

# On the structural transformation of Ni/BaH<sub>2</sub> during a N<sub>2</sub>-H<sub>2</sub> chemical looping process for ammonia synthesis: a joint *in situ* inelastic neutron scattering and first-principles simulation study

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## Abstract

The demand for decarbonizing the ammonia industry by using renewable energy has invoked increasing research interests into catalyst development for effective N<sub>2</sub> reduction under mild conditions. Hydride-based materials are among some of the emerging catalysts for ammonia synthesis at ambient pressure and low temperatures (<673 K). A recent chemical looping process based on Ni/BaH<sub>2</sub> showed the most promise as it can realize ammonia production at a temperature as low as 373 K and under ambient pressure. However, the chemical transformation of the hydride catalyst at the molecular level remains unclear in this process. In this work, we report detailed *in situ* neutron spectroscopy and diffraction investigations along with first-principles simulations on the structural transformation of Ni/BaH<sub>2</sub> during the nitridation and hydrogenation steps in the chemical looping process for ammonia synthesis. It was shown that a ball-milling process of the starting Ni/BaH<sub>2</sub> could significantly decrease the size of BaH<sub>2</sub> and increase the density of defects, thus potentially enhancing the reactivity of the hydride. The evolution from BaH<sub>2</sub> to barium imide (BaNH) was evidenced in the inelastic neutron scattering (INS) and neutron diffraction results during the N<sub>2</sub> reaction step. During the hydrogenation study, in addition to the recovery of BaH<sub>2</sub>, a **possible** intermediate species, N-deficient barium imide, was also detected. In comparing the N<sub>2</sub> and H<sub>2</sub> reaction steps, the neutron results indicate that the hydrogenation step is more difficult than the nitridation step, confirming the facile N<sub>2</sub> fixation property of Ni/BaH<sub>2</sub> catalyst in ammonia synthesis.

**Keywords:** chemical looping, ammonia synthesis, Ni/BaH<sub>2</sub>, neutron scattering, first-principles simulations

## Introduction

As a crucial component in fertilizers, ammonia is also a promising carrier for hydrogen storage because of its high volumetric and gravimetric hydrogen densities and potential as clean fuel for fuel cells and combustion engines.[1,2] Traditionally, a Haber-Bosch (HB) process is used as the main industrial route for ammonia production in which nitrogen and hydrogen gases react catalytically under harsh operating temperatures (673-873 K) and pressure (20-40 MPa), consuming 1~2% of world annual energy output.[3-6] Because ammonia synthesis at low temperatures is thermodynamically favorable, new catalytic processes have been explored for ammonia synthesis in a more sustainable and environmentally benign way, i.e., under much milder conditions than the HB process. Recent notable progress in heterogeneous catalysis has been witnessed in the utilization of advanced materials including nitrides,[7,8] electrides,[9,10] hydrides[11-14] including oxyhydrides[14,15] as the catalysts or supports, which enabled ammonia synthesis at temperatures below 673 K and at ambient pressure.

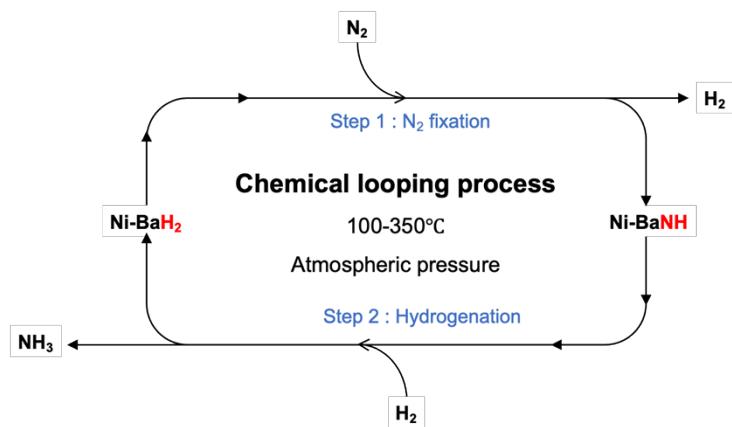
Among the emerging catalysts, alkali and alkaline hydride-based materials hold great promise in ammonia synthesis due to their appreciable reaction rate at mild conditions.[4-6] For example, the Chen group showed the first example of using hydrides such as LiH as co-catalysts working synergistically with transition metals including Cr, Mn, Fe and Co for ammonia synthesis at 423 – 623 K that outperformed the Cs-promoted Ru catalyst, one of the most active NH<sub>3</sub> synthesis catalysts by 2—3 times at 573 K and 12—20 times at 523 K.[11] Recently, a chemical looping process comprised of nitrogen fixation and hydrogenation steps over Ni/BaH<sub>2</sub> was demonstrated by the same group for ammonia synthesis at a temperature as low as 373 K and atmospheric pressure with the rate more than one order of magnitude higher than that of the commercial catalyst (Cs-Ru/MgO).[16] In these studies based on alkali and alkaline hydrides, the facile transformation among imides (NH<sup>2-</sup>), amide (NH<sub>2</sub><sup>-</sup>), and hydrides (H<sup>-</sup>) plays an essential role in the N-H bond formation and NH<sub>3</sub> production. NH<sub>x</sub> hydrogenation was considered the rate-limiting step (RDS) over these hydrides instead of the N<sub>2</sub> activation step on traditional ammonia synthesis catalysts.[11] For example, in the chemical looping process, it was proposed (see **Figure 1**) that the hydride anion in BaH<sub>2</sub> is a strong reducing agent and has a chemical potential that drives the N fixation step by providing electrons to N<sub>2</sub> to break the triple bond of N<sub>2</sub> and to form the imide intermediate (BaNH). During the hydrogenation step, the hydrogen undergoes disproportionation to hydridic and protic H atoms to bond with Ba and nitrogen atoms to form

