

# Straintronic Effect on Phonon-Mediated Superconductivity of Nb<sub>2</sub>CT<sub>2</sub> (T = O, S, Se, or Te) MXenes

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ABSTRACT: The electronic structures, phonon dispersions, and electron-phonon coupling of Nb<sub>2</sub>CT<sub>2</sub> (T = O, S, Se, or Te) MXenes were investigated via first-principles calculations. Different models of Nb<sub>2</sub>CT<sub>2</sub> were constructed and the results show the low-energy models of Nb<sub>2</sub>CT<sub>2</sub> are intrinsic phonon-mediated superconductors. Of the four Nb<sub>2</sub>CT<sub>2</sub> MXenes, Nb<sub>2</sub>CO<sub>2</sub> MXene exhibits the largest superconducting critical temperature ( $T_c$ ) of 14.43 K. The existence of soft modes induced by Kohn anomalies and the contribution of Nb atoms to the Fermi level lead to strong electron-phonon coupling ( $\lambda = 0.92$ ) in Nb<sub>2</sub>CO<sub>2</sub> MXene. The  $T_c$  of Nb<sub>2</sub>CO<sub>2</sub> is further

enhanced by biaxial tensile strain and reaches up to 18.28 K under 4% tensile strain. The predicted  $T_c$  of Nb<sub>2</sub>CS<sub>2</sub> is 4.5 K, which is comparable with experimental data. These findings will further stimulate the search for superconducting MXenes.

## 1. Introduction

Ultrathin two-dimensional (2D) materials have generated much interest among the scientific community due to their versatile advantageous properties that hold potential for next-generation technologies. MXene, a diverse large group of 2D materials, are especially attractive because they may serve as the building blocks for novel functional materials and new devices.<sup>1</sup> Since the first MXene (2D Ti<sub>3</sub>C<sub>2</sub>) was reported in 2011,<sup>2</sup> more than 30 MXenes have been experimentally synthesized and over 100 theoretically predicted.<sup>3</sup> The electronic structures of MXenes vary widely from metallic to insulating, and MXenes have been used to produce materials with properties relevant to a variety of applications such as energy storage,<sup>4-7</sup> transparent conductors and heaters,<sup>8-11</sup> photo-thermal therapy,<sup>12-15</sup> photocatalysts,<sup>16-21</sup> thermoelectrics,<sup>22-26</sup> superconductors,<sup>27-28</sup> ferromagnets,<sup>29-30</sup> topological insulators,<sup>31-33</sup> and more.

MXenes are generated by selectively etching the A component from the MAX phase; M indicates an early transition metal, A is an element from the third or fourth main group, and X represents carbon or nitrogen. Due to the unsaturated transition metal layer and unpaired electrons in the precursor MAX, the surface of MXenes always absorbs various functional groups, yielding the general formula  $M_{n+1}X_nT_x$ , where  $n$  can vary from 1 to 4 and  $T_x$  denotes the surface terminations for the outermost layers of the transition metal. The composition and configuration of the surface terminations strongly influence the physical and chemical properties of MXenes. Terminated by O (OH) and F, Nb<sub>2</sub>CT<sub>x</sub> MXenes have a better reversible capacity

than  $Ti_3C_2$  and  $Ti_2C$ , making them as promising electrode materials for Li-ion batteries.<sup>34</sup> Furthermore, the capacity of delaminated  $Nb_2CT_x$  and carbon nanotube (CNT) composites when used as Li-ion battery electrodes are larger than that of multilayer  $Nb_2CT_x$ .<sup>35</sup>  $Nb_2C$  MXene covered with  $Nb_2O_5$  can be used as a photocatalyst for hydrogen evolution, and has a better photocatalytic capacity than  $Nb_2O_5$ .<sup>36</sup> The structural, mechanical and electronic properties of  $Nb_2CCl_2$  and  $Nb_3C_2Cl_2$  MXenes were investigated by first-principles calculations. Both  $Nb_2CCl_2$  and  $Nb_3C_2Cl_2$  are nonmagnetic metals.<sup>37</sup> Recently, Kamysbayev *et al.* introduced a general strategy to modify the termination of MXenes.<sup>38</sup> Unexpectedly, S- and Se-terminated  $Nb_2C$  MXenes exhibit superconductivity and the superconducting critical temperatures ( $T_c$ ) of  $Nb_2CS_2$  and  $Nb_2CSe$  are 6.4 K and 4.5 K, respectively.

