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# In Situ High-Temperature Raman Spectroscopy of $\text{UCl}_3$ : A Combined Experimental and Theoretical Study

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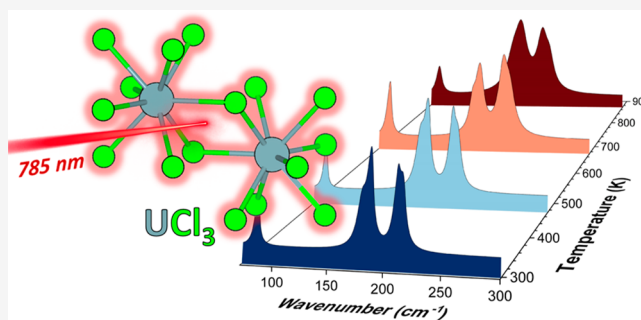


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Supporting Information

**ABSTRACT:** Uranium trichloride ( $\text{UCl}_3$ ) has received growing interest for its use in uranium-fueled molten salt reactors and in the pyrochemical processing of used fuel. In this paper, we report for the first time the experimentally determined Raman spectra of  $\text{UCl}_3$ , at both ambient condition and *in situ* high temperatures up to 871 K. The frequencies of five of the Raman-active vibrational modes ( $\nu_i$ ) of  $\text{UCl}_3$  exhibit a negative temperature derivative ( $(\partial\nu_i/\partial T)_P$ ) with increasing temperature. This red-shift behavior is likely due to the elongation of U–Cl bonds. The average isobaric mode Grüneisen parameter ( $\gamma_{ip} = 0.91 \pm 0.02$ ) of  $\text{UCl}_3$  was determined through use of the coefficient of thermal expansion published in Vogel et al. (2021) and the  $(\partial\nu_i/\partial T)_P$  values determined in this study. These results are in general agreement with those calculated here by density functional theory (DFT+U). Finally, a comparison of the ambient band positions of  $\text{UCl}_3$  to those of isostructural lanthanide (La–Eu) and actinide chlorides (Am–Cf) has been made.



## 1. INTRODUCTION

Growth in nuclear power is anticipated as an environmentally conscious, carbon-free energy source. A significant problem encountered with the light water reactors (LWRs) of generation II and III nuclear power plants is, however, that they generate transuranium byproducts. These byproducts are often the longest lived radioisotopes<sup>1</sup> and of the greatest concern from the perspective nonproliferation. Hence, these radioelements have been a significant focus in nuclear waste disposal strategies.<sup>2–4</sup> Conventionally, the disposal strategy of such wastes is through either a mined geological repository or a deep drilled borehole.<sup>5–7</sup> Another approach of dealing with the generation of nuclear waste is to design a nuclear reactor that can utilize more of the input fuel and consume the transuranium elements. This would significantly decrease the amount of radioactive waste produced and increase the environmental benefits of the carbon-free nuclear power. In the generation IV nuclear power plant designs, molten salt reactors (MSRs) have all these attributes. An example of such a reactor is at the Accelerator Research Laboratory (ARL), Texas A&M University (TAMU), where a method for accelerator-based destruction of actinides in molten salt (ADAM) to burn transuranium elements has been developed.<sup>8</sup> Although most of the research on MSR technologies has been toward fluoride-based fuel salts,<sup>9–11</sup> chloride-based fuel salts have higher solubility for actinides, can operate at lower temperatures, and

have higher volumetric heat capacity, while possessing good thermal conductivity and low viscosity.<sup>12–16</sup> All of these properties make chloride-based MSRs ideal for burning transuranium elements.<sup>12,13</sup> Despite this, there are still considerable knowledge gaps in the structures and properties of chloride salts at high temperatures.

Uranium trichloride ( $\text{UCl}_3$ ) is of significant interest as a fuel in MSRs. Recent advances have been made through a combination of both experimental (neutron scattering, neutron imaging, calorimetry, and dilatometry)<sup>15,17,18</sup> and computational (density functional theory)<sup>19</sup> methods.  $\text{UCl}_3$  is a prototypical compound, crystallizing in the hexagonal  $P6_3/m$  structure,<sup>20</sup> and nearly 100 other isostructural compounds have been reported in the Inorganic Crystal Structure Database (ICSD).<sup>18</sup> However, despite the importance of this phase, its Raman spectrum, to the best of our knowledge, has not been reported in the literature. In this study, we conducted first-ever Raman spectroscopic measurements of  $\text{UCl}_3$  from 298 to 871 K. Five Raman-active modes of  $\text{UCl}_3$  have been revealed, and

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the variations of their frequencies as a function of temperature have been determined. The obtained results were validated by density functional theory (DFT) calculations. Furthermore, these results allowed us to determine the isobaric mode Grüneisen parameters ( $\gamma_{ip}$ ) of the five Raman modes of  $\text{UCl}_3$ .

## 2. EXPERIMENTAL METHODS

**2.1. Sample Preparation.** Because  $\text{UCl}_3$  is very hygroscopic and is subjected to oxygen pollution in air, all sample preparation was done in an inert argon atmosphere drybox, where the  $\text{O}_2$  and moisture levels were monitored and maintained at  $\leq 1$  ppm.  $\text{UCl}_3$  was synthesized from uranium metal by conversion to hydride ( $\text{UH}_3$ ) and subsequent chlorination to  $\text{UCl}_3$ .<sup>21</sup> The phase purity of  $\text{UCl}_3$  was verified by powder X-ray diffraction and melt point analysis via differential scanning calorimetry, the details of which are presented in our earlier works.<sup>17,18</sup> A  $\text{UCl}_3$  powder sample ( $<5$  mg) was placed in a silica ( $\text{SiO}_2$ ) capillary (Technical Glass Inc.), which was approximately 8 mm in length, with a 1 mm inner diameter, a 2 mm outer diameter, and had a wall thickness of 0.5 mm. The capillary containing  $\text{UCl}_3$  was transferred out of the argon atmosphere drybox under vacuum, and the other end of the capillary was then sealed with an Arizona Hydrogen Flame Generator MG-300  $\text{H}_2$ – $\text{O}_2$  torch.

**2.2. Raman Spectroscopy.** Raman spectroscopic measurements were performed using a Horiba LabRAM HR Evolution microscope equipped with an Olympus 5X objective in backscattering/reflection mode. The instrument was calibrated with a silicon wafer using the first-order Si line at  $520.7\text{ cm}^{-1}$ . The measurements were conducted by using a TOPTICA Photonics 785 nm diode laser to avoid the fluorescence signals, which would be present in the Raman spectra collected using a lower wavelength laser (e.g., 532 nm). The maximum output of the 785 nm laser is 300 mW. The laser outputs were filtered to 50% of the maximum power (150 mW). The system is equipped with a grating of 600 grooves/mm, which yields an effective resolution of  $<1\text{ cm}^{-1}$ . Raman spectra of the vibrational bands of  $\text{UCl}_3$  were collected from 70 to  $300\text{ cm}^{-1}$ . Each spectrum is the result of three accumulations per window with the time of each accumulation being between 30 and 120 s. The total time of counting for each spectrum was approximately 5–20 min, depending on the counting time. Additionally, to verify that no U–O vibrational modes were present at frequencies  $>300\text{ cm}^{-1}$ , a set of spectra were collected from 300 to  $1000\text{ cm}^{-1}$  at 298, 473, 672, and 871 K. The above Raman spectroscopic measurement procedures using the same instrument have been described previously.<sup>22–24</sup>

For *in situ* high-temperature Raman spectroscopic measurements, the capillary containing  $\text{UCl}_3$  was placed in a Linkam TS1500 heating stage, and the temperature was increased from room temperature to 871 K (Figure S1). The heating rate was 10 K/min, and a thermal equilibration period of at least 5 min was adopted prior to recording a Raman spectrum at each temperature. Above 871 K, the quality of the collected Raman spectra diminished, and no vibrational bands of  $\text{UCl}_3$  could be observed. This loss in signal was due to the emitted electromagnetic radiation of the ceramic heating crucible of the Linkam TS1500 stage, and thus, 871 K was selected as the upper temperature limit for our measurements.