$\text{BaH}_2$  and ammonia, respectively.[16] Interestingly, the physically mixed late 3d metal (such as Ni, Co, and Fe) with metal hydride increases its  $\text{N}_2$  dissociation/hydrogenation ability, up to 6 times higher than the metal hydride itself [16,12], indicating a synergism between the hydride and the metal components. Although the role of metal hydride has been studied to understand the kinetics beyond the nitrogen fixation and hydrogenation steps of the chemical looping process, there is still a lack of understanding of the mechanism at the molecular level via spectroscopy by *in situ* monitoring the cycling between the metal hydride and imide species, and any potential intermediates during the looping process.

Neutron scattering is highly sensitive to hydrogen compared to other scattering techniques such as X-ray or electron scattering. It has been demonstrated as a powerful tool to probe the molecular structure of alkali and alkaline hydride, imide, and amide species via neutron diffraction and especially inelastic neutron scattering (INS).[9,17-22] Taking advantage of neutron scattering techniques and using  $\text{Ni/BaH}_2$  as a model catalyst, this work aims to elucidate the detailed mechanism of the chemical transformation of  $\text{BaH}_2$  in the  $\text{N}_2\text{-H}_2$  chemical looping process using both INS and neutron diffraction. First-principles calculations combined with lattice and molecular dynamic simulations are employed to interpret the neutron spectroscopy results and to understand the structures and their dynamics. It is clearly shown that both stoichiometric and non-stoichiometric  $\text{BaH}_2$  and  $\text{BaNH}$  species are involved during the cycling process for ammonia synthesis. The result provides additional insights into the structural dynamics of hydride-based catalysts in the chemical looping process for ammonia synthesis.

## Experimental

Barium hydride (American Elements, 99.7 % metal basis) and nickel powder (Sigma-Aldrich, nanopowder <100nm, >99% trace metals basis) were used as received. The mixture of



**Figure 1.** The proposed mechanism for the Chemical looping process for ammonia synthesis over  $\text{Ni/BaH}_2$ , modified from Ref. [16].

nickel powder and barium hydride were prepared via ball milling at a ratio of 1:1 (2 g of  $\text{BaH}_2$  and 2 g of Ni powder). The powders were mixed in a glass vial and the resulting mixture was placed in a vial with tungsten carbide caps and one steel ball for ball-milling with a high-energy SPEX mill. All operations were conducted in a dry helium glove box ( $\text{H}_2\text{O} < 0.1 \text{ ppm}$ ,  $\text{O}_2 < 0.1 \text{ ppm}$ ). Ball-milling was performed overnight ( $\sim 18 \text{ hr}$ ), alternating between 30 minutes periods of ball-milling and 5 minutes of cooling time (no ball-milling), corresponding to just under 16 hours of ball-milling time. X-Ray diffraction (XRD) experiments of the samples before and after ball milling were performed using a PANalytical X’Pert Pro MPD diffractometer with  $\text{Cu-K}\alpha$  radiation ( $\lambda = 1.5418 \text{ \AA}$ ). The ball-milled sample was recovered and placed in an aluminum sample holder for the following neutron scattering experiment.