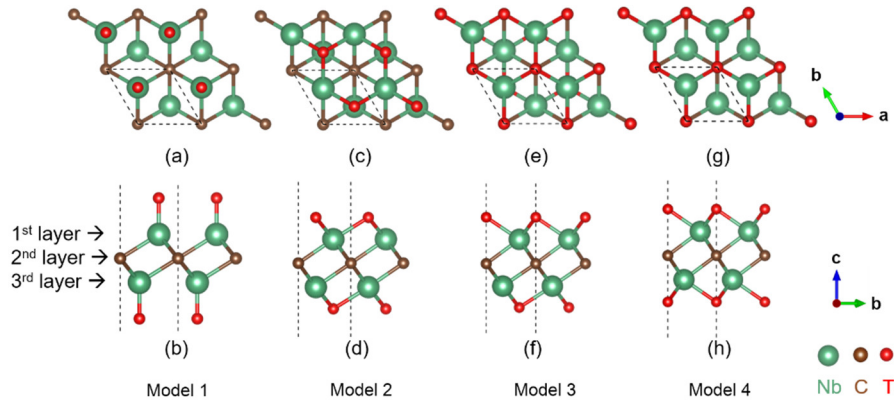
Inspired by the discovery of superconductivity in  $Nb_2CS_2$  and  $Nb_2CSe$ , we systematically investigated  $Nb_2CT_2$  ( $T = O, S, Se, \text{ or } Te$ ) via first-principles calculations using density functional perturbation theory. We built four structure models for each  $Nb_2CT_2$  based on different surface termination configurations. We studied the electronic properties, thermodynamic stability, and superconducting properties of the energy-favorable structural models of  $Nb_2CT_2$ .  $Nb_2CO_2$ ,  $Nb_2CS_2$ ,  $Nb_2CSe_2$ , and  $Nb_2CTe_2$  were found to exhibit superconductivity with  $T_c$  of 14.43 K, 4.54 K, 0.53 K, and 0.27 K, respectively. The  $T_c$  of  $Nb_2CO_2$  is further enhanced by external tensile strain, and reaches 18.28 K at 4% tensile strain.

## 2. Computational methods

Structural optimization, electronic band structure, phonon vibration, and electron-phonon coupling calculations were conducted within the density functional perturbation theory (DFPT) framework implemented in the Quantum-Espresso (QE) package.<sup>39</sup> We employed the

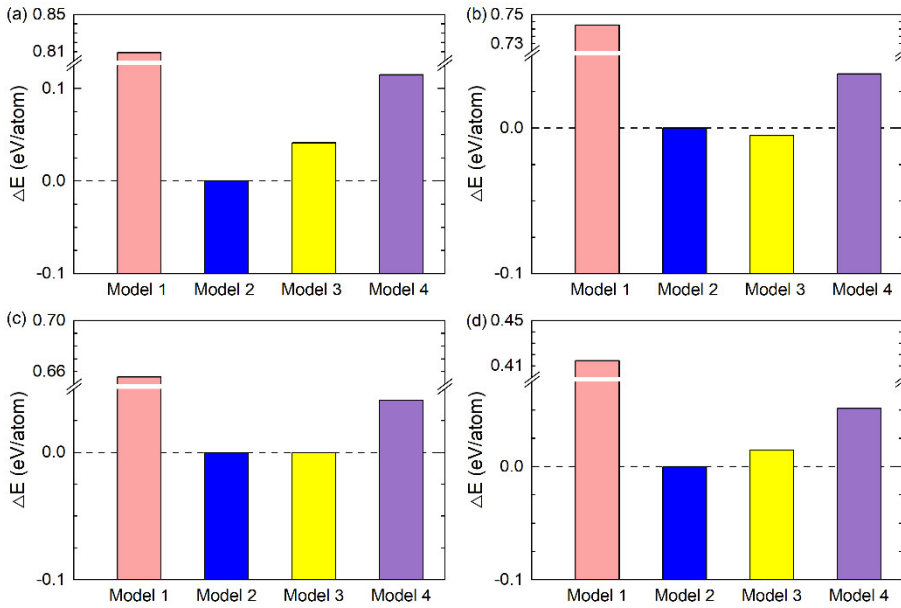
generalized gradient approximation (GGA) exchange correlation functional of the Perdew-Burke-Ernzerhof (PBE) type and norm-conserving pseudopotentials to mimic the interactions between electrons and ions.<sup>40-42</sup> The kinetic energy cutoff of plane wave and charge density were taken as 80 Ry and 560 Ry, respectively. To eliminate the interlayer interaction created by the periodic boundary condition, we set the vacuum layer to 20 Å. We adopted a  $12 \times 12 \times 1$   $k$ -point grid with a Gaussian smearing width of 0.02 Ry to simulate the Brillouin zone (BZ) integrations to calculate the self-consistent electron charge density. Convergence in the plane-wave basis set was confirmed by employing higher kinetic energy cutoffs. The electron-phonon coupling was checked carefully on a series  $q$ -point mesh and convergence was achieved with a  $6 \times 6 \times 1$   $q$ -point mesh using individual electron-phonon coupling matrix elements obtained with a  $12 \times 12 \times 1$   $k$ -point mesh.

