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PAPER

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Hierarchical Top-down Bottom-up Calibration with Consideration for Uncertainty and Inter-scale Discrepancy of Peierls Stress of bcc Fe

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ABSTRACT

A hierarchical multiscale model of plasticity in single crystal bcc Fe, framed at the mesoscale pertaining to a statistically representative set of dislocations for each slip system, relies on specification of a set of parameters which are informed using both bottom-up (atomistic) and top-down (experimental) information. We term this specification process a “connection” between bottom-up and top-down pathways to inform the mesoscale model. The connection is considered in the presence of error, uncertainty, and discrepancy between the models. We expand upon a previously developed reconciled top-down and bottom up calibration method to account for anticipated discrepancy between information from different length scales. The results of a previously formulated likelihood-based connection test of the multiscale model suggest a “missing link” may be responsible for part of the inter-scale discrepancy. In this case, this link is assumed to be a relation between the unit-process (single dislocation line) of coordinated kink-pair nucleation on a screw dislocation segment and the many-body dislocation process which manifests in the onset of experimentally observed plastic deformation in a single crystal. A physics-informed discrepancy layer is formulated to improve the connection in the

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3 presence of additional persistent uncertainties. This physics-informed hypothesis testing is
4 demonstrated as an alternative to fully data-driven search methods, particularly applicable to
5 datasets of limited size that are often encountered in such multiscale material modeling
6 applications. Discrete dislocation dynamics simulations are used to investigate the effect of the
7 Peierls stress on straight screw dislocations and in the operation of Frank-Read sources.
8 Discussion concerns the critical role of uncertainty quantification in providing a basis for this
9 form of incremental hypothesis testing.
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20 **1 INTRODUCTION**

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24 Computational modeling of materials involves simulations at a variety of length and time
25 scales (Andrade et al., 2011; Blanc et al., 2002; Ellis and McDowell, 2017; McDowell, 2010,
26 2010; Roters et al., 2010). Increasingly, multiple scales of modeling are used in conjunction to
27 capture connections across scales between mechanisms and performance. This is typically
28 approached by using a set of distinct, scale-specific models. The complexity of hierarchical
29 multiscale model (HMM) development suggests that methods for improving plausible models
30 are needed to arrive at suitable multiscale formulations. The development of such HMMs
31 requires inter-scale connections. These connections are made in the presence of uncertainty
32 (Koslowski and Strachan, 2011; Rizzi et al., 2012).
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46 The practice of evaluating a connection in a HMM between top-down (TD) and bottom-
47 up (BU) information pathways is a recent development (Tallman et al., 2017). In the approach,
48 discrepancies in the TD and BU estimates of model parameters which reflect given physical
49 quantities are reconciled through the use of a statistical likelihood framework. The application of
50 the framework in the reconciliation and later evaluation of the TD and BU pathways constitutes
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3 an empirical or data-driven method. The method quantifies the average magnitude of the
4 distances in the parameter domain across which the combined TDBU posterior density is
5 distributed. Reducing this average distance, and thereby the variance in the probabilistic
6 parameter distributions, is necessary to the identifying and refining of connections between
7 scales.
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16 Physical knowledge can be used to refine a connection between length scales. In cases
17 where inconsistencies can be explained through the physical differences between models of a
18 given quantity, the effects of those differences can be quantified and accounted for in the
19 reconciliation process. For instance, in the simulation of yielding in bcc Fe, atomistic predictions
20 and top-down experimental fits of the Peierls stress in a continuum crystal plasticity model differ
21 by a factor of 2 to 3 (Gröger and Vitek, 2007). The difference may be a result of configurations
22 of a network of dislocations which are susceptible to glide at resolved shear stresses in
23 experiments, which can be well below the values determined through atomistic simulations of
24 nucleation of double kinks on a single segment of a screw dislocation using periodic boundary
25 conditions. The influence of many-body dislocation configurations on glide has been further
26 explored with discrete dislocation dynamics (DDD) modeling (Arsenlis et al., 2012).
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42 In this work, a physics-based discrepancy layer is introduced to capture discrepancy
43 between physical descriptions at different scales such that the model calibration based on TD and
44 BU information pathways is improved. It is demonstrated with a crystal plasticity (CP) model,
45 where parameters are calibrated with discrepancy between TD (experimental observations) and
46 BU (atomistic simulations) descriptions. The discrepancy layer is applied to capture the
47 discrepancy between the modeling scales with regard to one of the parameters (thermal slip
48 resistance). DDD simulations are used to illustrate the discrepancy captured. It is shown that the
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3 inter-scale discrepancy treatment can improve precision of the inter-scale connection. This
4 connection refinement, using additional physical considerations and uncertainty quantification to
5 allow for model discrepancy, contrasts with traditional search methods in this application.
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10 11 **2 Background**