INS experiments were performed at the VISION beamline (BL-16B) of Spallation Neutron Source (SNS) at Oak Ridge National Laboratory. An empty aluminum can was measured at 5 K to collect the background of the INS spectrum. The Ni/ $\text{BaH}_2$  sample (4 g) was then loaded in the can and further exposed to 393 K under vacuum to remove the remaining moisture in the sample. After such treatment, the blank sample was measured at 5 K. After measurement, the sample was heated to 573 K and exposed to 3 bar of  $\text{N}_2$  at 573 K for 2 hr. The sample was then pumped shortly to remove any gaseous hydrogen-containing species at 573 K and exposed to 3 bar of  $\text{N}_2$  for an additional 2 hr at the same temperature. The sample was shortly pumped at 573 K to remove any gaseous and adsorbed species and then cooled down to 373 K and further pumped down to base vacuum level. After such treatment, the sample was cooled down to 5 K to collect the INS spectrum. The sample was further exposed to 3 bar of  $\text{H}_2$  at 573 K for 2 hr and then shortly pumped at 573 K and cooled down to room temperature followed by pumping down to remove unreacted  $\text{H}_2$ . It was quenched to 5 K to collect the INS spectrum. Neutron diffraction patterns were also collected for the sample during the measurement of the INS spectra. All gas loading operations at the neutron beam line were performed *in situ* with the sample undisturbed in the sample holder, which allows potential quantitative analysis of the hydrogen content in the sample during the different steps.

First-principles simulations including lattice dynamics (LD) and molecular dynamics (MD) simulations were conducted using Vienna Ab initio Simulation Package (VASP).[23] The calculation used Projector Augmented Wave (PAW) method [24,25] to describe the effects of core electrons, and Perdew-Burke-Ernzerhof (PBE) [26] implementation of the Generalized

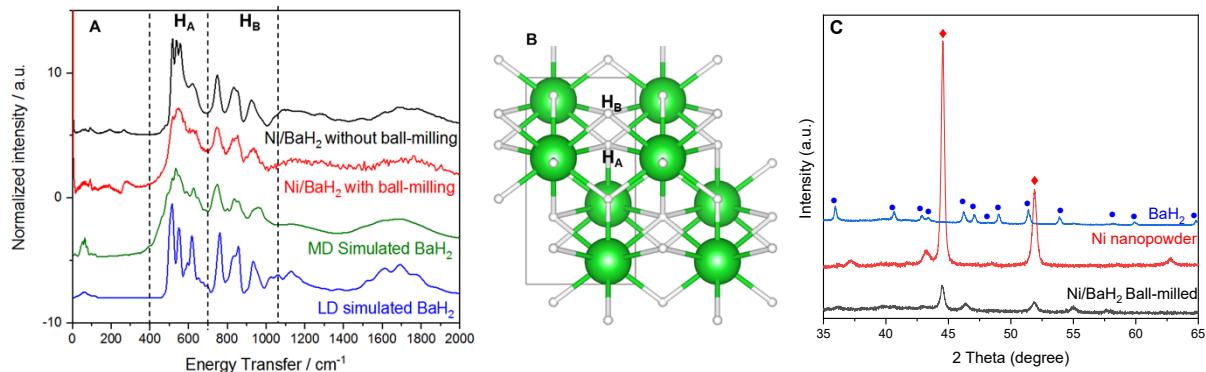
Gradient Approximation (GGA) for the exchange-correlation functional. The energy cutoff was 500 eV for the plane-wave basis of the valence electrons. The lattice parameters and atomic coordinates from the literature were used as the initial structure. The total energy tolerance for electronic energy minimization was  $10^{-8}$  eV, and for structure optimization it was  $10^{-7}$  eV. The maximum interatomic force after relaxation was below 0.005 eV/Å. The vibrational eigenfrequencies and modes were then calculated by solving the force constants and dynamical matrix using Phonopy.[27] The OCLIMAX software[28] was used to convert the DFT-calculated phonon results to the simulated INS spectra. MD simulations were performed with a lower cutoff energy (400 eV), at 300 K, with a time step of 1.0 fs.

## Results and Discussion

### *Impact of ball milling on the structure of Ni/BaH<sub>2</sub>*

It was shown by the Chen group [16] that adding nickel nanoparticles (NPs) to BaH<sub>2</sub> can not only lower the onset temperature in both the nitridation and hydrogenation steps, but also increase the reaction rates in both steps, especially the nitridation step (up to 30 times) relative to the bare BaH<sub>2</sub> in the chemical looping process. These promotional effects were attributed to the addition of Ni functionality, which was introduced via the ball milling process. However, in addition to creating intimate contact between Ni NPs and BaH<sub>2</sub>, it is unclear if ball milling plays an extra role for example in impacting the structure of BaH<sub>2</sub>. In this context, we first investigate the INS spectra of Ni/BaH<sub>2</sub> with and without the ball milling procedure and the results are shown in **Figure 2A**.