### 3. Results and discussion



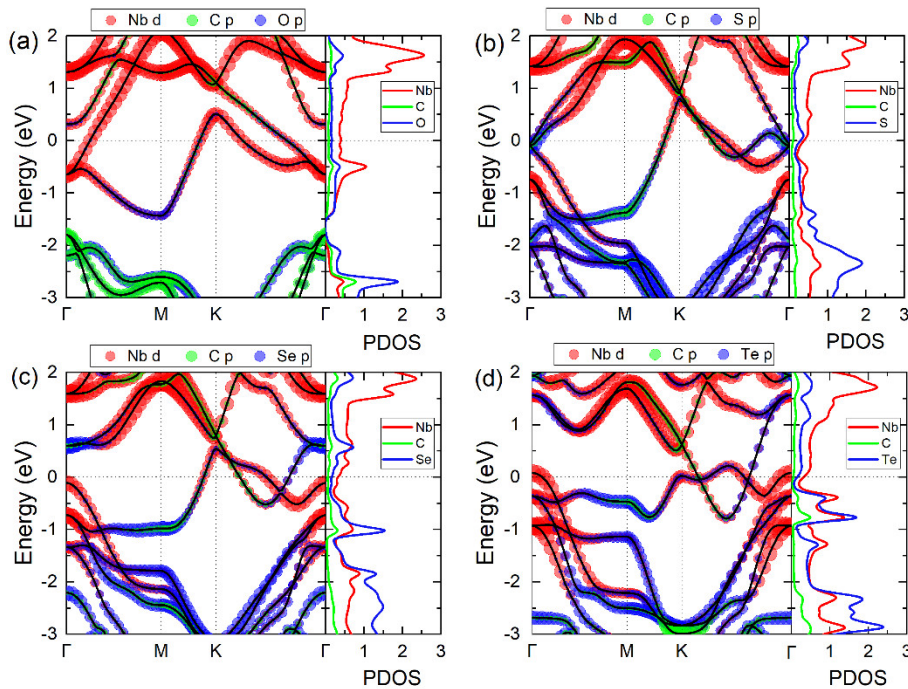
**Figure. 1** Schematic diagram of Nb<sub>2</sub>CT<sub>2</sub> MXenes structure. (a), (c), (e), and (g) Top view of Model 1, Model 2, Model 3, and Model 4. (b), (d), (f), and (h) Side view of Model 1, Model 2, Model 3, and Model 4. Light green, brown, and red balls represent Nb, C, and T atoms, respectively.

There are several passivated configurations for surface termination of Nb<sub>2</sub>CT<sub>2</sub>. Following previous work,<sup>37, 43-44</sup> we considered four possible surface configurations for Nb<sub>2</sub>CT<sub>2</sub>, as shown in Figs. 1(a)-(h), which were termed Model 1 to Model 4. An unpassivated Nb<sub>2</sub>C system consists of three atomic layers within a hexagonal cell: two Nb layers and one C layer. In Model 1, one chalcogenide atom (T) is positioned at the top site in the first Nb layer and another T atom is positioned at the top site in the third Nb layer, as shown in Figs. 1(a) and 1(b). For Model 2, the T atom connected to the first and third Nb layer locate on the top site of the third and first Nb layer, respectively. In Model 3, the position of first T atom is similar to that of Model 2, whereas the other T atom locates on the top site of the middle C layer. In Model 4, the T atoms are located on the top site of the middle C layer on both sides, as shown in Figs. 1(g) and 1(h). Details of structural properties of these four models of Nb<sub>2</sub>CT<sub>2</sub> MXenes are provided in the Supporting Information.



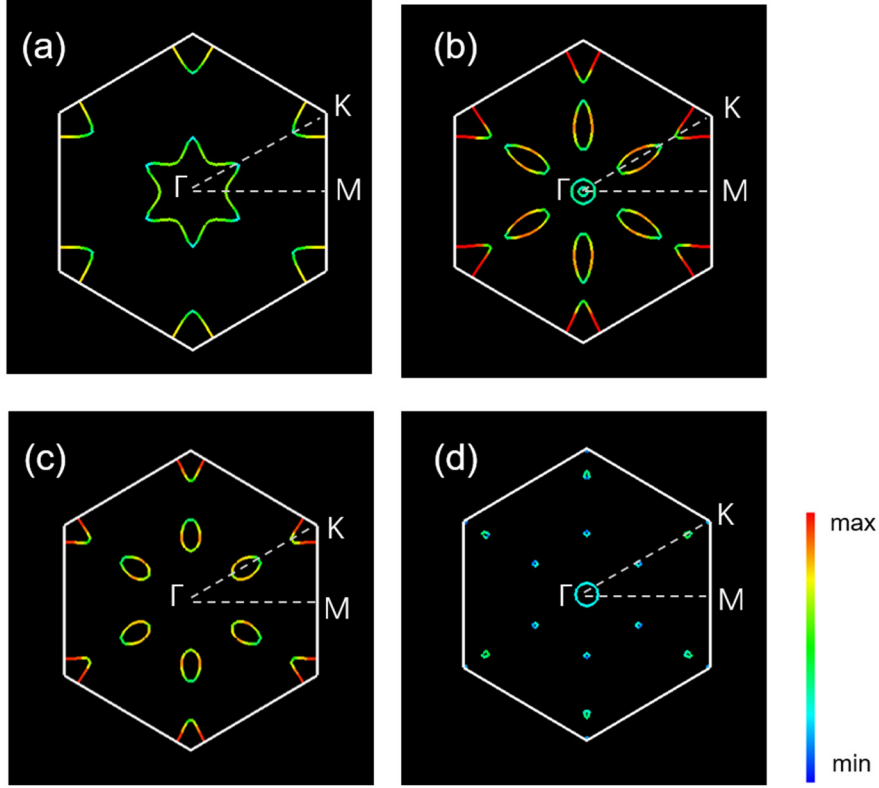
**Figure. 2** Energy comparison of four models for (a) Nb<sub>2</sub>CO<sub>2</sub>, (b) Nb<sub>2</sub>CS<sub>2</sub>, (c) Nb<sub>2</sub>CSe<sub>2</sub>, and (d) Nb<sub>2</sub>CTe<sub>2</sub>. The total energy of Model 2 for each Nb<sub>2</sub>CT<sub>2</sub> is set to 0.0 eV. The dash lines are guide to the eye.