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15 Hierarchical multiscale models (HMM) of materials depend on assumed connections
16 between independently validated models at different length and time scales (McDowell, 2010;
17 Narayanan et al., 2014; Tallman et al., 2017). In a HMM, a mapping between parameters is used
18 to establish this connection. In materials modeling, uncertainty affects any model and thus
19 affects the connections between models (Arendt et al., 2012; Chernatynskiy et al., 2013). The
20 concept of missing physics (Smith, 2013) has been used in model discrepancy approaches
21 (Brynjarsdóttir and O'Hagan, 2014; Ling et al., 2014). In propagation studies, the effects of
22 parametric uncertainties are quantified for the case where a HMM connection is held fixed
23 (Peherstorfer et al., 2018; Rizzi et al., 2012). The confluence of parameter and model-form
24 uncertainty that arises in HMM connection-making requires multiple distinct and relevant
25 sources of information to disentangle, in addition to a strategy for isolating the connection. Here,
26 hypothesis testing, where a presupposed mechanism is used instead of a best-fit, supports the
27 isolation of connection-effects in this complicated uncertainty condition, where uncertainty of
28 multiple model-forms and parameters must be considered at once.
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48 An estimate of the Peierls stress of a bcc Fe crystal may be approached from both
49 continuum and atomistic scales. In a continuum crystal plasticity model, a Kocks-type activation
50 enthalpy-based flow rule for dislocation glide can be calibrated to experiments (cf. Patra et al.,
51 2014). This flow rule includes the thermal slip resistance as a parameter, whose value reflects the
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3 effective level of stress required to activate slip in certain dislocation arrangements present in the
4 single crystal. Atomistic simulations examine a single screw dislocation exhibiting a coordinated
5 kink-pair nucleation process (Narayanan et al., 2014). The energy barrier at zero applied stress is
6 used to inform predictions of the Peierls stress. In this study, the dissimilarity of the two
7 mechanisms through which the Peierls stress is estimated by either continuum or atomistic
8 treatment is used to refine a connection made between the two pathways for estimation in an
9 empirical sense.
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20 The evaluation of a connection in a HMM is often made under data-scarce conditions.
21 Consequentially, model evaluation techniques that rely on partially withholding data (e.g. cross-
22 validation (Braga-Neto and Dougherty, 2004; Efron and Gong, 1983)) are more challenging to
23 implement. In the case of the current application, omitting a proportion of the available data
24 would result in substantial variations in maximum likelihood estimates of parameters. When data
25 are plentiful, such variations indicate an unreliable model. In data-scarce scenarios, the variations
26 are largely a consequence of the small number of data points. Nevertheless, calibrations and
27 evaluations are needed under data-scarce conditions just as urgently as in the other case, and
28 such conditions are often brought about by virtue of constraints on cost of experiments and/or
29 simulations. This work builds on previous efforts to evaluate a connection under data-scarce
30 conditions by including a model discrepancy layer.
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46 Missing physics is a noteworthy cause of model discrepancy. It can either reflect
47 limitation of a given model form or may reflect a key missing spatial or temporal level in the
48 HMM. In the Bayesian calibration of physics models (Kennedy and O'Hagan, 2001), treatment
49 of model discrepancy is critical to minimizing bias (Brynjarsdóttir and O'Hagan, 2014) and
50 maximizing the identifiability (Ling et al., 2014) of calibration parameters, and is accomplished
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3 through including an explicit additive term that adjusts model response or by adjusting
4 parameters to account for discrepancy (Sargsyan et al., 2015). Discrepancy formulations rely on
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6 informative priors to yield results, and in the case of missing physics approaches, the
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8 hypothetical impact of those physics is used to provide those informative priors. In the current
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10 work, discrepancy between the two length scales is attributed to differences between the analysis
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12 of a single dislocation at the atomistic scale and the longer length scale dislocation
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14 configurations that give rise to observed experimental behavior in bcc Fe. The formulation of a
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16 discrepancy layer is informed by this prior assumption, as opposed to a least-squares fitting. In
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18 this work, we examine a discrepancy formulation that involves a mapping between parameters of
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20 the BU and TD model, rather than introducing an explicit additive discrepancy correction to the
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22 model output.
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30 Separately from model discrepancy methods, various latent variable modeling approaches
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32 have been developed to capture hidden causal relationships. These include statistical structural
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34 equation modeling (Bentler and Weeks, 1980), Kalman filtering (Meinhold and Singpurwalla,
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36 1983), non-parametric hidden Markov modeling (Gordon et al., 1993), Gaussian process latent
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38 variable modeling (Titsias and Lawrence, 2004), causal modeling (Pearl, 2009), and their
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40 extensions. Instead of using these approaches to identify the coefficient C_p between TD and BU
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42 parameter spaces in Eq. (2), the value of coefficient is obtained from the scaling factor identified
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44 by Groger and Vitek (2007), which was derived according to physical knowledge about the
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46 mutual interaction between emitted dislocations.
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52 Frank-Read dislocation sources in bcc Fe have been explored (Gröger and Vitek, 2007)
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54 for the potential explanation of the Peierls stress and the related discrepancy found between its
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3 atomistic predictions and experimentally-informed continuum estimates. To approach this
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5 problem, we appeal to physical arguments that may explain why screw dislocations in networks
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7 at mesoscales may activate at applied stress levels much lower than the atomistic estimates of the
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9 Peierls barrier. One such mesoscale model of Gröger and Vitek (2007) assumes that mixed
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11 character dislocations will glide to form loops within a certain distance of the source. Thereafter,
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13 the screw segments which come to dominate the activity of the loops beyond that distance are
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15 assumed to glide at resolved stresses below the Peierls stress due to assistance from the stress
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17 fields of the mixed character loops in their wake. In order for the Frank-Read source mechanism
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19 to operate, the screw segments that form on the first generated dislocation must not prevent the
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21 dislocation from completing a glide loop. In this work, DDD simulations are performed to
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23 examine the Frank-Read source as a mesoscopic explanation of discrepancy in the Peierls stress
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25 between scales.
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32 DDD simulations allow a variety of dislocation configurations to be examined (Amodeo
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34 and Ghoniem, 1990; Sobie et al., 2017a, 2017b). In this work, the simulations are used to
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36 investigate the two dislocation structure idealizations relevant to the scale discrepancy argument
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38 of Gröger and Vitek (2007). These include a simulation of periodic parallel straight screw
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40 dislocations and a simulation of periodically placed Frank-Read sources. The high atomistic
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42 prediction of Peierls stress for screw segments is included in the simulations. An in-depth DDD
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44 study of this problem is left to future work, as the intent here is to explain the concept of the
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46 discrepancy layer.
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3 Methodology

The previously used top-down (TD) and bottom-up (BU) likelihood functions are augmented with an inter-scale discrepancy layer. In this section, the likelihood functions are described and the inter-scale discrepancy is formulated. The data from TD and BU are combined using the method of reconciliation previously detailed (Tallman et al., 2017), which multiplicatively combines surrogate model-based TD likelihood functions and BU penalty functions of Gaussian form to inform a TDBU posterior for parameters. In the TDBU calibration approach, a standard likelihood formulation for the top-down model (or its surrogate) is multiplied by a function which penalizes large differences between BU and TD parameters. The total likelihood represents how much the top down calibration process is informed and influenced by bottom up parameters. Thus, it is not a direct combination of TD and BU data but an updating of TD data by how much information the BU parameters bring to the TD calibration. A brief summary of the method is provided here.

