	<i>cw</i>	<i>i</i>	<i>i'</i>
q	68 $\pm$ 7	61 $\pm$ 9	66 $\pm$ 7	46 $\pm$ 10	72 $\pm$ 3
00.2	46 $\pm$ 9	5 $\pm$ 1	21 $\pm$ 5	35 $\pm$ 2	



Bioapatite crystal orientations in blue shark centrum. (left) Energy dispersive diffracted intensity in 3D (colors indicate intensity along different anatomical directions). (right) Schematic indicating crystal c-axes in different portions of the centrum.