After relaxing the lattice constants and the atomic positions, the relative total energies per atom for each  $\text{Nb}_2\text{CT}_2$  MXene system are shown in Fig. 2. To clearly illustrate the energy differences, we set the total energy per atom for Model 2 of each system to be 0.0 eV. From Fig. 2, we can see that Model 2 is the most stable configuration of  $\text{Nb}_2\text{CT}_2$  MXenes with O and Te as terminal groups, which is consistent with previous results.<sup>37</sup> On the other hand, Model 3 is the most stable configuration for  $\text{Nb}_2\text{CT}_2$  MXenes with S and Se; the energy differences between Model 3 and Model 2 for the two systems are within 5 meV/atom. However, Model 3 is nonmetal, except for  $\text{Nb}_2\text{CO}_2$  system, as shown in the Supporting Information. Therefore, we focused on Model 2 in the following study of electronic and vibrational properties and possible superconductivity.

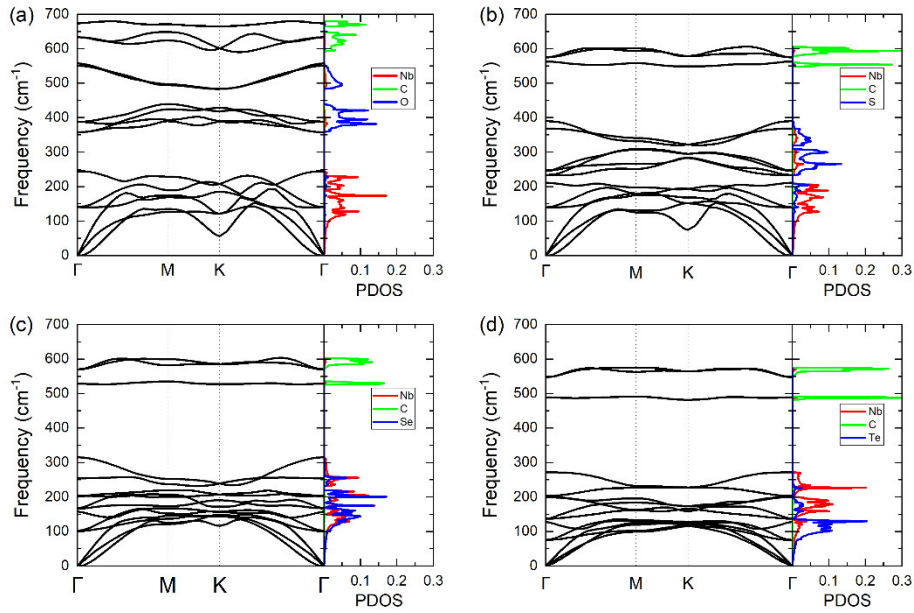


**Figure. 3** Electronic orbital-resolved band structure and projected density of states of Model 2 for (a) $\text{Nb}_2\text{CO}_2$ , (b)  $\text{Nb}_2\text{CS}_2$ , (c)  $\text{Nb}_2\text{CSe}_2$ , and (d)  $\text{Nb}_2\text{CTe}_2$ . The Fermi level indicated by the dotted line is set to 0 eV.

The electronic orbital-resolved band structures and projected density of states (PDOS) for  $\text{Nb}_2\text{CT}_2$  are plotted in Fig. 3. Clearly, all of the Model 2 of  $\text{Nb}_2\text{CT}_2$  are metallic with several bands crossing the Fermi level. Moreover, the orbitals from the Nb atoms mostly contribute to the DOS at the Fermi level and the smallest contribution comes from the orbitals of the C atoms. The PDOS of the Nb atom at the Fermi level of  $\text{Nb}_2\text{CT}_2$  decrease as the mass of T atom increases. For  $\text{Nb}_2\text{CO}_2$ , the Nb  $4d$  orbitals dominate the Fermi level, while the C  $2p$  and O  $2p$  orbitals have a limited contribution. For the  $\text{Nb}_2\text{CS}_2$ ,  $\text{Nb}_2\text{CSe}_2$ , and  $\text{Nb}_2\text{CTe}_2$  systems, there is strong hybridization between the Nb and T atoms below the Fermi level from -3 eV to -1 eV. Two bands cross the Fermi level in  $\text{Nb}_2\text{CO}_2$ , as shown in Fig. 3(a). The band crossing the Fermi level along the  $\Gamma$ -M and K- $\Gamma$  path creates a hexagram Fermi sheet around the  $\Gamma$  center, whereas the band crosses the Fermi level along the M-K and K- $\Gamma$  path forms a K-centered electron pocket, which corresponds to an arc-type Fermi sheet surrounding the vertexes of the first BZ, as shown in Fig. 4(a). Compared with  $\text{Nb}_2\text{CO}_2$ , more bands cross the Fermi level in  $\text{Nb}_2\text{CS}_2$ . These bands form four types of Fermi sheet, i.e., two  $\Gamma$ -centered circles, six ellipses along the  $\Gamma$ -K path, and six arcs around the K vertexes. The six arc Fermi sheets around the K vertexes have relatively large Fermi velocities, as shown in Fig. 4(b). The electronic band structure of  $\text{Nb}_2\text{CSe}_2$  is similar to that of  $\text{Nb}_2\text{CS}_2$  in the vicinity of the Fermi level, except for the two bands around the  $\Gamma$  point, which result in a relatively simple Fermi surface for  $\text{Nb}_2\text{CSe}_2$  (Fig. 4c). For  $\text{Nb}_2\text{CTe}_2$ , three bands cross the Fermi level, as shown in Fig. 3(d). These crossing bands form a  $\Gamma$ -centered electron pocket and two Dirac cones along the  $\Gamma$ -K path, and most of these bands are flat and thus result in a low Fermi velocity, as shown in Fig. 4(d).