Data from experiments on the temperature and crystallographic loading orientation dependent stress-strain behavior of single crystal bcc Fe (Keh, 1965; Spitzig and Keh, 1970) are gathered. The yield strength (proportional limit) is identified in the experimental data for use in calibrating a set of parameters of the flow rule, i.e.,

$$\dot{\gamma}^{\alpha} = \begin{cases} \dot{\gamma}_0 \exp\left(-\frac{\Delta F_g}{kT} \left(1 - \left(\frac{\tau_f^{\alpha} - s_a^{\alpha}}{s_t^{\alpha}}\right)^p\right)^q\right); & \text{for } \tau_f^{\alpha} > s_a^{\alpha} \\ 0; & \text{for } \tau_f^{\alpha} \leq s_a^{\alpha} \end{cases} \quad (1)$$

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3 in a crystal plasticity model, where g^a is the crystallographic shearing rate on slip system a , g_0
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5 is the pre-exponential factor, DF_g is the activation enthalpy of thermally activated dislocation
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7 glide at zero resolved shear stress, t_f^a is the driving force on the slip system which includes non-
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9 Schmid stresses, s_a^a is the athermal resistance to slip, s_t^a is the thermal resistance to slip, and p
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11 and q are shape parameters.
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18 The CP model is executed for a set of values for each calibration parameter. A set of
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20 second-order polynomial regression models is used to interpolate across the calibration
21
22 parameter space. Each of these models defines a relationship between calibration parameters and
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24 the predicted yield strength of the CP model at a specific temperature and crystallographic
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26 orientation, i.e.,
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$$32 \quad f_i(\theta_1, \dots, \theta_{N_\theta}) = \alpha_0 + \sum_{j=1}^{N_\theta} \alpha_j \theta_j + \sum_{j=1}^{N_\theta} \sum_{k=j}^{N_\theta} \alpha_{N_\theta-1+j+k} \theta_k \theta_j, \text{ for } T_i, g_i \quad (2)$$

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37 where α are regressive coefficients, θ are calibration parameters, T_i is the i-th temperature, and
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39 g_i is the i-th orientation of i experimental observations of the response. Through a standard sum
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41 of squared errors comparison with the experimental values of yield strength, likelihood functions
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43 are generated in terms of the calibration parameters. These likelihood functions are used as the
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45 basis of TD information in calibration parameter space. The BU information is contained in a
46
47 reference estimate of parameters, labeled as ‘reference’ to signify that the estimate originates
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49 externally to the model being calibrated. The reference estimate is used to inform some penalty
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51 functions in a form which can be combined with the TD likelihood functions.
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3 The new model discrepancy layer changes the relationship between the BU and TD
4 parameter spaces. In the previous study, the TD and BU parameter spaces were defined as
5 identical; in other words, the same set of mesoscale CP model parameters were estimated via
6 information from both BU and TD pathways. In the present study, the additional information
7 involving discrepancy relates to the Peierls stress alone; accordingly, the model discrepancy
8 layer here is demonstrated with a single parameter: the thermal slip resistance, s_t . A linear
9 relation between the BU and TD parameter spaces is defined as
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$$s_{t,BU} = C_p s_{t,TD} \quad (3)$$

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21 where $s_{t,BU}$ is the BU-defined thermal slip resistance and $s_{t,TD}$ is the TD-defined thermal slip
22 resistance.
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30 The value $C_p = 2$ is taken from the calculations of Gröger and Vitek (2007) of the stress
31 barrier for dislocation glide in a configuration of Frank-Read sources at a dislocation density of
32 the order of 10^{12}m^{-2} . Their calculations, based in the isotropic elastic theory of dislocations,
33 reflect the effects of interactions of Frank-Read emissions on the effective barrier to dislocation
34 glide. While the Peierls barrier of screw dislocations is unchanged, the configuration of
35 dislocations allows glide to occur at a steady state, at resolved shear stresses of one half the
36 Peierls barrier of screw dislocations. The annihilation of emitted dislocations occurs at a given
37 radius from the Frank-Read sources (which depends on dislocation density). Dislocations are
38 emitted as mixed character and are assumed to become pure screw after gliding for some
39 distance. If there exist N_s screw dislocations at distances x_i from a source and N_m mixed
40 character dislocations at distances $y_k < x_i \forall i$ the i -th screw dislocation will glide when
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$$\sigma_a + \frac{\mu b}{2\pi} \sum_{\substack{j=1 \\ j \neq i}}^{N_s} \frac{1}{x_i - x_j} + \frac{\mu b}{2\pi\alpha} \sum_{k=1}^{N_m} \frac{1}{x_i - y_k} + \frac{\mu b}{2\pi\alpha} \frac{1}{x_i} \geq \sigma_p \quad (4)$$

where σ_a is the applied stress, σ_p is the Peierls barrier of screw b is the Burgers vector of the emitted dislocations, μ is the shear modulus, and α is a constant. The stress exerted by the mixed character dislocations on the screw dislocations allows the Frank-Read source to be active at applied stressed below σ_p . The value $C_p = 2$ taken in this work corresponds to one of many dislocation densities and source spacings considered in the work of Gröger and Vitek (2007).

In the HMM formulation, the value $C_p = 2$ is intended to account for the difference in the Peierls stress levels, between the prediction from atomistic simulations (Narayanan et al., 2014) of purely screw character dislocations, as high as 1000 MPa, and the effective value gathered from experimental observations (Keh, 1965; Patra et al., 2014) which governs larger patterns of dislocations, as low as 390 MPa. The other four parameters (g^a , DF_g , p , and q) remain identical in the calibration set from BU to TD. Given the fixed nature of the relationship that we assume from Eq. 2, the rest of our analysis considers the improvement to the TDBU calibration rendered by this discrepancy approach. After the adjustment, the remaining steps of the TDBU reconciliation are executed.