The spectrum from a simple physical mixture of Ni/BaH<sub>2</sub> shows several sharp peaks below 1050 cm<sup>-1</sup> with a few broad features above 1100 cm<sup>-1</sup>, similar to what was reported for well-crystallized BaH<sub>2</sub>.[29] The sharp peaks can be divided into two regions as marked as A (400 – 700 cm<sup>-1</sup>) and B (700 – 1100 cm<sup>-1</sup>) in the spectra due to the vibrational modes of two types of hydrides in BaH<sub>2</sub> lattice. The lower energy branch A is associated with the vibrations of the 5-coordinated H (denoted as H<sub>A</sub> site) in an approximate square-pyramidal structure. In contrast, branch B is due to the vibrations of the 4-coordinated H (denoted as H<sub>B</sub> site) in a tetragonal structure, as shown in the illustration in **Figure 2B**. The subpeaks in each branch are due to the rattling of the hydride ions along different directions. Similar to previous work[29], the LD simulated spectrum matches generally well with that of the crystalline BaH<sub>2</sub> but some discrepancies exist especially in the A



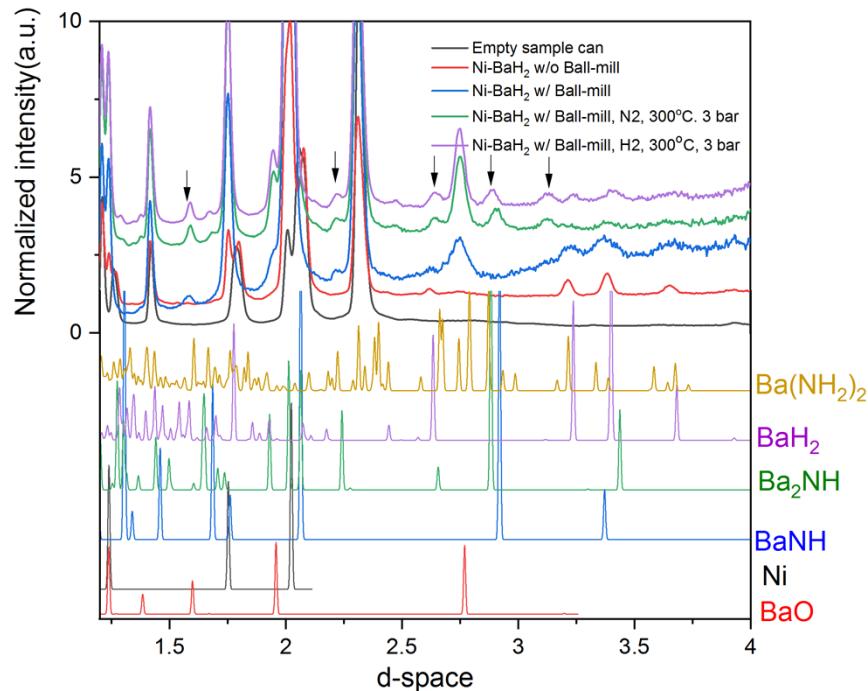
**Figure 2.** (A) INS spectra of Ni/BaH<sub>2</sub> before and after ball milling. The LD and MD simulated BaH<sub>2</sub> spectra are also shown for comparison. (B) Illustration of the two types hydride ions in the BaH<sub>2</sub> structure. Green ball: Ba; white ball: H. (C) XRD patterns of BaH<sub>2</sub>, Ni nanopowder and ball-milled Ni/BaH<sub>2</sub>.

branch region, likely due to the complex local potential energy profile and anharmonicity associated with the 5-coordinated H. MD simulation was further employed to probe the potential energy profile and the trajectory was converted to simulated INS spectrum. As shown in **Figure 2A**, compared to the LD simulation, the spectral features including relative peak intensity and frequency from MD simulation are now closer to those in the measured spectra from BaH<sub>2</sub>, indicating that anharmonicity indeed plays a role, at least for the 5-coordinated H. The peaks from the MD simulation are also generally broader than those observed in the experiment, probably due to the higher temperature used in the simulation which was intended to partially include the zero-point energy fluctuation of H atoms, while using a classical model, but would also introduce artifacts caused by larger Ba displacements (equivalent to more disorder Ba coordinates).

The INS spectrum from Ni/BaH<sub>2</sub> after ball milling shows similar spectral contour to that of the physical mixture. Still, with a significant peak broadening, a spectral feature very similar to that of the MD simulated one. This interesting observation indicates that the ball milling process results in a breakdown of the particle size and disordering of the BaH<sub>2</sub> and thus less crystallinity. This is confirmed by the XRD patterns (**Figure 2C**) collected before and after ball milling: the size of BaH<sub>2</sub> decreases significantly from 78 to 27 nm. Such a change can lead to increased defects in BaH<sub>2</sub> and thus can enhance its chemical reactivity, potentially an additional contributing factor to the improved performance of Ni/BaH<sub>2</sub> in the chemical looping process in ammonia synthesis in addition to the role of Ni NPs. Other possible contributing factors for the peak broadening include the formation of Ni hydride or BaNi hydride species during the high energy ball milling process.

Unfortunately, some of the major peaks from Ni hydride[30] and presumed bimetallic BaNi hydrides would be buried under the features from  $\text{BaH}_2$  and thus cannot be identified definitely. Furthermore, the neutron diffraction patterns (reported below) do not show clear evidence for the potential presence of these hydrides after the ball milling. Thus, it is not likely that new hydride species are formed during the ball milling process, and the INS spectral profile (broadening) change is mostly related to the size change and creation of defects in  $\text{BaH}_2$ .