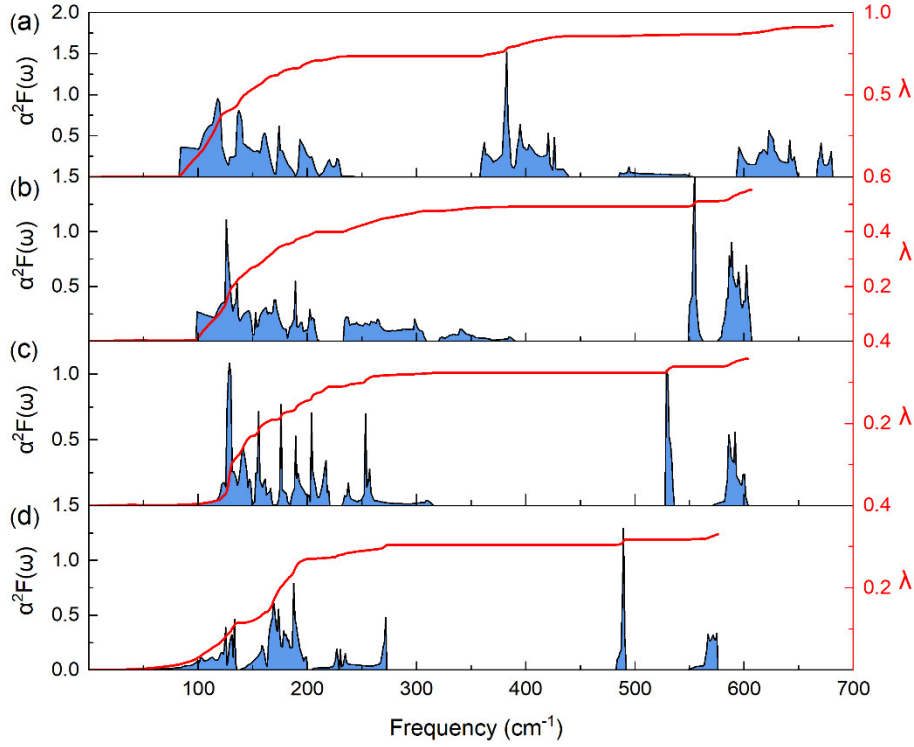


**Figure. 4** Fermi surface in the first Brillouin zone of Model 2 for (a)  $\text{Nb}_2\text{CO}_2$ , (b)  $\text{Nb}_2\text{CS}_2$ , (c)  $\text{Nb}_2\text{CSe}_2$ , and (d)  $\text{Nb}_2\text{CTe}_2$ . The variation of the color of the Fermi sheet is proportional to the magnitude of the Fermi velocity.



**Figure. 5** Phonon dispersion and projected density of states of Model 2 for (a) Nb<sub>2</sub>CO<sub>2</sub>, (b) Nb<sub>2</sub>CS<sub>2</sub>, (c) Nb<sub>2</sub>CSe<sub>2</sub>, and (d) Nb<sub>2</sub>CTe<sub>2</sub>.

Figure 5 presents the phonon spectrum and phonon projected density of states of Nb<sub>2</sub>CT<sub>2</sub> MXenes. The absence of imaginary phonon frequencies confirm that these 2D Nb<sub>2</sub>CT<sub>2</sub> systems are thermodynamically stable and could be synthesized under appropriate experimental conditions. The phonon spectra of Nb<sub>2</sub>CO<sub>2</sub> and Nb<sub>2</sub>CS<sub>2</sub> are distinctly different to the two other systems. The vibration frequencies of Nb<sub>2</sub>CO<sub>2</sub> and Nb<sub>2</sub>CS<sub>2</sub> can be divided into three parts, i.e., the low-, intermediate-, and high-frequency ranges. For Nb<sub>2</sub>CO<sub>2</sub>, the low-frequency range (below 250 cm<sup>-1</sup>), intermediate-frequency range (350 ~ 550 cm<sup>-1</sup>), and high-frequency range (above 600 cm<sup>-1</sup>) are respectively dominated by Nb, O, and C atoms. Due to the large atomic mass of S, the intermediate-frequency vibration moves downward, which enhances the interaction between the Nb and S atoms, especially from 150 cm<sup>-1</sup> to 400 cm<sup>-1</sup>. The T-related frequency range moves further downward in the Nb<sub>2</sub>CSe<sub>2</sub> and Nb<sub>2</sub>CTe<sub>2</sub> systems, which leads to disappearance of the intermediate-range frequency. It is noteworthy that there is a dip in the phonon dispersion curve around the K point of the first BZ of Nb<sub>2</sub>CT<sub>2</sub> MXenes, as shown in Fig. 5. This soft phonon mode at the K point is known as Kohn anomaly.<sup>45-48</sup> The Kohn anomaly is due to anomalous screening of the electrons around the Fermi level and further yields significant coupling between electrons and phonons. It occurs in a particular wave vector of phonon  $q$  in the BZ, which connects two electronic states on the Fermi surface:  $k$  and  $k'$ , satisfying  $k' = k + q$ . The Kohn anomaly takes place at the K point for an acoustic phonon mode with a low frequency of ~ 57 cm<sup>-1</sup>, ~ 75 cm<sup>-1</sup>, and ~ 116 cm<sup>-1</sup> for Nb<sub>2</sub>CO<sub>2</sub>, Nb<sub>2</sub>CS<sub>2</sub>, and Nb<sub>2</sub>CSe<sub>2</sub> MXenes, respectively. The Kohn anomaly is apparent in Nb<sub>2</sub>CO<sub>2</sub> and Nb<sub>2</sub>CS<sub>2</sub>, and becomes weaker in Nb<sub>2</sub>CSe<sub>2</sub>, and then not present in Nb<sub>2</sub>CTe<sub>2</sub>.