The likelihood functions are combined multiplicatively with the penalty functions in a constrained likelihood function (CLF). The addition of the discrepancy layer forces the decision of which parameter basis (TD, BU, or another) to use to formulate the CLF. The maximum likelihood estimate is unaffected by the choice; however, the inferred values of parameter

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3 uncertainty based on the CLF will vary. The uncertainty estimates are calculated in terms of
4 calibration parameters normalized to intervals, i.e., $[0,1]$. This prevents the units of the
5 parameters from influencing outcomes. However, it also leads to a situation where the results
6 depend on the values chosen to determine admissible parameter ranges. Given that the goal of
7 this study is to determine if a physics-informed discrepancy layer is sufficient to allow a
8 connection to be made between TD and BU data, it is appropriate to maintain consistent
9 admissible ranges to make such a comparison.
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21 The connection test is performed for the model which includes the discrepancy layer. In
22 the formulation of this test, the discrepancy layer adjusts the BU reference parameter estimate
23 which is incorporated in the calculation of TDBU parameter uncertainty. The difference in the
24 parameter values is shown in Table 1. The original BU values are used in both cases and are
25 translated into a TD native basis using Eq. (2) in the case of the discrepancy inclusive approach.
26 In the approach without discrepancy, the TD basis is arrived at using a one-to-one
27 correspondence.
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38 Table 1. The original and discrepancy adjusted reference estimates used as the BU penalty.
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$\hat{\theta}_{BU}^{ref}$	g_0	DF_g	p	q	s_t
Original	$3.19 \cdot 10^7 s^{-1}$	0.57 eV	0.67	1.18	1040 MPa
Discrepancy Adjusted	$3.19 \cdot 10^7 s^{-1}$	0.57 eV	0.67	1.18	520 MPa

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3 The TDBU calibration method requires evaluations of the uncertainty in TD information
4 through an experimental variance term, S_{exp}^2 for each data point and the uncertainty in BU
5 information through the variance of the penalty, S_p^2 . Both of these parameters influence the
6 results of a connection test, as the test checks for uncertainty reduction, and a lower uncertainty
7 level in inputs leads to a smaller tolerance for mismatch in data sources. In the extreme case,
8 fully certain TD observations mean that new BU data do not add additional information or value.
9 This sensitivity to input uncertainties, as is common in Bayesian methods, is a useful capacity of
10 the method. Given that not all data are accompanied by uncertainty estimates, additional effort
11 may be needed to ensure the test results appropriately reflect the available information. The
12 connection test involves quantifying the cost and benefit of using BU information to inform the
13 TD parameters. The connection may be very helpful or may be not helpful at all (e.g. the TD and
14 BU parameters conflict or completely disagree with each other).

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17 Here, a summary of the connection test is given, which is described in more detail
18 elsewhere (Tallman et al., 2017). The BU penalty functions and TD likelihood functions are
19 included in a constrained likelihood function (CLF). This CLF is raised to an exponent, the value
20 of which is informed using the assumed cost of the connection, u . A value of $u = 1$ reflects no
21 connection cost where the TD and BU data agree, and a value of $u = 0$ reflects an infinite
22 connection cost, where complete disagreement exists between the TD and BU data. The result is
23 normalized to satisfy the law of total probability over the admissible parameter space to give a
24 probability density function (PDF) in the parameter space. A quadrature-based Bayesian
25 approach is used to calculate the PDF(θ). The calibration variance is calculated by finding the
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square Euclidean distance of the PDF from the centroid of the PDF. These equations are given in Table 2.

Table 2. Equations defining the terms of the TDBU connection test.

$Penalty(\hat{\theta}) = \sum_i^{\hat{\theta} \text{ in } \hat{\theta}} \left(\frac{\hat{\theta}_i - \theta_{i,ref}}{\sigma_p} \right)^2,$ $SSE(\theta) = \sum_{j=1}^{Data} \frac{(f_j(\theta) - Y^E(\mathbf{x}_j))^2}{\sigma_{exp,j}^2}$	<p>Penalty functions and sum square error calculations using surrogates f_j in place of CP model, physical parameter value \mathbf{x} at data point j, experimental response Y^E</p>
$CLF(\theta) = \exp \left[-\frac{1}{2} (Penalty(\theta) + SSE(\theta)) \right]$	<p>Constrained Likelihood Function</p>
$u = \frac{1}{SSE(\hat{\theta}) + Penalty(\hat{\theta})}$	<p>Connection cost u is evaluated at the maximum likelihood estimate, $\hat{\theta}$</p>
$PDF(\theta) = Const. \times \exp \left[-\frac{u}{2} (Penalty(\theta) + SSE(\theta)) \right]$	<p>Probability density function</p>
$\sigma_\theta^2 = \int_{\theta_{min}}^{\theta_{max}} PDF(\theta) Dist^2(\theta, \bar{\theta}) d\theta$	<p>Calibration variance S_q^2 evaluated from centroid of parameter space, $\bar{\theta}$</p>

DDD simulations were pursued to provide a mechanistic description of the scale discrepancy layer introduced in this work. The DDD simulations are performed using a fast-Fourier transform accelerated DDD code (Bertin et al., 2015). Edge dislocations in bcc Fe have been shown to be highly mobile at low applied stresses (Monnet and Terentyev, 2009). The

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3 simulations assign the Peierls stress to be 1 GPa for screw dislocation segments, consistent with
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5 atomistic modeling results, and 10 MPa for edge and mixed character dislocation segments. An
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7 overdamped mobility law is used for all dislocation segments. The simulations are carried out
8
9 using periodic boundary conditions. Shear stress is applied at three levels: the stress at which a
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11 straight screw dislocation begins to glide (here 1 GPa), 75% (750 MPa), and 50% of that stress
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13 (500 MPa). The simulation volume is a cube of 10,000 unit cells in length (2.87 nm). A Frank-
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15 Read source of width 160 nm is assumed.
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20 The DDD simulations of a Frank-Read source are carried out using a prismatic
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22 dislocation loop. The glissile portions of the loop are initially pure edge dislocation segments on
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24 $\frac{1}{2}\langle 111 \rangle \{110\}$ slip systems. As a result, these segments are initially highly mobile under resolved
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26 shear stress. The simulations are carried out to examine whether or not the screw segments which
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28 form as the dislocation bows out will inhibit the formation of glide loops. This step is identified
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30 as critical to the feasibility of the explanation provided by Gröger and Vitek (2007).
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39 **4 Results**

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43 The impact of the discrepancy layer is evaluated in this section. To provide a point of
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45 comparison, the uncertainty estimates made in previous work without a model discrepancy
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47 treatment are included (Tallman et al., 2017). As mentioned above, DDD simulations of a Frank-
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49 Read source in bcc Fe are also included to highlight the assumptions which underlie the
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51 mechanistic description of the scale discrepancy used to motivate this work.
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The parameter S_p in the penalty function (listed in Table 2) controls the influence of the BU information on the probability density of the calibration parameters. As a result, the comparison of the variance of the TDBU calibration and the TD calibration depends strongly on the value of S_p . In Figure 1, the calibration variance σ_θ^2 is shown as a function of S_p for both connection formulations considered in this work. The calibration variance is a monotonic function for the case of TDBU with the discrepancy layer, decreasing as BU information is included. For the TDBU without the discrepancy layer, the parameter uncertainty increases as BU information is included, up until the BU information begins to outweigh the TD information (at S_p values < 0.1).