Neutron diffraction patterns collected during the INS experiment also support the change occurred to  $\text{Ni}/\text{BaH}_2$  during ball milling. As shown in **Figure 3**, the diffraction peaks at  $\sim 2.61$ ,  $3.22$ ,  $3.40$  and  $3.65$   $\text{\AA}$  due to  $\text{BaH}_2$  are apparently broadened after the ball milling process. Notably, new peaks are observed at  $1.60$ ,  $1.95$ , and  $2.75$   $\text{\AA}$  after ball milling. These peaks are identified to be due to  $\text{BaO}$ . It is possible that the trace water or  $\text{O}_2$  ( $<0.1$  ppm) in the glove box system slowly oxidizes a portion of  $\text{BaH}_2$  to  $\text{BaO}$  during the long ball milling process ( $\sim 18$  hr). Such a  $\text{BaO}$  phase also seems to be present in the XRD pattern of  $\text{Ni}/\text{BaH}_2$  reported by Chen and coworkers [16] (maybe in different amount from our case). Since the peaks due to the  $\text{BaO}$  phase only become sharper during the cycling process due to the annealing effect, it does not seem that  $\text{BaO}$  plays any



**Figure 3.** neutron diffraction patterns collected for  $\text{Ni}/\text{BaH}_2$  with and without ball milling, and during the  $\text{N}_2$ - $\text{H}_2$  cycling process. The peaks from several standards including  $\text{Ba}(\text{NH}_2)_2$ ,  $\text{BaH}_2$ ,  $\text{Ba}_2\text{NH}$ ,  $\text{BaNH}$ ,  $\text{Ni}$ , and  $\text{BaO}$  are also shown at the bottom.

role in the chemical looping process. Instead,  $\text{BaH}_2$  is the major player, as shown by the changes of the neutron diffraction peaks and the INS spectra (shown below) during the cycling process.

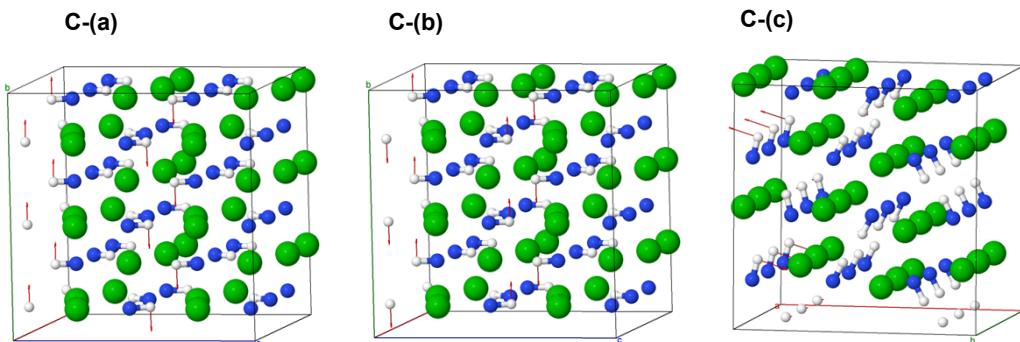
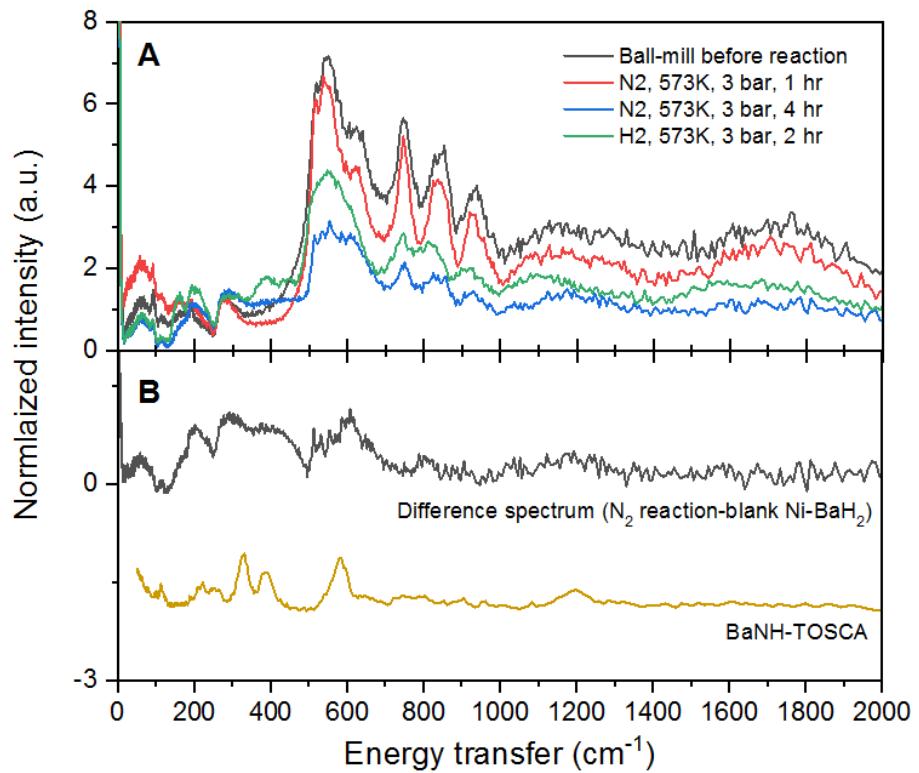
#### *Structural changes in the $\text{N}_2$ reaction step*