The temperature of the heating stage was calibrated against the  $\alpha$  to  $\beta$  transition temperature of quartz ( $\text{SiO}_2$ ;  $T_{\alpha-\beta} = 847.5\text{ K}$ ). The same heating rate (10 K/min) and thermal equilibration period (5 min) as for  $\text{UCl}_3$  measurements were used. For these calibration measurements, a single crystal of high purity natural quartz (Hot Springs, Arkansas) was powdered with an agate mortar and pestle, and the powders were placed into the heating stage on a sapphire window. The  $\alpha$  to  $\beta$  quartz transition was deemed complete once the  $A_1$  mode at  $355\text{ cm}^{-1}$  disappeared.<sup>25</sup> This was done by thermally cycling the quartz sample above (871 K) and below (836 K) the transition temperature (847.5 K)<sup>26</sup> several times. The Raman spectroscopic measurements for the temperature calibration were done with the same Horiba system but used a 532 nm Nd:YAG laser. The maximum output of the 532 nm laser is 200 mW. The laser outputs were filtered to 50% of the maximum power (100 mW). Raman spectra of the

vibrational bands of quartz were collected from 70 to  $600\text{ cm}^{-1}$ . Each spectrum is the result of three accumulations per window with the time of each accumulation being between 10 and 30 s. The total time of counting for each spectrum was approximately 5 min.

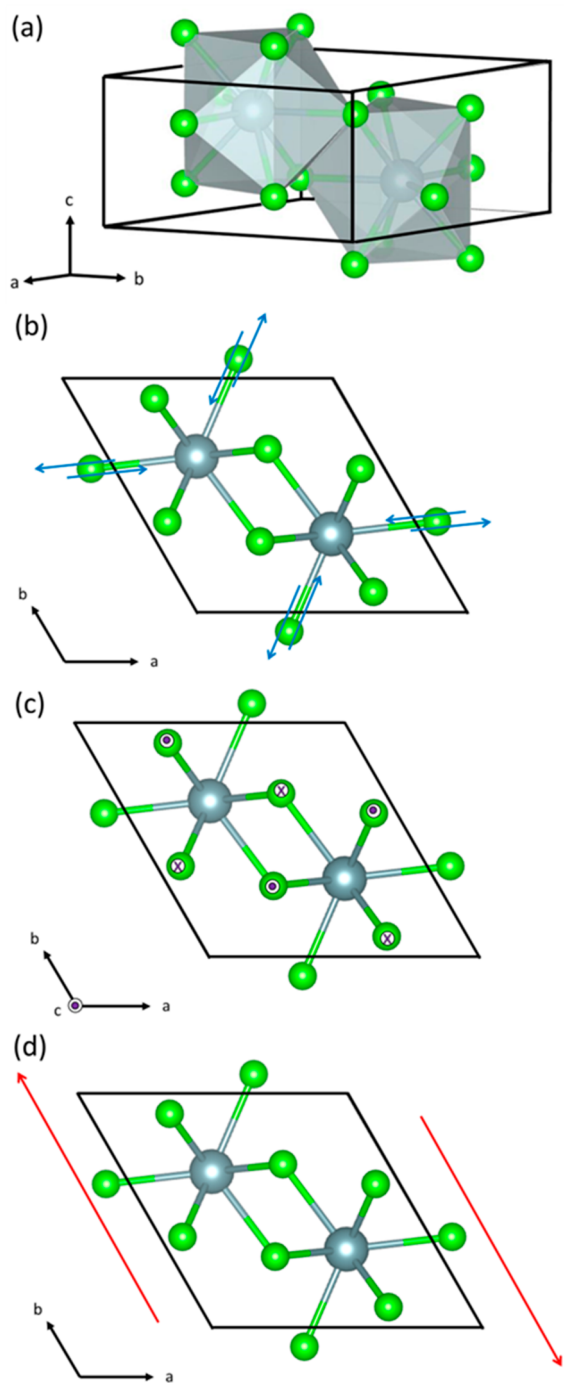
**2.3. Density Functional Theory Calculations.** Our calculations were performed with the use of density functional theory (DFT) and projector augmented-wave (PAW) method<sup>27</sup> as implemented in the Vienna Ab initio Simulation Package (VASP).<sup>28</sup> The Perdew–Burke–Ernzerhof (PBE)<sup>29</sup> functional of generalized gradient approximation (GGA) was used to represent the exchange–correlation interaction. The van der Waals (vdW) interaction was included in the calculations with the DFT-D3 method of Grimme.<sup>30</sup> A  $\Gamma$ -centered  $7 \times 7 \times 13$  k-point grid and plane-wave cutoff energy of 500 eV were employed for structural optimization.<sup>27</sup> The energy convergence was set to  $10^{-6}$  eV, and the residual force on each atom was less than 0.01 eV/Å. A Hubbard on-site correction of 4.0 eV was applied on the 5f orbitals of uranium atoms.<sup>31,32</sup> The predicted lattice constants of  $\text{UCl}_3$  are  $a = 7.40\text{ Å}$  and  $c = 4.31\text{ Å}$ .

The Phonopy code<sup>33</sup> was employed to compute the vibrational frequencies. The input force constants for Phonopy were derived using density functional perturbation theory (DFPT)<sup>34</sup> employing a  $1 \times 1 \times 2$  supercell. To assess the temperature-dependent behavior of the vibrational frequencies, the thermal expansion of  $\text{UCl}_3$  was analyzed across various temperatures using the quasi-harmonic approximation (QHA).<sup>18,35</sup> The free energy as a function of volume at each temperature and therefore the equilibrium volume as a function of temperature were obtained from the vibrational free energy and the equation of states at 0 K.

## 3. RESULTS

The crystal structure of  $\text{UCl}_3$  possesses a hexagonal symmetry with the  $P6_3/m$  space group and  $C_{6h}$  point group.<sup>20</sup> The unit cell contains two formula units.<sup>20,36</sup>  $\text{U}^{3+}$  is 9-fold coordinated to  $\text{Cl}^-$ , creating the  $[\text{UCl}_9]$  polyhedron with a tricapped trigonal prismatic geometry (Figure 1).<sup>37</sup> Factor group analysis of this space group predicts the occurrence of six Raman-active bands,  $\Gamma_{\text{Raman}} = 2A_g + E_g + 3E_{2g}$ .<sup>36</sup> While the  $A_g$  mode corresponds to the motion of  $\text{Cl}^-$  in the *ab*-plane of the structure (Figure 1a), the  $E_{1g}$  mode corresponds to the  $\text{Cl}^-$  motion out of the *ab*-plane along the *c*-axis (Figure 1b).<sup>36</sup> The  $E_{2g}$  mode can be classified as a translational mode of the  $\text{U}^{3+}$  and  $\text{Cl}^-$  ions that move together parallel to the *ab*-plane.<sup>36</sup>

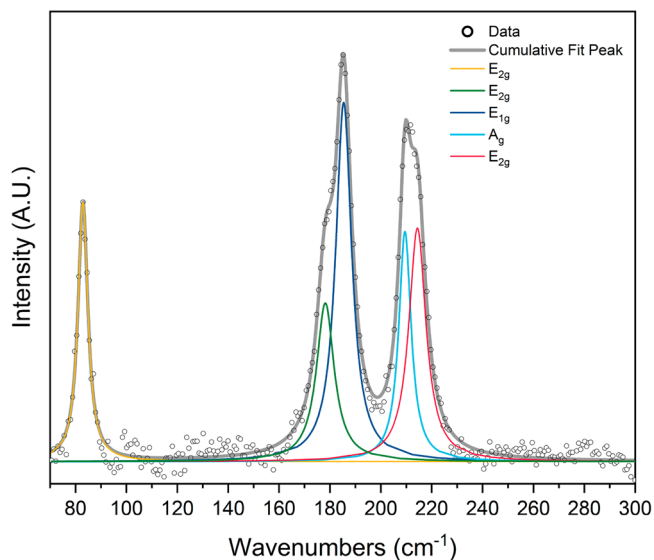
Band component analysis of the  $\text{UCl}_3$  Raman spectra was made by means of the Peakfit function within the OriginPro 2021 software suite. The background of each spectrum was determined using the second-derivative method with up to 20 anchor points that were connected by nonlinear interpolation of a B-spline function.<sup>38</sup> The peaks were fitted using a Lorentz function with the minimum number of components.<sup>38</sup> The raw uncorrected spectra are presented in the electronic supplement (Figure S3). A total of 12 temperature-dependent Raman spectra were deconvoluted. A representative deconvolution is shown in Figure 2. The adjusted  $R^2$  values from the fits range from 0.971 to 0.993, with a mean adjusted  $R^2$  of 0.986. The results from deconvolutions are listed in Table 1. At ambient conditions, the Raman spectrum of  $\text{UCl}_3$  has five bands at 83.6, 178.7, 186.1, 210.3, and  $215.5\text{ cm}^{-1}$  (Table 1). Based on our DFT calculations (see below), the Raman bands of  $\text{UCl}_3$  can be assigned as follows: the vibrational bands at 83.6 and  $215.5\text{ cm}^{-1}$  correspond to the  $E_{2g}$  mode, the bands at 178.7 and  $210.3\text{ cm}^{-1}$  belong to the  $A_g$  mode, and the  $186.1\text{ cm}^{-1}$  band corresponds to the  $E_{1g}$  mode. It is predicted that there exists a sixth Raman mode ( $\bar{E}_{2g}$ ), but its intensity is too low to be detected.<sup>39</sup> The observed Raman modes and assignments are in good accordance with those of the previously studied