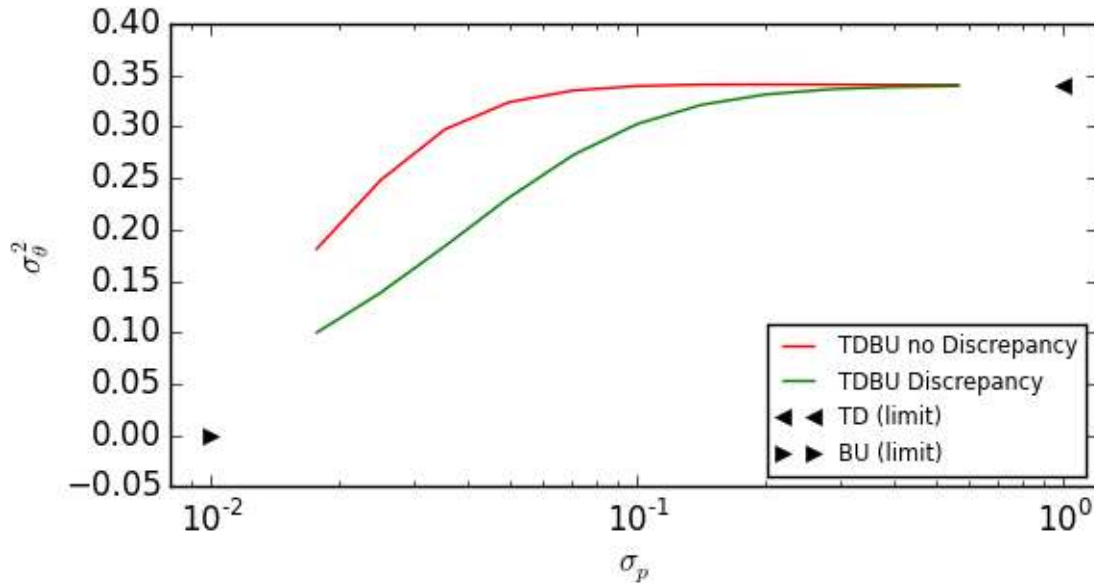


Figure 1. Plot of the calibration variance as function of the penalty parameter. Limits are plotted for context.

DDD simulations were carried out at 3 applied shear stress levels, 1000, 750, and 500 MPa. The limited mobility of screw dislocation segments is observed to control the operation of the Frank-Read source. Snapshots of the dislocation lines are taken normal to the glide plane at

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3 arbitrary times during the simulations to highlight the mechanism present at each stress level.
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5 These images are included in Figure 2. The dislocation line in Figure 2a continues to bow out,
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7 but the applied stress does not drive screw segments to glide. In Figures 2b and 2c, the applied
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9 stress is sufficiently high to cause screw segments to glide such that the Frank-Read source may
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11 generate glide loops. In Figure 2c, the screw dislocations are generated as a result of the glide
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13 loops annihilating at the periodic boundary in the direction of the edge dislocation motion. These
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15 screw dislocations were able to glide as necessary for the Frank-Read source to operate, however
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17 the simulation was not continued long enough to approach conditions of source saturation. Still,
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19 they illustrate the key point of the mesoscopic configuration controlling the effective value of the
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21 lattice resistance that one would interpret from experiments based on operation of distributed
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23 Frank-Read sources. The results support the discrepancy layer assignment $C_p = 2$ in Eq. (2)
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25 qualitatively if not quantitatively.
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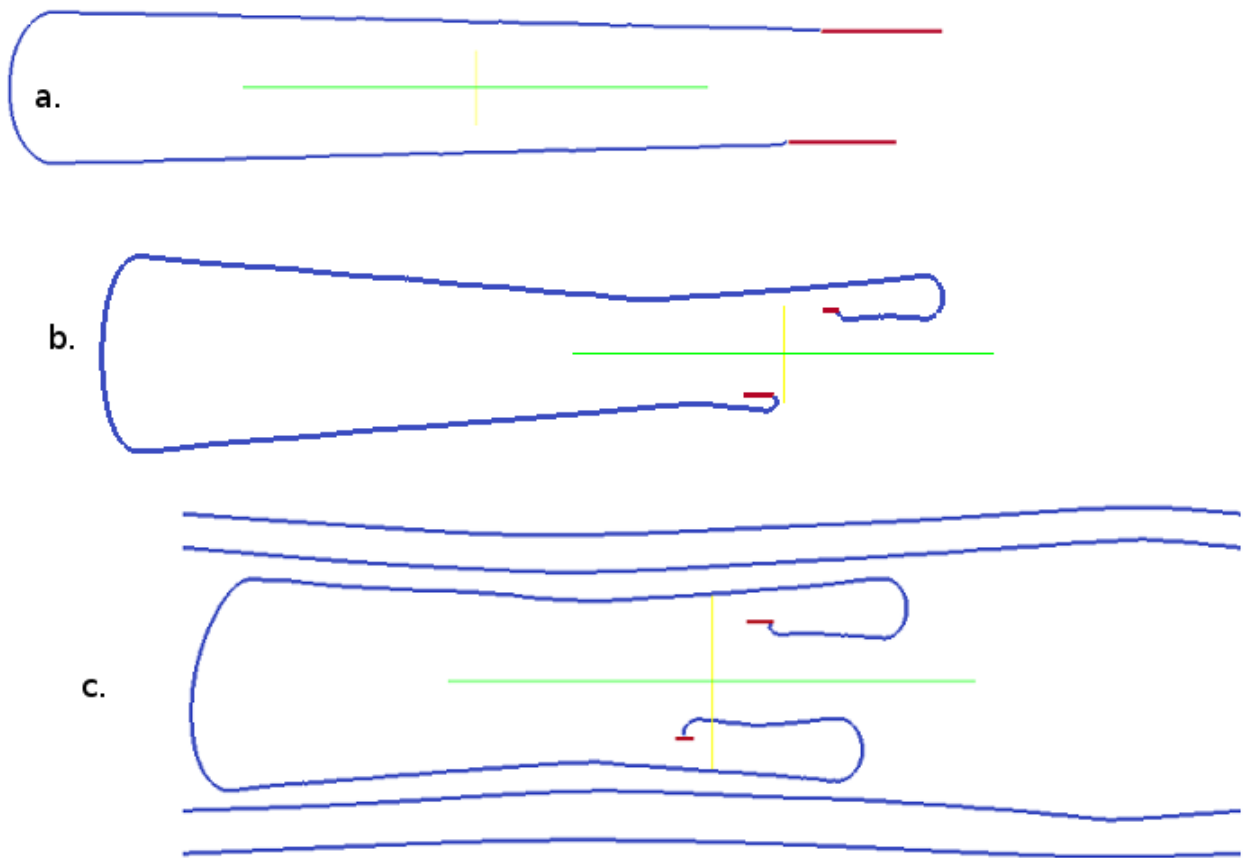