**Figure 4A** shows the INS spectra of ball-milled  $\text{Ni/BaH}_2$  before and after a nitridation step at 573 K after different reaction times. Because of the INS cell configuration, i.e., the contact of gases with the sample is in a stationary mode, 3 bar of  $\text{N}_2$  was used to facilitate the reaction instead of the typically used ambient pressure in the chemical looping.[16] After 1 hr reaction, some 20% intensity decline is observed for the spectrum from  $\text{Ni/BaH}_2$ , indicating the consumption of  $\text{BaH}_2$ . After 4 hr reaction, a significant change in spectral intensity is observed, suggesting a successful reaction between  $\text{N}_2$  and  $\text{Ni/BaH}_2$  has occurred. The most likely reaction between  $\text{N}_2$  and  $\text{BaH}_2$  (R1) was proposed previously[16] as follows:



Indeed, a subtraction of the INS spectra before (scaled down to match the intensity in the A and B branch regions) and after the  $\text{N}_2$  reaction step (4 hr data) results in a spectrum with features similar to that of reported barium imide ( $\text{BaNH}$ ) [19] which is also plotted in **Figure 4B**, implying the formation of  $\text{BaNH}$  species in the nitridation step over  $\text{Ni/BaH}_2$ , consistent with the previous observation from X-ray diffraction [16]. The modes associated with the major INS bands of the imide species are illustrated in **Figure 4C**. Interestingly, the intensity decline from the two types of hydride sites,  $\text{H}_A$  and  $\text{H}_B$ , does not show the evident difference after either 1 or 4 hr reaction in  $\text{N}_2$ , indicating a similar reactivity of the two types of lattice hydrides towards  $\text{N}_2$  in  $\text{Ni/BaH}_2$ .

In the corresponding neutron diffraction patterns shown in **Figure 3**, a few extra diffraction peaks appear at  $\sim 1.68$ ,  $2.22$ ,  $2.64$ ,  $2.88$ , and  $3.12$   $\text{\AA}$  after the  $\text{N}_2$  reaction step. The simulated patterns from  $\text{BaH}_2$ ,  $\text{BaNH}$ ,  $\text{Ba}_2\text{NH}$  and  $\text{Ba}(\text{NH}_2)_2$  are also included for comparison. Apparently,  $\text{BaNH}$  and  $\text{BaH}_2$  can account for most of the peaks, confirming the formation of  $\text{BaNH}$  in the nitridation step. Noticeably, the diffraction intensity decreases after the  $\text{N}_2$  reaction step and regains after the  $\text{H}_2$  step, which is indicative of H abstraction and addition into the sample since the neutron diffraction intensity is mostly from the incoherent scattering of hydrogen atoms.



**Figure 4.** A) INS spectra of Ni/BaH<sub>2</sub> before reaction, after N<sub>2</sub> reaction for 1 and 4 hr at 573 K, and after H<sub>2</sub> reaction for 2 hr at 573 K. B) comparison of difference INS spectrum after N<sub>2</sub> reaction with that reported for BaNH. C) Illustration of representative vibrational modes in BaNH for the INS bands at (a) 200, (b) 350, and (c) 600  $\text{cm}^{-1}$ . Green ball: Ba; Blue ball: N; while ball: H.

#### *Structural changes in the H<sub>2</sub> reaction step*

The INS spectrum from H<sub>2</sub> reaction step after the N<sub>2</sub> step over Ni/BaH<sub>2</sub> is also included in **Figure 4A**. An obvious increase in the spectral intensity is noted relative to the spectrum after N<sub>2</sub> reaction, suggesting the incorporation of hydrogen species into the sample. Considering the

concomitant increase of intensity in the A and B branches in the spectrum, the formation of  $\text{BaH}_2$  can be expected from the hydrogenation of imide as the following:

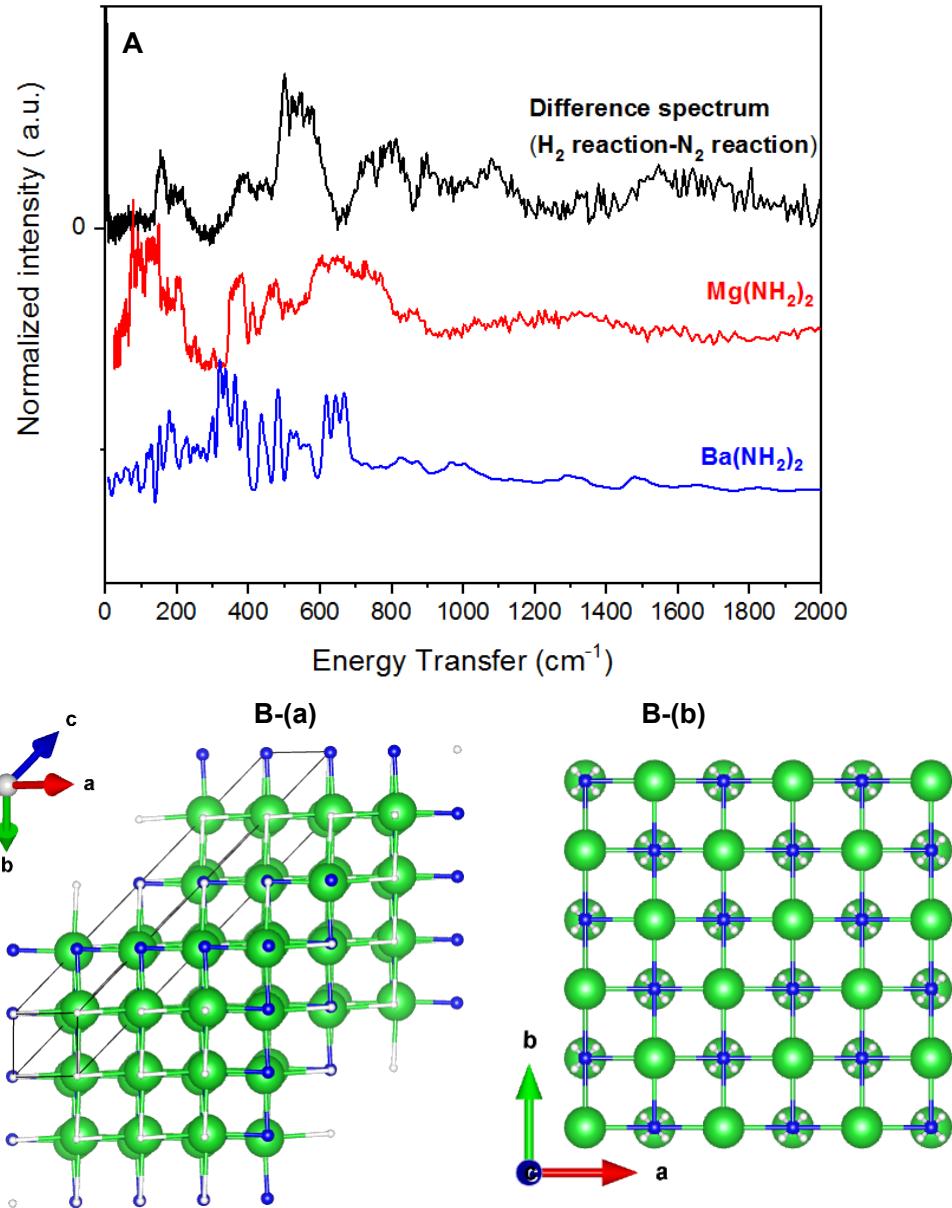


However, the spectral features are somewhat different from the starting  $\text{BaH}_2$ , implying potential formation of new species in addition to  $\text{BaH}_2$  during the hydrogenation step. A difference spectrum is obtained to identify such new species by subtracting the spectrum (after  $\text{H}_2$  reaction) from the one after  $\text{N}_2$  reaction (scaled to similar intensity) and plotted in **Figure 5A**. The main features from 500 to 1200  $\text{cm}^{-1}$  are similar to those of  $\text{BaH}_2$ , again confirming barium hydride production from imide hydrogenation. **It is notable that the peak positions are slightly different from those of pure  $\text{BaH}_2$  (Figure 2A), likely due to a modified local potential around H sites in  $\text{BaH}_2$  in the chemical looping process.** Yet the presence of some extra spectral features below 500  $\text{cm}^{-1}$  not found in  $\text{BaH}_2$  spectrum suggests the formation of new species during the imide hydrogenation. Considering amide is a possible intermediate in the hydrogenation of imide to hydride (see reactions R3-R4), the INS spectra from  $\text{Ba}(\text{NH}_2)_2$  are compared with the obtained difference spectrum in **Figure 5A**.