**Figure 1.** Visualization of the Raman-active modes of  $\text{UCl}_3$ . (a) depicts how  $\text{U}^{3+}$  is 9-fold coordinated to  $\text{Cl}^-$ , creating the  $[\text{UCl}_9]$  polyhedron with a tricapped trigonal prismatic geometry. (b) depicts the  $A_g$  mode and how the  $\text{Cl}^-$  ion moves parallel to (blue arrow) the  $ab$ -plane. (c) depicts the  $E_g$  mode and how the  $\text{Cl}^-$  ion moves in (purple dot) and out (purple x) along the  $c$ -axis; and (d) depicts the  $E_{2g}$  mode and how the  $\text{U}^{3+}$  and  $\text{Cl}^-$  ions translationally move together (red arrow) parallel to the  $ab$ -plane.<sup>67</sup>

isostructural lanthanide ( $\text{LnCl}_3$ , where  $\text{Ln} = \text{La}, \text{Ce}, \text{Pr}, \text{Pm}, \text{Sm}, \text{Eu}, \text{Gd}$ ) and actinide ( $\text{AnCl}_3$ , where  $\text{An} = \text{Am}, \text{Cm}, \text{Cf}$ ) trichlorides.<sup>36,37,40–46</sup>

It was found that the capillary sealing methodology successfully prevented degradation of the  $\text{UCl}_3$  sample via oxidation and/or hydration (Figure S4). This was evident, as there are no U–O vibrational modes present in any of the



**Figure 2.** Representative deconvolution of a background subtracted Raman spectrum of  $\text{UCl}_3$  recorded at 298 K. The adjusted  $R^2$  of the deconvolution is 0.988.

spectra that were recorded between 300 and  $1000\text{ cm}^{-1}$ ; in uranium oxychloride ( $\text{UO}_2\text{Cl}_2$ ), there is a strong temperature-dependent vibrational band located between 700 and  $760\text{ cm}^{-1}$ .<sup>47</sup> Additionally, there are no vibrational features near  $450\text{ cm}^{-1}$ , which occurs in uranium dioxide  $[(\text{U}^{4+}\text{O}_2)^0]$ , and near  $775\text{ cm}^{-1}$ , which occurs in uranyl  $[(\text{U}^{6+}\text{O}_2)^{2+}]$  hydrates and oxyhydrates.<sup>48–50</sup>

Our DFT calculations indicated that there are six Raman-active modes as well as three infrared (IR)-active modes (Figure S2). At 0 K, the DFT predicted Raman spectrum of  $\text{UCl}_3$  would exhibit bands at 73.6, 158.4, 168.3, 174.7, 201.6, and  $204.5\text{ cm}^{-1}$  (Table 2). The band assignments are as follows: the vibrational bands at 73.6, 158.4, and  $204.5\text{ cm}^{-1}$  correspond to the  $E_{2g}$  mode, the bands at 168.3 and  $201.6\text{ cm}^{-1}$  belong to the  $A_g$  mode, and the  $174.7\text{ cm}^{-1}$  band corresponds to the  $E_{1g}$  mode. Note that the  $E_{2g}$  mode at  $158.4\text{ cm}^{-1}$  was not observed experimentally, probably because its intensities were below the detection limit. In addition, there are three infrared (IR)-active modes at 126.4, 132.6, and  $187.7\text{ cm}^{-1}$ . The IR band at  $126.4\text{ cm}^{-1}$  is assigned to the  $A_u$  mode, while those at 132.6 and  $187.7\text{ cm}^{-1}$  are assigned to the  $E_{1u}$  mode. These IR-active modes need experimental verification by far-IR spectroscopy.

## 4. DISCUSSION

### 4.1. Vibrational and Bonding Characteristics of $\text{UCl}_3$ .

Our experimental and computational results show that all of the Raman bands (Figures 3 and 4) exhibit a red-shift with increasing temperature. These thermally induced frequency shifts are likely a result of the lengthening of the U–Cl bonds in the structure of  $\text{UCl}_3$ . There was no emergence or disappearance of vibrational bands in the temperature range investigated, suggesting no apparent phase transitions. This observation of no apparent phase transitions is consistent with the previous high-temperature neutron diffraction and differential scanning calorimetry experiments.<sup>17,18</sup> The variations of vibrational frequencies for all measurable Raman modes with temperature are illustrated in Figure 4. All of the vibrational modes systematically exhibit a negative temperature depend-

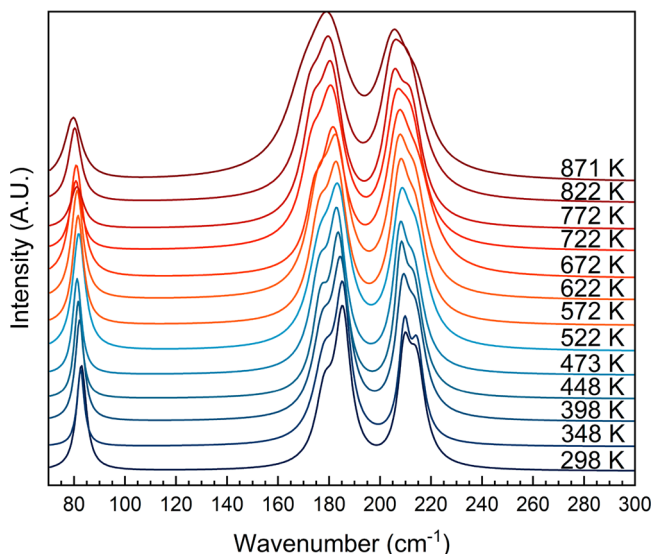


**Table 1.** Raman Band Positions of  $\text{UCl}_3$  from 298 to 871 K Obtained from Deconvolutions of *In Situ* High-Temperature Raman Spectra