**Figure. 6** Eliashberg spectral function  $\alpha^2F(\omega)$  and frequency-dependent integrated electron-phonon coupling constant  $\lambda(\omega)$  for (a)  $\text{Nb}_2\text{CO}_2$ , (b)  $\text{Nb}_2\text{CS}_2$ , (c)  $\text{Nb}_2\text{CSe}_2$ , and (d)  $\text{Nb}_2\text{CTe}_2$ .

The Eliashberg spectral function,  $\alpha^2F(\omega)$ , and integrated electron-phonon coupling constant,  $\lambda(\omega)$ , are presented in Fig. 6. Similarly to the phonon PDOS, the distribution of  $\alpha^2F(\omega)$  in  $\text{Nb}_2\text{CO}_2$  and  $\text{Nb}_2\text{CS}_2$  are different to that of the  $\text{Nb}_2\text{CSe}_2$  and  $\text{Nb}_2\text{CTe}_2$  systems.  $\text{Nb}_2\text{CO}_2$  possesses the largest integrated electron-phonon coupling of 0.92 of the four systems. The low-frequency (Nb-dominated) range, intermediate-frequency (O-dominated) range, and high-frequency (C-dominated) range contribute 79.6%, 13.3%, and 7.1% of the total  $\lambda$ , respectively.  $\text{Nb}_2\text{CS}_2$  has a relatively small  $\lambda$  (0.55). The contribution of phonon vibration to the total  $\lambda$  is similar in  $\text{Nb}_2\text{CS}_2$  and  $\text{Nb}_2\text{CO}_2$ .  $\text{Nb}_2\text{CSe}_2$  and  $\text{Nb}_2\text{CTe}_2$  have small  $\lambda$ , as shown in Table 1. Vibration of Nb and T atoms contribute 90.3% and 92.2% of the total  $\lambda$  for  $\text{Nb}_2\text{CSe}_2$  and  $\text{Nb}_2\text{CTe}_2$ , respectively. The large  $\lambda$  of  $\text{Nb}_2\text{CO}_2$  motivates us to reinspect the Fermi surface and phonon dispersion. We found that the

large arc-type Fermi sheet around the K point and the apparent Kohn anomaly at the K point of first BZ for Nb<sub>2</sub>CO<sub>2</sub>. Thus, the coupling between the electrons coming from the arc-type Fermi sheet and the softening phonon modes induced by Kohn anomaly results to a large  $\lambda$ . Moreover, the above mentioned electrons and phonon modes mainly come from Nb atoms, which is consistent with the analysis of Eliashberg spectral function.

**Table. 1** Logarithmic average frequency  $\omega_{\log}$ , electron-phonon coupling constant  $\lambda$ , and superconducting critical temperature  $T_c$  for four Nb<sub>2</sub>CT<sub>2</sub> MXenes.

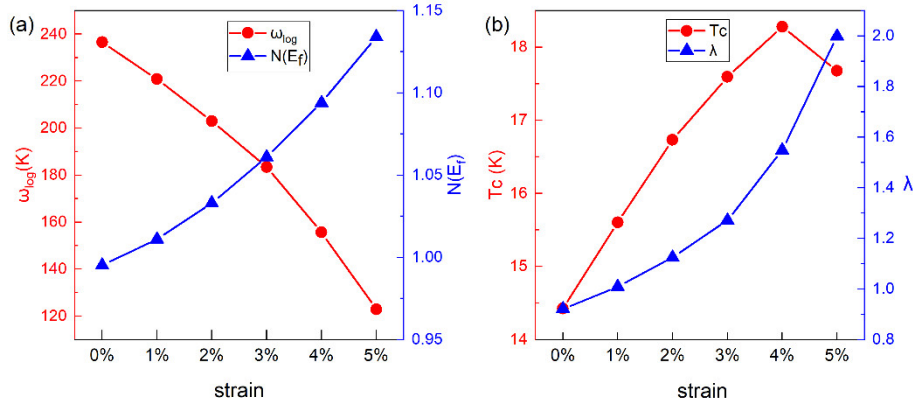
	$\omega_{\log}$ (K)	$\lambda$	$T_c$ (K)
Nb <sub>2</sub> CO <sub>2</sub>	236.51	0.92	14.43
Nb <sub>2</sub> CS <sub>2</sub>	258.60	0.55	4.54
Nb <sub>2</sub> CSe <sub>2</sub>	254.74	0.36	0.53
Nb <sub>2</sub> CTe <sub>2</sub>	237.45	0.33	0.27