Figure 2. Snapshots of DDD simulations of Frank-Read sources under applied shear stress of 500 MPa (a.), 750 MPa (b.), and 1000 MPa (c.).

5 Discussion

The benefits of the discrepancy layer are discussed. The dislocation configurations which allow a reduction in the thermal resistance to slip in bcc Fe are explored in this section. Additionally, the difficulty of adjusting connections in HMMs without informative prior estimates is addressed. Atomistic modeling which would increase the precision of predictions is suggested.

The addition of the discrepancy layer allows for the BU information to contribute to a reduction of uncertainty in the HMM. In previous work (Tallman et al., 2017), the combination

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3 of TD and BU information was concluded to increase uncertainty, once connection cost was
4 considered. Here, the discrepancy layer allows for a reconciliation of TD and BU information
5 which amounts to a reduction of the parametric uncertainty, as seen in Figure 1. At the preferred
6 value of $S_p = 0.1$, the value for the TDBU uncertainty estimate, $S_{q,\text{TDBU}}^2 = 0.30$, is a reduction
7 from the TD only estimate, $S_{q,\text{TD}}^2 = 0.34$. The interpretation of this result follows.

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17 The immediate result of the including the discrepancy layer is to achieve a connection
18 which adds mesoscale information to the calibration of the HMM. This result is predicated on the
19 uncertainty of each data source (TD via S_{exp}^2 and BU via S_p^2). As more data or more reliable
20 data become available, the initial uncertainties associated with the TD-only and BU-only
21 calibrations may reduce. The surrogate model used to define likelihood functions in parameter
22 space might lead to reductions in initial uncertainty. All of these reductions make further
23 reductions to uncertainty from a TDBU calibration more difficult. In other words, as the model
24 improves, the connection must also improve to continue to be of value.

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37 A longer-term result is the suggested importance of the discrepancy between the
38 atomistic parameter estimates and the continuum values in HMM development. This is to say
39 that more investigation may return analogous discrepancy formulations for other parameters in
40 the original TDBU calibration parameter set, further reducing uncertainty in connections.
41 Furthermore, the development of HMM connections can be incremental and driven by the TDBU
42 connection test. The framework proposed here helps to guide the modeler to seek specific
43 arguments and mechanisms for discrepancy to improve the BU and TD model connection, so it
44 very much resides in the realm of physics-based modeling.

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3 The choice of connection basis parameters is not resolved in this study. The example
4 shown used the TD parameters as a basis to allow the admissible parameter space to remain
5 consistent with a comparison case. The complication arising from adjusting the admissible
6 parameter space is that, in addition to providing a finite space for parameter calibration, the
7 admissible parameter space also provides a means of normalizing the magnitude of variations in
8 parameter values. To allow for comparisons of information connection scenarios which support
9 different admissible parameter ranges, or even different parameter spaces, a different
10 normalization scheme is needed. To provide a consistent basis for evaluating parameter
11 variations, independent of units, a basis can be made in a model of effects, i.e., the size of a
12 parameter value change is considered relative to a change incurred in a quantity of interest (QoI)
13 which is some function of the parameters, the model response, or a combination thereof. An
14 example is made using model response as the QoI. The unit of normalization is approximated

$$\theta_{norm} \simeq E \left[\frac{\Delta \theta}{\Delta Y} \right] \text{ for all } \theta \text{ in } [\theta_{min}, \theta_{max}] \quad (5)$$

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38 where q_{norm} is the normalizing value for calibration parameter q , and Y is the model response.
39 By normalizing changes in parameters to measured effects, the assumption that changes in
40 parameters have magnitudes relative to the admissible ranges may be relaxed.
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47 In cases where different parameter sets are considered, an approach to make a like
48 comparison may require the consideration of the effects of each parameter in terms of similarity.
49 In this case, a QoI with at least as many dimensions as parameters is best. Taken at some
50 representative number of points in parameter space, the similarity of effects is calculated as
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$$Sim(q_1, q_2) = \frac{1}{\|DY^2\|} \frac{DY}{Dq_1} \cdot \frac{DY}{Dq_2} \quad (6)$$

where the similarity value ranges from 0 to 1. The application of the TDBU calibration to connections between parameter spaces of different shapes and sizes is left to future work.

The DDD simulations were carried out at very low effective dislocation densities ($\sim 3 \cdot 10^{10} \text{ m}^{-2}$). At higher dislocation densities, the attraction between dislocation dipoles would contribute and assist sessile segments in gliding. This suggests that the Frank-Read source explanation of the Peierls stress discrepancy is likely to be plausible at realistic dislocation densities (greater than $1 \cdot 10^{12}$), where dislocations may find sinks, despite the lack of activity shown in the DDD simulations of this work at lower applied stress (500 MPa). An investigation of the Frank-Read source within more realistic microstructures is left to future work.