| temp (K) | $E_{2g}$ ( $\text{cm}^{-1}$ ) | $A_g$ ( $\text{cm}^{-1}$ ) | $E_{1g}$ ( $\text{cm}^{-1}$ ) | $A_g$ ( $\text{cm}^{-1}$ ) | $E_{2g}$ ( $\text{cm}^{-1}$ ) | adj $R^2$ |
|----------|-------------------------------|----------------------------|-------------------------------|----------------------------|-------------------------------|-----------|
| 298      | $83.6 \pm 0.1$                | $178.7 \pm 0.2$            | $186.1 \pm 0.1$               | $210.3 \pm 0.1$            | $215.4 \pm 0.2$               | 0.990     |
| 348      | $83.3 \pm 0.1$                | $178.6 \pm 0.3$            | $185.5 \pm 0.1$               | $209.9 \pm 0.2$            | $215.4 \pm 0.3$               | 0.987     |
| 398      | $82.9 \pm 0.1$                | $178.2 \pm 0.3$            | $185.0 \pm 0.1$               | $209.4 \pm 0.2$            | $215.0 \pm 0.7$               | 0.989     |
| 448      | $82.4 \pm 0.1$                | $177.2 \pm 0.3$            | $184.4 \pm 0.2$               | $208.8 \pm 0.3$            | $214.4 \pm 0.7$               | 0.991     |
| 522      | $81.9 \pm 0.1$                | $176.3 \pm 0.4$            | $183.7 \pm 0.2$               | $208.1 \pm 0.2$            | $214.2 \pm 0.4$               | 0.993     |
| 572      | $81.6 \pm 0.1$                | $176.0 \pm 0.3$            | $183.2 \pm 0.2$               | $207.7 \pm 0.2$            | $214.0 \pm 0.5$               | 0.987     |
| 622      | $81.3 \pm 0.1$                | $175.6 \pm 0.4$            | $183.0 \pm 0.3$               | $207.2 \pm 0.3$            | $213.7 \pm 0.6$               | 0.986     |
| 672      | $80.9 \pm 0.1$                | $174.1 \pm 0.4$            | $181.7 \pm 0.3$               | $207.1 \pm 0.4$            | $214.6 \pm 0.9$               | 0.984     |
| 722      | $80.7 \pm 0.1$                | $173.1 \pm 0.5$            | $181.0 \pm 0.3$               | $206.0 \pm 0.4$            | $212.3 \pm 0.6$               | 0.986     |
| 772      | $81.1 \pm 0.1$                | $173.5 \pm 0.4$            | $180.9 \pm 0.3$               | $205.2 \pm 0.3$            | $211.5 \pm 0.4$               | 0.984     |
| 822      | $80.3 \pm 0.1$                | $172.2 \pm 0.5$            | $180.2 \pm 0.4$               | $204.8 \pm 0.5$            | $211.5 \pm 0.7$               | 0.985     |
| 871      | $79.7 \pm 0.1$                | $169.9 \pm 1.7$            | $179.8 \pm 0.8$               | $205.3 \pm 0.6$            | $214.3 \pm 1.0$               | 0.985     |

**Table 2.** Raman Band Positions of  $\text{UCl}_3$  from 0 to 800 K Predicted by DFT Calculations

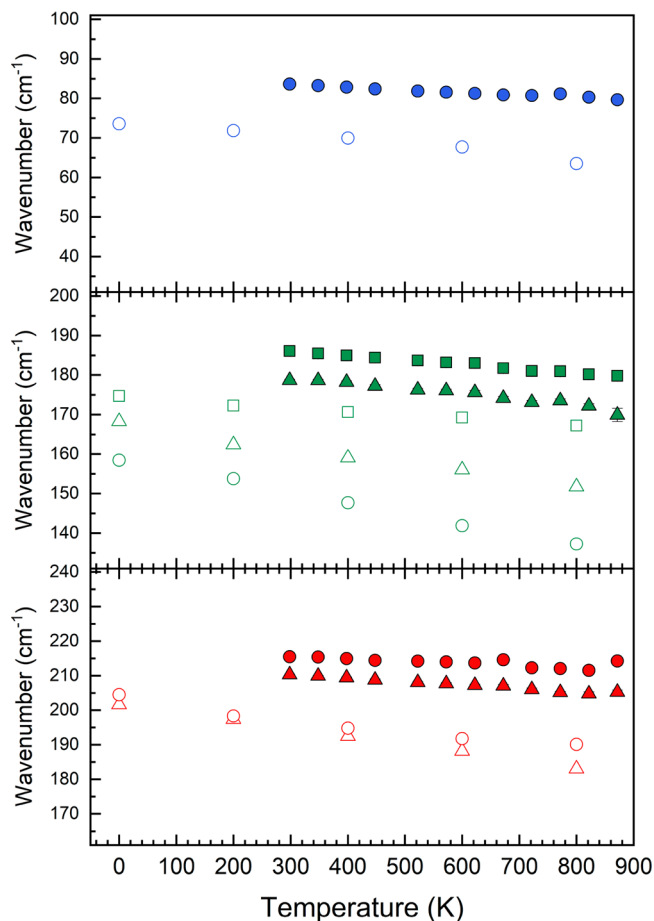
| temp (K) | $E_{2g}$ ( $\text{cm}^{-1}$ ) | $A_g$ ( $\text{cm}^{-1}$ ) | $E_{1g}$ ( $\text{cm}^{-1}$ ) | $A_g$ ( $\text{cm}^{-1}$ ) | $E_{2g}$ ( $\text{cm}^{-1}$ ) |
|----------|-------------------------------|----------------------------|-------------------------------|----------------------------|-------------------------------|
| 0        | 73.6                          | 158.4                      | 168.3                         | 174.7                      | 201.6                         |
| 200      | 71.8                          | 153.8                      | 162.4                         | 172.3                      | 197.3                         |
| 400      | 70.0                          | 147.7                      | 159.1                         | 170.6                      | 192.4                         |
| 600      | 67.7                          | 141.9                      | 156.0                         | 169.2                      | 188.2                         |
| 800      | 63.5                          | 137.2                      | 151.7                         | 167.2                      | 183.0                         |

**Figure 3.** Selected Raman spectra of  $\text{UCl}_3$  from 298 to 871 K. The presented spectra are the cumulative fits resulting from the deconvolutions at different temperatures.

ency by means of a linear regression,  $\nu_i = a_i + b_i T$ . The fitted temperature coefficients ( $b_i$ ) of the Raman modes are listed in Table 3.

The Grüneisen parameter ( $\gamma$ ) is a dimensionless quantity which is useful in quantifying the relationship between the thermal and elastic properties of a solid.<sup>51</sup> The temperature coefficients ( $b_i$ ) of the Raman modes (Table 3) can be utilized to calculate the isobaric mode Grüneisen parameter ( $\gamma_{ip}$ ), as it is equivalent to  $\left(\frac{\partial \nu_i}{\partial T}\right)_p$  through the following relationship:

$$\gamma_{ip} = \left(-\frac{1}{\alpha \nu_i}\right) \left(\frac{\partial \nu_i}{\partial T}\right)_p \quad (1)$$

**Figure 4.** Variation of the frequencies of Raman modes of  $\text{UCl}_3$  with temperature determined experimentally and computationally. Circles indicate the  $E_{2g}$  modes, squares indicate the  $E_{1g}$  mode, and triangles indicate the  $A_g$  modes. Shaded shapes are experimentally determined values, and hollow shapes are computationally determined values.

where  $\nu_i$  is the vibrational frequency of the  $i$ th vibrational band and  $\alpha$  is the isobaric thermal expansion coefficient, which is defined by  $\alpha = \left(\frac{1}{V}\right) \left(\frac{\partial V}{\partial T}\right)_p$ . The isobaric thermal expansion of  $\text{UCl}_3$  was determined by Vogel et al., using *in situ* high-temperature neutron diffraction, to be  $(6.32 \pm 0.16) \times 10^{-5} \text{ K}^{-1}$ <sup>18</sup> and was used in the derivation of the  $\gamma_{ip}$  (Table 3). The average value of  $\gamma_{ip}$  for  $\text{UCl}_3$  was found to be  $0.91 \pm 0.02$ .