According to the Allen-Dynes modified version of the McMillan equation:<sup>49-50</sup>

$$T_c = \frac{\omega_{\log}}{1.2} \exp \left[ -\frac{1.04(1 + \lambda)}{\lambda - \mu^*(1 + 0.62\lambda)} \right]$$

we estimated the  $T_c$  of Nb<sub>2</sub>CT<sub>2</sub> using the Coulomb pseudopotential  $\mu^* = 0.10$ , as shown in Table 1. Clearly, Nb<sub>2</sub>CO<sub>2</sub> possesses the largest  $T_c$  (14.43 K) of the four Nb<sub>2</sub>CT<sub>2</sub> MXenes systems, while Nb<sub>2</sub>CTe<sub>2</sub> has a relatively small  $T_c$  of 0.27 K. It is worth noting that the reported  $T_c$  of Nb<sub>2</sub>CS<sub>2</sub> is 6.4 K, which is good agreement with our theoretical data (4.54 K).<sup>38</sup> We note that superconductivity was not experimentally observed in Nb<sub>2</sub>CO<sub>2</sub>, probably as the experimental sample had a different structure to that used in the present calculations. The experimental sample

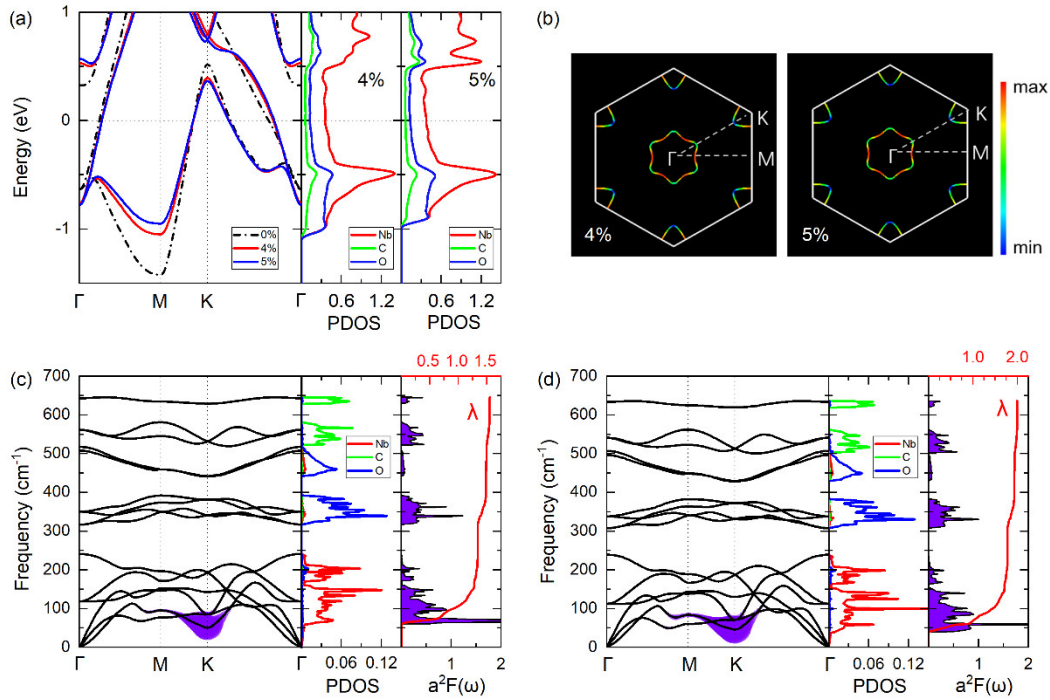
was cold-pressed pellets, while the present theoretical calculations were based on a multilayer atomic structure of only five angstroms thickness.



**Figure. 7** Effect of biaxial strain on (a)  $\omega_{\log}$  and  $N(E_f)$ , and (b)  $\lambda$  and  $T_c$  for  $\text{Nb}_2\text{CO}_2$ .

Strain has been proved to be an effective approach to tune electronic and dynamic properties of 2D materials.<sup>51-57</sup> Introducing strain is a mature and important technology in nanomaterials, as well as MXenes.<sup>58-59</sup> The strain effects on the superconductivity of 2D materials have been extensively investigated theoretically within the framework of BCS theory.<sup>47, 60-67</sup> The superconductivity was predicted to be enhanced by biaxial strain in some 2D materials, such as 2D phosphorus carbide<sup>47</sup> and 2D tungsten boride.<sup>63</sup> Here, we introduced biaxial strain to the  $\text{Nb}_2\text{CO}_2$  MXene, which has the largest  $T_c$  in the absence of strain among the four systems, and investigated the strain effects on superconductivity. Biaxial strain ( $\delta$ ) ranging from -5% to 5% was applied to  $\text{Nb}_2\text{CO}_2$  at intervals of 1% to simulate the conditions of sample growth on a substrate.  $\delta$  was calculated as  $\delta = (a - a_0)/a_0$ , where  $a_0$  is the lattice constant in the absence of strain, and negative and positive values indicate compressive and tensile strain, respectively. Imaginary phonon frequencies were observed for  $\text{Nb}_2\text{CO}_2$  under -5% to -1% strain, which means that compressive strain destroys the stability of  $\text{Nb}_2\text{CO}_2$  MXene. Figure 7 presents the variations in the superconductivity-related parameters of