There is need for atomistic investigation of stress-dependent mobility of mixed character dislocations in bcc Fe. The Frank-Read source was shown in this work to be able to operate under an applied stress of 75% of the Peierls stress of screw dislocations. To refine this determination further, atomistic simulations will be needed. The DDD simulations rely upon the tangent vector of each dislocation segment to determine screw/edge character. As a result, the glide of the screw segments which occurred due to the propagation of mixed character regions of the dislocation line along screw segments was highly dependent on the angles (between the dislocation line and the Burgers vector) at which a segment was considered a screw segment. The DDD code is not atomistic, and hence cannot explicitly describe atomistic mechanisms such as kink-pair nucleation; such information is heuristically introduced in the segment mobility

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3 functions. The contribution of DDD simulations here is to match behavior as exhibited by
4 dislocations. To provide more accuracy in predicting the relation of the effective Peierls barrier
5 to Frank-Read sources to the Peierls stress of pure screw dislocations, atomistic simulations of
6 glide loops will be necessary at larger scale such that the DDD code can be made to match the
7 behavior shown in atomistic simulation.
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16 The discrepancy relation has explored physics-based calculations to restrict flexibility in
17 the method. The presence of discrepancy in TD and BU information would be trivial to deal with
18 if a discrepancy layer was allowed to modify the BU parameter space such that $\hat{\theta}_{BU} = \hat{\theta}_{TD}$. The
19 arguments against this practice is that such a treatment has no understanding to impart to the
20 modeler, and renders the information from BU to be without impact on the calibration itself. On
21 the contrary, the method employed here used a connection test to determine that a specific
22 multibody explanation of discrepancy in atomistic and continuum predictions of Peierls stress in
23 bcc Fe is sufficient to reconcile the BU and TD information in the current HMM.
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3 This paper describes objective technical results and analysis. Any subjective views or
4 opinions that might be expressed in the paper do not necessarily represent the views of the U.S.
5 Department of Energy or the United States Government.
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10 REFERENCES

11
12
13
14
15 Amodeo, R.J., Ghoniem, N.M., 1990. Dislocation dynamics. II. Applications to the formation of
16 persistent slip bands, planar arrays, and dislocation cells. *Phys. Rev. B* 41, 6968–6976.
17
18 <https://doi.org/10.1103/PhysRevB.41.6968>
19
20
21
22

23 Andrade, J.E., Avila, C.F., Hall, S.A., Lenoir, N., Viggiani, G., 2011. Multiscale modeling and
24 characterization of granular matter: From grain kinematics to continuum mechanics. *J. Mech.*
25 *Phys. Solids* 59, 237–250. <https://doi.org/10.1016/j.jmps.2010.10.009>
26
27
28
29
30

31 Arendt, P.D., Apley, D.W., Chen, W., 2012. Quantification of Model Uncertainty: Calibration,
32 Model Discrepancy, and Identifiability. *J. Mech. Des.* 134, 100908.
33
34 <https://doi.org/10.1115/1.4007390>
35
36
37
38

39 Arsenlis, A., Rhee, M., Hommes, G., Cook, R., Marian, J., 2012. A dislocation dynamics study
40 of the transition from homogeneous to heterogeneous deformation in irradiated body-centered
41 cubic iron. *Acta Mater.* 60, 3748–3757. <https://doi.org/10.1016/j.actamat.2012.03.041>
42
43
44
45
46

47 Bentler, P.M., Weeks, D.G., 1980. Linear structural equations with latent variables.
48 *Psychometrika* 45, 289–308. <https://doi.org/10.1007/BF02293905>
49
50
51

52 Bertin, N., Upadhyay, M.V., Pradalier, C., Capolungo, L., 2015. A FFT-based formulation for
53 efficient mechanical fields computation in isotropic and anisotropic periodic discrete dislocation
54
55
56
57
58

1
2
3 dynamics. *Model. Simul. Mater. Sci. Eng.* 23, 065009. <https://doi.org/10.1088/0965->
4
5 0393/23/6/065009
6

7
8
9 Blanc, X., Bris, C.L., Lions, P.-L., 2002. From Molecular Models¶to Continuum Mechanics.
10
11 *Arch. Ration. Mech. Anal.* 164, 341–381. <https://doi.org/10.1007/s00205-002-0218-5>
12

13
14 Braga-Neto, U.M., Dougherty, E.R., 2004. Is cross-validation valid for small-sample microarray
15
16 classification? *Bioinformatics* 20, 374–380. <https://doi.org/10.1093/bioinformatics/btg419>
17

18
19
20 Brynjarsdóttir, J., O’Hagan, A., 2014. Learning about physical parameters: the importance of
21
22 model discrepancy. *Inverse Probl.* 30, 114007. <https://doi.org/10.1088/0266-5611/30/11/114007>
23

24
25 Chernatynskiy, A., Phillpot, S.R., LeSar, R., 2013. Uncertainty Quantification in Multiscale
26
27 *Simulation of Materials: A Prospective. Annu. Rev. Mater. Res.* 43, 157–182.
28
29 <https://doi.org/10.1146/annurev-matsci-071312-121708>
30
31

32
33 Efron, B., Gong, G., 1983. A Leisurely Look at the Bootstrap, the Jackknife, and Cross-
34
35 Validation. *Am. Stat.* 37, 36–48. <https://doi.org/10.1080/00031305.1983.10483087>
36
37

38
39 Ellis, B.D., McDowell, D.L., 2017. Application-Specific Computational Materials Design via
40
41 Multiscale Modeling and the Inductive Design Exploration Method (IDEM). *Integrating Mater.*
42
43 *Manuf. Innov.* 6, 9–35. <https://doi.org/10.1007/s40192-017-0086-3>
44
45

46
47 Gordon, N.J., Salmond, D.J., Smith, A.F.M., 1993. Novel approach to nonlinear/non-Gaussian
48
49 Bayesian state estimation. *IEE Proc. F - Radar Signal Process.* 140, 107–113.
50
51 <https://doi.org/10.1049/ip-f-2.1993.0015>
52
53