**Table 3. Fitted Parameters  $a$  and  $b$  in  $\nu_i = a_i + b_i T$  from the Frequencies ( $\nu_i$ ) of High- $T$  Raman Bands of  $\text{UCl}_3$  and the Determined Isobaric Mode Grüneisen Parameters ( $\gamma_{ip}$ )**

| parameter                           | $E_{2g}$             | $A_g$                | $E_{1g}$             | $A_g$                | $E_{2g}$             |
|-------------------------------------|----------------------|----------------------|----------------------|----------------------|----------------------|
| $a_i$ ( $\text{cm}^{-1}$ )          | $83.70 \pm 0.07$     | $179.4 \pm 0.2$      | $186.37 \pm 0.06$    | $210.63 \pm 0.08$    | $215.7 \pm 0.2$      |
| $b_i$ ( $\text{cm}^{-1}/\text{K}$ ) | $-0.0068 \pm 0.0003$ | $-0.0124 \pm 0.0007$ | $-0.0111 \pm 0.0003$ | $-0.0101 \pm 0.0003$ | $-0.0064 \pm 0.0008$ |
| adj $R^2$                           | 0.982                | 0.964                | 0.993                | 0.988                | 0.858                |
| $\gamma_{ip}$                       | $1.29 \pm 0.06$      | $1.10 \pm 0.07$      | $0.94 \pm 0.03$      | $0.75 \pm 0.03$      | $0.46 \pm 0.06$      |

**Table 4. Force Constants of the Raman Modes of  $\text{UCl}_3$** 

| temp (K) | $E_{2g}$ (N/m)   | $E_{2g}$ (N/m)   | $E_{1g}$ (N/m)   | $A_g$ (N/m)      | $E_{2g}$ (N/m)   |
|----------|------------------|------------------|------------------|------------------|------------------|
| 298      | $12.72 \pm 0.01$ | $58.04 \pm 0.08$ | $62.97 \pm 0.03$ | $80.42 \pm 0.05$ | $84.42 \pm 0.09$ |
| 348      | $12.61 \pm 0.01$ | $58.02 \pm 0.08$ | $62.55 \pm 0.04$ | $80.12 \pm 0.06$ | $84.33 \pm 0.12$ |
| 398      | $12.49 \pm 0.01$ | $57.74 \pm 0.10$ | $62.22 \pm 0.05$ | $79.74 \pm 0.06$ | $84.01 \pm 0.12$ |
| 448      | $12.35 \pm 0.01$ | $57.10 \pm 0.11$ | $61.83 \pm 0.06$ | $79.25 \pm 0.11$ | $83.60 \pm 0.28$ |
| 522      | $12.19 \pm 0.01$ | $56.48 \pm 0.11$ | $61.34 \pm 0.06$ | $78.71 \pm 0.08$ | $83.43 \pm 0.16$ |
| 572      | $12.10 \pm 0.01$ | $56.35 \pm 0.11$ | $61.00 \pm 0.07$ | $78.46 \pm 0.09$ | $83.26 \pm 0.18$ |
| 622      | $12.01 \pm 0.01$ | $56.07 \pm 0.14$ | $60.90 \pm 0.10$ | $78.09 \pm 0.12$ | $83.03 \pm 0.21$ |
| 672      | $11.90 \pm 0.01$ | $55.14 \pm 0.13$ | $60.03 \pm 0.09$ | $77.97 \pm 0.14$ | $83.72 \pm 0.34$ |
| 722      | $11.85 \pm 0.02$ | $54.48 \pm 0.15$ | $59.57 \pm 0.10$ | $77.15 \pm 0.16$ | $81.94 \pm 0.25$ |
| 772      | $11.97 \pm 0.04$ | $54.74 \pm 0.14$ | $59.52 \pm 0.10$ | $76.56 \pm 0.11$ | $81.77 \pm 0.17$ |
| 822      | $11.72 \pm 0.02$ | $53.91 \pm 0.17$ | $59.01 \pm 0.13$ | $76.25 \pm 0.19$ | $81.36 \pm 0.28$ |
| 871      | $11.54 \pm 0.03$ | $52.48 \pm 0.51$ | $58.77 \pm 0.27$ | $76.61 \pm 0.22$ | $83.46 \pm 0.38$ |

The bond force constant ( $k$ ) represents the interaction between nearest atoms and is determined by both the length and angle of the bond. This constant is useful in the analysis of the lattice dynamics. A simplified way of approximating the bond force constant ( $k$ ) is through use of the Hooke's law:<sup>52</sup>

$$k = (2\pi c \nu_i)^2 \mu \quad (2)$$

where  $\nu_i$  is the vibrational frequency of the  $i$ th vibrational band,  $c$  is the velocity of light in a vacuum ( $2.998 \times 10^{10}$  cm/s), and  $\mu$  is the reduced mass of the vibrating atoms (e.g., U and Cl). Using eq 2, we calculated the  $k$  values of  $\text{UCl}_3$  as a function of temperature (Table 4). For all of the vibrational bands, the force constants were found to decrease with increasing temperature (Table 4). This behavior is probably due to the elongation of U–Cl bonds in the structure. This elongation results in decreases in the electronegative forces between the  $\text{U}^{3+}$  cation and  $\text{Cl}^-$  anion, thereby weakening the U–Cl bonds.

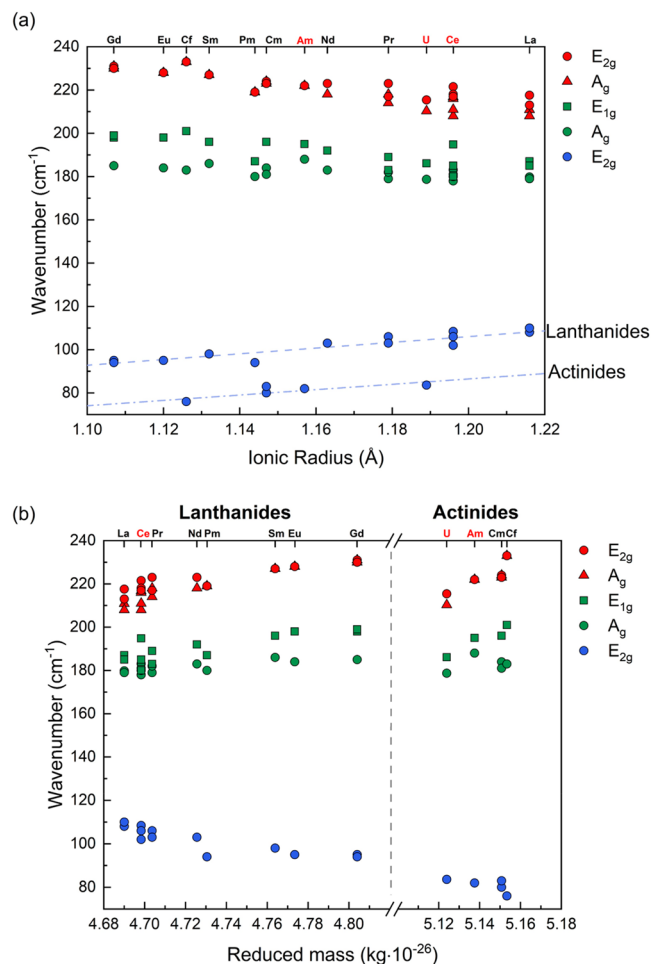
**4.2. Systematics in the Vibrational Properties of Isostructural Trichlorides.** As stated in the Introduction,  $\text{UCl}_3$  is the prototypical structure of nearly 100 compounds reported in the ICSD that include eight  $\text{LnCl}_3$  ( $\text{Ln} = \text{La}–\text{Gd}$ ) and eight  $\text{AnCl}_3$  ( $\text{An} = \text{UCl}_3–\text{EsCl}_3$ ) phases.<sup>18</sup> However, despite the importance of this structure, its Raman spectrum has not been reported until this study. Furthermore, it is interesting to compare the vibrational properties of  $\text{UCl}_3$  with those of other isostructural lanthanide ( $\text{LnCl}_3$ , where  $\text{Ln} = \text{La}, \text{Ce}, \text{Pr}, \text{Sm}, \text{Eu}, \text{Gd}$ ) and actinide ( $\text{AnCl}_3$ , where  $\text{An} = \text{Am}, \text{Cm}, \text{Cf}$ ) trichlorides.<sup>36,37,40–46</sup>

Many physical and chemical properties of isostructural inorganic materials can be evaluated through various empirically derived relationships.<sup>53–62</sup> These relationships link a specific property variable to a structural parameter of the isostructural phases and allow the study of the systematics of the said variable. One of these relationships would be looking at how the frequency of the Raman band changes across isostructural  $\text{LnCl}_3$  and  $\text{AnCl}_3$ . As Raman spectroscopy is a type of vibrational spectroscopy, the positions of the vibrational bands are sensitive to changes in the reduced mass

( $\mu$ ) and bond length.<sup>52,63,64</sup> Plotting the published vibrational band positions<sup>36,37,40–46</sup> of the  $\text{LnCl}_3$  and  $\text{AnCl}_3$  compounds against either the ionic radii (Figure 5a) or  $\mu$  (Figure 5b) of the trivalent f-metal center in a 9-fold coordination environment allows for determination of any systematic trend.<sup>65,66</sup>

The vibrational frequencies of Raman bands for all of the isostructural f-metal trichlorides, apart from the low-frequency  $E_{2g}$  mode, show near-linear dependencies with ionic radii (Figure 5a), where the smaller cations (i.e.,  $\text{Gd}^{3+}$ ,  $\text{Eu}^{3+}$ , and  $\text{Ce}^{3+}$ ) have higher wavenumbers than larger cations (i.e.,  $\text{La}^{3+}$ ,  $\text{Ce}^{3+}$ , and  $\text{U}^{3+}$ ). This suggests that the vibrational modes of these f-metal trichlorides are strongly influenced by lengths of the f-metal–chlorides bonds. One exception is that the low-frequency  $E_{2g}$  modes of  $\text{LnCl}_3$  and  $\text{AnCl}_3$  display two independent trends with respect to ionic radii (Figure 5a). Instead, the low-frequency  $E_{2g}$  mode shows a stronger dependency with  $\mu$  (Figure 5b), illustrating the strong influence that mass has on vibrational properties.