$\text{Nb}_2\text{CO}_2$  under tensile strain. Interestingly, the  $\lambda$  and DOS at the Fermi level  $N(E_f)$  increase monotonically with tensile strain, whereas the  $\omega_{\log}$  decreases monotonically with tensile strain. Due to interplay between  $\omega_{\log}$ ,  $\lambda$ , and  $N(E_f)$ , the  $T_c$  of  $\text{Nb}_2\text{CO}_2$  initially increases with tensile strain and reaches 18.28 K at 4% strain, and then decreases to 17.68 K at 5% tensile strain. In addition to biaxial strain, we also applied uniaxial or shear strains on the  $\text{Nb}_2\text{CO}_2$  MXene; however, most of the configurations of Model 2 of  $\text{Nb}_2\text{CO}_2$  MXene under uniaxial or shear strains become thermodynamically unstable. Therefore, we focused on biaxial strain and explored the mechanism of strain effects on superconductivity in the following text.



**Figure. 8** (a) Electronic band structure and PDOS, (b) Fermi surface of  $\text{Nb}_2\text{CO}_2$  MXenes under tensile strain. EPC-weighted phonon spectra, Phonon PDOS,  $\alpha^2F(\omega)$  and frequency-dependent integrated  $\lambda(\omega)$  of  $\text{Nb}_2\text{CO}_2$  MXenes under (c) 4% and (d) 5% strain. The weight in the phonon

spectrum is proportional to the electron-phonon coupling strength  $\lambda_{qv}$ , marked in purple. The subscript  $q$  represents the wave vector and  $v$  represents the phonon mode.

Figure 8 shows the electronic and thermodynamic properties of Nb<sub>2</sub>CO<sub>2</sub> under 4% and 5% tensile strain. The electronic bands that cross the Fermi level become slightly smoother, as shown in Fig. 8(a). The contribution of Nb atoms to the Fermi level increases as the strain increases. The shapes of the Fermi surface of Nb<sub>2</sub>CO<sub>2</sub> under 0%, 4%, and 5% strain are similar, as shown in Fig. 8(b). Under 4% and 5% tensile strain, the intermediate-frequency and high-frequency phonon vibrations for Nb<sub>2</sub>CO<sub>2</sub> move downward, and the low-frequency range (Nb-dominated) remains below 250 cm<sup>-1</sup>. This leads to a decrease in  $\omega_{\log}$ . The low-lying vibration mode controlled by Nb atoms contributes 86.4% and 88.9% of total  $\lambda$ , respectively. The projected  $\lambda_{qv}$  shows the soft mode at the K point contributes significantly to the total  $\lambda$ . Between 0% and 4% strain, the increase in  $\lambda$  correlates with  $N(E_f)$  and dominates the increase in the  $T_c$  of Nb<sub>2</sub>CO<sub>2</sub>. Between 4% and 5% strain, the reduction in  $\omega_{\log}$  originates from the downshift of the phonon vibrations under tensile strain, and overcomes the increase in  $\lambda$  and eventually decreases the  $T_c$  of Nb<sub>2</sub>CO<sub>2</sub> under 5% strain.

#### 4. Conclusions

In summary, the electronic and superconducting properties of Nb<sub>2</sub>CT<sub>2</sub> (T = O, S, Se, or Te) MXenes were investigated using first-principles calculations within the DFPT framework. We built four structural models for each Nb<sub>2</sub>CT<sub>2</sub> MXene based on different surface passivations, and then focused on the low-energy Model 2. Nb<sub>2</sub>CO<sub>2</sub> possesses a strong electron-phonon coupling ( $\lambda = 0.92$ ), which leads to the largest  $T_c$  among the four Nb<sub>2</sub>CT<sub>2</sub> MXenes. The existence of soft modes due to Kohn anomalies and the contribution from Nb atoms to the Fermi level account for the strong electron-phonon coupling of Nb<sub>2</sub>CO<sub>2</sub>. Using the Allen and Dynes modified McMillan

equation, we calculated the  $T_c$  of Nb<sub>2</sub>CO<sub>2</sub> is to be 14.43 K. Furthermore, the  $T_c$  of Nb<sub>2</sub>CO<sub>2</sub> is enhanced by tensile strain and increases by 27% at 4% tensile strain. At 5% tensile strain,  $T_c$  decreases to 17.68 K due to reduction of  $\omega_{\log}$  of phonon vibrations overcoming the increase in electron-phonon coupling. The results have great implications for the superconductivity of MXenes and other 2D systems.

## ASSOCIATED CONTENT

### **Supporting Information.**

The Supporting Information is available free of charge on the ACS Publications website.

Details of structural parameters, superconducting critical temperature, electronic band structures, and phonon spectrum of Nb<sub>2</sub>CT<sub>2</sub> MXenes. (PDF)

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## Notes

The authors declare no competing financial interest.

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