However, even though the reported spectrum of  $\text{Mg}(\text{NH}_2)_2$ [31] matches fairly well in the region below 500  $\text{cm}^{-1}$  with the features from the new species formed during the  $\text{H}_2$  step, the simulated spectrum from  $\text{Ba}(\text{NH}_2)_2$  (no experimentally measured spectrum available in the literature) does not resemble well. We then turn to the neutron diffraction pattern to examine the changes after  $\text{H}_2$  reaction. The most obvious change is a systematic shift of the  $\text{BaNH}$  diffraction peaks to a lower d-spacing, by about 0.7%. A possible explanation of this shift is the formation of N- deficient barium imide,  $\text{BaN}_{1-x}\text{H}_{1+y}$  ( $0 \leq x, y \leq 1$ ), as  $\text{H}_2$  can react away some of the N to form  $\text{NH}_3$ , and the imide is in a defective intermediate state as it is eventually converted to  $\text{BaH}_2$ . An interesting reference to compare with is  $\text{Ba}_2\text{NH}$ , which has a similar Ba arrangement as in  $\text{BaNH}$  but with some of the N and proton removed as illustrated in **Figure 5B**, resulting in a smaller average Ba-Ba distance and therefore a shift of the corresponding Bragg peaks to lower d- spacing (**Figure 3**). In our case, as  $\text{BaNH}$  is reacted with  $\text{H}_2$ , some of the N will be released (with protons, in the form of  $\text{NH}_3$ ), and the added H will be in the form of  $\text{H}^-$ , forming hydride with Ba. We believe such a similar local environment as in  $\text{Ba}_2\text{NH}$  will lead to reduced lattice constants, and

this intermediate phase  $\text{BaN}_{1-x}\text{H}_{1+y}$  ( $0 \leq x, y \leq 1$ ) is **likely** what we have observed in the neutron diffraction.



**Figure 5.** A). INS difference spectrum subtracted from  $\text{H}_2$  step from  $\text{N}_2$  step over Ni/ $\text{BaH}_2$ . The reported INS spectrum of  $\text{Mg}(\text{NH}_2)_2$  and simulated spectrum of  $\text{Ba}(\text{NH}_2)_2$  are also shown for comparison. B) Crystal structures of (a)  $\text{Ba}_2\text{NH}$  and (b) BaNH. This angle of view highlights the similarity between the two in terms of the overall rock-salt-like arrangement of Ba and N. The differences are mainly the occupancy of the N site, as well as the position and valence state of H. Green ball: Ba; Blue ball: N; while ball: H.

The formation of N-deficient barium imide intermediate was not observed in the previous study of chemical looping for ammonia synthesis using X-ray diffraction[16]. The contrast can be attributable to the different reaction conditions used in our neutron study and the high sensitivity of neutron spectroscopy to hydrogenous species, especially when the species is not fully crystallized. The observation of the mixture of  $\text{BaH}_2$  and possible  $\text{BaN}_{1-x}\text{H}_{1+y}$  ( $0 \leq x, y \leq 1$ ) in the hydrogenation step along with the not fully restored intensity of the INS spectrum (compared with the original  $\text{BaH}_2$  one), indicates that the hydrogenation step could be more difficult than the nitridation step. This is consistent with previous kinetic studies of ammonia synthesis over  $\text{BaH}_2$ -supported transition metals where the N-H hydrogenation rather than the  $\text{N}_2$  activation was considered the rate-limiting step for ammonia synthesis.[12] Of note, the coexistence of  $\text{H}_2$  and  $\text{NH}_3$  in the batch reactor used in our neutron study might also play a role for the incomplete hydrogenation of  $\text{BaNH}$ .

## Conclusions

The chemical looping process for ammonia synthesis over  $\text{Ni}/\text{BaH}_2$  was monitored by *in situ* neutron spectroscopy and neutron diffraction to reveal the structural evolution of  $\text{BaH}_2$  during the different steps. The interpretation of the INS spectra was assisted with first-principles simulations, including lattice dynamics and molecular dynamics simulations. The neutron results exhibited that the ball-milling process, even though producing some  $\text{BaO}$  phase that was shown to be a spectator for ammonia synthesis, can significantly reduce the crystalline size of  $\text{BaH}_2$  and thus potentially increase its reactivity in ammonia synthesis. The transformation from  $\text{BaH}_2$  to  $\text{BaNH}$  was successfully monitored by both INS and neutron diffraction during the  $\text{N}_2$  reaction step. The reverse to  $\text{BaH}_2$  was also evidenced by INS during the  $\text{H}_2$  step, while a likely intermediate phase, N-deficient barium imide, was also indicated by both INS and neutron diffraction. The incomplete hydrogenation of imide to hydride seems to imply that the hydrogenation of  $\text{NH}_x$  species is more difficult than the  $\text{N}_2$  fixation (nitridation) step, consistent with the reported kinetic studies over  $\text{Ni}/\text{BaH}_2$  for chemical looping production of ammonia.

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## Declarations

The authors declare no competing financial interest.

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