In terms of similarity in Raman spectra to the other isostructural metal trichlorides,  $\text{UCl}_3$  is most like  $\text{CeCl}_3$  and  $\text{AmCl}_3$ . The ionic radii of  $\text{U}^{3+}$  (1.189 Å)<sup>66</sup> and  $\text{Ce}^{3+}$  (1.196 Å)<sup>65</sup> are very similar (<0.60% difference). As a result, the frequencies for four of the five Raman-active modes of  $\text{UCl}_3$  and  $\text{CeCl}_3$  differ only by a maximum of 1.3%. On the other hand, the positions for the low-frequency  $E_{2g}$  mode of  $\text{UCl}_3$  and  $\text{CeCl}_3$  are dissimilar, as this mode is more affected by the mass of the metal center than the ionic radii (Figure 5b). Here  $\text{UCl}_3$  is most like  $\text{AmCl}_3$ , as the two metal centers are closer in atomic mass to one another than U is to Ce. One would expect that overall the Raman spectrum of  $\text{UCl}_3$  to be most similar to those of  $\text{NpCl}_3$  and  $\text{PuCl}_3$  because both the atomic masses of the metal centers and the ionic radii in a 9-fold coordination ( $\text{Np}^{3+} = 1.178$  Å and  $\text{Pu}^{3+} = 1.168$  Å)<sup>66</sup> are closer to one another. However, the Raman spectra of both  $\text{NpCl}_3$  and  $\text{PuCl}_3$  have not yet been reported in the literature, and a comparison cannot be made at this time.



**Figure 5.** Variation of the frequencies of Raman bands of  $\text{LnCl}_3$  and  $\text{AnCl}_3$  as a function of (a) ionic radius and (b) reduced mass of trivalent cation possessing 9-fold coordination.<sup>7,8,10–16,34,35</sup>

## 5. CONCLUSIONS

Raman spectroscopic measurements of crystalline  $\text{UCl}_3$  have been performed from 298 to 871 K, and the obtained results are in general agreement with those predicted by our DFT calculations. This work represents the first Raman spectroscopic study of  $\text{UCl}_3$  at both ambient and high temperatures. The band positions of  $\text{UCl}_3$  are comparable to those of  $\text{CeCl}_3$ , which possesses a similar ionic radius, and  $\text{AmCl}_3$ , which possesses a similar molecular mass. The frequencies ( $\nu_i$ ) of all five Raman-active modes were determined by deconvoluting the spectra using a Lorentz function. With increasing temperature, all five of the Raman-active vibrational modes exhibit a shift toward lower wavenumbers. This behavior is likely due to the lengthening of U–Cl bonds, which in turn causes a negative temperature derivative in vibrational frequency. The isobaric mode Grüneisen parameters of all five Raman modes of  $\text{UCl}_3$  were determined using the thermal expansion coefficient ( $\alpha$ ) of Vogel et al. (2021) and the temperature derivatives of  $\nu_i$  determined in this study. The average isobaric mode Grüneisen parameter was found to be  $0.91 \pm 0.02$ . These results shed light on the U–Cl bonding behavior of  $\text{UCl}_3$  at high temperatures and may potentially be used for modeling the speciation behavior of  $\text{UCl}_3$  in molten salt solvents.

## ■ ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.inorgchem.3c03139>.

Raman experimental setup, uncorrected Raman spectra, and calculated phonon dispersion of  $\text{UCl}_3$  (PDF)

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### Author Contributions

H.B., H.X., M.J.M., S.M.H., and A.C.S. designed the project. S.M.H., M.J.M., S.S.P., R.B., H.B., and A.C.S. oversaw sample preparation and characterization. *In situ* high-temperature Raman spectroscopic measurements were performed by A.C.S., with H.B. supporting the experiments. Spectral deconvolutions were performed by A.C.S. with H.B. and H.X. supporting the analyses. DFT calculations were performed by G.W. with



D.A.A. supporting the analyses. This manuscript was written through contributions of all authors. All authors have given approval to the final version of this manuscript.

## Notes

The authors declare no competing financial interest.