1
2
3 Gröger, R., Vitek, V., 2007. Explanation of the discrepancy between the measured and
4
5 atomistically calculated yield stresses in body-centred cubic metals. *Philos. Mag. Lett.* 87, 113–
6
7 120. <https://doi.org/10.1080/09500830601158781>

8
9
10
11 Keh, A.S., 1965. Work hardening and deformation sub-structure in iron single crystals deformed
12
13 in tension at 298°k. *Philos. Mag.* 12, 9–30. <https://doi.org/10.1080/14786436508224942>

14
15
16 Kennedy, M.C., O'Hagan, A., 2001. Bayesian calibration of computer models. *J. R. Stat. Soc.*
17
18 *Ser. B Stat. Methodol.* 63, 425–464. <https://doi.org/10.1111/1467-9868.00294>

19
20
21
22 Koslowski, M., Strachan, A., 2011. Uncertainty propagation in a multiscale model of
23
24 nanocrystalline plasticity. *Reliab. Eng. Syst. Saf.* 96, 1161–1170.
25
26 <https://doi.org/10.1016/j.ress.2010.11.011>

27
28
29
30 Ling, Y., Mullins, J., Mahadevan, S., 2014. Selection of model discrepancy priors in Bayesian
31
32 calibration. *J. Comput. Phys.* 276, 665–680. <https://doi.org/10.1016/j.jcp.2014.08.005>

33
34
35
36 McDowell, D.L., 2010. A perspective on trends in multiscale plasticity. *Int. J. Plast., Special*
37
38 *Issue In Honor of David L. McDowell* 26, 1280–1309.
39
40 <https://doi.org/10.1016/j.ijplas.2010.02.008>

41
42
43
44 Meinhold, R.J., Singpurwalla, N.D., 1983. Understanding the Kalman Filter. *Am. Stat.* 37, 123–
45
46 127. <https://doi.org/10.1080/00031305.1983.10482723>

47
48
49
50 Monnet, G., Terentyev, D., 2009. Structure and mobility of the $12\langle 111 \rangle \{112\}$ edge dislocation
51
52 in BCC iron studied by molecular dynamics. *Acta Mater.* 57, 1416–1426.
53
54 <https://doi.org/10.1016/j.actamat.2008.11.030>

1
2
3 Narayanan, S., McDowell, D.L., Zhu, T., 2014. Crystal plasticity model for BCC iron
4 atomistically informed by kinetics of correlated kinkpair nucleation on screw dislocation. *J.*
5 *Mech. Phys. Solids* 65, 54–68. <https://doi.org/10.1016/j.jmps.2014.01.004>
6
7

8
9
10 Patra, A., Zhu, T., McDowell, D.L., 2014. Constitutive equations for modeling non-Schmid
11 effects in single crystal bcc-Fe at low and ambient temperatures. *Int. J. Plast.* 59, 1–14.
12
13 <https://doi.org/10.1016/j.ijplas.2014.03.016>
14
15

16
17
18 Pearl, J., 2009. *Causality*. Cambridge University Press.
19

20
21
22 Peherstorfer, B., Willcox, K., Gunzburger, M., 2018. Survey of Multifidelity Methods in
23 Uncertainty Propagation, Inference, and Optimization. *SIAM Rev.* 60, 550–591.
24
25 <https://doi.org/10.1137/16M1082469>
26
27

28
29
30 Rizzi, F., Najm, H., Debusschere, B., Sargsyan, K., Salloum, M., Adalsteinsson, H., Knio, O.,
31 2012. Uncertainty Quantification in MD Simulations. Part I: Forward Propagation. *Multiscale*
32 *Model. Simul.* 10, 1428–1459. <https://doi.org/10.1137/110853169>
33
34

35
36
37 Roters, F., Eisenlohr, P., Hantcherli, L., Tjahjanto, D.D., Bieler, T.R., Raabe, D., 2010.
38 Overview of constitutive laws, kinematics, homogenization and multiscale methods in crystal
39 plasticity finite-element modeling: Theory, experiments, applications. *Acta Mater.* 58, 1152–
40
41 1211. <https://doi.org/10.1016/j.actamat.2009.10.058>
42
43
44

45
46
47 Sargsyan, K., Najm, H.N., Ghanem, R., 2015. On the Statistical Calibration of Physical Models.
48
49 *Int. J. Chem. Kinet.* 47, 246–276. <https://doi.org/10.1002/kin.20906>
50
51
52
53
54
55
56
57
58
59
60

1
2
3 Smith, R.C., 2013. Uncertainty Quantification: Theory, Implementation, and Applications.
4
5 SIAM-Society for Industrial and Applied Mathematics, Philadelphia.
6
7

8
9 Sobie, C., Capolungo, L., McDowell, D.L., Martinez, E., 2017a. Modal analysis of dislocation
10
11 vibration and reaction attempt frequency. *Acta Mater.* 134, 203–210.
12
13 <https://doi.org/10.1016/j.actamat.2017.02.005>
14
15

16
17 Sobie, C., Capolungo, L., McDowell, D.L., Martinez, E., 2017b. Scale transition using
18
19 dislocation dynamics and the nudged elastic band method. *J. Mech. Phys. Solids* 105, 161–178.
20
21 <https://doi.org/10.1016/j.jmps.2017.05.004>
22
23

24
25 Spitzig, W.A., Keh, A.S., 1970. Orientation and temperature dependence of slip in iron single
26
27 crystals. *Metall. Trans.* 1, 2751–2757. <https://doi.org/10.1007/BF03037811>
28
29

30
31 Tallman, A.E., Swiler, L.P., Wang, Y., McDowell, D.L., 2017. Reconciled Top-down and
32
33 Bottom-up Hierarchical Multiscale Calibration of bcc Fe Crystal Plasticity. *Int. J. Multiscale*
34
35 *Comput. Eng.* 15. <https://doi.org/10.1615/IntJMultCompEng.2017021859>
36
37

38
39 Titsias, M.K., Lawrence, N.D., 2004. Gaussian process latent variable models for visualisation of
40
41 high dimensional data, in: *Adv. in Neural Inf. Proc. Sys.*
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60