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## REFERENCES

- (1) Bruno, J.; Ewing, R. C. Spent Nuclear Fuel. *Elements* **2006**, *2* (6), 343–349.
- (2) Weber, W. J.; Ewing, R. C.; Vance, E. R.; Gregg, D.; Peugeot, S.; Wiss, T. Plutonium in Waste Forms. In *Plutonium Handbook*; Clark, D. L., Geeson, D. A., Hanrahan, Jr., R. J., Eds.; American Nuclear Society: 2019; pp 2349–2422.
- (3) Weber, W. J.; Ewing, R. C. Chapter 10: Ceramic Waste Forms for Uranium and Transuranium Elements. In *Uranium: Cradle to Grave*; Burns, P. C., Sigmon, G. E., Eds.; Mineralogical Association of Canada: 2013; pp 317–336.
- (4) Strzelecki, A. C.; Bourgeois, C.; Kriegsmann, K. W.; Estevenon, P.; Wei, N.; Szenknect, S.; Mesbah, A.; Wu, D.; Ewing, R. C.; Dacheux, N.; Guo, X. Thermodynamics of  $\text{CeSiO}_4$ : Implications for Actinide Orthosilicates. *Inorg. Chem.* **2020**, *59* (18), 13174–13183.
- (5) Goel, A.; McCloy, J. S.; Pokorny, R.; Kruger, A. A. Challenges with Vitrification of Hanford High-Level Waste (HLW) to Borosilicate Glass - An Overview. *J. Non-Crystalline Solids X* **2019**, *4*, 1–19.
- (6) Wegel, S.; Czempinski, V.; Oei, P.-Y.; Wealer, B. Transporting and Storing High-Level Nuclear Waste in the U.S.-Insights from a Mathematical Model. *Appl. Sci.* **2019**, *9* (12), 2437.
- (7) Weber, W. J.; Navrotsky, A.; Stefanovsky, S.; Vance, E. R.; Vernaz, E. Materials Science of High-Level Immobilization. *MRS Bull.* **2009**, *34*, 46–53.
- (8) McIntyre, P.; Assadi, S.; Badgley, K.; Baker, W.; Comeaux, J.; Gerity, J.; Kellams, J.; McInturff, A.; Pogue, N.; Phongikaroon, S.; Sattarov, A.; Simpson, M.; Sooby, E.; Tsvetkov, P. Accelerator-Driven Subcritical Fission in Molten Salt Core: Closing the Nuclear Fuel Cycle for Green Nuclear Energy. *AIP Conf. Proc.* **2013**, *1525* (April 2013), 636–642.
- (9) Rosenthal, M. W.; Kasten, P. R.; Briggs, R. B. Molten-Salt Reactors. History, Status, and Potential. *Nucl. Appl. Technol.* **1970**, *8* (2), 107–117.
- (10) Serp, J.; Allibert, M.; Beneš, O.; Delpech, S.; Feynberg, O.; Ghetta, V.; Heuer, D.; Holcomb, D.; Ignatiev, V.; Kloosterman, J. L.; Luzzi, L.; Merle-Lucotte, E.; Uhlíř, J.; Yoshioka, R.; Zhimin, D. The Molten Salt Reactor (MSR) in Generation IV: Overview and Perspectives. *Prog. Nucl. Energy* **2014**, *77*, 308–319.
- (11) LeBlanc, D. Molten Salt Reactors: A New Beginning for an Old Idea. *Nucl. Eng. Des.* **2010**, *240* (6), 1644–1656.
- (12) Taube, M. Fast Reactor Using Molten Chloride Salts as Fuel, Wuerenlingen, 1978.
- (13) Holcomb, D. E.; Flanagan, G. F.; Patton, B. W.; Gehin, J. C.; Howard, R. L.; Harrison, T. J. Fast Spectrum Molten Salt Reactor Options, 2011.
- (14) Lhermitte, C. R.; Parker, S. S.; Jackson, J. M.; Monreal, M. J. Communication— $\text{Mg}^{2+/0}$  as a Reliable Reference Electrode for Molten Chloride Salts. *J. Electrochem. Soc.* **2021**, *168* (6), 066501.
- (15) Long, A. M.; Parker, S. S.; Carver, D. T.; Jackson, J. M.; Monreal, M. J.; Newmark, D. A.; Vogel, S. C. Remote Density Measurements of Molten Salts via Neutron Radiography. *J. Imaging* **2021**, *7* (5), 88.
- (16) Sooby, E. S.; Nelson, A. T.; White, J. T.; McIntyre, P. M. Measurements of the Liquidus Surface and Solidus Transitions of the  $\text{NaCl-UCl}_3$  and  $\text{NaCl-UCl}_3\text{-CeCl}_3$  Phase Diagrams. *J. Nucl. Mater.* **2015**, *466*, 280–285.
- (17) Parker, S. S.; Long, A.; Lhermitte, C.; Vogel, S.; Monreal, M.; Jackson, J. M. Thermophysical Properties of Liquid Chlorides from 600 to 1600 K: Melt Point, Enthalpy of Fusion, and Volumetric Expansion. *J. Mol. Liq.* **2022**, *346*, 118147.
- (18) Vogel, S. C.; Andersson, D. A.; Monreal, M. J.; Jackson, J. M.; Parker, S. S.; Wang, G.; Yang, P.; Zhang, J. Crystal Structure Evolution of  $\text{UCl}_3$  from Room Temperature to Melting. *JOM* **2021**, *73* (11), 3555–3563.
- (19) Andersson, D. A.; Beeler, B. W. Ab Initio Molecular Dynamics (AIMD) Simulations of  $\text{NaCl}$ ,  $\text{UCl}_3$  and  $\text{NaCl-UCl}_3$  Molten Salts. *J. Nucl. Mater.* **2022**, *568*, 153836.
- (20) Zachariasen, W. H. Crystal Chemical Studies of the 5f-Series of Elements. I. New Structure Types. *Acta Crystallogr.* **1948**, *1* (5), 265–268.
- (21) Katz, J. J.; Rabinowitch, E. *The Chemistry of Uranium, Part I: The Element, Its Binary and Related Compounds*; McGraw-Hill: New York, 1951.
- (22) Strzelecki, A. C.; Reece, M. E.; Zhao, X.; Yu, W.; Benmore, C. J.; Ren, Y.; Alcorn, C. D.; Migdisov, A.; Xu, H.; Guo, X. Thermodynamics of Mixing HREE in Xenotime Solid Solution ( $\text{Er}_{(x)}\text{Yb}_{(1-x)}\text{PO}_4$ ). *ACS Earth Sp. Chem.* **2022**, *6*, 1375.
- (23) Strzelecki, A. C.; Zhao, X.; Baker, J.; Estevenon, P.; Barral, T.; Mesbah, A.; Popov, D.; Chariton, S.; Prakapenka, V.; Ahmed, S.; Yoo, C.; Dacheux, N.; Xu, H.; Guo, X. High-Pressure Structural and Thermodynamic Properties of Cerium Orthosilicates. *J. Phys. Chem. C* **2023**, *127*, 4225.
- (24) Kalintsev, A.; Migdisov, A.; Alcorn, C.; Baker, J.; Brugger, J.; Mayanovic, R. A.; Akram, N.; Guo, X.; Xu, H.; Boukhalfa, H.; Caporuscio, F. A.; Viswanathan, H.; Jove-Colon, C.; Wang, Y.; Matteo, E.; Roback, R. Uranium Carbonate Complexes Demonstrate Drastic Decrease in Stability at Elevated Temperatures. *Commun. Chem.* **2021**, *4* (1), 1–8.
- (25) Shapiro, S. M.; O'Shea, D. C.; Cummins, H. Z. Raman Scattering Study of the Alpha-Beta Phase Transition in Quartz. *Phys. Rev. Lett.* **1967**, *19* (7), 361–364.
- (26) Heaney, P. J. Structure and Chemistry of the Low-Pressure Silica Polymorphs. In *Silica: Physical Behavior, Geochemistry, and Materials Applications*; Heaney, P. J., Prewitt, C. T., Gibbs, G. V., Eds.; De Gruyter: 1994; Vol. 29, pp 1–40.
- (27) Kresse, G.; Joubert, D. From Ultrasoft Pseudopotentials to the Projector Augmented-Wave Method. *Phys. Rev. B - Condens. Matter Mater. Phys.* **1999**, *59* (3), 1758–1775.
- (28) Kresse, G.; Furthmüller, J. Efficient Iterative Schemes for Ab Initio Total-Energy Calculations Using a Plane-Wave Basis Set. *Phys. Rev. B* **1996**, *124* (20), 4053–4061.
- (29) Perdew, J. P.; Burke, K.; Wang, Y. Generalized Gradient Approximation for the Exchange-Correlation Hole of a Many-Electron System. *Phys. Rev. B - Condens. Matter Mater. Phys.* **1996**, *54* (23), 16533–16539.



- (30) Grimme, S. Semiempirical GGA-Type Density Functional Constructed with a Long-Range Dispersion Correction. *J. Comput. Chem.* **2006**, *27*, 1787–1799.
- (31) Anisimov, V. I.; Gunnarsson, O. Density-Functional Calculation of Effective Coulomb Interactions in Metals. *Phys. Rev. B* **1991**, *43* (10), 7570–7574.
- (32) Anisimov, V. I.; Zaanen, J.; Andersen, O. K. Band Theory and Mott Insulators: Hubbard U Instead of Stoner I. *Phys. Rev. B* **1991**, *44* (3), 943–954.
- (33) Togo, A.; Tanaka, I. First Principles Phonon Calculations in Materials Science. *Scr. Mater.* **2015**, *108*, 1–5.
- (34) Baroni, S.; de Gironcoli, S.; Dal Corso, A.; Giannozzi, P. Phonons and Related Crystal Properties from Density-Functional Perturbation Theory. *Rev. Mod. Phys.* **2001**, *73* (2), 515–562.
- (35) Togo, A.; Chaput, L.; Tanaka, I.; Hug, G. First-Principles Phonon Calculations of Thermal Expansion in  $\text{Ti}_3\text{SiC}_2$ ,  $\text{Ti}_3\text{AlC}_2$ , and  $\text{Ti}_3\text{GeC}_2$ . *Phys. Rev. B - Condens. Matter Mater. Phys.* **2010**, *81* (17), 1–6.
- (36) Schaack, G.; Koningstein, J. A. Phonon Frequencies of Rare Earth Trichlorides with Unit Cells of Different Dimensions. *J. Phys. Chem. Solids* **1970**, *31* (11), 2417–2420.
- (37) Zakir'yanova, I. D.; Salyulev, A. B. Raman Spectra of Crystalline and Molten  $\text{GdCl}_3$ . *Russ. Metall.* **2010**, *2010* (2), 112–115.
- (38) Hawthorne, F. C.; Waychunas, G. A. Spectrum-Fitting Methods. In *Spectroscopic Methods in Mineralogy and Geology*; Hawthorne, F. C., Ed.; Mineralogical Society of America: 1988; pp 63–98.
- (39) Daniel, J. F.; Wilmarth, W. R.; Begun, G. M.; Peterson, J. R. Raman Spectroscopic Studies of Gadolinium Trichloride as a Function of Temperature. *J. Crystallogr. Spectrosc. Res.* **1989**, *19* (1), 39–49.
- (40) Richman, I.; Satten, R. A.; Wong, E. Y. Lattice Vibrations of  $\text{LaCl}_3$  and  $\text{LaBr}_3$  from Vibronic Spectra. *J. Chem. Phys.* **1964**, *40* (5), 1451–1452.
- (41) Damen, T. C.; Kiel, A.; Porto, S. P. S.; Singh, S. The Raman Effects of  $\text{CeCl}_3$  and  $\text{PrCl}_3$ . *Solid State Commun.* **1968**, *6*, 671–673.
- (42) Wilmarth, W. R.; Begun, G. M.; Haire, R. G.; Young, J. P.; Peterson, J. R. The Effect of Pressure on the Raman and Absorption Spectra of  $\text{PrCl}_3$ ,  $\text{PrBr}_3$ , and  $\text{NdBr}_3$ . *Appl. Spectrosc.* **1989**, *43* (6), 1038–1045.
- (43) Wilmarth, W. R.; Begun, G. M.; Haire, R. G.; Peterson, J. R. Raman Spectra of  $\text{Pm}_2\text{O}_3$ ,  $\text{PmF}_3$ ,  $\text{PmCl}_3$ ,  $\text{PmBr}_3$ , and  $\text{PmI}_3$ . *J. Raman Spectrosc.* **1988**, *19*, 271–275.
- (44) Wilmarth, W. R.; Begun, G. M.; Haire, R. G.; Peterson, J. R. Raman Spectra of Selected Transuranium Trihalides in the Solid State. *J. Chem. Phys.* **1988**, *89* (8), 4666–4670.
- (45) Del Cul, G. D.; Haire, R. G.; Peterson, J. R. Effects of Pressure on the Luminescence, Raman and Absorption Spectra of  $^{248}\text{CmCl}_3$ . *J. Alloys Compd.* **1992**, *181* (1–2), 63–70.
- (46) Zakir'yanova, I. D.; Salyulev, A. B.; Khokhlov, V. A. Raman Spectroscopy Study of the Phase Transitions in Rare-Earth Metal Trichlorides. *Russ. Metall.* **2011**, *2011* (8), 754–759.
- (47) Tuffy, B. W.; Birkner, N. R.; Schorne-pinto, J.; Davis, R. C.; Mofrad, A. M.; Dixon, C. M.; Azizih, M.; Christian, M. S.; Lynch, T. J.; Bartlett, M. T.; Besmann, T. M.; Brinkman, K. S.; Chiu, W. K. S. Identification and Decomposition of Uranium Oxychloride Phases in Oxygen-Exposed  $\text{UCl}_3$  Salt Compositions. *J. Phys. Chem. B* **2023**, *127*, 6091.
- (48) Spano, T. L.; Olds, T. A.; McDonnell, M.; Smith, R.; Niedela, J. L.; Miskowicz, A.; Kapsimalis, R.; Shields, A. A. CURIUS: Compendium of Uranium Raman and Infrared Experimental Spectra. *Am. Mineral.* **2023**, DOI: 10.2138/am-2022-8738, in press.
- (49) Karcher, S.; Mohun, R.; Olds, T. A.; Weber, M.; Kriegsman, K.; Zhao, X.; Guo, X.; Corkhill, C.; Field, D.; McCloy, J. Benefits of Using Multiple Raman Laser Wavelengths for Characterizing Defects in a  $\text{UO}_2$  Matrix. *J. Raman Spectrosc.* **2022**, *53* (5), 988–1002.
- (50) Olds, T. A.; Karcher, S. E.; Kriegsman, K. W.; Guo, X.; McCloy, J. S. Oxidation and Anion Lattice Defect Signatures of Hypostoichiometric Lanthanide-Doped  $\text{UO}_2$ . *J. Nucl. Mater.* **2020**, *530*, 151959.
- (51) Anderson, O. L. *Equations of State of Solids for Geophysics Ceramic Science*; Charnock, H., Dewey, J., Morris, S. C., Navrotsky, A., Oxburgh, E. R., Price, R. A., Skinner, B. J., Eds.; Oxford University Press: 1995; Vol. 31.
- (52) Carter, R. L. Vibrational Spectroscopy. In *Molecular Symmetry and Group Theory*; Wiley: 1997; pp 164–200.
- (53) Navrotsky, A. Systematic Trends and Prediction of Enthalpies of Formation of Refractory Lanthanide and Actinide Ternary Oxide Phases. *Ceram. Trans.* **2001**, No. 119, 137–146.
- (54) Helean, K. B.; Navrotsky, A.; Lumpkin, G. R.; Colella, M.; Lian, J.; Ewing, R. C.; Ebbinghaus, B.; Catalano, J. G. Enthalpies of Formation of U-, Th-, Ce-Brannerite: Implications for Plutonium Immobilization. *J. Nucl. Mater.* **2003**, *320* (3), 231–244.
- (55) Navrotsky, A. Thermochemical Insights into Refractory Ceramic Materials Based on Oxides with Large Tetravalent Cations. *J. Mater. Chem.* **2005**, *15* (19), 1883–1890.
- (56) Qi, J.; Guo, X.; Mielewczyk-Gryn, A.; Navrotsky, A. Formation Enthalpies of  $\text{LaLnO}_3$  (Ln = Ho, Er, Tm and Yb) Interlanthanide Perovskites. *J. Solid State Chem.* **2015**, *227*, 150–154.
- (57) Helean, K. B.; Ushakov, S. V.; Brown, C. E.; Navrotsky, A.; Lian, J.; Ewing, R. C.; Farmer, J. M.; Boatner, L. A. Formation Enthalpies of Rare Earth Titanate Pyrochlores. *J. Solid State Chem.* **2004**, *177* (6), 1858–1866.
- (58) Guo, X.; Tiferet, E.; Qi, L.; Solomon, J. M.; Lanzirrotti, A.; Newville, M.; Engelhard, M. H.; Kukkadapu, R. K.; Wu, D.; Ilton, E. S.; Asta, M.; Sutton, S. R.; Xu, H.; Navrotsky, A. U(v) in Metal Uranates: A Combined Experimental and Theoretical Study of  $\text{MgUO}_4$ ,  $\text{CrUO}_4$ , and  $\text{FeUO}_4$ . *Dalt. Trans.* **2016**, *45* (11), 4622–4632.
- (59) Chen, F.; Ewing, R. C.; Clark, S. B. The Gibbs Free Energies and Enthalpies of Formation of  $\text{U}^{6+}$  Phases: An Empirical Method of Prediction. *Am. Mineral.* **1999**, *84* (4), 650–664.
- (60) Sverjensky, D.; Shock, E. L.; Helgeson, H. C. Prediction of the Thermodynamic Properties of Aqueous Metal Complexes to 1000 C and 5 Kb. *Geochim. Cosmochim. Acta* **1997**, *61* (7), 1359–1412.
- (61) Anderson, D. L.; Anderson, O. L. The Bulk Modulus-Volume Relationship for Oxides. *J. Geophys. Res.* **1970**, *75* (17), 3494–3500.
- (62) Xu, H.; Zhao, Y.; Zhang, J.; Wang, Y.; Hickmott, D. D.; Daemen, L. L.; Hartl, M. A.; Wang, L. Anisotropic Elasticity of Jarosite: A High-P Synchrotron XRD Study. *Am. Mineral.* **2010**, *95* (1), 19–23.
- (63) Harris, D. C.; Bertolucci, M. D. *Symmetry and Spectroscopy: An Introduction to Vibrational and Electronical Spectroscopy*; Oxford University Press: 1978.
- (64) Bishop, D. M. Molecular Vibrations. In *Group Theory and Chemistry*; Clarendon Press: 1973; pp 164–195.
- (65) Shannon, R. D. Revised Effective Ionic Radii and Systematic Studies of Interatomic Distances in Halides and Chalcogenides. *Acta Crystallogr., Sect. A* **1976**, *32* (5), 751–767.
- (66) Lundberg, D.; Persson, I. The Size of Actinoid (III) Ions - Structural Analysis vs. Common Misinterpretations. *Coord. Chem. Rev.* **2016**, *318*, 131–134.
- (67) Momma, K.; Izumi, F. VESTA 3 for Three-Dimensional Visualization of Crystal, Volumetric and Morphology Data. *J. Appl. Crystallogr.* **2011**, *44* (6), 1272–1276.